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A comparison study of boron emitter passivation by silicon oxide and a PECVD silicon nitride stack

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Abstract

Surface recombination of minority charge carriers is a significant loss mechanism in crystalline-Si (c-Si) solar cells. For n^+ (phosphorous) doped regions a sufficient surface passivation is achieved by using a silicon nitride (SiN_x) film deposited by Plasma Enhanced Chemical Vapour Deposition (PECVD). However, on p^+ boron doped regions, SiN_x does not passivate the surface. Instead, a stack of layers comprising of a thin silicon dioxide (SiO_2), or aluminium oxide (Al_2O_3), and a SiN_x film is commonly used to passivate boron doped regions.

In this study we investigated the passivation quality of boron doped emitters by varying the composition of $\text{SiO}_2/\text{SiN}_x$ stack layers. For this purpose, n -PERT (passivated emitter, rear totally-diffused) solar cells with boron doped front side emitter and phosphorous doped back-surface-field (BSF), as well as symmetrical boron doped structures, were fabricated on 6-inch n -type wafers.

The results show that the optimum passivation is achieved by a stack layer of a thermal SiO_2 , with a thickness of at least 10 nm, and a SiN_x layer with a low refractive index. The chemical composition of SiN_x capping layer plays an important role for surface passivation. A more Si-rich SiN_x layer show significant degradation in surface passivation of the stack, due to the increase in the density of interface states (D_{it}) and fixed positive charges (Q_{tot}) at the interface.

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Keywords: Solar cell; boron emitter; n -PERT; passivation; NAOS

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1. Introduction

Surface passivation of heavily doped p^+ emitters is required in order to realize cost-effective industrial high-efficiency solar cells based on n -type silicon wafers. The boron emitter can be passivated by a dielectric having a low density of interface states (D_{it}), potentially in combination with a high fixed negative charge density. However conventional SiN_x is not suitable for p^+ surface passivation [1]. One of the reasons for the relatively poor surface passivation of p^+ surfaces by SiN_x is related to the very high fixed charge density in PECVD SiN_x . It is attributed to the band gap defects that SiN_x form during deposition, which are more detrimental to p^+ Si [2, 3]. It was observed that the fixed charge density can be reduced by adding a thin silicon oxide layer between the crystalline silicon and the SiN_x film. Another approach to passivate p^+ layer is based on using thin Al_2O_3 dielectric with the negative charge as a surface passivation layer and SiN_x as an antireflection coating (ARC) layer [4-7].

2. Experimental part

2.1. Sample preparation

In this work we investigated the passivation of boron doped emitters with different $\text{SiO}_2/\text{SiN}_x$ stack layers. For this purpose we used, on the one hand, an n -PERT cell with a homogeneous diffused front boron emitter and a phosphorous back-surface-field (BSF) and on the other hand, symmetrical boron diffused test structure. Both device structures were fabricated on 6-inch n -type monocrystalline Si wafers with base resistivity of 2.5 Ωcm . Standard industrial processes were used, which included wet chemical alkaline texturization, cleaning by HCl, HF, and HF/O_3 solutions, diffusion in quartz tube furnaces containing phosphoryl chloride (POCl_3 ; for n^+ BSF) or boron tribromide (BBr_3 ; for p^+ emitter). For the solar cell devices the back side was chemically polished and the n^+ BSF diffusion was passivated by a thermal $\text{SiO}_2/\text{SiN}_x$ stack. Screen printing and firing through of commercial silver containing pastes were applied to both sides for metallization. A schematic cross-section of the studied solar cells is presented in the Fig. 1.

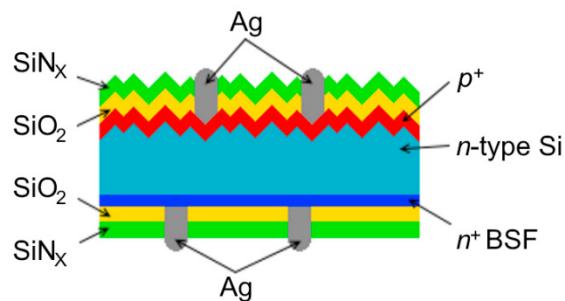


Fig. 1. Schematic cross-section of the investigated n -PERT cells.

2.2. Passivation stacks

All solar cells were processed identically, with the exception of boron emitter passivation. For the boron emitter 5 different passivation stacks were investigated and compared, as detailed in Table I. First, the stacks G1 and G2 aimed to investigate the effect of SiO_2 interfacial layer thickness on the passivation quality of the $\text{SiO}_2/\text{SiN}_x$ stack. A symmetrical boron diffused test structure was used, in addition to the solar cells, to find the optimum thickness of the SiO_2 in the stack. Here, a thermal SiO_2 was grown in situ during the BBr_3 diffusion on boron emitter surface with a thickness of about 40 nm. Subsequently the thick SiO_2 layer was partially etched (thinned) in HF (2%) solution to yield various thicknesses, followed by a 62 nm PECVD SiN_x antireflection coating (ARC; $n \approx 2$) deposition of the capping layer to complete the passivation stack.

Then, the stacks G1 and G3 aimed to compare different SiO_2 interface passivation layers, whereas the SiN_x capping layer was kept the same (ARC; $n \approx 2$). Stack G3 contained a 1.5 nm chemical SiO_2 interfacial layer grown

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