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# Photo-effect on the electromotive force in two-compartment hydrogen peroxide-photofuel cell



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ARTICLE INFO	A B S T R A C T
Keywords:	Mesoporous $TiO_2$ nanocrystalline film was coated on fluorine-doped tin oxide electrode (mp-TiO <sub>2</sub> /FTO) by the
Hydrogen peroxide	doctor blade method, and Prussian blue film was electrodeposited on FTO (PB/FTO). A two-compartment H <sub>2</sub> O <sub>2</sub> -
itanium dioxide	PEC with the structure of mp-TiQ_/ETO (photoanode) $[0.1 \text{ M KC}]_{0}$ (pH 3)[Nafion (separator)] $[0.1 \text{ M KC}]_{0}$ , with
Prussian blue	0.1 M H.O. (pH 3)DB/ETO (cathoda) was constructed. Under UV light irradiation, the two compartment DEC
Photofuel cell	provides short-circuit current $(J_{sc}) = 1.14 \text{ mA cm}^{-2}$ and open-circuit potential $(V_{oc}) = 952 \text{ mV}$ much greater
Electromotive force	
Photocatalyst	than the values of $J_{sc} = 0.197$ mA cm <sup>-2</sup> and $V_{oc} = 607$ mV for the one-compartment cell. Importantly, the
	electromotive force for the two-compartment H <sub>2</sub> O <sub>2</sub> -PFC exceeds the thermodynamic value of 534 mV calculated
	from the electrode potentials for the half-reactions.

#### 1. Introduction

Photofuel cell (PFC) using TiO<sub>2</sub> as the photoanode has recently emerged as a new type of fuel cells [1–6]. Besides the usability of biomass and bio-related compounds as the fuel and the excellent stability, the PFC can be expected to provide high electromotive force (EMF) due to the input light energy. In practice, the EMF reported for the PFCs (EMF<sub>pfc</sub>) using various fuels is significantly smaller than the thermodynamic value (EMF<sub>th</sub>), e.g., the EMF<sub>pfc</sub> for the ethanol-PFC is 43% of the EMF<sub>th</sub> [2]. More recently, we have developed a one-compartment H<sub>2</sub>O<sub>2</sub>-PFC, but the EMF<sub>pfc</sub> still remains ~70% of the EMF<sub>th</sub> [7]. There are two possible reasons for the low EMF<sub>PFC</sub>. One is the loss of the excited electrons in the CB of TiO<sub>2</sub> (e<sup>-</sup><sub>CB</sub>) due to the trapping by the H<sub>2</sub>O<sub>2</sub>-surface complex [8] during the transport to the electron-collecting electrode (Eq. 1) [9], while the surface complex gives visiblelight responsibility to the cell [7].

$$H_2O_2\cdots TiO_2 + 2H^+ + 2e^-_{CB} \rightarrow 2H_2O + TiO_2$$
 (1)

The other is the large overpotential for  $H_2O_2$  reduction at the cathode [10]. Previously, Itaya et al. showed that Prussian blue film deposited on fluorine-doped tin oxide (PB/FTO) possesses electrocatalytic activity for the reduction of  $O_2$  and  $H_2O_2$  [11]. Further, The research group of Shaegh and Nguyen has recently reported a  $H_2O_2$ -FC using PB as the cathode material [12].

To solve these two problems simultaneously, we constructed a twocompartment  $H_2O_2$ -PFC with the structure of mesoporous TiO<sub>2</sub> nanocrystalline film coated on fluorine-doped tin oxide electrode (mp-TiO<sub>2</sub>/

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FTO, photoanode) $|0.1 \text{ M KClO}_4|$ Nafion $|0.1 \text{ M KClO}_4$  with  $0.1 \text{ M H}_2\text{O}_2$  (pH 3)|PB film-deposited FTO electrode (PB/FTO, cathode). Here we show for the first time that the EMF<sub>pfc</sub> far exceeds the EMF<sub>th</sub>.

#### 2. Experimental methods

A paste containing anatase TiO2 particles with mean size of 20 nm (PST-18NR, Nikki Syokubai Kasei) was coated on FTO electrode (sheet resistance =  $10 \Omega$ /square) by the doctor blade technique. The as-coated film was heated at 773 K for 1 h to form mp-TiO<sub>2</sub>/FTO. The specific surface area and pore volume of  $\rm mp\mathchar`-TiO_2$  were determined by nitrogen adsorption-desorption isotherms at 77 K with a micromeritics automatic surface area and porosimetry analyzer (TriStar 3000, Shimadzu). PB films were electrodeposited on FTO (PB/FTO) from an aqueous solution of ferric-ferricyanide [13]. A three-electrode photoelectrochemical cell with the structure of FTO working electrode a mixed aqueous solution of 0.02 M FeCl<sub>3</sub> and 0.02 M K<sub>3</sub>[Fe(CN)<sub>6</sub>] (200 mL)|Ag/AgCl reference electrode (TOA-DKK)|GC counter electrode was fabricated. PB films were electrodeposited on FTO under a constant current of  $-40 \,\mu A \,\mathrm{cm}^{-2}$ . The film thickness ( $l_{PB}$ ) was controlled by the electrodeposition time  $(t_{ed})$ . Scanning electron microscopic (SEM) observation was carried out by Hitachi S-4800 Type II at an applied voltage of 20 kV.

