



## Short communication

# Constructing interfacial contact for enhanced photocatalytic activity through BiOIO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> nanoflake heterostructure

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## ABSTRACT

The highly efficient BiOIO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> with unique nanoflake heterostructures were synthesized through hydrothermal method by simply depositing BiOIO<sub>3</sub> flake on the surface of g-C<sub>3</sub>N<sub>4</sub>. Photocatalytic ability of nanoflakes was tested by oxidation of gaseous Hg<sup>0</sup> under 18 W and 24 W LED light irradiation. The g-C<sub>3</sub>N<sub>4</sub>/BiOIO<sub>3</sub> composite with the molar ratio of 10:1 exhibited that the Hg<sup>0</sup> removal efficiency reached up to 93.7%. The remarkable photocatalytic performance of the as-prepared catalysts can be attributed to the interfacial contact heterostructure formed between BiOIO<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub>, which constructs an intimate interfacial interaction between the two components. Moreover, well-matched band structure can efficiently lead to the separation of photogenerated charges.

## 1. Introduction

Graphitic-phase carbon nitride (g-C<sub>3</sub>N<sub>4</sub>), as a metal-free inorganic photocatalyst, has received many researches [1]. In further studies, scientists found that g-C<sub>3</sub>N<sub>4</sub> has the characteristics of two-dimensional conjugate structures and can be used as good substrate materials [2,3]. BiOIO<sub>3</sub>, as a kind of promising bismuth based photocatalyst, has gained much attention nowadays because it is a new noncentrosymmetric and internal polar compound with Aurivillius-type (Bi<sub>2</sub>O<sub>2</sub>)<sup>2+</sup> layers, exhibiting many superior properties [4–6]. However, both g-C<sub>3</sub>N<sub>4</sub> and BiOIO<sub>3</sub> have the same disadvantage, i.e., the recombination of photo-generated charge carriers is too fast [7,8]. The relatively limited absorption under visible-light is another drawback for BiOIO<sub>3</sub> [9]. To overcome these shortages, scientists explored many methods such as element doping and co-catalyst modification [10–12]. Recently, formation of heterojunction [13,14] with other semiconductors is considered to be another promising way to improve the performance of g-C<sub>3</sub>N<sub>4</sub> by obtaining unique nanostructures and fast charge separation owing to the presence of built-in electric field at the heterostructures interface. Therefore, emphasis has been given on designing novel 2D nanosheet-based heterostructure systems benefiting from 2D structural feature. In particular, due to the notable advantages of the sufficient and tight contact and the improved charge transfer mobility, 2D-2D heterostructures are more than expectation as compared with the 0D-2D and 1D-2D heterostructures [15–18]. Here, we issue a facile strategy

for creating 2D layered BiOIO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> heterostructure with intimate interfacial contact via regulating the amount of BiOIO<sub>3</sub> deposited on the surface of g-C<sub>3</sub>N<sub>4</sub>. The built in electric field coupled with BiOIO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> nanoflake heterostructure has been able to achieve such high efficiency. Mercury, a kind of element with high toxicity, strong volatility and persistent durability in bioaccumulation [19], is difficult to be removed by conventional methods because of high oxidation-reduction potentials. The as-prepared 2D-2D heterostructure BiOIO<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> nanoflakes are used to oxidizing elemental mercury (Hg<sup>0</sup>) to evaluate its photocatalytic performance. Meanwhile, the possible reaction mechanism is elaborated on the basis of characterization and experimental results.

## 2. Experimental section

## 2.1. Materials and reagent

Bi(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O, KIO<sub>3</sub> and C<sub>3</sub>H<sub>6</sub>N<sub>6</sub> were obtained from Guoyao Chemical Reagent Co. Ltd. In this study, all chemicals were used as received and all solvents were prepared with ultrapure water.

2.2. Preparation of g-C<sub>3</sub>N<sub>4</sub>/BiOIO<sub>3</sub>

Pure g-C<sub>3</sub>N<sub>4</sub> powders were obtained by an improved calcination process. Concretely, 10 g of melamine were added to ceramic crucible

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with a cap, and then maintained at 550 °C for 2 h under the argon atmosphere protection with a heating rate of 5 °C/ min.

