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# Competition of XA and L2<sub>1</sub>B ordering in Heusler alloys $Mn_2CoZ$ (Z = Al, Ga, Si, Ge and Sb) and its influence on electronic structure

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#### A R T I C L E I N F O

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#### ABSTRACT

Competition between the highly-ordered XA structure and disordered L2<sub>1</sub>B structure in Heusler alloys  $Mn_2CoZ$  (Z = Al, Ga, Si, Ge, Sb) has been investigated. The relative stability of the two structures strongly depends on the main group element Z. When Z belongs to Al or Ga, the XA structure is stabler, but when Z belongs to Si, Ge or Sb, the L2<sub>1</sub>B structure gains stability and is lower in energy. This is related to the different number of valence electrons in main group element Z, which influences the DOS structure near the Fermi level and changes N(E<sub>F</sub>). The energy difference  $\Delta E$  between the XA and L2<sub>1</sub>B structures may be used to estimate the tendency to form L2<sub>1</sub>B structure in different Heusler alloys qualitatively. A large negative  $\Delta E$  is preferable to retard the A-C site disorder and retain the highly-ordered XA structure. That is just the case in Mn<sub>2</sub>CoAl. A robust half-metallicity is observed in Mn<sub>2</sub>CoAl and Mn<sub>2</sub>CoGa, their spin gapless semiconducting character will be destroyed and replaced by a half-metallic state if L2<sub>1</sub>B disorder occurs. Finally, these results suggest that the L2<sub>1</sub>B structure should be considered together with XA structure when discussing the electronic structure of "inverse" Heusler alloys.

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#### 1. Introduction

In recent years, the Mn<sub>2</sub>-based Heusler alloys have attracted much attention for their interesting properties and possible applications in many technical fields. One important application of Mn<sub>2</sub>-based Heusler alloys is spintronic materials, till now quite a few Mn<sub>2</sub>-based Heusler alloys have been reported to be half-metals or spin gapless semiconductors (SGSs). Some typical examples are Mn<sub>2</sub>VAl, Mn<sub>2</sub>VSi, Mn<sub>2</sub>FeZ (Z = Al, Sb), Mn<sub>2</sub>CoZ (Z = Al, Ga, Si, Sb), Mn<sub>2</sub>CuSb, Mn<sub>2</sub>ZrSi, etc. [1–12]. The half-metallic materials have a 100% spin polarization of the conduction electrons at the Fermi level E<sub>F</sub> and are of great importance in spintronics. SGS is an intermediate state between the well known half-metallic ferromagnets and gapless semiconductors. In case of SGS, one spin channel has an open band gap at E<sub>F</sub> like a half-metal but the other spin channel has a zero-width gap like a gapless semiconductor [5,13], thus the conducting electrons or holes are not only 100% spin

http://dx.doi.org/10.1016/j.intermet.2016.10.001 0966-9795/© 2016 Elsevier Ltd. All rights reserved. polarized but also easily excited. Among these  $Mn_2$ -based Heusler alloys,  $Mn_2CoZ$  (Z = Al, Ga, In, Si, Ge, Sn, Sb) are particularly interesting, for they are not only predicted to be half-metals/SGSs theoretically, but also can be realized experimentally [4,5,14]. It has been found that the unique properties of  $Mn_2CoZ$  alloys are strongly related to the preferred occupation of Mn and Co atoms in the cubic lattice of Heusler alloys [15].

Heusler alloys crystallize in a highly-ordered cubic structure and have a stoichiometric composition of  $X_2YZ$ , where X and Y are transition metal elements, and Z is a main group element. In Heusler alloys there are four Wyckoff-positions namely A (0, 0, 0), B (0.25, 0.25, 0.25), C (0.5, 0.5, 0.5) and D (0.75, 0.75, 0.75), respectively. The transition metal elements X, Y enter A, B, C sites and main group element Z always enters D sites in the cubic lattice. The site preference of transition metal elements is usually determined by the number of their valence electrons: atoms with more electrons tend to occupy the A and C positions while the atoms with fewer electrons prefer the B position [16]. So in Mn<sub>2</sub>CoZ the most possible structure is that one Mn and one Co occupy (A, C) sites and the other Mn enters the B site. In previous literature, people usually assume that Mn and Co occupy the A and C sites respectively [4,17],







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which results in a highly-ordered XA structure ( $Hg_2CuTi$ -type, space group No.216), as shown in Fig. 1.

