ARTICLE IN PRESS

Advanced Powder Technology

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Advanced Powder Technology xxx (2018) xxx-xxx

Contents lists available at ScienceDirect

Advanced Powder Technology

journal homepage: www.elsevier.com/locate/apt





Effect of Mn doping concentration on structural, vibrational and magnetic properties of NiO nanoparticles

Kiran N. Patel^{a,*}, M.P. Deshpande^a, Krishna Chauhan^a, Piyush Rajput^a, Vivek P. Gujarati^a, Swati Pandya^a, Vasant Sathe^b, S.H. Chaki^a

10 ^a Department of Physics, Sardar Patel University, Vallabh Vidyanagar 388120, Gujarat, India 11 ^b UGC-DAE Consortium for Scientific Research, Khandwa Road, Indore 452017, India

ARTICLE INFO

Article history: 18 Received 4 March 2018 19 Received in revised form 20 June 2018 20 Accepted 22 June 2018 21 Available online xxxx

- 22 Keywords:
- 23 Mn-doped NiO
- 24 Raman spectroscopy
- 25 Ferromagnetism
- 26 27 Diluted magnetic semiconductor

ABSTRACT

The $Ni_{1-x}Mn_xO(x = 0.00, 0.02, 0.04 \text{ and } 0.06)$ nanoparticles were synthesized by chemical precipitation route followed by calcination at 500 °C for 4 h. The prepared samples were characterized by energy dispersive analysis of X-rays (EDAX), powder X-ray diffraction (XRD), transmission electron microscopy (TEM), Raman spectroscopy, Fourier transform infrared spectroscopy (FT-IR) and vibrating sample magnetometer (VSM). Rietveld refinement of XRD data confirms the structural phase purity and XRD patterns are well indexed to NaCl like rock salt fcc crystal structure with Fm-3m space group. The particle size of Mn doped samples is found to be less than that of pure NiO sample. However, the particle size increases slightly on increasing the Mn concentration due to surface/grain boundary diffusion. The vibrational properties of the synthesized nanoparticles were investigated by Raman and FT-IR spectroscopy. The results of room temperature magnetization (M-H) and temperature dependent magnetization (M-T) measurements are explained with a core-shell model. The synthesized nanoparticles show weak ferromagnetic and super-paramagnetic like behavior at room temperature.

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1. Introduction

47 Recently, transition metal ion doped oxide diluted magnetic semiconductors (DMS) have attracted huge attention in the field 48 of advanced spintronic, magneto-electronic and magneto-opto-49 electronics [1]. Huge research has been demonstrated on magnetic 50 51 or non-magnetic transition metal ion doped oxide DMSs which 52 exhibit room temperature ferromagnetism (RTFM) with Curie tem-53 perature well above room temperature for their potential applica-54 tion in spintronics [2–4]. One of the most intensively investigated 55 members of metal oxide family, NiO possesses anti-ferromagnetic 56 behavior below Neel temperature of about 523 K [5,6]. NiO nanoparticles with size smaller than 31 nm leads to large anoma-57 lous magnetic moments with noticeable coercivity and loop shift 58 59 at low temperature [7]. The ferromagnetism has been demonstrated in NiO nano-crystalline sample having grain size of 5 nm 60 which is believed to be due to missing bonds and lattice distortion 61 [8]. Although, pure NiO is a Mott- Hubbard insulator with a wide 62 band gap of about 4 eV in bulk form, it shows p-type semiconduct-63

ing behavior due to presence of Ni⁺² vacancies or on doping of cations [9–12].

The room temperature ferromagnetism in pure and Fe doped 66 NiO nanoparticles is reported by Mishra et al. [13] and Khemprasit 67 et al. [14]. Swatsitang et al. [15] reported theoretical calculations 68 on the magnetic properties of Mn-doped NiO using density func-69 tional theory with generalized gradient approximation (GGA) and 70 plane wave basis. They have shown ferromagnetism in Mn-doped 71 NiO and also shown increase in net magnetization with increase 72 of Mn concentration up to 9.375%. Mallick et al. [16] have synthe-73 sized nano-crystalline $Ni_{1-x}Mn_xO$ (x = 0, 0.01, 0.03 and 0.05) with 74 average grain size of 21-28 nm by a chemical route using nickel 75 76 nitrate hexahydrate and manganese acetate and calcined at 500 °C. They have observed anti-ferromagnetic behavior at 10 K in pris-77 tine and 1% Mn doped NiO nanoparticles and super-78 79 paramagnetism at room temperature in higher doped samples with average blocking temperature of the order of 180 K. Raja 80 et al. [17] have synthesized $Ni_{1-x}Mn_xO$ (x = 0.00, 0.01, 0.02 and 81 0.03) nanoparticles by chemical co-precipitation method and 82 found single phase up to 2% Mn-doping concentration. However, 83 RTFM was not observed for their samples (x < 0.02). Anandan 84 et al. [18] have synthesized $Ni_{1-x}Mn_xO$ (x = 0.00, 0.01, 0.02 and 85

