

## Invited Paper

# Fabrication of vertical thin-GaN light-emitting diode by low-temperature Cu/Sn/Ag wafer bonding

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## ABSTRACT

Vertical thin-GaN LED was successfully fabricated on the GaN LED epi-layers grown on the patterned-sapphire substrate with the pyramidal pattern by low-temperature Cu/Sn/Ag wafer bonding at 150 °C. An inverted pyramidal pattern formed on the n-GaN surface after the GaN epi-layer was transferred onto Si wafer, which resulted from the pyramidal pattern on the patterned-sapphire substrate. The inverted pyramidal pattern has an equivalent function with roughening the n-GaN surface. With higher inverted pyramidal pattern coverage, the light extraction efficiency can be greatly enhanced. In addition, we found that the 4-fold increase (from 13.6% to 53.8%) in the pyramidal pattern coverage on patterned-sapphire substrate only gives the GaN LED epi-layer about 5.7% enhancement in the internal quantum efficiency.

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## 1. Introduction

Recently, a new LED structure (vertical-GaN LED) was developed for the high-power LED applications [1–4]. The two key processes of vertical-GaN LEDs are (1) wafer bonding and (2) laser-lift off (LLO) techniques. First, the GaN/sapphire wafer has to be bonded with the transferring Si wafer. Then, the sapphire wafer was stripped off by KrF excimer laser (248 nm). Finally, the initial GaN LED epi-layers were transferred onto the Si wafer. So far, vertical thin-GaN LED are fabricated by using the GaN LED epi-layer grown on the planer sapphire substrate. The flat n-GaN/sapphire interface seems having better process window for focusing the laser beam during the laser lift-off process. Recently, the patterned-sapphire substrate techniques have been widely used in GaN-based LEDs [5,6]. With the break-through of the patterned-sapphire substrate technique, the efficacy of high-brightness GaN-based LEDs has been driven to a record-high of 150 lm/W [7,8]. The efficacy enhancement of GaN-based LEDs with the patterned-sapphire substrate technique is generally attributed to the improvement in both light extraction efficiency and internal quantum efficiency [9–13]. The regular patterns created on the sapphire substrate, which counteracts the effect of the total internal reflection (TIR) at the GaN/sapphire interface [9]. And, the enhancement in the internal quantum efficiency benefits from the reduction of threading dislocations by possible lateral growth of GaN epi-layer on the patterned-sapphire substrate [9,14–17].

Owing to the advantages of the patterned-sapphire substrate mentioned above, it would be of interest to fabricate the vertical

thin-GaN LED chips with the GaN LED epi-layers grown on the patterned-sapphire substrate. For fabricating the vertical thin-GaN LED, the GaN LED epi-layers wafer first has to be bonded with a transferring substrate wafer by wafer bonding. Then, the patterned-sapphire substrate wafer would be stripped off. As a result, the GaN LED epi-layers can be transferred onto the desired transferring substrate, such as, Si wafer. How do the vertical thin-GaN LED fabrication processes affect the merits of the GaN LED epi-layers on the patterned-sapphire substrate has not yet been studied. For example, after the patterned-sapphire substrate being stripped off, the n-GaN layer with the inverted pattern would become the emitting surface. So, how does the inverted pattern on the n-GaN surface affect the light extraction efficiency ( $\eta_{LEE}$ ), is an important issue to be understood. In this study, we fabricate the vertical thin-GaN LED chips with the GaN LED epi-layers grown on the patterned-sapphire substrate and study the effect of the inverted pattern on the performance of the vertical thin-GaN LED.

