



Effects of Cyclohexane- Monoxime and Dioxime on the Electrodeposition of Cobalt



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ABSTRACT

Electrodeposition of cobalt was studied in presence of two additives, cyclohexane oxime (CHO) and cyclohexane dioxime (CHD), which differ by one oxime group. Cyclic voltammetry (CV), potentiostatic deposition with pulsed rotation rates, and galvanostatic deposition with injections of additives were carried out to understand the formation and breakdown of intermediate species that suppress cobalt electrodeposition. While strong and fast suppression was observed for CHD on cobalt deposition, the suppression effect of CHO was much less pronounced. A negative differential resistance in CV was observed for CHD due to a breakdown of suppression, which was related to the reduction of absorbed Co-CHD chelate species. A critical rotation rate was defined for CHD as the minimum rotation required to enable a fully suppressed deposition state. An inversely proportional relation between additive concentration and the square root of critical rotation rate demonstrated an equilibrium between the diffusion of the intermediate in solution and the consumption of intermediate by incorporation into deposit. While a pair of adjacent oxime groups was necessary for a strong fast suppression effect, bulkier molecular structures and stronger chelating resulted in stronger suppression.

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1. Introduction

Copper back-end-of-line (BEOL) interconnect was invented in late 1990's [1,2] and has been used since then for its superior conductivity, reliability and scalability. However, as the dimension of interconnect structures approaches or even falls below the electron mean free path of copper, the copper resistivity in such very fine structures exponentially increases [3–8]. This is because the electron scattering at interfaces between copper and surrounding materials and at grain boundaries can occur more often than the intrinsic electron scattering and dominate the material resistivity. Therefore, materials with much shorter electron mean free path are of interest [9] because of the insensitivity to scaling even if they have a higher bulk resistivity than copper. Among a list of candidate materials, cobalt has been identified as a front runner [10,11] because of its very short mean free path and relatively low cost.

Electrodeposition of copper interconnects has been studied for more than a decade and the mechanism of a so-called super conformal electrodeposition (or superfilling) has been well

understood [12–14]. A preferential deposition at the bottom of a trench results in the defect free formation of interconnect structures. Superfilling of copper are generally accepted as a result of a synergistic interaction of two organic additives during copper electrodeposition [12–19]. An accelerator promotes the deposition rate at the bottom of the trench with respect to the top, where the rate is suppressed by a suppressor. A concentration gradient of suppressor can be present across the depth of the trench, resulting in a lower concentration of suppressor to facilitate the accelerator at the bottom of the trench [19–22]. Such synergistic mechanism between two-component additive system has been exercised for a few other metals [23–26].

On the other hand, a negative differential resistance (NDR), observed as an s-shaped cyclic voltammetry [27–30], has also been found to enable preferential deposition at the bottom of structures due to a positive feedback between deposition and adsorbate consumption (less suppression). Two stable steady states at a single deposition potential can be achieved in such systems. Bottom up deposition in deep silicon via is a result of two different deposition steady states, i.e, fast deposition at the bottom and suppressed deposition at the top [30–32].

A list of different organic additives was previously studied for the superfilling of iron group metals in to sub-50 nm interconnect

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