

# Large-scale synthesis of ternary $\text{Sn}_5\text{SbP}_3/\text{C}$ composite by ball milling for superior stable sodium-ion battery anode



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## ABSTRACT

Alloy-based materials (i.e. Sn, Sb, P) are promising candidates for sodium-ion battery (SIB) anodes, but they suffer from capacity decay during charge/discharge cycling due to the pulverization caused by their huge volume change. Nanostructures can slow down the capacity fade, but most of the synthesis methods of such nanostructured anodes are difficult to scale-up. Herein, a ternary  $\text{Sn}_5\text{SbP}_3/\text{C}$  composite was fabricated by a green, low cost, one-step and easily scalable ball-milling of elementary Sn, Sb, P, and C. The microstructure of the ball-milled powders consists of micrometric agglomerates of active nano  $\text{Sn}_4\text{P}_3$  and  $\text{SnSb}$  and Sn particles. Carbon in the composite acts as a conducting matrix, and it does not only benefit to the ball milling efficiency, but also benefit to the cycle life of the electrode. Each of the active  $\text{Sn}_4\text{P}_3$  and  $\text{SnSb}$  and Sn phases in the composite functions mutually as a buffer for the others. As a result, this ternary composite anode delivers a good capacity of  $352 \text{ mA h g}^{-1}$  at the current density of  $2 \text{ Ag}^{-1}$ , which is notably higher than that of the binary  $\text{Sn}_4\text{P}_3/\text{C}$  and  $\text{SnSb}/\text{C}$  composites produced under the same conditions.

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## 1. Introduction

Sodium-ion batteries (SIBs) have been attracting great interest recently due to the low cost and abundance of sodium resources, especially for applications in stationary energy storage [1–16]. The successful commercialization of SIBs does not only calls for the development of electrode materials with enhanced performance, but also scalable synthesis protocols [17]. Alloy-based materials are believed to be very promising candidates for SIB anodes due to their high gravimetric and volumetric capacities, and slightly higher thermodynamic potential [1,18,19]. For instance, group 14 (Sn,  $\text{Na}_{15}\text{Sn}_4$ ,  $847 \text{ mA h g}^{-1}$ ) and 15 elements (P,  $\text{Na}_3\text{P}$ ,  $2596 \text{ mA h g}^{-1}$ ; Sb,  $\text{Na}_3\text{Sb}$ ,  $660 \text{ mA h g}^{-1}$ ) based alloys [14,20,21] have been intensively studied recently and have showed promising properties. Nevertheless, large volume changes (>200%) are inevitable for these alloy anodes during sodiation/desodiation. The huge volume expansion will lead to continuous pulverization, particle cracking, and subsequent separation of the active materials from the current collector and result in fast capacity fade [1,2,22].

Nanomaterials with designed and optimized structures can slow down the capacity fade. The use of these nanosized materials by industry in the near future is unlikely, however, because of their complicated synthesis procedures, very high synthesis costs, high surface reactivity, low tap density, and difficulties for large-scale synthesis. In contrast, ball milling is a simple, cheap, and easily scalable synthesis method, and hence, it has a potential for industrial-scale application. Binary composite systems such as  $\text{Sn}_4\text{P}_3$  and  $\text{SnSb}$  formed via ball milling have been shown to have significantly better electrochemical performance over pristine Sn, Sb, or P.<sup>20–23</sup> The electrochemical performance of the ball milled binary  $\text{Sn}_4\text{P}_3$  or  $\text{SnSb}$  is still far from adequate for practical applications, however [23–26].

Herein, we report a ternary  $\text{Sn}_5\text{SbP}_3/\text{C}$  composite synthesized via ball-milling, which contains  $\text{Sn}_4\text{P}_3$ ,  $\text{SnSb}$  and Sn nanoparticles (NPs), which are in intimate contact with each other and form heterojunctions in a conductive carbon matrix. The ternary composite displays superior electrochemical performance to the binary  $\text{Sn}_4\text{P}_3/\text{C}$  or  $\text{SnSb}/\text{C}$  composite. It was found that the Sn and  $\text{SnSb}$  in the composite facilitate the electron transfer and  $\text{Sn}_4\text{P}_3$  provides high sodium storage capacity. The  $\text{SnSb}$  phase also promotes a superior stable cycling performance by the formation of an intermediate amorphous phase  $\text{Na}_x\text{Sb}$  [23,27–29]. It seems that the superior electrochemical performance of the ternary

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$\text{Sn}_5\text{SbP}_3/\text{C}$  composite is resulted from a synergetic effect of the multi phased nanostructure on a conductive network. Importantly ball milling is a scalable process and a large volume of composite architecture electrodes can be easily produced for practical applications.

