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⁴ Synthesis of highly crystalline hexagonal cesium tungsten bronze $\frac{1}{2}$ nanoparticles by flame-assisted spray pyrolysis

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ABSTRACT

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44 1. Introduction

 Recently, near-infrared shielding materials have become highly desired for applications in solar control windows for automobiles and buildings to reduce the energy consumption of air condition-48 ing. Noble metal nanoparticles (e.g., Ag $[1]$ or Au $[2]$), semiconduc-49 tor oxides (e.g., indium tin oxide (ITO) $[3]$ or antimony tin oxide (ATO) [\[4\]](#page--1-0)), black compounds, and rare-earth hexaborides show remarkably strong absorption of near-infrared light owing to the effects of localized surface plasmon resonances [\[5\].](#page--1-0)

53 Tungsten bronze nanoparticles, such as tungsten trioxide doped 54 with alkali metals, are promising candidates for applications to 55 near-infrared shielding $[6,7]$. Takeda and Adachi found that tung-56 sten bronze nanoparticles $(M_xWO_3, M = Na, Tl, Rb, and Cs)$ pro-57 duced by solid state reactions and milling methods have 58 excellent optical properties $[8]$. The incorporation of the cations 59 inside the $WO₃$ crystal structure also introduces free electrons, 60 which are essential for enhancing the localized surface plasmon 61 resonance. Hexagonal cesium tungsten bronze $(Cs_{0.32}WO_3)$ 62 nanoparticles are regarded as particularly promising for solar con-

process in this paper has several advantages, including a short reaction time and the ability to yield prod- 36 ucts with high purity and good energy efficiency. Furthermore, the $C_{50,32}WO_3$ nanoparticles produced in 37
this research showed a remarkable near-infrared shielding ability with a 97.7% cut-off at 1500 nm this research showed a remarkable near-infrared shielding ability with a 97.7% cut-off at 1500 nm.

Highly crystalline and hexagonal single-phase cesium tungsten bronze $(Cs_{0.32}WO_3)$ nanoparticles were 32

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trol window applications. Cesium ions have a plasmon resonant 63 frequency that produces strong absorption in the near-infrared 64 region and high transmittance in the visible region with a sharp 65 gradient between the two regions [\[9\]](#page--1-0). 66

To date, $Cs_{0.32}WO_3$ nanoparticles have been synthesized by 67 solid state $[8,10]$, hydrothermal $[11]$, water controlled-release 68 solvothermal $[12]$, and thermal plasma $[13]$ synthesis methods. 69 Among these approaches, the solid state synthesis method has 70 been widely used to produce $Cs_{0.32}WO_3$ nanoparticles on an indus- 71 trial scale. However, this method involves many elaborate and 72 multi steps, including heating under a H_2/N_2 atmosphere (e.g., at 73 approximately 550 °C for 1 h), under a N_2 atmosphere (e.g., at 74 approximately 800 \degree C for 1 h), and mechanical grinding (e.g., for 75 6 h) [\[8\].](#page--1-0) In addition, contamination occurring during the grinding 76 step has also become a serious problem. Therefore, it is necessary 77 to develop a simple and energy-efficient method for producing 78 $Cs_{0.32}WO₃$ nanoparticles that is feasible for the large-scale synthe- 79 sis of materials required for practical applications. 80

Flame-assisted spray synthesis (FASP) shows great promise as a 81 process for continuous production of nanoparticles at a high rate. 82 In the FASP process, the particle size, crystal size, and morphology 83 of nanoparticles can be controlled by adjusting the gas flow rates 84 and precursor concentration $[14,15]$. Owing to these advantages, 85 a wide range of materials, from simple oxides to more complex 86

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successfully synthesized by a flame-assisted spray pyrolysis followed by annealing under a reducing 33 gas atmosphere. The resulting $Cs_{0.32}WO_3$ nanoparticles featured a pure hexagonal $Cs_{0.32}WO_3$ phase with 34
a bigh crystallinity and homogeneous chemical composition Unlike conventional methods the proposed 35 a high crystallinity and homogeneous chemical composition. Unlike conventional methods, the proposed

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87 functional materials, can be produced by the FASP method 88 $[16-19]$. In this work, for the first time we report the synthesis 89 of $Cs_{0.32}WO_3$ nanoparticles with the use of the FASP method by 90 varying the methane (fuel) gas flow rates and investigate the 91 effects of the flame temperature on the physical and optical prop-92 erties of the so-formed nanoparticles. The as-synthesized $Cs_{0.32}$ -93 WO₃ nanoparticles were annealed at various temperatures under 94 an H₂/Ar atmosphere to achieve a single $Cs_{0.32}WO₃$ phase. The 95 near-infrared absorption properties of the as-synthesized $Cs_{0.32}$ - 96 WO₃ nanoparticles before and after annealing were evaluated.

