Advanced **Powder Technology** 

# [Advanced Powder Technology xxx \(2017\) xxx–xxx](https://doi.org/10.1016/j.apt.2017.12.016)

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# Advanced Powder Technology

journal homepage: [www.elsevier.com/locate/apt](http://www.elsevier.com/locate/apt)

# <sup>2</sup> Original Research Paper

# Fabrication of  $MoS<sub>2</sub>$ -graphene modified with  $Fe<sub>3</sub>O<sub>4</sub>$  particles  $\frac{7}{5}$  and its enhanced microwave absorption performance

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## article info

1 4 2 6 15 Article history:<br>16 Received 29 Ju 16 Received 29 July 2017<br>17 Received in revised for 17 Received in revised form 14 December 2017 18 Accepted 19 December 2017<br>19 Available online xxxx Available online xxxx

- 20 Keywords:<br>21 Graphene
- 21 Graphene<br>22  $MoS<sub>2</sub>$
- $\frac{22}{23}$  MoS<sub>2</sub> Nanocomposites

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24 Microwave absorption properties 25

#### ABSTRACT

To obtain high-performance microwave absorbers, the 3D hierarchical structure material (MoS<sub>2</sub>/ 27<br>Fe<sub>3</sub>O<sub>a</sub>/graphene) has drawn much attention due to its superior structure and properties. Here 3D archi- 28  $Fe<sub>3</sub>O<sub>4</sub>/graphene$ ) has drawn much attention due to its superior structure and properties. Here, 3D architecture like flower structure of  $MoS<sub>2</sub>/graph$ ene was synthesized by one-pot hydrothermal. Fe<sub>3</sub>O<sub>4</sub> particles 29 were grown on the petals of  $MoS_2$ /graphene flower structure through co-precipitation route. 30<br>Electromagnetic (EM) wave absorbing results of MoS<sub>2</sub>/Fe<sub>2</sub>O<sub>4</sub>/graphene demonstrate the maximum reflec- 31 Electromagnetic (EM) wave absorbing results of MoS<sub>2</sub>/Fe<sub>3</sub>O<sub>4</sub>/graphene demonstrate the maximum reflec-<br>tion loss (R<sub>i</sub>) is -45.8 dB at 5.9 GHz for a coating of 2.5 mm which is significantly enhanced compared 32 tion loss  $(R_L)$  is  $-45.8$  dB at 5.9 GHz for a coating of 2.5 mm, which is significantly enhanced compared 32<br>with MoS<sub>2</sub>/graphene, in terms of absorption intensity and absorbing bandwidth. The absorption band-<br>33 with  $MoS<sub>2</sub>/graphene$ , in terms of absorption intensity and absorbing bandwidth. The absorption bandwidth less than  $-10$  dB can achieve almost 7.4 GHz (4.6–9.7 GHz, 15.7–18 GHz) with a thickness of 34<br>2.5 mm The excellent EM wave properties of MoS-/Fe-0./graphene are ascribed to the suitable impe-<br>35 2.5 mm. The excellent EM wave properties of  $MoS<sub>2</sub>/Fe<sub>3</sub>O<sub>4</sub>/graphene$  are ascribed to the suitable impe-<br>dance matching and multi-polarization. Consequently it is believed that the 3D hierarchical structure 36 dance matching and multi-polarization. Consequently, it is believed that the 3D hierarchical structure 36<br>of MoS-/Fe-Q /graphene can serve as a potential FM wave absorber and can be used in practical 37 of  $MoS<sub>2</sub>/Fe<sub>3</sub>O<sub>4</sub>/graphene$  can serve as a potential EM wave absorber and can be used in practical 37<br>38 applications. 38

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# 44 1. Introduction

 Currently, electromagnetic (EM) absorption materials have attracted more and more attention due to serious EM pollution, which results from increasing use of radar equipment and wireless 48 system and has severely threatened human health  $[1-3]$ . The high- performance EM wave absorbers must satisfy the need for strong absorption, light weight, thin thickness and wide bandwidth on the condition that high impedance matching, excellent synergistic 52 effect  $[4-6]$ . Traditionally, microwave absorbers are mainly com- prised of metal oxides. Unfortunately, the traditional absorbers face lots of challenges, such as high density and low permittivity, which are barriers to prevent the development of high- performance EM wave absorber. Compared with metal oxides as magnetic materials, dielectric material has been developed to fur- ther improve microwave absorption performance because of its light-weight, higher permittivity [\[7,8\]](#page--1-0).