* Corresponding author. E-mail address: kirankumar_patel_physics@spuvvn.edu (K.N. Patel).

https://doi.org/10.1016/j.apt.2018.06.018

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Please cite this article in press as: K.N. Patel et al., Effect of Mn doping concentration on structural, vibrational and magnetic properties of NiO nanoparticles, Advanced Powder Technology (2018), https://doi.org/10.1016/j.apt.2018.06.018

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86 0.03) nanoparticles by precipitation route and calcined at 500 °C. 87 They have found anti-ferromagnetic to super-paramagnetic transi-88 tion up to 2% Mn doping concentration whereas 3% Mn doped sam-89 ple exhibits weak ferromagnetism at room temperature. Recently, 90 Layek et al. [19] have synthesized single phase $Ni_{1-x}Mn_xO$ (x = 0.00, 0.02, 0.04 and 0.06) nanoparticles up to 6% Mn-doping by 91 92 low temperature hydrothermal method and calcined at 400 °C. 93 They have observed RTFM in their samples but net magnetization 94 decreases with increase of Mn-doping concentration which contradicts with theoretical calculations done by Swatsitang et al. They 95 96 also concluded that observed RTFM is an intrinsic property of the 97 material and was not due to any other impurity phases. Bharathy et al. [20] have synthesized single phase $Ni_{1-x}Mn_xO$ (x = 0.0, 0.1, 98 0.2 and 0.3) nanoparticles up to 30% Mn-doping by sol-gel method. 99 100 Their samples show weak ferromagnetic behavior at room temper-101 ature. Similarly, Sankar et al. [21] have synthesized $Ni_{1-x}Mn_xO(x)$ 102 = 0.0, 0.1, 0.2, 0.3 and 0.4) nanoparticles and studied their optical and photocatalytic properties. Therefore, it can be seen that syn-103 thesis route plays an important role in structural and subsequently 104 on magnetic properties of Mn-doped NiO nanoparticles. Therefore, 105 106 a simple, efficient and reproducible method to synthesize Mn-107 doped NiO nanoparticles having RTFM with significant magnetiza-108 tion is yet to reach the desire grade.

109 In this present article, we demonstrate the structural, morpho-110 logical, vibrational and magnetic properties of $Ni_{1-x}Mn_xO(x = 0.00)$, 111 0.02, 0.04 and 0.06) nanoparticles synthesized by chemical precip-112 itation route with the addition of polymer. Herein, room temperature ferromagnetism is achieved and doping assisted enhanced 113 magnetism is realized for $Ni_{1-x}Mn_xO$ (x = 0.00, 0.02, 0.04 and 114 115 0.06) nanoparticles.

2. Experimental details 116

117 2.1. Synthesis

118 $Ni_{1-x}Mn_xO$ (x = 0.00, 0.02, 0.04 and 0.06) nanoparticles were 119 synthesized by chemical precipitation technique [22] with the 120 addition of polyvinyl-pyrrolidone (PVP) which acts as a capping 121 agent. In the experiment, 20 ml of 0.25 M aqueous solution of 122 metal chloride salts using stoichiometric amount of nickel chloride hexahydrate (NiCl₂·6H₂O), manganese chloride tetra-hydrate 123 (MnCl₂·4H₂O) and 100 ml of 0.5 M aqueous solution of sodium 124 125 hydroxide (NaOH) were prepared in double distilled water separately. Then 0.4 g PVP was dissolved in NaOH solution and heated 126 127 at 45 °C. The metal chloride solution was added drop wise to the 128 heated NaOH solution under high speed stirring for 2 h. The light 129 green or brownish green solution depending on doping concentra-130 tion of Mn was observed and was left for 2 h. The resulting light 131 green or brownish green precipitates of $(Ni_{1-x}Mn_x(OH)_2)$ (x = 132 0.00, 0.02, 0.04 and 0.06) was filtered using whatman filter paper and washed twice with deionized water and ethanol to remove 133 the unreacted salts. The samples were dried in a hot air oven at 134 80 °C for 10 h. The obtained dried materials $(Ni_{1-x}Mn_x(OH)_2)$ is 135 136 then grounded with mortar and pestle. Then, the hydroxide powder of Ni_{0.98}Mn_{0.02}(OH)₂ was characterized by TGA/DTG/DTA using 137 138 Seiko SII-EXSTAR TG/DTA-7200 to study their thermal decomposition properties and to decide proper calcination temperature in 139 order to obtain Ni_{0.98}Mn_{0.02}O nanoparticles. 140

