Journal of Nuclear Materials 492 (2017) 239-243

Contents lists available at ScienceDirect

Journal of Nuclear Materials

journal homepage: www.elsevier.com/locate/jnucmat

Immobilization and bonding scheme of radioactive iodine-129 in silver tellurite glass



JOURNAL O

1927

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HIGHLIGHTS

- Chemically durable silver tellurite glass for immobilization of ¹²⁹I was developed.
- Silver tellurite glass can contain up to 11.21 wt% of iodine.
- There's no significant volatilization of iodine (<5%) during the vitrification.
- Local structure of iodine in silver tellurite glass is similar to β -AgI.

ARTICLE INFO

Article history: Received 31 January 2017 Received in revised form 14 May 2017 Accepted 19 May 2017 Available online 22 May 2017

Keywords: Sequestration Silver tellurite glass Radioactive iodine Vitrification XAS

1. Introduction

 129 I is a radioactive isotope that is generated during the fission reactions inside nuclear power plants, and can also be released during reprocessing of spent nuclear fuel [1]. 129 I has a long half-life (~1.57 \times 10⁷ y), high solubility and mobility in water, and high volatility, so immobilization of 129 I is a significant objective in management of nuclear wastes [2]. Most of 129 I is collected in the form of AgI on the surface of Ag-activated zeolite filters as the 129

G R A P H I C A L A B S T R A C T



ABSTRACT

Silver tellurite glasses with melting temperatures < 700 °C were prepared to immobilize the ¹²⁹I that normally volatilizes during high-temperature melting. Glasses have densities of $6.31 \pm 0.1 \text{ g/cm}^3$ and glass transition temperatures of 165 ± 3 °C that provide thermal stability at the disposal site. Iodine waste loading in glasses was as high as 12.64 wt% of all metallic elements and 11.21 wt% including oxygen. Normalized elemental releases obtained from the product consistency test were well below US regulation of 2 g/m². Iodines are surrounded by four Ag⁺ ions forming [Ag₄]³⁺ units that are further connected to tellurite network through bonds with non-bridging oxygens.

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moves through the off-gas treatment process. Silver ions capture I with good efficiency, and AgI has low solubility in water [3]. However, AgI must be firmly immobilized inside a chemically-stable matrix to prevent it from re-entering the environment by evaporation or chemical reactions.

Ceramic waste forms were made from AgI-containing sorbents and ceramic matrix using hot isostatic pressing (HIPing) to sequester ¹²⁹I. Although volatilization of iodine was minimized during the processing, remained as adsorbents without making any chemical bonding with the matrix [4,5]. Silver phosphate glasses have been investigated for immobilization of AgI wastes [6–8]. AgI content [AgI] in glasses can be increased to 60 mol%. In this case, the

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Table 1	
Nominal and analyzed comp	ositions of the glass (wt.%).

Starting Chemica	ıls	Nominal composition	Elements excluding oxygens	Nominal composition	Analyzed composition
TeO ₂		41.29	Те	37.18	35.99 ± 0.85
Ag ₂ O		25.69	Ag (combined)	38.16	39.38 ± 0.84
Bi ₂ O ₃		11.33	Bi	11.45	11.99 ± 0.40
AgI	Ag	9.97	_	_	_
	I	11.72	Ι	13.21	12.64 ± 0.02
Total		100.00	Total	100.00	100.00

glass transition temperature decreased to ~55 °C for below the control temperature (<150 °C) at the deep geological disposal sites [9]. Therefore, the glass waste forms should have $T_g > 150$ °C to avoid changes in flow properties or crystallization behavior in glasses [10]. Addition of 2 wt% of Al₂O₃ or Bi₂O₃ to silver phosphate glasses can improve their chemical durability but usually leads to crystallization [8]. Borosilicate glasses are not suitable for vitrification of ¹²⁹I because of their high melting temperatures (>1000 °C) and low iodine solubility of <1 wt% [11].

