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Therapeutic-designed electrospun bone scaffolds: Mesoporous bioactive nanocarriers in hollow fiber composites to sequentially deliver dual

⁵ 01 growth factors

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ABSTRACT

A novel therapeutic design of nanofibrous scaffolds, holding a capacity to load and deliver dual growth factors, that targets bone regeneration is proposed. Mesoporous bioactive glass nanospheres (MBNs) were used as bioactive nanocarriers for long-term delivery of the osteogenic enhancer fibroblast growth factor 18 (FGF18). Furthermore, a core-shell structure of a biopolymer fiber made of polyethylene oxide/ polycaprolactone was introduced to load FGF2, another type of cell proliferative and angiogenic growth factor, safely within the core while releasing it more rapidly than FGF18. The prepared MBNs showed enlarged mesopores of about 7 nm, with a large surface area and pore volume. The protein-loading capacity of MBNs was as high as 13% when tested using cytochrome C, a model protein. The protein-loaded MBNs were smoothly incorporated within the core of the fiber by electrospinning, while preserving a fibrous morphology. The incorporation of MBNs significantly increased the apatite-forming ability and mechanical properties of the core-shell fibers. The possibility of sequential delivery of two experimental growth factors, FGF2 and FGF18, incorporated either within the core-shell fiber (FGF2) or within MBNs (FGF18), was demonstrated by the use of cytochrome C. In vitro studies using rat mesenchymal stem cells demonstrated the effects of the FGF2-FGF18 loadings: significant stimulation of cell proliferation as well as the induction of alkaline phosphate activity and cellular mineralization. An in vivo study performed on rat calvarium defects for 6 weeks demonstrated that FGF2-FGF18-loaded fiber scaffolds had significantly higher bone-forming ability, in terms of bone volume and density. The current design utilizing novel MBN nanocarriers with a core-shell structure aims to release two types of growth factors, FGF2 and FGF18, in a sequential manner, and is considered to provide a promising therapeutic scaffold platform that is effective for bone regeneration.

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1. Introduction 54

Therapeutic scaffolds that possess the capacity to load and deli-55 56 ver therapeutically relevant molecules, including drugs, proteins 57 and genes, are considered to be promising platforms for the engineering of tissues, including bone [1]. While the approaches that 58 tailor the scaffolds intrinsically, including hybridization, minerali-59 60 zation and stiffness matching, can also be effective in bone regener-61 ation [2–6], the controlled delivery of extrinsic bioactive molecules 62 provides the scaffolds with additional therapeutic functions. The

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scaffolds play direct roles in triggering the functions of the stem/ progenitor cells by providing physical and biochemical environments for homing and recruiting cells when implanted in vivo [7]. Many methodologies have been developed to load therapeutic molecules effectively onto the scaffolding materials, including surface immobilization, affinity-driven tethering, direct incorporation within the structure and nano/microencapsulation [8-10].

Electrospun fibers have shown great potential as a scaffold platform for the repair and regeneration of a series of tissues, such as skin, muscle, nerve, cartilage and bone [11-14]. The fibrous morphology, which is tunable from hundreds to a few micrometers, largely mimics the native tissue extracellular matrices, thus favoring cellular adhesion and growth [15]. The electrospinning technique enables the formation of fibers with different morphologies, structures and compositions. In particular, for drug

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delivery purposes, fibers with a core–shell structure have shown great promise since relatively large quantities of therapeutic molecules can be encapsulated within the core while preserving the biological activity of the molecules [16,17]. Theoretically, the release of the encapsulated drugs can be controlled by altering the shell composition and thickness.

Here we also focus on this particular type of electrospun coreshell fibrous scaffold, with a view to developing a drug delivering scaffold for bone regeneration. Furthermore, we intend to deliver two different types of therapeutic molecules in a sequential manner. One type of drug will be released relatively quickly, while the other drug is released in a more sustained manner. This concept of sequential delivery has recently gained great interest for scaffold designs targeting tissue repair and regeneration, which generally requires the time-dependent action of multiple molecules [18–21].

94 We use fibroblast growth factor 2 (FGF2) as the fast-releasing 95 molecule while FGF18 is used as the slow-releasing one, aiming 96 at achieving an effective delivery system for bone regeneration. 97 FGF2 has long been shown to stimulate mitosis and proliferation 98 of cells, including those of fibroblasts and endothelial cells, which 99 play an important role in populating and maintaining such cells for the subsequent tissue repair process [22]. On the other hand, 100 FGF18, a growth factor in the FGF family that has been highlighted 101 102 relatively more recently, is known to stimulate cellular osteogene-103 sis by the up-regulation of bone morphogenetic protein 2 [23]. 104 Therefore, the co-delivery of both growth factors FGF2 and FGF18 105 in a sequential release pattern is considered to be effective for 106 the bone repair and regeneration process.

107 In order to realize sequential delivery, we first preload FGF18 108 within mesoporous bioactive glass nanospheres (MBNs), which 109 are then encapsulated within the core, where FGF2 is also loaded directly. The MBNs recently developed by our group have shown 110 great performance in loading and delivering therapeutic molecules, 111 including chemical drugs, proteins and genes [24-26]. Further-112 113 more, MBNs are an attractive nanomaterial platform for the regen-114 eration of hard tissues like bone and tooth because the 115 nanospheres have shown outstanding mineralization behavior. 116 which is mainly ascribed to the high surface bioreactivity [18]. 117 Furthermore, the possible release of Ca and Si ions from the nano-118 spheres would be beneficial for stimulating progenitor and stem 119 cells into an osteogenic lineage [27].