However, a new possible structure named L2<sub>1</sub>B may also be considered when discussing the electronic structure and magnetic properties of Heusler alloys Mn<sub>2</sub>CoZ. The L2<sub>1</sub>B structure evolves from the ordered XA structure [18,19]. In L2<sub>1</sub>B type Mn<sub>2</sub>CoZ, one Mn and one Co occupy the A and C sites randomly rather than respectively, which results in space group Fm-3m, the same to the  $L2_1$  structure. In Fig. 1, we compared the crystal structures of  $L2_1B$ and XA type Mn<sub>2</sub>CoAl as an example. In some Fe-based Heusler alloys, the L2<sub>1</sub>B structure is found to has a lower total energy compared with XA structure and is stabler [20]. In some Mn<sub>2</sub>-based Heusler alloys, the L2<sub>1</sub>B structure may increase the total energy slightly, but due to the effect of enthalpy of mixing, it can still be observed in actual samples at room temperature [19]. It has also been found that the L2<sub>1</sub>B structure can affect the electronic and magnetic properties of Heusler alloys obviously [21]. Thus it is quite meaningful to investigate the electronic structure and stability of half-metallicity in Heusler alloys under possible L2<sub>1</sub>B structure.

In this paper, we investigated the competition between L2<sub>1</sub>B and XA ordering in Heusler alloys  $Mn_2CoZ$  (Z = Al, Ga, Si, Ge, Sb). A close relation between the phase stability of L2<sub>1</sub>B structure and main group element Z has been observed. L2<sub>1</sub>B structure is more preferable in several  $Mn_2CoZ$  alloys. But the half-metallicity in them is robust under either XA or L2<sub>1</sub>B structure.

#### 2. Calculation methods

The electronic structures of  $Mn_2CoZ(Z = Al, Ga, Si, Ge, Sb)$  were calculated by using CASTEP code based on pseudopotential method with a plane-wave basis set [22,23]. The interactions between the atomic core and the valence electrons were described by the ultrasoft pseudopotential [24]. The electronic exchange–correlation energy was treated under the generalized-gradient-approximation (GGA) [25]. Supercell approach is used for the calculations of L2<sub>1</sub>B structure. The random occupation of Mn and Co at (A, C) sites is treated in a 16-atom supercell, which resulting in a chemical formula of Mn<sub>8</sub>Co<sub>4</sub>Z<sub>4</sub>. The detail can be found in Fig. 1. For all cases, a plane-wave basis set cut-off of 500 eV was used. A mesh of  $20 \times 20 \times 20$  or  $16 \times 16 \times 16$  *k*-points was employed for Brillouin zone integrations in the XA and L2<sub>1</sub>B structures, respectively. These parameters ensured good convergences for total energy. The convergence tolerance for the calculations was selected as a difference on total energy within  $1 \times 10^{-6}$  eV/atom.

#### 3. Results and discussions

#### 3.1. Competition between L2<sub>1</sub>B and XA ordering in Mn<sub>2</sub>CoZ

First, We investigated the site preference of Mn and Co in  $Mn_2CoZ$  (Z = Al, Ga, Si, Ge, Sb) and compared the phase stability of XA and L2<sub>1</sub>B structures. During structural optimization, we



**Fig. 1.** Crystal structures of XA and L2<sub>1</sub>B type Mn<sub>2</sub>CoAl, in the L2<sub>1</sub>B structure, Mn and Co occupy A, C sites randomly, the L2<sub>1</sub>B supercell indicates the exact structure used for calculation, in which Mn and Co atoms occupy four A and four C sites alternately.

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