

Investigation on I-V characteristics of current induced metal insulator transition in VO₂ device



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ARTICLE INFO

Article history:

Received 11 June 2017

Received in revised form

8 August 2017

Accepted 9 August 2017

Available online 12 August 2017

Keywords:

VO₂

Metal-insulator transition

I-V characteristics

ABSTRACT

The I-V characteristics of two terminal planar VO₂ film devices are investigated as the devices undergo the current induced metal insulator transition (I-MIT). The I-MIT occurred when the device resistivity reached $\sim 7 \Omega\text{cm}$, where metallic grains formed initial conductive current path within insulating matrix. The transition time needed for the I-MIT increased with increasing external resistance, R_{EXT} , connected to the device in series, i.e. $\sim 390 \mu\text{s}$ ($R_{\text{EXT}} = 5 \text{ k}\Omega$) to $\sim 1400 \mu\text{s}$ ($R_{\text{EXT}} = 20 \text{ k}\Omega$). The transition time is closely related to the RC time delay from capacitance discharge of the VO₂ device. During the I-MIT, the amount of discharge current was estimated as large as $\sim 100 \text{ mA}$, which was larger than the current just before the I-MIT. After the I-MIT, the current density decreased from $1.1 \times 10^6 \text{ A/cm}^2$ to $6.5 \times 10^5 \text{ A/cm}^2$, suggesting a large temperature changes up to $\sim 300^\circ\text{C}$.

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1. Introduction

Vanadium dioxide (VO₂) exhibits various dramatic physical phenomena, such as metal insulator transitions (MIT), optical reflectivity change, structural phase transition, and super elasticity, owing to the abrupt collapses of the subtle metastable insulating state among the competing interactions of lattice, spin, charge, and orbital degrees of freedom caused by diverse external stimuli such as changes in temperature, pressure, current, voltage, and light [1–5]. These unique physical properties of VO₂ make it one of the appealing candidates for future advanced materials in countless applications, including ultrafast sensors, Mott transistors, optical switching devices, and future memory devices [6–9].

Among the diverse stimuli that induce MIT in VO₂, changes in electric current and voltage are considered to be the most promising method for use in future VO₂ applications [10,11]. Temperature-induced MIT occurs in over the entire VO₂ area owing to the uniform temperature change of the sample, while the electrically-induced MIT occurs only in local conductive areas like

conduction channels placed inside of the insulating phase of VO₂ samples [12]. There are two competing models explaining the mechanism of the electrically induced MIT: Joule heating model and electric-field-induced breakdown model [7,13]. In the Joule heating model, the MIT occurs when the sample temperature is raised to the transition temperature by Joule heat. On the other hand, in the electric-field-induced breakdown model, the MIT occurs when the intense electric field above critical field $50 \text{ V}/\mu\text{m}$ frees its bound electrons in vanadium dimers. Although over forty years have been passed since the first report of electrically induced MIT [3], the primary origin of electrically induced MIT is still not well understood and many debates still continue. While numerous studies have been conducted on voltage-induced MIT [14,15], there are only a few studies on current-induced MIT (I-MIT) [16] despite the fact that both voltage and current are of almost equal importance in determining the nature of electrically-induced MIT.

In this study, a systematic investigation on the I-V characteristics of I-MIT in VO₂ thin film planar-type devices was conducted. Based on the current percolation model, we conclude that the I-MIT occurs when metallic phase grains are connected to form first conductive channel. The transition time for I-MIT was also found to be positively correlated with external resistance R_{EXT} , going from $\sim 390 \mu\text{s}$ at $R_{\text{EXT}} = 5 \text{ k}\Omega$ to $\sim 1400 \mu\text{s}$ at $R_{\text{EXT}} = 20 \text{ k}\Omega$. Conduction

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channel temperature was shown to have varied as much as 300 °C by adjusting conduction channel width.

2. Experiment

A 200-nm-thick VO₂ thin films were prepared on Al₂O₃ (0001) substrates by the annealing of sputter deposited vanadium metal films at 540 °C for two hours at oxygen annealing pressure of 160 mtorr, followed by cooling procedure at 120 mtorr. The preparation details of VO₂ thin films were reported elsewhere [17]. Upon the fabrication of VO₂ film, the VO₂ film strips in dimension 1 mm × 100 μm were fabricated with the standard optical lithography process, followed by lift-off processes. For electrical contacts, Au(50 nm)/Cr(10 nm) electrodes were prepared on the VO₂ film stripe using dc magnetron sputtering and conventional photolithography methods. The device length (L) was either 10 μm or 20 μm with a fixed width (W) of 100 μm. The temperature dependence of the resistivity of the VO₂ device was measured with custom-built heating stage in the temperature range from 30 °C to 100 °C. During the temperature-dependent resistivity measurements, a constant current of 10 μA was flowed in the VO₂ device. During the measurements, the temperature of the VO₂ device was controlled with the accuracy better than ± 0.5 °C. During current versus voltage measurements (I-V) and high speed voltage versus time (t) measurements, the external resistors, R_{EXT} = 5, 7.5, 10, 15, or 20 kΩ, were connected in series with the VO₂ device to protect the device and the measuring instruments. The high-speed V-t measurement of VO₂ devices were performed using a source meter (Keithley 2635A) with data collection rate up to 20,000 data per second. Current ramping rate during I-V measurements was set to 0.01 mA/s for all I-V measurement and this ensured the nearly steady state within the film devices. Optical microscopy with reflection geometry mode was applied to monitor the metallic and insulating phases of VO₂ before and after the I-MIT measurements. The microstructure and morphology of VO₂ devices were characterized by scanning electron microscopy (SEM; JSM7800F, JEOL).