In our experiment, 0.1 g  $g\text{-C}_3\text{N}_4$  was added into 60 mL deionized water, and then ultrasonic treated for 45 min to disperse the  $g\text{-C}_3\text{N}_4$  completely. After that, an appropriate amount of  $\text{KIO}_3$  and a stoichiometric amount of  $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$  were dissolved in the suspension and stirred continuously for 45 min. Then, the mixtures were added to a 100 mL Teflon-lined stainless autoclave. All the autoclaves were covered by seals and maintained at 150 °C for 18 h. After reaction, all the autoclaves are placed on the experimental table until they are cooled to room temperature naturally. All the products were filtered and washed by ethanol and ultrapure water for 5 times then dried at 80 °C for 12 h. The  $g\text{-C}_3\text{N}_4/\text{BiOIO}_3$  samples with different molar ratios at 10:0.5, 10:0.75, 10:1, 10:2 and 10:4 were prepared according to this method. The as-prepared composites were named as CNB-0.5, CNB-0.75, CNB-1, CNB-2, and CNB-4.  $\text{BiOIO}_3$  product was also prepared under the same conditions as a reference.

### 2.3. Photocatalytic activity test

The bench-scale experimental equipment is made up of three main parts, namely simulated on-line mercury analyzer, simulated flue gas ( $\text{N}_2$ ), and photocatalytic reactor, which is consistent with the one we used in our previous studies [20]. Two branches of nitrogen are released from cylinders and controlled by mass flow meters (MFC, CS200) with 0.2 and 1.0 L/min flow rates respectively, one of which with 0.2 L/min passed through mercury generator (PSA, UK) with stated temperature (50 °C). Then, the branch carrying mercury ( $\text{Hg}^0$ ) vapor is mixed with another branch in a mixing tank, then introducing into photocatalytic reactor. The gas-phase  $\text{Hg}^0$  concentration is measured by an on-line mercury analyzer (RA-915-M, Lumex, Russia). The catalysts (50 mg) were coated on quartz glass plates (75 mm × 75 mm) by using a dip-coating method, and then it was put into the photocatalytic reactor under 18 W and 24 W LED light (400–450 nm) irradiation, which is shown in Fig. S1. At the beginning of each test, the nitrogen passed through the bypass then was switched to the photocatalytic reactor till the  $\text{Hg}^0$  concentration reached the setting value and the on-line mercury analyzer was calibrated. After analyses, the exhaust gas is introduced into the activated carbon and  $\text{KMnO}_4$  solution bottle to prevent its emission before being expelled into the atmosphere.

The removal efficiency of  $\text{Hg}^0$  under 18 W and 24 W LED lamp illumination is calculated using the Eq. (1).

$$\eta_{\text{Hg}}(\%) = \frac{H_{\text{g}_{\text{in}}}^0 - H_{\text{g}_{\text{out}}}^0}{H_{\text{g}_{\text{in}}}^0} \times 100\% \quad (1)$$

where  $H_{\text{g}_{\text{in}}}^0$  and  $H_{\text{g}_{\text{out}}}^0$  are  $\text{Hg}^0$  concentration ( $\mu\text{g}/\text{m}^3$ ) at the inlet and outlet of the photocatalytic reactor respectively.

## 3. Results and discussions

### 3.1. Characterization of $\text{BiOIO}_3/g\text{-C}_3\text{N}_4$ heterostructure

Fig. 1. shows the typical XRD peaks of the  $g\text{-C}_3\text{N}_4$  sample and  $\text{BiOIO}_3/g\text{-C}_3\text{N}_4$  nanosheets with different molar ratio of  $\text{BiOIO}_3$ . Two peaks at 13.04° and 27.47° can be detected in the as-prepared  $g\text{-C}_3\text{N}_4$ , which can be used as the indicators of (100) and (002) crystal face respectively [21]. This result is coincident with the JCPDS card (No. 87-1526). With increasing the amount of  $\text{BiOIO}_3$ , the two peaks decrease in intensity obviously, which indicates that  $\text{BiOIO}_3$  has been successfully deposited on the  $g\text{-C}_3\text{N}_4$  surface. The  $\text{BiOIO}_3/g\text{-C}_3\text{N}_4$  samples exhibit a two phase constitution:  $g\text{-C}_3\text{N}_4$  and  $\text{BiOIO}_3$ . All the diffractions of the  $\text{BiOIO}_3$  sample can be indexed to the orthorhombic phase of  $\text{BiOIO}_3$  (ICSD #262019). In the composites, the peaks at  $2\theta = 27.34^\circ$  can be assigned to the (121) planes of  $\text{BiOIO}_3$ . The (121) diffraction is quite strong, which suggests that the preferential growth was along the (121)