141 Fig. 1 represents TGA, DTG and DTA curves of synthesized dried 142 hydroxide powders of Ni_{0.98}Mn_{0.02}(OH)₂ recorded for 5 °C min⁻¹ 143 heating rate in the N₂ environment from room temperature to 144 750 °C. The first step was observed at temperature below 150 °C 145 and corresponding weight loss is 12.84%. This step can be associ-146 ated with the elimination of chemically adsorbed and/or struc-147 turally bonded water molecules [23,24]. The second and major



Fig. 1. TGA, DTG and DTA curves of dried powder of Ni_{0.98}Mn_{0.02}(OH)₂.

step seen in the temperature range 200 °C to 300 °C, shows weight 148 losses of 17.51%. This step is related to the thermal decomposition 149 of Ni_{0.98}Mn_{0.02}(OH)₂ to Ni_{0.98}Mn_{0.02}O [23,24]. Further, thermal 150 decomposition of Ni_{0.98}Mn_{0.02}(OH)₂ to Ni_{0.98}Mn_{0.02}O leads to the 151 oxidation of Ni⁺² or Mn⁺² to a higher oxidation state like Ni_{0.98}-152 $Mn_{0.02}O_v$ (where y is stoichiometry) [25]. The third and final step 153 marked by sharp decrease in the slope of TGA curve up to 500 °C, 154 shows weight loss of 3.08%. This may correspond to the decompo-155 sition of non-stoichiometric Ni_{0.98}Mn_{0.02}O_v to Ni_{0.98}Mn_{0.02}O [23,25] 156 and no further weight loss was observed beyond 500 °C. Therefore, 157 the dried precipitates of $Ni_{1-x}Mn_x(OH)_2$ ((x = 0.00, 0.02, 0.04 and 158 0.06) was calcined in air at 500 °C for 4 h and black colored Ni_{1-x}-159 Mn_xO (x = 0.00, 0.02, 0.04 and 0.06) nanoparticles are obtained. 160

2.2. Characterization

The chemical composition of the synthesized $Ni_{1-x}Mn_xO$ (x = 162 0.00, 0.02, 0.04 and 0.06) samples was studied by EDAX. The struc-163 ture and phase purity of samples were characterized by XRD using 164 Rigaku powder X – ray diffractometer with Cu K_{α} radiation (λ = 1. 165 5418 Å). Further, morphology and the crystalline nature of the syn-166 thesized nanoparticles were checked by TEM using a Tecnai 20 167 electron microscope operated at 200 kV. Selected area electron 168 diffraction (SAED) patterns were also recorded for further struc-169 tural analysis. The vibrational properties of the synthesized 170 nanoparticles was studied by Raman spectroscopy using diode 171 laser (473 nm, 25 mW) as an excitation source at room tempera-172 ture. Further, the FT-IR spectra of the synthesized $Ni_{1-x}Mn_xO(x)$ 173 = 0.00, 0.02, 0.04 and 0.06) nanoparticles were recorded using Perkin Elmer GX instrument. The room temperature isothermal magnetization (M-H) measurements with an applied magnetic field up to 5 T and temperature dependent magnetization (M-T) measure-177 ments in the temperature interval 10-350 K with an applied mag-178 netic field of 500 Oe were done using 14T PPMS-vibrating sample 179 magnetometer (VSM). 180

3. Results and discussion

The EDAX spectra of $Ni_{1-x}Mn_xO$ (x = 0.00, 0.02, 0.04 and 0.06) 182 samples are shown in Fig. 2(a-d). The synthesized samples are sto-183 ichiometric and do not contain any other foreign impurity ele-184 ment/impurity. The weight and atomic percentage of the 185 $Ni_{1-x}Mn_xO$ (x = 0.00, 0.02, 0.04 and 0.06) samples is presented in 186 Table 1. 187

Fig. 3 represents powder XRD patterns of the synthesized Ni_{1-x}- Mn_xO (x = 0.00, 0.02, 0.04 and 0.06) nanoparticles along with Rietveld fit. The Rietveld refinement was carried out using full-prof software to determine lattice parameters. The XRD patterns of $Ni_{1-x}Mn_xO$ (x = 0.00, 0.02, 0.04 and 0.06) nanoparticles are well

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