## 2. Experimental procedures

Numerous patterning features produced on the patterned-sapphire substrate by either dry etching or wet etching processes, which includes circle cavities, square cavities, hemispheric bumps and trenched stripes, have been studied [1,9,18–20]. Yet, no matter what etching process is used to create the patterns, a hard-mask ( $\text{SiO}_2$  in most cases) lithographic process is required on the flat c-plane sapphire wafer. In this study, a mask-free wet-etching process was used to produce a so-called nature-patterned-sapphire substrate (n-pss). The n-pss wafers was done by immersing planar sapphire wafers into a mixing solution ( $\text{H}_3\text{PO}_4:\text{H}_2\text{SO}_4 = 3:1$ ) at 260 °C for 30 min, and 60 min. Fig. 1 shows the SEM image of the

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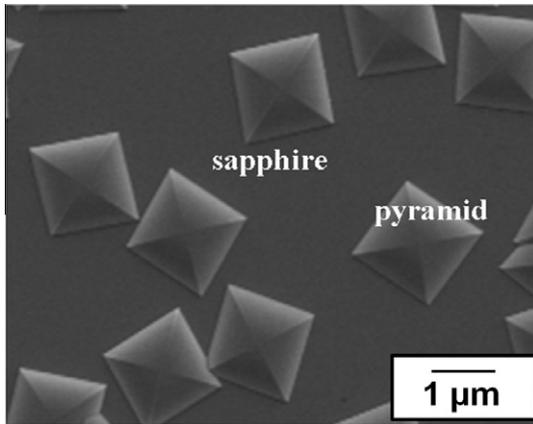


Fig. 1. SEM image of the pyramidal pattern on the etched sapphire wafer.

facet pyramids on the etched sapphire surface. The height and the width of the pyramid is about  $0.2 \mu\text{m}$  and  $1\text{--}2 \mu\text{m}$ , respectively. Then, GaN LED epi-layers were grown on the n-pss wafer with the pyramidal pattern by MOCVD process. The LED epi-layer structure includes a  $1.8 \mu\text{m}$  thick undoped GaN layer and a  $2.5 \mu\text{m}$  thick Si-doped n-type GaN cladding layer, an active region of  $450 \text{ nm}$  emitting wavelength with six periods of InGaN/GaN multiple quantum wells (MQWs), and a  $0.3 \mu\text{m}$  thick Mg-doped p-type GaN cladding layer. After MOCVD epitaxial process, the GaN epitaxial n-pss wafers are ready for the fabrication processes of the vertical thin-GaN LED chips.

For fabricating the vertical-GaN LED structure, an excellent reflective metal layer is essential for the p-contact scheme on the p-GaN layer. Ag is known to have a high reflectivity in the visible regime. Prior to the deposition of the Ag reflective layer, a  $20 \text{ \AA}$  Ni adhesion layer has to be deposited on the p-GaN surface. Then, a  $2\text{-}\mu\text{m}$  Ag reflective layer are deposited on the p-GaN surface of the GaN epitaxial/sapphire wafers. Beside the function of the reflective layer the thick  $2\text{-}\mu\text{m}$  Ag layer also serves as the bonding layer on the GaN/sapphire wafer side.

For the wafer bonding process, the Cu/Sn/Ag bonding system was studied in this present work. On the GaN/sapphire wafer side,  $2\text{-}\mu\text{m}$  Ag layer is deposited as the bonding layer to bond with the Si wafer. On the Si wafer side, Cu/Sn bonding metallizations are prepared. The Cu/Sn metallization on the Si wafers were sequentially deposited with a  $500\text{-}\text{\AA}$  Cr adhesion layer, a  $500\text{-}\text{\AA}$  Pt barrier layer, a  $500\text{-nm}$  Cu layer, and a  $2\text{-}\mu\text{m}$  Sn bonding layer. The above metallization structures were deposited by E-Gun deposition process. The wafer bonding process is described below. First, the Si wafer and GaN/sapphire sapphire wafer with proper bonding metallizations were loaded into a graphite bonding fixture. Both wafers were intimately contacting with a compressive pressure of  $2 \text{ MPa}$ . Then, the sapphire/Si wafer bonding pair is placed in a vacuum furnace with a base pressure of  $5 \times 10^{-2} \text{ torr}$  at  $150 \text{ }^\circ\text{C}$  for  $30\text{--}60 \text{ min}$ .

