

## Short Communication on “*In-situ* TEM ion irradiation investigations on $\text{U}_3\text{Si}_2$ at LWR temperatures”



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### ABSTRACT

The radiation-induced amorphization of  $\text{U}_3\text{Si}_2$  was investigated by *in-situ* transmission electron microscopy using 1 MeV Kr ion irradiation. Both arc-melted and sintered  $\text{U}_3\text{Si}_2$  specimens were irradiated at room temperature to confirm the similarity in their responses to radiation. The sintered specimens were then irradiated at 350 °C and 550 °C up to  $7.2 \times 10^{15}$  ions/cm<sup>2</sup> to examine their amorphization behavior under light water reactor (LWR) conditions.  $\text{U}_3\text{Si}_2$  remains crystalline under irradiation at LWR temperatures. Oxidation of the material was observed at high irradiation doses.

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### 1. Introduction

The tsunami and subsequent nuclear accident in Fukushima stimulated global efforts to search for nuclear fuels with enhanced accident tolerance. The accident tolerant fuel (ATF) program aims to replace the conventional  $\text{UO}_2$ -zirconium alloy solution with a novel fuel design that can withstand a loss of coolant accident (LOCA) for a significantly longer period of time [1]. Having superior thermal conductivity and uranium density compared to  $\text{UO}_2$  [2],  $\text{U}_3\text{Si}_2$  has been widely regarded as a promising ATF candidate [3,4]. To evaluate the qualification of  $\text{U}_3\text{Si}_2$  as a LWR fuel, its fuel performance, especially fission gas behavior, must be experimentally examined. In addition, development of advanced fuel performance codes that can precisely predict the fuel behavior of  $\text{U}_3\text{Si}_2$  in LWRs calls for experimental microstructural references and validations of fission gas behavior. Previous studies on arc-melted  $\text{U}_3\text{Si}_2$  at research reactor temperatures (<250 °C) show that ion irradiation completely amorphizes  $\text{U}_3\text{Si}_2$  at a very low dose level (approximately 0.3 dpa) [5]. As amorphization significantly alters the

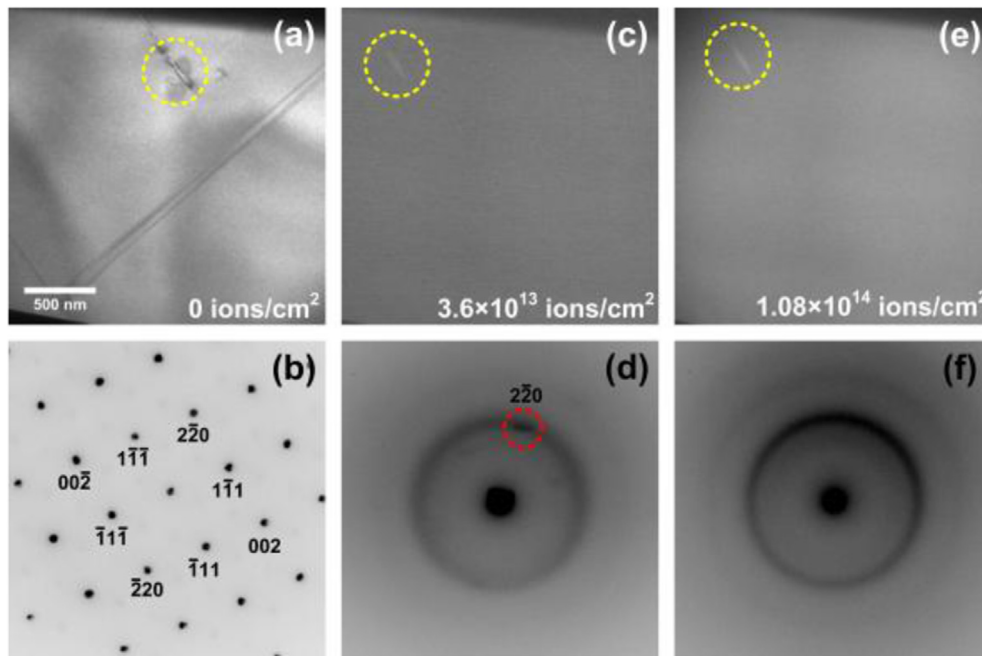
microstructure of the material, and therefore interferes with the evolution of fission gas bubbles [6,7], the radiation-induced amorphization behavior of  $\text{U}_3\text{Si}_2$  at LWR temperatures (from approximately 350 °C to 800 °C) must be well characterized. In addition, when cold pressing and sintering technique is used in fabrication, the microstructure of produced  $\text{U}_3\text{Si}_2$  fuel pellets features reduced grain size, and precipitation of secondary phases such as USi and  $\text{UO}_2$  [8]. This difference in microstructure may also lead to amorphization behaviors different from those in the arc-melted fuel material. Therefore, it is important to understand and characterize any difference in irradiation behavior of the  $\text{U}_3\text{Si}_2$  samples fabricated by the aforementioned methods. Although in-pile irradiation tests are essential to provide reliable experimental data describing material behavior in nuclear reactors, ion irradiation provides an inexpensive and time-saving option to obtain valuable qualitative information about irradiation effects on nuclear materials. In this regard, the combination of transmission electron microscopy (TEM) and *in-situ* ion irradiation makes it possible to capture the kinetics of radiation-induced microstructural modifications [9], including amorphization [5,10–12], in materials. Therefore, an *in-situ* TEM ion irradiation technique was utilized in this study to characterize the amorphization behavior of  $\text{U}_3\text{Si}_2$  at LWR conditions.

