Electrochimica Acta

journal homepage: <www.elsevier.com/locate/electacta>

Research Paper

Cobalt(II) dicarboxylate-based metal-organic framework for long-cycling and high-rate potassium-ion battery anode

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A R T I C L E I N F O

Article history: Received 6 July 2017 Received in revised form 7 September 2017 Accepted 15 September 2017 Available online 18 September 2017

Keywords: metal-organic framework cobalt potassium-ion battery anode redox activity

A B S T R A C T

Metal-organic frameworks (MOFs) are promising electrical storage materials due to their abundant electroactive components and large ion diffusion tunnels, however, there is no report on the employment of MOFs as electrodes for potassium-ion batteries (PIBs) hitherto. Here in this work, we firstly develop a cobalt(II) terephthalate-based layered MOF (referred as 'L-Co₂(OH)₂BDC', BDC = 1,4-benzenedicarboxylate) as an anode material for PIBs. The exceptional potassium storage performance of L-Co₂(OH)₂BDC is demonstrated with a high reversible capacity of 188 mAh g^{-1} after 600 prolonged cycles at 1 A g^{-1} , revealing the considerable foreground of MOFs as superior potassium storage anodes. Moreover, the redox chemistry investigation of L-Co₂(OH)₂BDC based on X-ray absorption near edge structure (XANES) and soft X-ray spectroscopy (sXAS) techniques substantiate that both Co centers and organic ligands participate in the potassium storage, and the coordination between oxygen ions and cobalt significantly ensures the reversibility of potassiation and depotassiation processes. This work represents a significant step forward for the intensive application of MOFs in rechargeable metal batteries.

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1. Introduction

Potassium-ion batteries (PIBs) have recently galvanized paramount attempts as one of the most promising alternatives to lithium-ion batteries (LIBs) for stationary energy storage, since K resources are much more ubiquitous and abundant in the crust of the Earth than the Li ones (2.09 wt\%) vs. $0.0017 \text{ wt\%})$ [\[1,2\].](#page--1-0) Simultaneously, the redox potential of K^+/K in aqueous media $(-2.93 \text{ V} \text{ vs. SHE})$ is much lower than that of Na⁺/Na $(-2.71 \text{ V} \text{ vs.}$ SHE) but closer to that of Li^{\dagger}/Li (-3.04 V vs. SHE) [3 -5]. Hence, PIBs have alluring potentials as low-cost rechargeable batteries with high voltage and high energy density. Recently, Komaba's group reported an inexpensive high-voltage (4V) K-ion full cell based on $K_{1.75}$ Mn[Fe^{II}(CN)₆]_{0.93}.0.16H₂O cathode and graphite anode and proved its feasibility as high energy density battery system [\[6\]](#page--1-0). Nevertheless, it is still a grand challenge to design appropriate materials that can reversibly store a substantial amount of much larger K-ions (1.38 Å) in a stable and fast manner. Up to now, carbonaceous materials such as graphite and hard carbon [7–[12\],](#page--1-0) alloy materials such as Sb/C, Sn/C and Sn_4P_3/C [\[13](#page--1-0)–15], and organic

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<http://dx.doi.org/10.1016/j.electacta.2017.09.090> 0013-4686/© 2017 Elsevier Ltd. All rights reserved. materials such as 3,4,9,10-perylene-tetracarboxylicacid-dianhydride and dipotassium terephthalate have been investigated as PIB anodes in succession [\[16,17\]](#page--1-0). Besides, highly reversible potassium metal cycling was also achieved for over 200 cycles by using potassium bis(fluoroslufonyl)imide (KFSI)-dimethoxyethane (DME) electrolyte recently [\[18\]](#page--1-0). However, most of the reports documented hitherto are still deficient to be applied in futuristic PIBs due to the limited reversible capacity and poor rate capability, as summarized in [Table](#page-1-0) 1.

Metal-organic frameworks (MOFs) or coordination network compounds (CNCs) are crystalline sponge-like materials formed by the covalently stitching of metal nodes and organic building blocks [\[19\].](#page--1-0) On the grounds of their abundant electroactive constituents and large ion diffusion channels, MOFs are intensively evaluated as electrical storage materials in LIBs [20–[23\]](#page--1-0), Li-S batterie[s\[24\]](#page--1-0), and electrochemical capacitors (ECs) in recent years [\[25](#page--1-0)–27]. In addition, Co 2,5-thiophenedicarboxylate and $[Co(L)(H_2O)]_n$.2nH₂O (L = 5-aminoisophthalic acid) MOFs have also been applied in advanced Na-ion batteries recently [\[28,29\]](#page--1-0). However, to the best of our knowledge, there is no report on the employment of MOFs as anode for PIBs until now, and the possibility of reversible K-storage in MOFs is still unclear.