Fig. 1 illustrates the schematic design of the present study, the 120 121 development of a novel therapeutic bone scaffold where FGF18preloaded MBNs are incorporated within the FGF2-loaded core-122 123 shell electrospun polymeric fiber. FGF2 is to be released initially 124 to stimulate cellular mitosis and possible angiogenesis, while 125 FGF18 is to be released more slowly to induce osteogenesis. In this 126 study, we describe the processing tools used for the therapeutic 127 scaffolds and investigate their physico-chemical properties, includ-128 ing the morphological development, mechanical properties, in vitro degradation and bone bioactivity. The release pattern of 129 the growth factors is monitored using a model protein for a long 130 period of a few months. The therapeutic efficacy of the FGF2/ 131 132 FGF18 delivered from the fiber scaffolds is addressed in terms of 133 the in vitro responses of rat mesenchymal stem cells (MSCs), 134 including growth and osteogenic differentiation, as well as in vivo bone formation in rat calvarium defects. 135

136 **2. Materials and methods**

137 2.1. Mesoporous bioactive glass nanospheres

138 MBNs were prepared in basic solution at room temperature 139 using water, ethanol and 2-ethoxyethanol as co-solvents and hexadecyltrimethylammonium bromide (CTAB) as a surfactant. In 140 a typical procedure, 1 g of CTAB was dissolved in an emulsion sys-141 tem composed of 150 ml of H₂O, 2 ml of aqueous ammonia, 10 ml 142 of 2-ethoxyethanol, 20 ml of ethanol and tetrahydrate calcium 143 nitrate (Ca(NO₃)₂·4H₂O). After the mixture had been vigorously Q5 144 stirred for 0.5 h at room temperature, tetraethyl orthosilicate was 145 quickly dripped into the mixture. The molar ratio of Ca:Si was 146 15:85. The resulting mixture was vigorously stirred at room tem-147 perature for 4 h. A white precipitate was obtained, filtered, washed 148 with pure water and dried in air at 60 °C for 24 h. The mixture was 149 then treated with 20 mg of ammonium nitrate solution in 40 ml of 150 deionized water at 60 °C overnight to remove any remaining CTAB. 151 The nanoparticle solution was centrifuged and washed with etha-152 nol and deionized water, then dried overnight under a vacuum to 153 obtain MBNs. 154

2.2. Electrospinning of core-shell hollow-structured composite fibers 155

The core-shell hollow-structured fiber was made of polycapro-156 lactone (PCL; Mw = 80,000; Sigma-Aldrich) and poly(ethylene 157 oxide) (PEO; Mw = 60,000; Sigma-Aldrich) as the outer and inner 158 parts. The MBNs were added to the PEG at varying contents (1, 3 159 and 5 wt.% with respect to PEO). The solvent used for PCL was a 160 co-solvent comprising chloroform and dimethyl formamide mixed 161 at a ratio of 4:1, and that for PEO was distilled water. PCL was dis-162 solved at 10% and PEO was dissolved at 4%. Coaxial electrospinning 163 was conducted to prepare PCL as the shell and PEO (+MBN) as the 164 core using a syringe-like apparatus with an inner needle placed 165 coaxially inside an outer one. The feed rates were 4.0 and 0.5 ml h⁻¹, 166 respectively, for the outer and the inner part. Each solution loaded 167 into the syringe was injected through a spinneret needle under 168 the application of high voltage power (Gamma High Voltage 169 Research, Ormond Beach) onto a slowly rotating metal collector. 170 The distance between the needle and the collector was set to 171 15 cm, and the collection was carried out over 2 h. The collected 172 fiber mesh was vacuum-dried overnight. In particular, to load the 173 proteins and growth factors within the fiber, the solution, at a 174 concentration 100 μ g ml⁻¹ in phosphate-buffered saline (PBS), was 175 mixed with the PEO solution, which was adequate for electrospinning. 176

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2.3. Characterizations

The morphology and the nanostructure of the samples were 178 examined respectively by field-emission scanning electron micros-179 copy (SEM; Tescan, Mira II LMH) and high-resolution transmission 180 electron microscopy (TEM; JEM-3010, JEOL, operating at 300 kV). 181 The chemical bond structure was investigated by attenuated total 182 reflectance-Fourier transform infrared spectroscopy (ATR-FTIR) 183 at a resolution of 4 cm^{-1} in the range from 4000 to 400 cm^{-1} using 184 a Varian 640-IR spectroscope with a GladiATR diamond crystal 185 accessory (PIKE Technologies). The thermal behavior of the sam-186 ples was investigated by thermogravimetric analysis (TGA; TGA 187 N-1500, Scinco, operating at a heating rate of $10 \circ C \min^{-1}$). The 188 mesopore structure was analyzed according to N2 adsorption-189 desorption measurements using an automated surface area and 190 pore size analyzer (Quadrasorb SI, Quantachrom Instruments). 191 The specific surface area was estimated according to the Bru-192 nauer-Emmett-Teller (BET) method, and the pore size distribution 193 was determined using the non-local density functional theory 194 (NLDFT) method. The surface electrical properties were investi-195 gated with *ξ*-potential measurements (Zetasizer Nano ZS, Malvern 196 Instruments), and the ξ-potential was measured in water at 25 °C 197 at a pH of 7 with an applied field strength of 20 V cm^{-1} . The 198 instrument automatically calculates the electrophoretic 199 mobility (U), and the ξ -potential is obtained according to the 200

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