60 Since graphene was extracted from graphite by an ordinary 61 adhesive tape in 2004, it has developed rapidly in many fields such 62 as capacitor  $[9,10]$ , lithium-ion battery  $[11]$ , photocatalyst  $[12,13]$ , 63 sensor  $[14,15]$ . Importantly, as a thinnest 2-D material, graphene is a candidate of EM wave absorption materials, which is ascribed to 64 abundant defects and functional groups of its surface  $[16,17]$ . How- 65 ever, the high permittivity of graphene has limited the improving 66 of EM wave absorption performances due to the mismatched 67 impedance effect. Therefore, all kinds of dielectric and magnetic 68 materials have been introduced into graphene surface to lower 69 the permittivity, results in enhanced EM wave performances. For 70 instance, Wang et al. reported that PANI nanorod arrays grew on 71 the surface of N-doped graphene and the highest  $R_L$  could be up  $72$ to  $-38.8$  dB for a thickness of 3 mm [\[18\]](#page--1-0). The graphene@polyani-<br>line fim composite was studied by Liu and the maximum  $R_1$  could  $= 74$ line fim composite was studied by Liu and the maximum  $R<sub>L</sub>$  could achieve almost  $-41.4$  dB, the absorption bandwidth below  $-10$  dB  $-75$  was 4.2 GHz [19]. Fu et al. synthesized nanostructured graphene/ $-76$ was 4.2 GHz  $[19]$ . Fu et al. synthesized nanostructured graphene/ NiFe<sub>2</sub>O<sub>4</sub> nanorod composite and found that the composite reached 77<br>-29.2 dB with a coating of 2 mm which was better than that of 78  $-29.2$  dB with a coating of 2 mm, which was better than that of 78 pure graphene [20]. As a representative dielectric material MoS<sub>2</sub> and pure graphene  $[20]$ . As a representative dielectric material, MoS<sub>2</sub> possesses multilayer structure like graphene and indirect energy 80 band of 1.2 eV, which has received increasing attention and been 81 used as an attractive 2-D material analogous to graphene  $[21]$ . Fur- 82 thermore,  $MoS<sub>2</sub>/graphene$  composite with 3-D architecture 83 nanosheets can prevent the adjacent layers of graphene from 84 restacking, which provides favourable properties [\[22\].](#page--1-0) Nowadays, 85 the composite of  $MoS<sub>2</sub>/graphene$  is widely used in the area of 86 energy storage [\[23,24\]](#page--1-0), hydrogen evolution [\[25\],](#page--1-0) microwave 87

## <https://doi.org/10.1016/j.apt.2017.12.016>

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Please cite this article in press as: Y. Wang et al., Fabrication of MoS<sub>2</sub>-graphene modified with Fe<sub>3</sub>O<sub>4</sub> particles and its enhanced microwave absorption performance, Advanced Powder Technology (2017), <https://doi.org/10.1016/j.apt.2017.12.016>

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 absorption [\[26\].](#page--1-0) As far as we know, the dielectric properties and 89 microwave absorption performance of  $MoS<sub>2</sub>/graph$ ene composite modified with magnetic particles have never been reported. The introduction of magnetic particles can improve the equilibrium between permittivity and permeability, which is beneficial to enhanced EM wave absorption performances.

94 In this study, the 3-D architecture of  $MoS<sub>2</sub>/Fe<sub>3</sub>O<sub>4</sub>/graphene$ 95 composite was fabricated and its enhanced EM wave absorption 96 mechanisms were investigated.

# 97 2. Experimental

98 Graphene oxide (GO) was produced according to the previous 99 literature process  $[27]$ . The synthesis of MoS<sub>2</sub>/graphene was car-100 ried out as follows. In detail, 2.19 g sodium molybdate (Na<sub>2</sub>MoO<sub>4</sub>) 101 and 2.07 g thiourea were dissolved in 70 mL GO aqueous solution 102 by stirring to form a homogeneous solution. Then, the mixture 103 was placed to a 100 mL Teflon-lined steel autoclave and kept at 104 200  $\degree$ C for 24h. After cooling down to room temperature naturally, 105 the product ( $MoS<sub>2</sub>/graphene$ ) was washed with ethanol and deion-106 ized water, and dried at  $60^{\circ}$ C overnight.

