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Annealing temperature dependent catalytic water oxidation activity of iron oxyhydroxide thin films

P. T. Babar^a, B. S. Pawar^b, A. C. Lokhande^a, M. G. Gang^a, J. S. Jang^a, M. P. Suryawanshi^a, S. M. Pawar^b, Jin Hyeok Kim^{a,*}

^a Optoelectronic Convergence Research Center, Department of Materials Science and Engineering, Chonnam National University, Gwangju 500-757, South Korea ^b Division of Physics and Semiconductor Science, Dongguk University, Seoul 100-715, South Korea

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ABSTRACT

Nanostructured iron oxyhydroxide (FeOOH) thin films have been synthesized using an electrodeposition method on a nickel foam (NF) substrate and effect of air annealing temperature on the catalytic performance is studied. The as-deposited and annealed thin films were characterized by X-ray diffraction (XRD), X-ray photoelectron spectroscopy (XPS), field emission scanning electron microscopy (FE-SEM) and linear sweep voltammetry (LSV) to determine their structural, morphological, compositional and electrochemical properties, respectively. The as-deposited nanostructured amorphous FeOOH thin film is converted into a polycrystalline Fe₂O₃ with hematite crystal structure at a high temperature. The FeOOH thin film acts as an efficient electrocatalyst for the oxygen evolution reaction (OER) in an alkaline 1 M KOH electrolyte. The film annealed at 200 °C shows high catalytic activity with an onset overpotential of 240 mV with a smaller Tafel slope of 48 mV/dec. Additionally, it needs an overpotential of 290 mV to the drive the current density of 10 mA/cm² and shows good stability in the 1 M KOH electrolyte solution.

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1 1. Introduction

With the increasing consumption of fossil fuels and growing 2 concern about global warming, renewable energy sources must be-3 come a viable option for fossil fuel technology [1]. Solar and wind 4 5 are primary renewable energy sources; however these sources are intermittent and require conversion into renewable fuels [2]. Elec-6 7 trochemical water splitting has been considered as an efficient path to produce hydrogen fuels and to store electricity from re-8 newable sources, such as wind and sunlight [3]. Electrochemical 9 water splitting has two main half-cell reactions, one at the cath-10 ode where hydrogen evolution reaction (HER) occurs and another 11 at the anode where the oxygen evolution reaction (OER) occurs [4]. 12 The HER is relatively easy; however the OER is more complicated 13 14 and kinetically sluggish as it requires more overpotential than the 15 theoretical potential value (1.23 V) [5]. Therefore, an electrocatalyst is needed to reduce overpotential and to increase the reac-16 tion rate [6]. Currently, precious metals such as Ru and Ir-based 17 oxides exhibit good OER activity. However, due to their scarcity and 18 19 high-cost, their practical applications are greatly hampered [7]. Ac-

* Corresponding author. E-mail addresses: jinhyeok@jnu.ac.kr, jinhyeok@chonnam.ac.kr (J.H. Kim). cordingly, tremendous efforts have been undertaken to develop in-
expensive, highly durable and readily available electrocatalysts for
use as electrode materials in the OER to replace noble metal cata-
lysts. Recently, great progress has been made in developing oxides
and hydroxides based on earth abundant transition metals (Ni, Co,
Fe, Mn) as OER electrocatalysts [8–15].20

Among the various transition metal oxides/hydroxides inves-26 tigated as OER electrocatalysts, iron-based compounds can be 27 considered as better alternatives for RuO₂ and IrO₂ due to being 28 inexpensive, earth abundant and environmental friendly [16]. The 29 iron oxides and hydroxides can be synthesized by various methods, 30 such as chemical bath deposition (CBD) [17], successive ionic layer 31 adsorption and reaction (SILAR) [18], hydrothermal [19], spray 32 pyrolysis [20] and electrodeposition [21]. Among these methods, 33 electrodeposition is a simple, scalable and low temperature versa-34 tile method that allows control over the thickness of the deposited 35 layer simply by changing either the current or voltage [21]. The 36 major advantage of this deposition method is that the material 37 can be deposited on the desired conductive substrate without the 38 need of a binder. Recently, iron oxide/hydroxide were used as OER 39 electrocatalysts for water splitting. Trotochaud et al. [22] studied 40 the OER electrocatalytic properties of spin coated FeO_x films and 41 reported 405 mV as the onset potential. Smith et al. [23] syn-42 thesized amorphous FeO_x using photochemical metal organic 43