Several ionically-conductive glass systems that use TeO₂-, V₂O₅and MoO₃-based glasses have been developed [12–15]. In all of these glasses, preparation of homogeneous glasses with high ionic conductivities requires high [AgI]. These glasses can bear up to 80 mol% AgI, and therefore may provide applications in immobilization of ¹²⁹I. We have focused on glasses in the tellurite TeO₂-Ag₂O-AgI system, with addition of Bi₂O₃ to increase the chemical durability [16]. Tellurite glasses normally show good chemical durability [17], and are therefore, suitable waste forms for immobilization of ¹²⁹I. Melting temperatures of tellurite glasses (<700 °C) are low enough to prevent excessive volatilization of AgI during melting. The goal of this study was to develop new glass waste forms to sequester AgI wastes generated by pyrochemical processing of spent fuels. TeO2-based glasses prepared can accommodate >10 wt% I, and show good chemical and thermal durabilities. Chemical bonding status of I with the glass network was investigated using X-ray absorption spectroscopy (XAS). The oxidation state of I is -1; it forms a bond with Ag⁺ with coordination of ~4, which is similar to that in borosilicate and borate glasses [18,19].

2. Experimental methods

The glass (Table 1) was prepared by weighing and mixing powders of TeO₂, Ag₂O, Bi₂O₃ and AgI (with purities > 99%). Batches were melted at 700 °C for 30 min in alumina crucibles in ambient atmosphere, then the melts were quenched by pouring onto brass molds in air. X-ray fluorescence (XRF, PANalytical axios minerals) spectroscopy was used to analyze the glass composition quantitatively. Density was measured at room temperature by the Archimedes method using diethyl phthalate as a medium. T_g was determined using a differential thermal analyzer (DTA, Shimadzu DTG/TA-60) at a heating rate of 10 °C/min.

To evaluate the chemical durability of the glass, a product consistency test (PCT) was performed following the procedures

Table 2 Concentration [C (ppm)] and normalized elemental releases [r (g/m^2)] of elements from the glass after the 7-day PCT.

Element	Concentration (ppm)	Normalized elemental releases (g/m ²)
Te Ag Bi I	10.9 2.3×10^{-2} 2.6×10^{-3} 6.2×10^{-2}	$\begin{array}{c} 3.9 \times 10^{-2} \\ 8.0 \times 10^{-5} \\ 3.0 \times 10^{-5} \\ 6.5 \times 10^{-4} \end{array}$

described in American Society for Testing and Materials (ASTM) [20]. Glasses were crushed using mortar and pestle and sieved to collect powders with diameters of 75–150 μ m. Powders were ultrasonically washed with deionized water (DIW) and ethanol to remove fine particles and impurities, then dried at 90 °C for 12 h. Samples (1.5 g) of powders were immersed in 15 mL of DIW in a Teflon vessel and kept at 90 ± 2 °C for 7 days in an oven. The solution was filtered through a syringe with a 0.45- μ m filter. Concentrations of elements in the solution were analyzed using inductively coupled plasma mass spectroscopy (ICP-MS, NexION 350D, Perkin-Elmer SCIEX). Normalized elemental releases [g/m^2] were calculated as

$$r_i = \frac{C_i}{f_i(A/V)} \tag{1}$$

where c_i is concentration of element *i* in the solution [ppm = g/m³ in deionized water], f_i is mass fraction of element *i* in the glass, and A/V is ratio of the glass powder surface area to solution volume [m⁻¹]. A/V is 849 m⁻¹ when density of glass is 6.31 g/cm³.

lodine K-edge X-ray absorption spectra were collected at beam line 10C of the Pohang Light Source (PLS) at the Pohang accelerator laboratory, South Korea. Spectra were collected in fluorescence mode with a Si (1 1 1) monochromator crystal. Spectra of crystalline standards (AgI, KI, KIO₃) were collected to adjust the lodine K-edge energy $E_0 = 33,169$ eV, then spectra of the glasses were recorded from 200 eV below the iodine K-edge to 1000 eV above the edge. Spectra thus collected were processed using Athena software to remove background and noise, then radial



Fig. 1. lodine K-edge XANES spectra of three crystalline materials and silver tellurite glass.

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