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Flux of silver-carbonate membranes for post-combustion CO₂ capture: The effects of membrane thickness, gas concentration and time



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ABSTRACT

In this paper, we systematically studied the effects of membrane thickness, CO₂ concentration, and operating time on the CO₂ flux density of an Al₂O₃-surface-modified and carbon-black-pore-former derived silver-carbonate membrane. It was found that the CO₂ transport through a silver-carbonate membrane exhibited a critical thickness of 0.84 mm, below which no flux enhancement could be achieved. The CO₂ flux was also observed to be proportional to the difference of ($P_{\text{CO}_2}P_{\text{O}_2}^{1/2}$) at the two reacting surfaces, and not the commonly assumed logarithmic partial pressure difference. The CO₂ flux tested against time at 650 °C increased by 200% for the first 160 h, followed by a gradual decrease. At the 326-h marker, the flux was still the same as the original value when the test was started. Overall, the use of carbon black as a pore-former and Al₂O₃ as a surface modifier for silver-carbonate membranes has been proven effective and necessary to achieve a higher and more stable CO₂ flux density.

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1. Introduction

Capture of post-combusted CO₂ from predominant fossil-fueled power plants is of technological importance to the stabilization of atmospheric CO₂ concentration and mitigation of global warming and climate change. The conventional solvent and sorbent based CO₂ capture technologies are costly, cumbersome and energy intensive [1–4]. With the existing material systems and engineering designs, it seems difficult to overcome the grand challenges in cost and energy penalty for years to come. Development of new, cost-effective and energy-efficient CO₂ capture technologies is, therefore, imperative.

Recently, we demonstrated high-flux silver-carbonate mixed electron and carbonate-ion conducting (MECC) membranes suitable for post-combustion CO₂ capture. This new class of electrochemical membranes in theory can exclusively and continuously separate CO₂ from a CO₂ source such as flue gas at high temperatures [5–7]. The working principle of the membrane is based on a concomitant migration of CO₃²⁻ and e⁻ under a chemical gradient of CO₂ and O₂ across the membrane, thus transporting CO₂ and O₂ from higher concentration to capture side. The captured CO₂-O₂ has been suggested to convert into a mixture of easily separable CO₂ and H₂O mixture by combusting in a syngas. The large amount of heat can be recovered to produce steam for electricity [8]. It has also been suggested to recycle back to the combustion chamber for oxy-combustion. Therefore, MECC membrane has great potential to play

an important role in stabilizing the atmospheric CO₂ concentration by capturing CO₂ emitted from the existing fossil-fueled power plants. Since the capture process is continuous and operated at elevated temperatures, the new membranes have great potential to be a cost-effective and energy-efficient CO₂ capture technology. We also showed that the long-term stability of the membrane can be significantly improved by modifying the surface of porous silver networks with a thin layer of Al₂O₃ [9].

In previous study, the microcrystalline methylcellulose has been used as a pore-former to fabricate the porous silver matrix. Based on the knowledge we have learned from mixed oxide-ion and carbonate-ion conductor (MOCC), smaller pore size and uniform porous structure are among keys to achieving high CO₂ flux density [10–13]. In this study, we further expand the previous study by investigating the effects of membrane thickness, CO₂ concentration and time on the flux of CO₂ permeation. While the same Al₂O₃ surface modification on silver network was kept for the present work, a new pore-former carbon black, which is one of the widely used pore-former [14,15], was tested with a goal to create a better connected porous silver network.

2. Experimental procedure

2.1. Synthesis of silver-carbonate membranes

A two-step approach was employed to synthesize the dual-phase silver-carbonate MECC membranes. The porous silver

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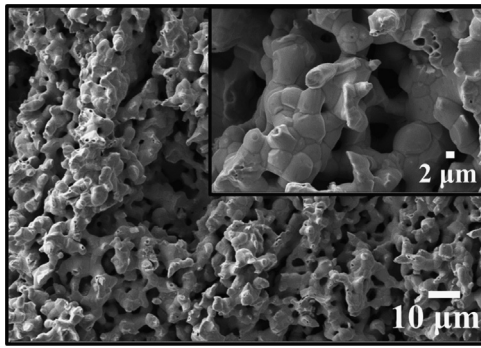


Fig. 1. The cross-sectional view of microstructure of a porous Ag matrix created by carbon black as a pore former.

matrix was first fabricated by intimately mixing silver powders (99.9%, Alfa Aesar) with carbon black as a pore-former (Alfa Aesar) in a ratio of 60:40 (vol%). The powder was ball-milled (Mix/Mill 8000M, Spex Sample Prep) and uniaxially pressed at 200 MPa into pellets using a static mold presser (18 mm in diameter), followed by firing at 650 °C for 2 h. After that, the sample was loaded into the vacuum impregnation unit (CitoVac, Struers). An Al_2O_3 colloidal solution (0.05 μm , Alfa Aesar) with 5 wt% concentrations was then infiltrated into the porous silver scaffold for one time under a vacuum condition. After drying and firing at 400 °C for 2 h, the coated silver membrane was immersed into a carbonate melt containing a eutectic mixture of Li_2CO_3 ($\geq 99\%$, Alfa Aesar) and K_2CO_3 ($\geq 99\%$, Alfa Aesar) in 62:38 (mol%). The detail about this procedure can be found in our previous work [10,16–18].