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deposition and reported 320 mV as the onset overpotential. Liu et 44 45 al. [24] synthesized α -Fe₂O₃ nanorods by a hydrothermal method on carbon cloth which showed 330 mV as the onset potential in an 46 47 alkaline electrolyte. Zhao et al. [10,16] reported the electrodeposition of iron-based compounds on an ITO substrate in aqueous and 48 non-aqueous media. It has been observed that the aqueous de-49 posited iron oxide shows an onset potential of 408.6 mV, whereas 50 iron oxyhydroxide deposited in the latter medium shows an onset 51 52 potential of 470 mV. Cao et al. [25] have electrodeposited an iron based thin film on an ITO substrate with an onset potential of 53 54 480 mV. Zou et al. [26] have electrodeposited an iron (oxy) hydroxide thin film on an Au substrate with an onset potential of 350 mV. 55 Recently, three dimensional (3D) porous nanostructured nickel 56 57 foam (NF) has been used for depositing efficient electrocatalyst materials due to its high surface area, excellent conductivity, and 58 high number of exposed active sites favoring sufficient accessibility 59 of the electrolyte for higher electrochemical performance [27,28]. 60

In this manuscript, we have synthesized iron oxyhydroxide 61 (FeOOH) thin films on nickel foam (NF) substrates by an electrode-62 position method followed by air annealing at different tempera-63 tures. The impact of air annealing on the structural, morphologi-64 cal, and electrochemical properties of the thin films is studied. The 65 66 FeOOH thin film exhibits a lower overpotential of 290 mV at a current density of 10 mA/cm² in 1 M KOH electrolyte solution. In ad-67 dition, it exhibits excellent stability in the electrolyte for 11 h, in-68 dicating the promising material for OER electrocatalyst. 69

70 2. Experimental

The iron oxyhydroxide (FeOOH) thin films were synthe-71 sized using an electrodeposition method on the NF substrate 72 73 from an aqueous electrolytic bath containing 0.01 M iron nitrate $(Fe(NO_3)_3 \bullet 9H_2O)$ at room temperature. The electrochemical cell 74 75 contained a saturated calomel electrode (SCE) as the reference electrode, platinum as the counter electrode and NF with a deposi-76 tion area of 1×1 cm² as the working electrode. Analytical reagent 77 grade (AR) chemicals used for precursor solution preparation (sup-78 79 plied by Sigma-Aldrich). Prior to electrodeposition, NF was first 80 cleaned with 1 M HCl solution with ultra-sonication for 15 min to remove the nickel oxide layer. After that, the NF was washed with 81 acetone, ethanol and doubly distilled water for 15 min and then 82 dried at 60 °C for 24 h. The adherent and uniform films were de-83 posited at -1.0 V vs. SCE in potentiostatic mode for 300 s using a 84 WonATech, WMPG1000 Multichannel Potentiostat/Galvanostat ver. 85 1.11. After film deposition, the deposited films were rinsed with 86 double distilled water and then dried at room temperature. The 87 films were then annealed in an air atmosphere at different an-88 nealing temperature ranging from 100 °C to 400 °C for 2 h. The re-89 sulting films were denoted as Fe-OH/NF-100, Fe-OH/NF-200, Fe-90 OH/NF-300, Fe-OH/NF-400, respectively. The as-deposited and an-91 nealed films were used for further characterization. 92