## 2. Experimental Section

### 2.1. Preparation of $\text{Sn}_5\text{SbP}_3/\text{C}$ ternary composite

$\text{Sn}_5\text{SbP}_3/\text{C}$  powder was directly synthesized by ball milling the raw materials of Sn (Aldrich,  $\geq 99.8\%$ ), Sb (Aldrich,  $\geq 99.5\%$ ), red phosphorus (Aldrich,  $\geq 99\%$ ) and carbon black. The weight ratio of Sn: Sb: P: C in the composite is 64.26:13.18:10.06:12.5. The ball milling was conducted in a planetary QM-1SP2 ball mill for 30 h at 500 rpm. A stainless steel jar and stainless steel balls of 10 mm in diameter are used. The powder-to-ball weight ratio was 1:30. For comparison, binary composites  $\text{SnSb}/\text{C}$  and  $\text{Sn}_4\text{P}_3/\text{C}$  were also synthesized under the same conditions. The weight ratios are Sn: Sb: C = 43.19: 44.31: 12.5 and Sn: P: C = 73.11: 14.39: 12.5 in the binary composites  $\text{SnSb}/\text{C}$  and  $\text{Sn}_4\text{P}_3/\text{C}$  respectively. The storage and handling of all the samples were performed in an Ar filled glovebox (MBraun Unilab).

### 2.2. Materials characterization

The phasic composition and information of the as-prepared powders were characterized by powder X-ray diffraction (XRD) on a GBC MMA diffractometer with a  $\text{Cu K}\alpha$  radiation at a scanning rate of  $1^\circ \text{ min}^{-1}$ . The particle morphology and size of the prepared powder materials were characterized on a JEOL JSM-7500FA field-emission scanning electron microscope (FESEM) and on a JEOL ARM-200F cold field emission and aberration-corrected transmission electron microscope (TEM).

### 2.3. Electrochemical measurements

Electrodes were fabricated using a slurry-coating method. The synthesized materials ( $\text{Sn}_5\text{SbP}_3/\text{C}$ ,  $\text{Sn}_4\text{P}_3/\text{C}$  and  $\text{SnSb}/\text{C}$ ) were mixed with Super P carbon black and carboxymethyl cellulose (CMC) in the weight ratio of 8:1:1 to form slurry with deionized (DI) water. Then, the slurry was coated on copper foil and dried in a vacuum oven at  $80^\circ \text{C}$  for 12 h. Coin-type (CR2032) cells were assembled in an argon-filled glove box ( $\text{H}_2\text{O} < 0.1 \text{ ppm}$ ,  $\text{O}_2 < 0.1 \text{ ppm}$ ), with 1 M  $\text{NaClO}_4$  in a mixture of propylene carbonate (PC) with 5 wt. % fluoroethylene carbonate (FEC) as the electrolyte. Cyclic

voltammetry (CV) tests were conducted on a Bio-logic VMP-3 electrochemical workstation at a scan rate of  $0.05 \text{ mV s}^{-1}$ . The cells were galvanostatically charged-discharged between 0.01 and 2.0 V versus  $\text{Na}/\text{Na}^+$  at various current densities on a Land battery tester. The mass loading of active material ( $\text{Sn}_5\text{SbP}_3$ ,  $\text{Sn}_4\text{P}_3$  and  $\text{SnSb}$ ) was over  $0.78 \text{ mg cm}^{-2}$ . The specific capacity was calculated based on the weight of pure  $\text{Sn}_5\text{SbP}_3$ ,  $\text{Sn}_4\text{P}_3$  and  $\text{SnSb}$  active materials.