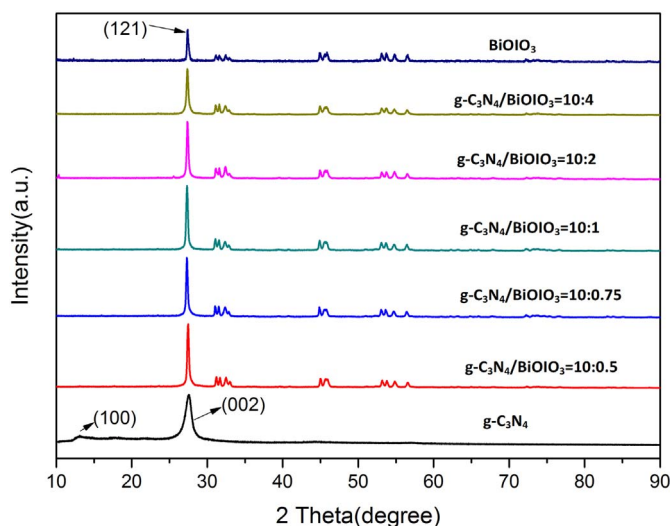


Fig. 1. XRD patterns of the as-prepared  $g\text{-C}_3\text{N}_4$  and  $g\text{-C}_3\text{N}_4/\text{BiOIO}_3$  composites.

direction. The sharp and intense diffraction peaks illustrate that the  $\text{BiOIO}_3$  has a good degree of crystallinity. It shows that there are no impurities formed between  $\text{BiOIO}_3$  and  $g\text{-C}_3\text{N}_4$  because no other phases can be detected in the composites. It is noteworthy that the intensity of characteristic peaks increases first and then decreases with molar doping ratio, and the maximum intensity is at molar ratio 10:1, which is in agreement with the experimental results.

The SEM images of  $g\text{-C}_3\text{N}_4$  and  $g\text{-C}_3\text{N}_4/\text{BiOIO}_3$  composites with diverse molar ratios are demonstrated in Fig.S2. In Fig.S2.a,  $g\text{-C}_3\text{N}_4$  displays a typically smooth morphology with a lamellar structure. From Fig.S2.b-f, it can be seen clearly that  $\text{BiOIO}_3$  nanosheets were deposited uniformly on the surface of the  $g\text{-C}_3\text{N}_4$  after the hydrothermal reaction. When the molar ratio of  $g\text{-C}_3\text{N}_4/\text{BiOIO}_3$  is 10:0.5, the sporadic  $\text{BiOIO}_3$  nanosheets distribute on the surface of  $g\text{-C}_3\text{N}_4$ . When the molar ratio of  $g\text{-C}_3\text{N}_4/\text{BiOIO}_3$  reaches to 10:0.75, a thin layer of  $\text{BiOIO}_3$  nanosheets attach to the  $g\text{-C}_3\text{N}_4$  surface. The  $\text{BiOIO}_3$  nanosheets are accumulated and agglomerated on the  $g\text{-C}_3\text{N}_4$  surface when further increases the amount of  $\text{BiOIO}_3$  (10:2). When  $\text{BiOIO}_3$  nanosheets are large in amount (10:4), the  $g\text{-C}_3\text{N}_4$  is covered by the  $\text{BiOIO}_3$  nanosheets with high-density. Thus, it could be confirmed that the  $\text{BiOIO}_3$  effectively distributed on the surface of  $g\text{-C}_3\text{N}_4$  and formed a large amount of heterostructures with the  $g\text{-C}_3\text{N}_4$  nanosheets. This result can also be obtained from the XRD and SEM characterization. The specific surface area and porosity of the  $g\text{-C}_3\text{N}_4$ ,  $\text{BiOIO}_3$  and a series of  $\text{BiOIO}_3/g\text{-C}_3\text{N}_4$  catalysts were accessed by  $\text{N}_2$  absorption-desorption isotherms and BJH pore-size distribution respectively, and the results are shown in Fig.S3 and Table S1. All the as-prepared samples belong to type IV in BDDT classification shape of isotherms with a  $\text{H}_3$ -type hysteresis loop in the IUPAC classification [22], which indicates the presence of plate slit and mesoporous structure. The results are in good consistent with the SEM patterns.

The TEM and HRTEM images of CNB-1 composite are shown in Fig. 2. In Fig. 2a, it can be obviously seen that the materials are composed of two parts. Fig. 2b shows obvious  $\text{BiOIO}_3$  lattice fringes corresponding to (121) plane of orthorhombic phase. The one with fringe spaces of 0.282 nm is in accordance with the (002) crystal plane of  $g\text{-C}_3\text{N}_4$ . The above-mentioned lattice fringes are in good agreement with XRD patterns of the samples. Moreover, the close interfacial contact between  $g\text{-C}_3\text{N}_4$  and  $\text{BiOIO}_3$  implies the heterostructure is well-formed between them, which are helpful for the migration of charges, thus can improve the photocatalytic efficiency.

The optical properties of as-prepared samples were investigated by the UV-vis diffuse reflectance spectroscopy. In Fig. 3a, the  $g\text{-C}_3\text{N}_4$  shows the absorption edge is around 530 nm. The pure  $\text{BiOIO}_3$  scarcely

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