3. Results and discussion

Fig. 1 shows the temperature dependence of the resistivity of VO₂ devices with length (L) of 10 μm and 20 μm. The VO₂ devices

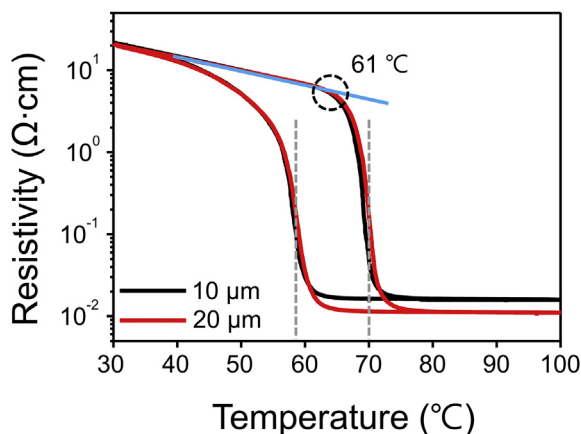


Fig. 1. (a) Temperature dependence of the resistivity of VO₂ film devices with the lengths of 10 μm (black line) and 20 μm (red line). Vertical lines on the graph represent the transition temperature during heating ($T_{c\uparrow}$) and cooling ($T_{c\downarrow}$), determined from the maximum slope of the graph. The resistivity begins to drop rapidly at ~61 °C. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

exhibit typical MIT behavior from a low temperature insulating state to a high temperature metallic state with a hysteresis loop occurred during heating and cooling cycles. The transition temperatures during heating and cooling, determined from the positions of maximum slope in the temperature dependence of the device resistivity, were ~69 °C and ~58 °C, respectively, indicated with dotted line in Fig. 1. The insulating resistivity at 30 °C, $\rho(30\text{ °C})$, of VO₂ devices with L = 10 μm and 20 μm were the same magnitude of ~21 Ωcm, while the metallic resistivity at 100 °C, $\rho(100\text{ °C})$, of the devices were ~16 mΩcm for 10 μm long device and ~11 mΩcm for 20 μm long device, respectively. The slope of insulating resistivity showed exponential temperature dependence, i.e. a linear slope in $\log(\rho)$ vs. T plot in Fig. 1. On the other hand, the resistivity begun to drop significantly when the temperature reached at 61 °C, e.g. the resistivity reached ~6.5 Ωcm, indicated by the dotted circle in Fig. 1. This temperature is ~8 °C lower than the transition temperature during heating up (69 °C). The discussion on this observation will be followed later.

Fig. 2(a) illustrates a schematic circuit diagram constructed for the current controlled I-V and high speed V-t measurements of the planar VO₂ device with Au/Cr electrodes. The single-loop circuit was composed of a VO₂ device (L of either 10 or 20 μm at fixed width W = 100 μm), a R_{EXT}, and a current source. Fig. 2(b) shows SEM micrograph of VO₂ film in the device, which display the closely packed grains with the average grain size of ~180 nm. Fig. 2(c) shows the scanning electron microscope (SEM) micrographs of a VO₂ films device with L = 20 μm and the enlarged view of the interface between VO₂ device and electrodes is shown in Fig. 2(d), showing a sharp interface formed between VO₂ device and electrodes. The VO₂ device was consisted of a wide strip on top of a Al₂O₃ substrate with Au/Cr contact electrodes.

We measured I-V characteristics of VO₂ devices at room temperature (not shown) and however the VO₂ devices often suffered permanent damages after the repeated measurements because of the high voltage appeared on the device during the measurement. Thus all the data is taken at 40 °C to ensure the device reliability during repeated measurements. Fig. 3. (a) shows current controlled

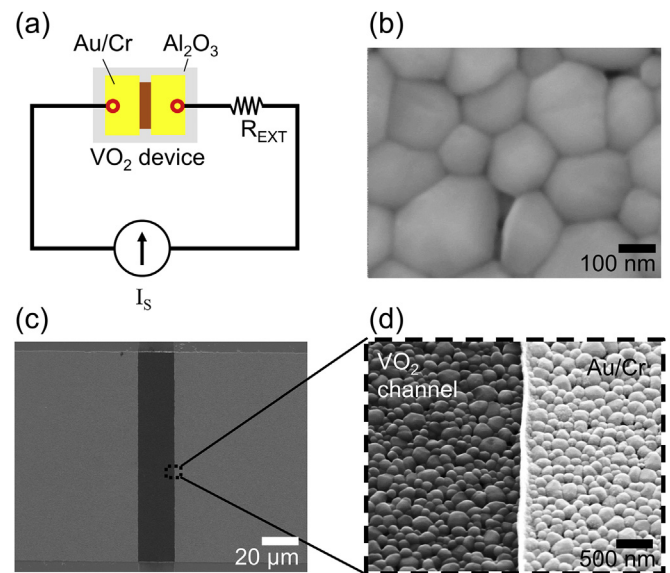


Fig. 2. (a) Schematic circuit drawing for the current induced MIT and high-speed measurements of the VO₂ film devices (device length = 20 μm) with Au (50 nm)/Cr(10 nm) contact electrodes. I_s : current source and R_{EXT}: external resistance. (b) Scanning electron micrograph of a VO₂ film in the VO₂ device (c) Scanning electron micrograph of a VO₂ device (L = 20 μm). (d) Scanning electron micrograph of the VO₂ device near the Au/Cr electrode (enlarged view of (c) near the contact electrode).

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