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MOVPE growth of violet GaN LEDs on β -Ga₂O₃ Substrates

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Abstract—We report that a H₂-free atmosphere is essential for the initial stage of metalorganic vapour phase epitaxy (MOVPE) growth of GaN on β -Ga₂O₃ to prevent the surface from damage. A simple growth method is proposed that can easily transfer established GaN growth recipes from sapphire to β -Ga₂O₃ with both (-201) and (100) orientations. This method features a thin AlN nucleation layer grown below 900 °C in N₂ atmosphere to protect the surface of β -Ga₂O₃ from deterioration during further growth under the H₂ atmosphere. Based on this, we demonstrate working violet vertical light emitting diodes (VLEDs) on n-conductive β -Ga₂O₃ substrates.

Index Terms—B3. Light emitting diodes, B1. Oxides, A3. Metalorganic vapor phase epitaxy, A1. Nucleation

I. INTRODUCTION

III-nitride semiconductor devices, such as light emitting diodes (LEDs), laser diodes (LDs), high electron mobility transistors (HEMTs), and field effect transistors (FETs), have attracted attention from both scientific and industrial communities for decades [1]. Some of their applications have already emerged in our daily life due to easy usage, high efficiencies and thus low carbon dioxide emission such as LED lighting, laser diodes for projector application, and efficient power convertors in electric vehicles. In general, nitride based devices are either heteroepitaxially grown on sapphire, Si, SiC etc. or homoepitaxially on GaN or AlN substrates. While sapphire is available in high quality at low costs it is not electrically conductive. Si, SiC, and GaN on the other hand are non-transparent in the deep ultra violet (UV) region. For devices with high Al content, a conductive substrate with transparency up to the deep UV and low lattice mismatch is

desired.

Ga₂O₃ is an attractive substrate for III-nitride semiconductors since it is a transparent semiconducting oxide (TSO) [2] transparent down to 260 nm [3]. Its conductivity can be adjusted according to the demands [4]. Stable β -phase Ga₂O₃ bulk substrates can be achieved by the Czochralski method (CZ) [5, 6], edge-defined film-fed growth (EFG) [7], and floating zone growth (FZ) [4]. Bulk Ga₂O₃ substrates with high structural quality are commercially available up to 2 inch. β -Ga₂O₃ has a monoclinic structure with lattice constants of $a = 12.214 \text{ \AA}$, $b = 3.037 \text{ \AA}$, $c = 5.798 \text{ \AA}$, and $\beta = 103.83^\circ$ resulting in an in-plane lattice mismatch between c-plane GaN and β -Ga₂O₃ of 2.6-4.7 % [8]. The availability of bulk β -Ga₂O₃ single crystals with good electrical and optical properties [9-14] led to a demonstration of blue LEDs on β -Ga₂O₃ substrates [15, 16]. Deposition of a SiN_x mask on the substrate [15] or a thick AlN buffer layer [17] were employed to achieve high quality of the obtained structures. However, thick AlN layer would hinder the vertical transport of carriers while SiN_x potentially could lower the efficiency of light extraction from UV LEDs.

In this paper, we report on the thermal stability of (-201) β -Ga₂O₃ substrates in low pressure MOVPE in different growth environments by in-situ optical reflectance. Based on an optimized growth atmosphere, we have developed a growth procedure for GaN on both (-201) and (100) β -Ga₂O₃ substrates. Furthermore, vertical LED structures are demonstrated on conductive β -Ga₂O₃ substrate.

II. EXPERIMENTS

β -Ga₂O₃ substrates used in the experiments are commercially available (-201) β -Ga₂O₃ wafers obtained by the EFG method with intentional Sn doping (n-type) [7] and (100) β -Ga₂O₃ substrates were prepared from crystals grown by the CZ method, which are available both semi-insulating and n-type states [5, 6]. They are all in stable β -phase. Thermal stability experiments were performed in an Aix200/4-RF-S reactor at 200 hPa using an Epicurve TT in-situ monitoring system, which allows to study the reflectance of the surface at 405 nm, 650 nm and 950 nm as well as the curvature during growth. In a first step, the (-201) β -Ga₂O₃ substrates were placed on a sapphire wafer in order to prevent contamination of the susceptor and heated up in different environments to 1150°C. β -Ga₂O₃ samples A, B and C were heated-up in pure H₂, pure N₂ atmosphere and N₂ + 2 % NH₃ atmospheres, respectively at temperatures 600, 710, 830, 880, 940, 990, 1050, 1100 and

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