After the GaN/sapphire wafer is bonded with the Si wafer, the back-side of the sapphire wafer was irradiated by KrF  $248 \text{ nm}$  excimer laser. A thin GaN buffer-layer right above the sapphire substrate absorbs the energy of the incident excimer laser and decomposed to Ga droplets and  $\text{N}_2$  gas. As a result, the GaN epi-layer can be striped-off from the initial grown sapphire wafer. The by-product of the Ga metal droplets resided on the n-GaN surface, which would affect the subsequent analysis on the transferred GaN epi-layer. The dilute HCl acid solution (10%) was used to remove Ga droplets on the n-GaN surface. Then, the suitable n-contact metal Cr/Pt/Au pad can be fabricated on the cleaned n-GaN epi-layer. Fig. 2 illustrates the finished structure of a vertical

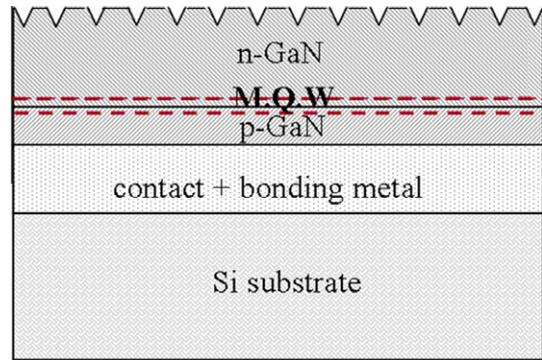


Fig. 2. The finished structure of a vertical thin-GaN LED chip on the Si substrate.

thin-GaN LED chip on the Si substrate. The chip size is about  $1 \text{ mm} \times 1 \text{ mm}$ . After the process of n-contact pad, the finished LED chips were measured by an integral sphere measurement system.

### 3. Results and discussions

Fig. 3 shows the SEM cross-sectional images on the bonding interface at  $150 \text{ }^\circ\text{C}$  for  $30\text{--}60 \text{ min}$ . The SEM examination result

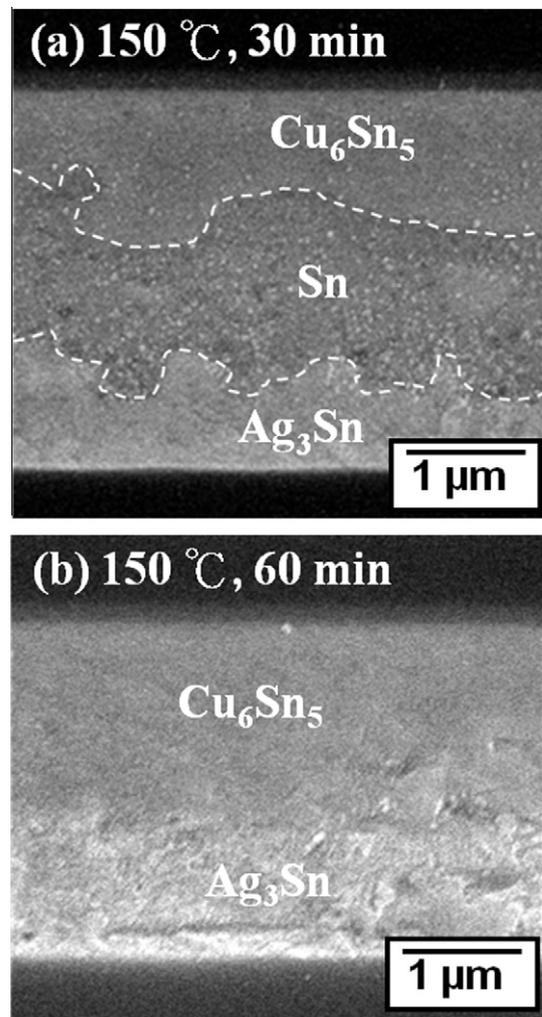


Fig. 3. The SEM cross-sectional images on the bonding interface at  $150 \text{ }^\circ\text{C}$  for (a)  $30$  and (b)  $60 \text{ min}$ .

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