## 3. Results and discussion

Fig. 1a presents the X-ray diffraction (XRD) patterns of the ternary  $\text{Sn}_5\text{SbP}_3/\text{C}$  and the binary  $\text{Sn}_4\text{P}_3/\text{C}$  and  $\text{SnSb}/\text{C}$  composites. The ternary  $\text{Sn}_5\text{SbP}_3/\text{C}$  composite contains three intermetallic and metallic phases:  $\text{Sn}_4\text{P}_3$  (ICSD#15014),  $\text{SnSb}$  (ICSD#52303) and Sn (ICSD#40037). In contrast, only  $\text{Sn}_4\text{P}_3$  or  $\text{SnSb}$  was observed in the binary composite  $\text{Sn}_4\text{P}_3/\text{C}$  or  $\text{SnSb}/\text{C}$ , respectively. The carbon black is not observed in the XRD patterns due to its amorphous features, but it can be clearly identified by Raman analysis, as shown in Fig. S1 in the Supporting information. Also, no bonds of C with Sn, Sb, and/or P are observed in X-ray photoelectron spectroscopy (XPS) analysis (Fig. S2). Fig. 1b presents the a Rietveld refinement profile for the  $\text{Sn}_5\text{SbP}_3/\text{C}$  composite, performed by using the GSAS-II software package [30]. All the peaks in the XRD pattern of the ternary  $\text{Sn}_5\text{SbP}_3/\text{C}$  can be indexed those of hexagonal  $\text{Sn}_4\text{P}_3$  ( $R\bar{3}m$ ), cubic  $\text{SnSb}$  ( $Fm\bar{3}m$ ), and tetragonal Sn ( $I4_1/amd$ ). The lattice parameters obtained from the Rietveld refinement are  $a = 4.0283 \text{ \AA}$ ,  $c = 35.6999 \text{ \AA}$  for the  $\text{Sn}_4\text{P}_3$ ,  $a = 6.1278 \text{ \AA}$  for the  $\text{SnSb}$  and  $a = 5.8456 \text{ \AA}$ ,  $c = 3.1903 \text{ \AA}$  for the Sn respectively. The amount of each phase in the  $\text{Sn}_5\text{SbP}_3$  was calculated as  $65.75 \pm 3.14 \text{ wt\%}$   $\text{Sn}_4\text{P}_3$ ,  $18.54 \pm 0.62 \text{ wt\%}$   $\text{SnSb}$ , and  $15.71 \pm 0.67 \text{ wt\%}$  Sn.

The as-milled  $\text{Sn}_5\text{SbP}_3/\text{C}$ ,  $\text{Sn}_4\text{P}_3/\text{C}$  and  $\text{SnSb}/\text{C}$  powders were characterized by SEM and the results are shown in Fig. 2. It can be seen that the powders in all three samples show irregular shapes in their morphology. The agglomerated and micro sized particles are the main constituent of the powders, and there are also numerous primary nanoparticles about 50 nm–100 nm in size (Fig. 2a–c). The mixed structure of the micro- and nano- particles may allow full contact of the electrolyte with the active materials and hence facilitate an efficient ion transportation [31]. It can also be seen that Sn, Sb, and P are distributed uniformly in the  $\text{Sn}_5\text{SbP}_3/\text{C}$  composite according to the energy dispersive X-ray spectroscopy (EDX) (Fig. 2d). EDX results for the  $\text{Sn}_4\text{P}_3/\text{C}$  and  $\text{SnSb}/\text{C}$  are shown in Fig. S3 and the compositions of the three different powders are listed in Table S1.

Fig. 3 shows a TEM image, and a HRTEM image taken from a micron particle of the  $\text{Sn}_5\text{SbP}_3/\text{C}$  composite. The micron particle is

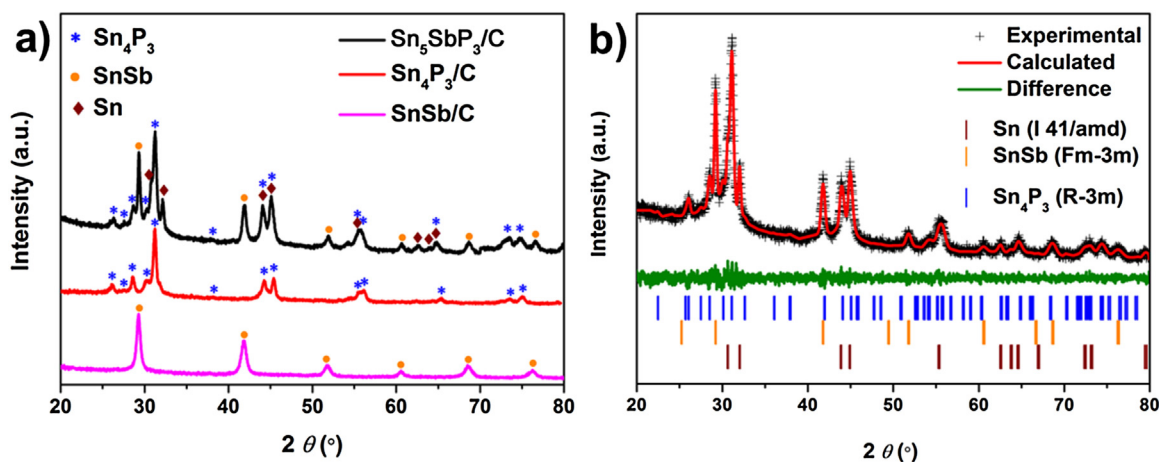


Fig. 1. a) XRD patterns of  $\text{Sn}_5\text{SbP}_3/\text{C}$  composite,  $\text{Sn}_4\text{P}_3/\text{C}$  and  $\text{SnSb}/\text{C}$ ; b) Rietveld refinement profile for the  $\text{Sn}_5\text{SbP}_3/\text{C}$  composite.

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