107 The fabrication of  $MoS<sub>2</sub>/Fe<sub>3</sub>O<sub>4</sub>/graphene composite was pre-$ 108 pared by one-pot co-precipitation process. First,  $0.2 g$  MoS<sub>2</sub>/-109 graphene composite was dispersed in 80 mL distilled water via 110 ultrasound treatment for 20 min. Second,  $0.14$  g FeCl<sub>3</sub> $\cdot$ 6H<sub>2</sub>O and  $111$  0.1032 g FeCl<sub>3</sub> $\cdot$ 4H<sub>2</sub>O were added into the above solution and dis-111  $0.1032$  g FeCl<sub>2</sub>.4H<sub>2</sub>O were added into the above solution and dis-<br>112 solved slowly. Then. 1 M NaOH solution was added dropwise to solved slowly. Then, 1 M NaOH solution was added dropwise to 113 the above suspension until  $pH = 11$ , and the mixture was prepared 114 by stirring at 80 °C for 3 h. Finally, the products  $(MoS<sub>2</sub>/Fe<sub>3</sub>O<sub>4</sub>)-$ 115 graphene) were washed with ethanol and deionized water, and 116 dried at  $60 °C$  for 24 h.

 The crystal phase was tested by X-ray powder diffraction (Ger- man Bruker D8 using Cu-Ka radiation). The FESEM (Quanta 600FEG) and TEM (JTM-2100) were used to observe the morphol- ogy and nanostructure. The XPS spectrum was recorded in an ESCALAB 250 X-ray photoelectron spectrometer to explore the valence state of element. The magnetization hysteresis curves of the samples were tested to obtain coercive force and saturation magnetization by a vibrating sample magnetometer (VSM, Lake 125 Shore7307). The electromagnetic parameters ( $\varepsilon'$ ,  $\varepsilon''$ ,  $\mu'$  and  $\mu''$  val- ues) were recorded with coaxial-line method using a vector net- work analyzer (HP8720ES) from 2 GHz to 18 GHz, with a product to paraffin weight ratio of 3:7 and pressed into a ring shape (OD 7 mm and ID 3.04 mm).

#### 130 3. Results and discussion

131 The crystal structures of  $MoS<sub>2</sub>/graphene$ ,  $MoS<sub>2</sub>/Fe<sub>3</sub>O<sub>4</sub>/graphene$ 132 composite were investigated by XRD measurements, as shown in 133 Fig. 1. As for  $MoS<sub>2</sub>/graph$ ene shown in Fig. 1a, the diffraction peaks 134 of 14.2°, 33.5°, 39.6° and 58.9° can be corresponded to  $(0\ 0\ 2)$ ,  $(1\ 0\ 0\ 0\ 0)$ 135 0), (1 0 3) and (1 1 0) planes, suggesting the hexagonal phase  $MoS<sub>2</sub>$ 136 (JCPDS No. 37-1492) [\[21,22\]](#page--1-0). For  $MoS<sub>2</sub>/Fe<sub>3</sub>O<sub>4</sub>/graphene composite$ 137 in Fig. 1b, it can be seen that except for the existence of  $MoS<sub>2</sub>$ 138 peaks, the diffraction peaks of  $Fe_3O_4$  at 30.3°, 35.5°, 43.4°, 57.2°<br>139 and 62.9° are assigned to the (2 2 0), (3 1 1), (4 0 0), (5 1 1) and and 62.9 $^{\circ}$  are assigned to the (2 2 0), (3 1 1), (4 0 0), (5 1 1) and 140  $(440)$  planes, match well with Fe<sub>3</sub>O<sub>4</sub> (JCPDS No. 19-0629) [\[28\].](#page--1-0) 141 In particular, the peaks of graphene can be not observed in the 142 two samples, which may be ascribed to the decreasing stacking 143 of graphene sheets by the introduction of  $MoS<sub>2</sub>$  and  $Fe<sub>3</sub>O<sub>4</sub>$  [\[29\].](#page--1-0) 144 Furthermore, the impurity peaks are not detected, indicating the 145 high purity of all the samples.

146 [Fig. 2](#page--1-0) shows XPS spectra of  $MoS<sub>2</sub>/Fe<sub>3</sub>O<sub>4</sub>/graphene$  composite. 147 From [Fig. 2](#page--1-0)a, the wide scan XPS spectrum confirms the existence 148 of S 2p, Mo 3d, C 1s, O 1s and Fe 2p, according with the experimen-



Fig. 1. XRD patterns of  $MoS<sub>2</sub>/graphene$  (a) and  $MoS<sub>2</sub>/Fe<sub>3</sub>O<sub>4</sub>/graphene$  (b).

