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Porous NiCo₂O₄ nanowires supported on carbon cloth for flexible asymmetric supercapacitor with high energy density

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ABSTRACT

Recently, binary metal oxides have been considerably researched for energy storage since it can provide higher electrical conductivity and electrochemical activity than single components. Besides, rational arrays structure design can effectively enhance the utilization of active material. In this article, we synthesize a porous NiCo₂O₄ nanowires arrays, which were intimate contact with flexible carbon cloth (CC) by a facile hydrothermal reaction and calcination treatment. The rational array structures of NiCo₂O₄ facilitate the diffusion of electrolyte and effectively increase the utilization of active material. The as-obtained NiCo₂O₄@CC electrode exhibits a high capacitance of 1183 mF cm⁻² and an outstanding capacitance retention of 90.4% after 3000 cycles. Furthermore, a flexible asymmetric supercapacitor (ASC) using NiCo₂O₄@CC as positive electrode and activated carbon cloth (ACC) as negative electrode was fabricated, which delivers a large capacitance of 750 mF cm⁻² (12.5 F cm⁻³), a high energy density of 0.24 mWh cm⁻² (3.91 mWh cm⁻³), as well as excellent cycle stability under different bending states. These remarkable results suggest that as-assembled NiCo₂O₄@CC//ACC ASC is a promising candidate in flexible energy storage applications.

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1. Introduction

The increasing depletion of fossil fuel resources and growing environment crisis requires the development of clean and sustainable electrochemical rechargeable systems [1–4]. Compared with lithium-ion batteries, supercapacitor can charge and discharge at a much faster rate, therefore they have advantageous in high power density energy storage requirements [5–7]. Recently, in order to satisfy the demands for portable electronic devices, the development of flexible and wearable supercapacitors [8] becoming more and more urgent. A critical component of flexible supercapacitor is the flexible electrode material, which can maintain the electrochemical performance while endure large levels of bend, stretch or even compress.

Various flexible nanostructured electrode materials and supercapacitor configuration have been researched. Such as 3D paper-like graphene by the hard template-directed ordered assembly [9], highly compressible PANI-SWCNT-sponge electrode by “dipping

and drying” strategy and subsequent chemical oxidation polymerization [10], high-strength graphene composite film by molecular level coupling [11], highly flexible free-standing epidermal supercapacitor with 1 micron thickness [12] etc. However, most of these reports are nanocarbon-based flexible electrodes and symmetric configurations. To get an enhanced energy density of supercapacitor, pseudocapacitor or design an asymmetric configuration is considered to be efficient strategies. Pseudocapacitor exhibits a higher capacitance and energy density compared with EDLCs supercapacitors owing to the fast redox reactions. If utilization of two different materials as positive and negative electrode to construct an ASC device, a higher energy density will be easily obtained [13] according to the equation $E = 1/2CV^2$ [14]. Up to now, a series of pseudocapacitor and ASCs have been reported, for example, Wu et al. fabricated a high performance pseudocapacitor with 2D heterostructure film of ultrathin thiophene (TP) and electrochemically exfoliated graphene (EG) nanosheets, which exhibit impressive volumetric capacitance, high energy density and power density [15]. Chodankar et al. synthesis a 2D ultrathin nanoflakes of MnO₂ film on the flexible stainless foil, the pseudocapacitor shows excellent energy density of 23 Wh kg⁻¹ and power density of 1.9 kW kg⁻¹ [16]. Zheng et al. constructed a

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40 planar ASC with MnO₂/poly (3,4-ethylenedioxythiophene)-poly
41 (styrenesulfonate) (MP) as positive electrode and electrochemically
42 exfoliated graphene as negative electrode, which exhibit volumetric
43 energy density of 8.6 mWh cm⁻³ [17]. Xie et al. reported
44 a ASC with CoNi-LDH as positive electrode and activated carbon
45 as negative electrode, which delivers a high energy density of
46 46.59 Wh kg⁻¹ at a power density of 107 W kg⁻¹ [18] etc.

47 Pseudocapacitive characteristics electrode materials, such as
48 Co₃O₄ [6], NiMoO₄ [19], CoMoO₄ [20], ZnCo₂O₄ [21] and ZnWO₄
49 [22] have been widely researched. Among them, NiCo₂O₄ is re-
50 garded as one of the most attractive electrode materials owing
51 to its higher electrical conductivity and superior electrochemical
52 properties [23,24]. However, owing to the severe aggregation of
53 oxide nanoparticles and the short diffusion distance of electrolyte
54 into electrodes (~20 nm), only the materials surface of NiCo₂O₄
55 can participate in the electrochemical reactions, thus leading
56 to unsatisfactory specific capacitance [25]. In order to boost the
57 utilization of active materials, a rational arrays structure design
58 [26,27] by directly grown the electroactive materials on current
59 collector will facilitate electron and ion transport and benefit for
60 the full utilization of electrode materials. In addition, the use of
61 polymer binder and conductive additives can be avoided, thus re-
62 duce the contact resistance between current collector and elec-
63 trode materials. At present, some NiCo₂O₄ nanostructures grown
64 on CC have been prepared [28,29], to the best of our knowledge,
65 the study on a NiCo₂O₄@CC electrode used in a flexible ASC has
66 seldom been reported to date.