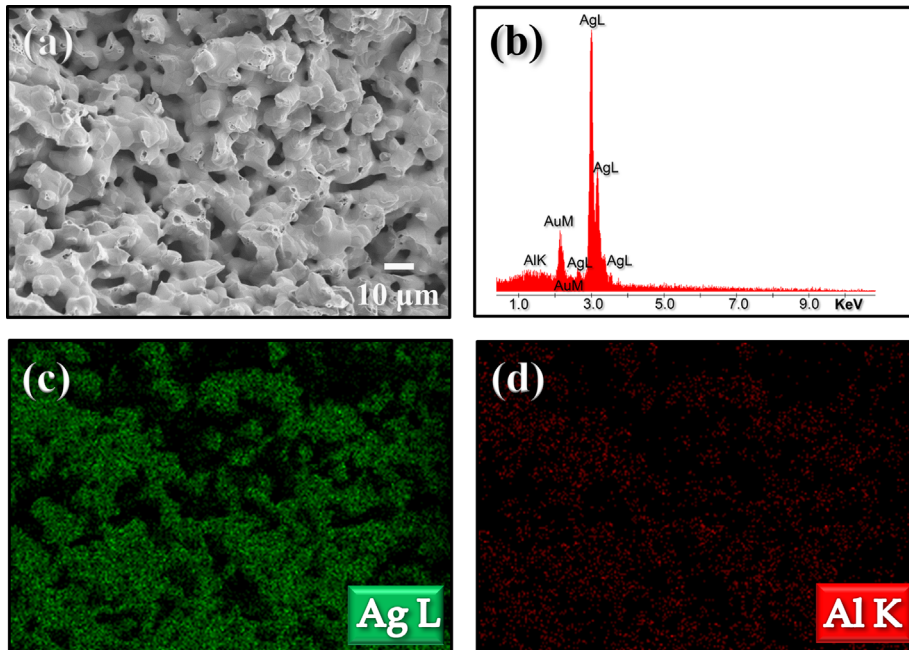


Fig. 2. (a) Microstructure and (b) EDS compositions of a porous Ag matrix coated with 5% Al_2O_3 colloidal; elemental distributions of Ag matrix with 5% Al_2O_3 colloidal (c) Ag mapping and (d) Al mapping.

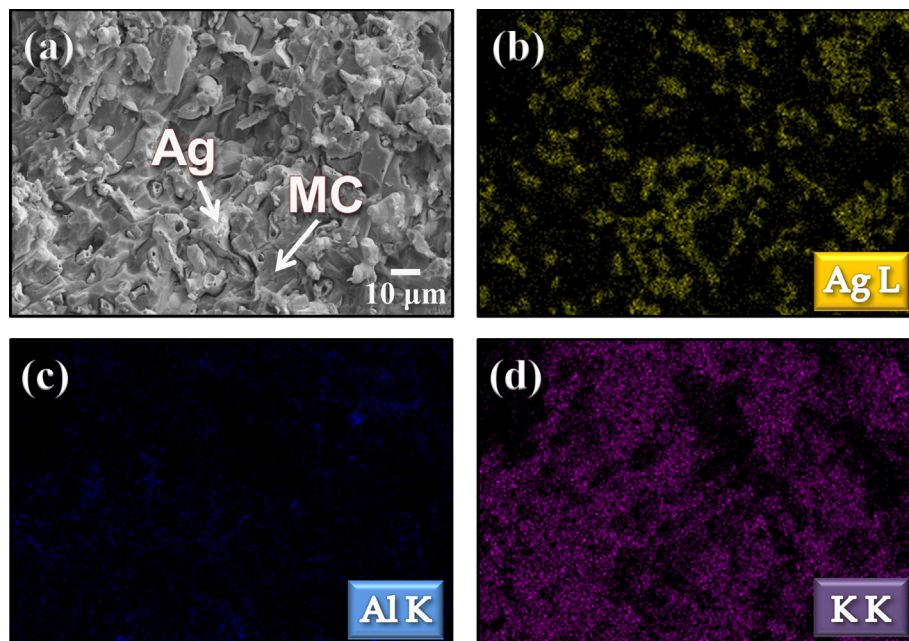


Fig. 3. Microstructure and elemental distributions of silver-carbonate MECC membrane decorated with Al_2O_3 coating (5% colloidal) before testing (a) SEM image; (b) Ag mapping; (c) Al mapping; and (d) K mapping. Note: Li is not detectable by EDS.

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