tal theoretical value. As demonstrated in  $Fig. 2b$ , the C1s spectrum  $149$ can be separated into two peaks, originating form  $C-C/C = C$  150  $(284.5 \text{ eV})$  in the aromatic rings,  $C=O$   $(286.1 \text{ eV})$  in oxygen- 151 containing bound  $[30]$ . The extremely weak peak of C- $\overline{O}$  suggests 152 that the GO has been successfully reduced into graphene. The 153 peaks at 711.4 and 725.3 eV [\(Fig. 2](#page--1-0)c) are attributed to Fe  $2p3/2$  154 and Fe 2p1/2, which illustrates the existence of  $Fe^{2+}$  and  $Fe^{3+}$ , prov- 155 ing the formation of  $Fe<sub>3</sub>O<sub>4</sub>$ . The O 1s peaks [\(Fig. 2](#page--1-0)e) at 530.1 and 156 531.4 eV correspond to the metal-oxygen bonds [\[31\].](#page--1-0) From the 157 Mo 3d spectrum in [Fig. 2](#page--1-0)d, the Mo 3d band can be divided into 158 two peaks located at 229.1 and 232.3 eV, which corresponds to 159 Mo 3d5/2 and Mo 3d3/2 [\[21,22\],](#page--1-0) revealing the formation of  $Mo^{4+}$ . 160 The peaks at 161.8 and 162.9 eV are ascribed to the S  $2p3/2$  and 161 S 2p1/2 of divalent sulfide ions  $(S^{2-})$ , respectively, as shown in 162 [Fig. 2f](#page--1-0) [\[32\]](#page--1-0). The characteristic spectra in [Fig. 2d](#page--1-0) and f prove the 163 existence of  $MoS<sub>2</sub>$ . 164

FESEM and TEM were employed to observe the morphology and 165 microstructure of the as-prepared samples. From Fig.  $3(a, b)$ , it is 166 obvious that  $MoS<sub>2</sub>/graphene$  composite displays a homogeneous 167 flower-like structure, which is composed of intertwined sheet- 168 like structure. The flower-like structure can enlarge the specific 169 surface area and provide more absorption sites, which is beneficial 170 to the microwave absorption performances. As can be observed in 171 [Fig. 3](#page--1-0)(c, d), during the co-precipitation route,  $Fe<sub>3</sub>O<sub>4</sub>$  nanoparticles 172 with a size of around 20–30 nm were grown in the cavity of  $MoS<sub>2</sub>/-173$ graphene. The adding of  $Fe<sub>3</sub>O<sub>4</sub>$  particles does not destroy the 3D 174 flower-like structure. TEM was used to further obtain microstruc- 175 ture information of the samples. From Fig.  $4(a, b)$ , MoS<sub>2</sub>/graphene 176 composite displays a hydrangea-like structure and the particle size 177 agrees with FESEM analysis. It is found from [Fig. 4](#page--1-0)c that the edge of 178  $MoS<sub>2</sub>/graphene$  is wrinkled and transparent. As depicted in [Fig. 4](#page--1-0) 179 (d–f), we can observe clearly that the transparent edge of  $MoS<sub>2</sub>/-$  180 graphene is covered with  $Fe<sub>3</sub>O<sub>4</sub>$  particles with a uniform distribu- 181 tion. Based on the above analysis, it can be inferred that the 182  $MoS<sub>2</sub>/Fe<sub>3</sub>O<sub>4</sub>/graphene composite were successfully fabricated.$  183

The magnetic properties are crucial to EM wave absorption per- 184 formances and the magnetic parameters of  $Fe<sub>3</sub>O<sub>4</sub>$ , MoS<sub>2</sub>/Fe<sub>3</sub>O<sub>4</sub>/- 185 graphene were tested at room temperature by VSM. As shown in 186 [Fig. 5](#page--1-0), both the two samples display a typical superparamagnetic 187 behavior, which possess negligible remanence and coercivity. The 188 saturation magnetization  $(M_S)$  values of Fe<sub>3</sub>O<sub>4</sub>, MoS<sub>2</sub>/Fe<sub>3</sub>O<sub>4</sub>/- 189 graphene are 84.1 emu/g and 33.4 emu/g, respectively. The great  $190$ difference in  $M<sub>S</sub>$  values of the two samples may be ascribed to 191 the presence of graphene and  $MoS<sub>2</sub>$ . 192

The EM wave absorption performances of the  $MoS<sub>2</sub>/graphene$ , 193  $MoS<sub>2</sub>/Fe<sub>3</sub>O<sub>4</sub>/graphene$  were investigated at 2–18 GHz and the 194

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