67 Here, a flexible solid-state ASC with NiCo₂O₄@CC as the posi-
68 tive electrode and ACC as the negative electrode in PVA/KOH gel
69 electrolyte was successfully fabricated. The positive electrode ma-
70 terial NiCo₂O₄ with hierarchical porous nanowires structure were
71 grown on CC with the aid of hydrothermal treatment combined
72 with subsequent calcination process. We studied the structural
73 evolution process of NiCo-precursor@CC to NiCo₂O₄@CC with the
74 increase of calcinations temperature, the results suggest that the
75 spinel NiCo₂O₄ phase began to form at the calcination temper-
76 ature of 250 °C and completely formed at 300 °C. In this pa-
77 per, ACC was used as negative electrode in an flexible ASC for
78 the first time due to their inherent porous structure and excel-
79 lent mechanical flexibility. Our optimal ASC device could be op-
80 erated at a high voltage of 1.5 V and exhibited a high areal ca-
81 pacitance of 750 mF cm⁻² (12.5 F cm⁻³), high energy density
82 of 0.24 mWh cm⁻² (3.91 mWh cm⁻³) and power density of
83 0.75 mW cm⁻² (12.51 mW cm⁻³), as well as excellent cycle stabil-
84 ity of 72.5% capacitance retention under different bending states
85 after 3000 cycles. The stored energy density of our ASC device in-
86 creased at least 6996 % when the voltage window increase from
87 0.6 V to 1.5 V. Such ASC device may open up opportunities for ap-
88 plications in flexible, lightweight and wearable electronics.

89 2. Experimental

90 2.1. Preparation of NiCo₂O₄@CC

91 CC (thickness: 0.36 mm, unit weight: 120 g m⁻²) was cut into
92 small pieces (2 cm × 3 cm), set aside. Co(NO₃)₂ · 6H₂O (1.16 g),
93 Ni(NO₃)₂ · 6H₂O (0.58 g), and urea (0.6 g) were added into DI wa-
94 ter (40 mL) under stirring for 15 min, then transferred to a 50 mL
95 Teflon-lined stainless autoclave. A piece of CC was putting into the
96 above solution, after hydrothermal reaction heated at 120 °C for
97 4 h, CC was taken out and rinsed with DI water and ethanol to
98 remove the other impurities. Finally, the dried product was an-
99 nealed in a quartz tube at different temperatures ranging from 200
100 to 500 °C for 2 h.

2.2. Fabrication of NiCo₂O₄@CC//ACC solid-state ASC

101

102 The solid-state ASC was fabricated by using NiCo₂O₄@CC as
103 positive electrode and ACC as negative electrode with PVA/KOH
104 gel as electrolyte. In a typical process, 1.8 g KOH was added into
105 30 mL PVA solution (3 g PVA, 30 mL water) and heated at 80 °C
106 until the solution became clear. NiCo₂O₄@CC and ACC electrodes
107 were coated with a thin layer PVA/KOH gel electrolyte and assem-
108 bled together.

2.3. Characterization

109

110 XRD spectroscopy (Bruker D8 Advance X-ray powder diffrac-
111 tiometer) and X-ray photoelectron spectroscopy (AXIS Ultra DLD
112 spectrometer instrument) were carried out to examine the phase
113 structures of the products. Thermogravimetric analysis (TGA) was
114 performed on a NETZSCH STA409PC instrument. The morphology
115 and mapping of the products were characterized by scanning elec-
116 tron microscopy (SEM, JSM-7001F), X-ray energy-dispersive spec-
117 tra (EDAX) connected with SEM equipment and transmission elec-
118 tron microscope (TEM, JEM-2010). The BET surface area and
119 the pore size distribution were measured at 77 K on a nitrogen
120 physisorption apparatus (JW-BK 122 W). Barrett-Joyner-Halenda
121 (BJH) model was used to determine the pore size distribution of
122 NiCo₂O₄@CC product. Density functional theory (DFT) was used to
123 calculate the pore size distribution of activated carbon cloth.

2.4. Electrochemical measurements

124

125 The electrochemical performances of single electrodes were
126 characterized by a three-electrode configuration in 6 M KOH
127 aqueous electrolyte, with Pt foil as the counter electrode and
128 Hg/HgO as the reference electrode. NiCo₂O₄@CC and ACC (both
129 1.4 cm × 1 cm) were used directly as the working electrode. By
130 TGA tests (Fig. S1), the mass loading of NiCo₂O₄ on CC was around
131 1.2 mg cm⁻². The areal capacitances (C_A, mF cm⁻²) and gravimet-
132 ric capacitances (C_M, F g⁻¹) were calculated from the galvanostatic
133 discharge curves by the following equations::

$$134 C_A = 1000I\Delta t/S\Delta V \quad (1)$$

$$135 C_M = I\Delta t/m\Delta V \quad (2)$$

136 where *I* (A) is the discharge current, Δt (s) is the discharge time,
137 *S* (cm²) is the area of single electrode, *m* (g) is the mass of active
138 materials in single electrode, for positive electrode, *m* (g) is the
139 mass of NiCo₂O₄, for negative electrode, *m* (g) is the mass of ACC,
140 and ΔV (V) is the voltage window during the discharge process.

141 As for an ASC, the positive and negative electrode should follow
142 the relationship $q_+ = q_-$. For each electrode, the charge (*q*) stored
143 depends on the gravimetric capacitances (C_M), the voltage window
144 during the discharge process (ΔV) and the mass of each electrode
(*m*) according to the following equation [14]:

$$145 q = C_M \times \Delta V \times m \quad (3)$$

146 Therefore, in order to get the charge balance, the mass loading
147 between the positive and negative electrode will follow the equa-
148 tion below [14]:

$$149 m_+/m_- = C_{M-} \times \Delta V_- / C_{M+} \times \Delta V_+ \quad (4)$$

150 Based on the gravimetric capacitances values and voltage window
151 for two electrodes, the suitable mass ratio of NiCo₂O₄@CC/ACC was
152 expected to be 1:2.18.

153 The electrochemical characterizations of the solid-state ASC
154 were performed in a two-electrode system. The areal capacitance,
155 areal energy density and power density, volumetric capacitance,

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