In this article, we firstly report a cobalt(II) terephthalate-based layered MOF (hereafter referred as 'L-Co₂(OH)₂BDC', BDC = 1,4-
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The comparison of electrochemical performance between L-Co₂(OH)₂BDC and other reported anode materials for PIBs.

benzenedicarboxylate) as an anode material for K-ion coin cells. The as-fabricated L-Co₂(OH)₂BDC is able to deliver high reversible capacities of 246 and 214 mAh g^{-1} with Coulombic efficiencies approaching 100% at 100 and 200 mA g^{-1} , respectively. More spectacularly, a high reversible capacity of 188 mAh g^{-1} after 600 prolonged cycles is still achieved at 1Ag^{-1} , which is the best one ever reported for PIB anodes. Furthermore, the possible redox mechanism of L -Co₂(OH)₂BDC during potassiation/depotassiation process was proposed via synchrotron-based X-ray absorption near edge structure (XANES) and soft X-ray spectroscopy (sXAS) analyses, and the results unravel that both Co centers and Ocontaining organic ligands participate in the potassium storage. Our work demonstrated here will open up a new train of thoughts for the wide application of MOFs in various electrical storage systems.

2. Experimental section

2.1. Synthesis of L-Co₂(OH)₂BDC MOF

All chemicals and solvents were purchased from commercial suppliers and used received. The L-Co₂(OH)₂BDC MOF was fabricated through a typical solvothermal approach. Firstly, dimethylformamide (34 mL), ethanol (10 mL), and water (10 mL) were blended into a 100 mL Teflon-lined high pressure autoclave. Then, 0.75 mmol terephthalic acid (H_2BDC) (0.25 g) and 1.50 mmol $CoCl₂·6H₂O$ (0.357 g) were added to the above solution and dissolved under ultrasonication. Afterwards, the autoclave was capped tightly and placed in an oven at 140° C for 48 h without stirring. Finally, the resultant crystalline materials were harvested by pumping filtration and rinsed with dimethylformamide and water for 3 \sim 6 times, followed by vacuum desiccation at 110 °C for 24 h.

2.2. Materials Characterization

Scanning electron microscopy (SEM) micrographs were observed using a S-4800 (HITACHI, Japan) scanning electron microscope operated at 10 kV. Transmission electron microscopy (TEM) images were taken on a Tecnai G2 F20 electron microscope operated at 200 kV. Powder X-ray diffraction (PXRD) patterns were conducted on a Holland Panalytical PRO PW3040/60 diffractometer, with Cu-K α radiation source (λ = 1.5418 Å). Continuous-wave (CW) X-band EPR spectrum was performed at 2 K with a Bruker EMX plus 10/12 spectrometer, microwave power (2 mW), modulation amplitude (2 G), modulation frequency (100 kHz), sweep

time (128 s), conversion time (16 ms) and time constant (163.84 ms) were reasonably set. X-ray photoelectron spectroscopy (XPS) spectra were carried out on an ESCALAB 250Xi X-ray photoelectron spectrometer operating at 150W, with Al–K α radiation ($hv = 1486.6 eV$). Thermogravimetric analysis (TGA) was performed by a STA 449 F3 Jupiter simultaneous thermoanalyzer at a ramping rate 10° C/min from 25 to 800 $^{\circ}$ C in air atmosphere. N₂ sorption isotherms were obtained at 77K with an Autosorb-IQ automated gas sorption analyser, the specific surface area was calculated through Brunauer-Emmett-Teller (BET) method. Tap density was measured with a homemade tapping apparatus (Shenzhen Sannuo Co., China).

2.3. Electrochemical Measurements

CR2032 coin cells, which were assembled in an Ar-filled glove box $(0₂< 0.1$ ppm, $H₂O < 0.1$ ppm), were used to investigate electrochemical performance of L-Co₂(OH)₂BDC MOF. The working electrodes were composed of the as-prepared L-Co₂(OH)₂BDC active materials, acetylene black, and carboxymethyl cellulose (CMC) binder in the weight ratio of 60:30:10. The loading amount of active materials was 0.8-1.2 mg cm $^{-2}$. The potassium metal was employed as both counter electrode and reference electrode. The electrolyte was made up with $1 M KPF_6$ dissolved in 1, 2dimethoxyethane (DME), and $40 \mu L$ of electrolyte was used in each coin cell. Whatman glassy fiber (GF-D) was employed as the separator. Galvanostatic charge/discharge measurements were conducted in the voltage range of $0.20-3.00$ V (vs. K^+/K) on a LAND CT2001A battery instrument. All the specific capacities were calculated based on the mass of L-Co₂(OH)₂BDC. Cyclic voltammetry (CV) was recorded using a CHI 660a electrochemical workstation in the potential window of $0.20-3.00$ V (vs. K^+/K). Electrochemical impedance spectra (EIS) were also carried out on a CHI 660a electrochemical workstation with the frequency range of 0.01 Hz to 100KHz.

2.4. XANES and sXAS Characterizations

Soft X-ray absorption spectroscopy (sXAS) was monitored at Shanghai Synchrotron Radiation Facility (BL08U). Total electron yield (TEY) spectra at O K-edge were recorded via detecting the electric current caused by excited electrons from the cycled L- $Co₂(OH)₂BDC$ samples at various SOC. X-ray absorption near edge structure (XANES) spectra were monitored at Shanghai Synchrotron Radiation Facility (BL15U). The energy resolution ($\Delta E/E$) of this facility was 2×10^{-4} , and the maximum photon flux was about

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