93 The structural and chemical properties of the as-deposited 94 and annealed thin films were studied using high resolution X-ray diffraction (XRD) with Ni-filtered CuK α radiation (k α = 1.54056 Å) 95 (X_pert PRO, Philips, Eindhoven, Netherlands) and X-ray photoelec-96 tron spectroscopy (XPS, VG Multilab 2000, Thermo VG Scientific, 97 UK) with a monochromatic Mg-K α (1253.6 eV) radiation source, re-98 spectively. The surface morphology and compositional analysis of 99 the films was measured using an FE-SEM (field emission scanning 100 101 electron microscope, Model: JSM-6701F, JEOL, Japan) attached to an energy-dispersive X-ray analysis (EDAX) system. Electrochemi-102 cal measurements were made using a standard three electrode cell 103 configuration with a 1 M KOH electrolyte solution using the as-104 deposited/annealed films as a working electrode, platinum as the 105 counter electrode and SCE as the reference electrode. To evaluate 106 the OER activities of all the films, linear sweep voltammetry (LSV) 107



Fig. 1. XRD patterns of as-deposited and annealed FeOOH thin films at different temperatures.

was performed to acquire the polarization curve. All the potentials 108 recorded in this work were vs. reversible hydrogen electrode (RHE) 109 by the Nernst equation $E_{\text{RHE}} = E_{\text{SCE}} + (0.059 \times \text{pH}) + 0.24$ and the 110 overpotential (η) was obtained by the equation: $\eta = E_{\text{RHE}} - 1.23$ V 111 [29]. 112

3. Results and discussion

Fig. 1 shows the XRD patterns of the as-deposited and air an-114 nealed thin films in the range of 100-400 °C for 2 h. The XRD spec-115 tra reveal the amorphous nature of the as-deposited and the an-116 nealed films up to 200 °C. However, for the films annealed above 117 200 °C, weak diffraction peaks corresponding to the (104), (110), 118 (113), (024) and (116) planes are observed in the spectra indicat-119 ing the formation of a crystalline α -Fe₂O₃ hematite phase. The ob-120 served crystalline phase matches with the JCPDS-89-0548 card and 121 is in good agreement with the reported results [30, 31]. The in-122 tense and sharp peaks (indicated by *) in the XRD spectra are at-123 tributed to the NF substrate. Thus, it is clear that the crystallinity 124 of the Fe₂O₃ films is enhanced with increasing the annealing tem-125 perature. 126

Fig. 2 shows the XPS spectra for the as-deposited and annealed 127 thin films. In the survey spectra, the photoelectron peaks with 128 binding energies (BEs) of 724 eV, 710 eV, 530 eV and 284 eV are 129 observed, which could be assigned to the Fe $2p_{1/2}$, Fe $2p_{3/2}$, O 1s, 130 C 1s, respectively (Fig. 2(a)). The high resolution XPS spectra for 131 the Fe 2p and O 1s regions for all the films are shown in Fig. 2(c, 132 d). The two major peaks of Fe 2p are centered approximately at 133 711.5 eV and 724.53 eV for the as-deposited film [32]. Interestingly. 134 the BE values decreased with an increase in the annealing tem-135 perature, and the spectrum shows 710.7 eV and 724.12 eV for the 136 film annealed at 400 °C [33]. The characteristic satellite peak is as-137 sociated with Fe $2p_{3/2}$ and is located approximately at 8 eV higher 138 than main Fe $2p_{3/2}$ peak [34]. The as-deposited film shows lowest 139 BE peaks at 529.5 eV, 531 eV and 532.5 eV in the O 1s region and 140 can be assigned to O^{2-} , OH and chemically or physically adsorbed 141 water respectively. The presence of -OH group suggests that the 142 as-deposited film contain FeOOH phase [35]. Also Fe-OH/NF-100 143 and Fe-OH/NF-200 °C samples show similar peaks. The ratio of OH 144 to O^{2-} is approximately 1, which indicates the presence of FeOOH 145 phase in these samples. However, as annealing temperature in-146 creases, the -OH group peak is disappeared due to evaporation 147 of hydroxide content from the film and single phase α -Fe₂O₃ is 148 formed. Also, O²⁻ peak is slightly shifted monotonically toward the 149 lower energy side. 150

Fig. 3 shows the SEM images of the as-deposited and annealed 151 iron-based compound thin films deposited on NF substrates at different annealing temperatures. The as-deposited (Fig. 3(b)) and annealed films below 200 °C (Fig. 3(c, d)) show smooth, dense uni-154

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