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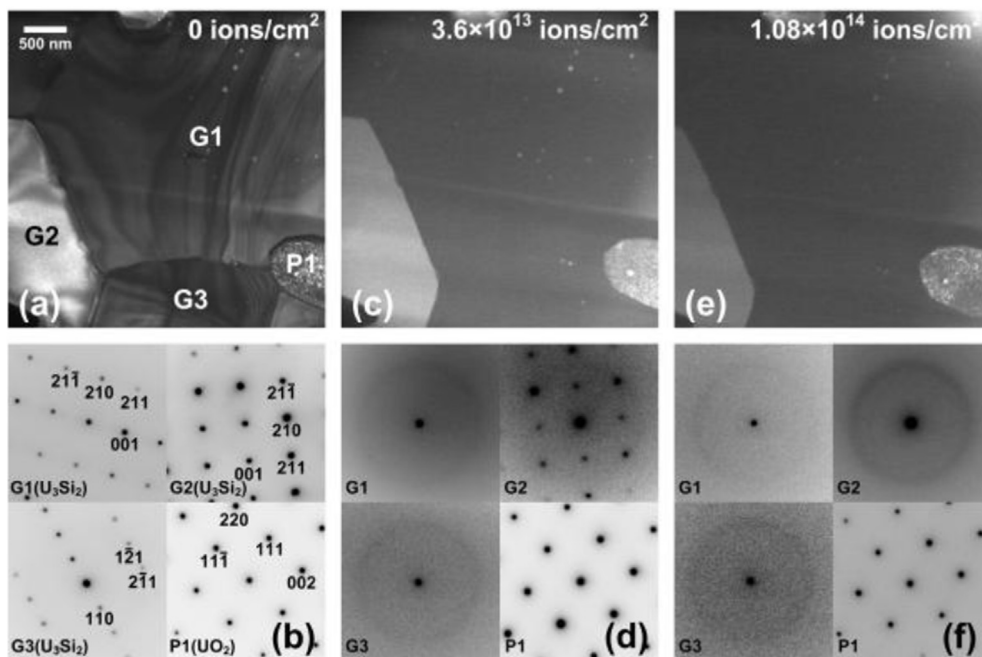
## 2. Experiments

Fine uranium powder (92.5 wt%) and silicon powder (7.5 wt%) were mixed and pressed at 225 MPa. The powder mixture was then arc melted to produce  $U_3Si_2$  ingots. To fabricate  $U_3Si_2$  pellets, the arc-melted ingots were further comminuted into fine powder. The

$U_3Si_2$  powder was then cold pressed and sintered in an Ar atmosphere to form fuel pellets using the same technique as for samples irradiated in-pile for the ATF-1 campaign. The details of this fabrication can be found in Ref. [8]. The arc-melted ingots are relatively pure, containing only 1.5 vol%  $U_3Si$  precipitates, whereas the sintering technique introduced  $USi$  and  $UO_2$  secondary phases



**Fig. 1.** Amorphization of arc-melted  $U_3Si_2$  irradiated by 1 MeV Kr at room temperature: (a)/(b) Prior to ion irradiation, the arc-melted specimen contains stacking faults and shows clear  $U_3Si_2$  diffraction pattern; (c)/(d) the majority of the  $U_3Si_2$  phase becomes amorphous at  $3.6 \times 10^{13}$  ions/cm<sup>2</sup> (0.1 SRIM dpa), eliminating the preexistent stacking faults; (e)/(f) full amorphization occurs at  $1.08 \times 10^{14}$  ions/cm<sup>2</sup> (0.3 SRIM dpa).



**Fig. 2.** Amorphization of cold pressed and sintered  $U_3Si_2$  irradiated by 1 MeV Kr at room temperature: (a)/(b) prior to ion irradiation, three crystalline  $U_3Si_2$  grains (G1, G2, and G3) and one  $UO_2$  precipitate (P1) were identified in the specimen; (c)/(d) at  $3.6 \times 10^{13}$  ions/cm<sup>2</sup> (0.1 SRIM dpa), G1 and G3 become completely amorphous, whereas G2 has distinguishable diffraction pattern, showing various amorphization rates depending on crystallographic orientation; (e)/(f) At  $1.08 \times 10^{14}$  ions/cm<sup>2</sup> (0.3 SRIM dpa), all three grains become completely amorphous, while the  $UO_2$  precipitate remains stable and crystalline during the ion irradiation at room temperature.

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