97 2. Experimental

98 2.1. Experimental setups

 A schematic diagram of the experimental setups for the prepa-100 ration of the $Cs_{0.32}WO_3$ nanoparticles is shown in Fig. 1. The FASP system [Fig. 1(a)] consisted of an ultrasonic nebulizer (NE-U17, Omron Healthcare Co., Ltd., Tokyo, Japan; operated at 1.7 MHz) for droplet formation, a diffusion flame burner (see Supporting Information Fig. SI-1), a glass flame reactor, and a bag filter for par- ticle collection. Two precursors, i.e., an aqueous solution of 0.01 106 mol/L ammonium tungstate pentahydrate (ATP, $(NH_4)_{10}(W_12O_{41})$.
107 5H₂O: Kanto Chemical Co., Inc., Japan: purity 88–90%) and an aque- $5H₂O$; Kanto Chemical Co., Inc., Japan; purity 88–90%) and an aque-108 ous solution of 6.45 g/L cesium carbonate $(Cs_2CO_3; Sigma-Aldrich$ Co., USA; purity 99.9%) were simultaneously supplied into the ultrasonic nebulizer by peristaltic pumps. The mixture of both pre- cursors was immediately nebulized to prevent precipitation, and the generated droplets were subsequently fed into the central tube 113 of the diffusion flame burner with a carrier gas $(N_2; 3 L/min)$. 114 Methane (CH_4) was used as the fuel gas, and its flow rate was var-115 ied from 0.5 to 3.0 L/min to investigate the effects of flame temper- ature on the obtained nanoparticles. The flow rate ratio of oxygen $(O₂)$ to methane was maintained at 2.5 to ensure complete com-
118 bustion. The as-synthesized nanoparticles were collected in a bag bustion. The as-synthesized nanoparticles were collected in a bag filter. During the experiment, the temperature inside the particle collector was maintained at 200 \degree C to avoid water condensation. 120 To produce a single phase $Cs_{0.32}WO_3$, the as-synthesized particles 121 were annealed inside a tubular furnace $[Fig. 1(b)]$ $[20,21]$. The 122 FASP-made nanoparticles were heated for 1 h at various tempera- 123 tures at a heating rate of 400 °C/h under a 5%-H₂/Ar atmosphere 124 based on the previously reported literatures $[8,10,11]$. After 125 annealing, the nanoparticles were allowed to cool naturally. 126

Supplementary data associated with this article can be found, in 127 the online version, at [https://doi.org/10.1016/j.apt.2018.07.001.](https://doi.org/10.1016/j.apt.2018.07.001) 128

2.2. Characterization 130

A scanning electron microscope (SEM; S-5200, Hitachi, Tokyo, 131 Japan; operated at 5–20 kV) and a transmission electron micro-
132 scope (TEM; JEM-3000F, JEOL Ltd., Tokyo, Japan; operated at 133 297 kV) were used to investigate the size, morphology, and crystal 134 structure of the as-synthesized nanoparticles. The crystal struc- 135 tures of the $Cs_{0.32}WO_3$ nanoparticles were also determined by 136 X-ray diffraction (XRD; D2 PHASER, 40 kV and 30 mA, Bruker Corp., 137 USA). The ATP powder and precipitates of the precursor mixture 138 were characterized by thermogravimetric analysis (TGA; TGA-60, 139 Shimadzu, Japan; at a heating rate of $10 °C/min$ and N_2 carrier 140 gas flow rate of 50 mL/min). To characterize their optical perfor- 141 mance, the as-synthesized $Cs_{0.32}WO_3$ nanoparticles were dispersed 142 in methyl isobutyl ketone at a concentration of 0.02 wt% $[9]$. Opti- 143 cal measurements were performed using a UV–Vis-NIR spec- 144 trophotometer (Model V-670, JASCO Corporation, Japan). 145

3. Results and discussion 146 and 147 and 147 and 147 and 147 and 147 and 147

3.1. Effects of flame temperature on the formation and physical 147 properties of $Cs_{0.32}WO_3$ nanoparticles 148

In general, the flame temperature of the reaction field in the 149 FASP process is influenced by the flow rates of methane and oxy-
150 gen [\[22\]](#page--1-0). Generally, a higher methane flow rate results in a higher 151

Fig. 1. Schematic diagram of experimental setup for (a) flame-assisted spray synthesis and (b) annealing processes.

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