A two-compartment H<sub>2</sub>O<sub>2</sub>-PFC with the structure of mp-TiO<sub>2</sub>/FTO (apparent area =  $1.0 \times 1.0 \text{ cm}^2$ )|0.1 M KClO<sub>4</sub>|separator|0.1 M KClO<sub>4</sub> and 0.1 M H<sub>2</sub>O<sub>2</sub>|PB/FTO was constructed using Nafion (Aldrich) as the separator. PB is stable in acidic solution, while it easily dissolves into

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Fig. 1. (A) Crystal structure of PB. (B) Plots of the thickness of the PB film thickness versus electrodeposition time. The inset shows a SEM image for the cross-section of PB/FTO.

alkaline solution. Then, the pH of the catholyte was regulated at 3. For comparison, one-compartment H<sub>2</sub>O<sub>2</sub>-PFC with the structure of mp-TiO<sub>2</sub>/FTO (apparent area =  $1.0 \times 1.0 \text{ cm}^2$ )|0.1 M NaClO<sub>4</sub> and 0.1 M H<sub>2</sub>O<sub>2</sub> (pH 3)|PB/FTO was fabricated. The current (*J*)-cell voltage (*V*<sub>cell</sub>) curves for the cells were measured using a potentio/galvanostat (HZ-7000, Hokuto Denko) under irradiation of UV-light ( $\lambda$  = 365 ± 25 nm, *I*<sub>365</sub> = 31.2 mW cm<sup>-2</sup>) by a UV-LED (HLV-24UV365-4WNRBTNJ, CCS) at 298 K.

#### 3. Results and discussion

SEM observation confirmed that mp-TiO<sub>2</sub> film with thickness of 4.0 µm is formed on FTO. The specific surface area and pore volume of mp-TiO<sub>2</sub> were determined to be  $17.28 \text{ m}^2 \text{ g}^{-1}$  and  $0.72 \text{ cm}^3 \text{ g}^{-1}$ , respectively. Pore size distribution analysis by the Barret-Joyner-Halenda (BJH) method afforded a peak pore size of ~20 nm [14]. As shown in Fig. 1A, PB possesses a cubic unit cell dimension of 10.2 Å to form unique nanopores. The electrodeposition of PB on FTO induces a diffraction peak at  $2\theta = 17.58^{\circ}$  assignable to the diffraction from its (200) plane (ICDD 00-052-1907). SEM image for a PB/FTO sample in the inset in Fig. 1B exhibits that a fairly uniform PB film is deposited on FTO. The thickness of PB film ( $l_{PB}$ ) increases in proportion to  $t_{ed}$  according to the following equation of  $l_{PB}(\mu m) = 0.0895t_{ed}/min$ .

Two-compartment H2O2-PFC with the structure of mp-TiO2/ FTO 0.1 M KClO<sub>4</sub> (pH 3) Nafion 0.1 M KClO<sub>4</sub> with 0.1 M H<sub>2</sub>O<sub>2</sub> (pH 3)|PB/FTO ( $l_{PB} = 0.90 \,\mu$ m), and the cell performance was compared with that for one-compartment cell with the structure of mp- $TiO_2/FTO|0.1 M KClO_4$  with  $0.1 M H_2O_2$ (pH 3)|PB/FTO  $(l_{PB} = 0.90 \,\mu\text{m})$ . In each cell, no current flowed in the dark. Fig. 2A shows photocurrent (J)-cell voltage ( $V_{cell}$ ) curves for the H<sub>2</sub>O<sub>2</sub>-PFCs were measured under UV-light irradiation at 298 K. The one-compartment cell provides a short-circuit current  $(J_{sc})$  of 0.197 mA cm<sup>-2</sup> and an open-circuit voltage ( $V_{oc}$ ) of 607 mV. Strikingly, in the two-compartment cell, the  $J_{sc}$  and  $V_{oc}$  values increase to 1.14 mA cm<sup>-2</sup> and 952 mV, respectively. Fig. 2B shows power density as a function of J for the cells. Both the curves exhibit volcano-shaped ones with a peak of 0.0493 mW cm<sup>-2</sup> at J = 0.140 mA cm<sup>-2</sup> for the one-compartment cell and a peak of 0.247 mW cm<sup>-2</sup> at J = 0.635 mA cm<sup>-2</sup> for the twocompartment cell. Fig. 2C shows plots of  $J_{sc}$  and  $V_{oc}$  for the two-compartment  $H_2O_2$ -PFC as a function of  $l_{PB}$ . The  $J_{sc}$  increases with increasing  $l_{\rm PB}$  to reach a constant value of ~1.1 mA cm<sup>-2</sup> at  $l_{\rm PB} > 0.1 \,\mu{\rm m}$ , while the  $V_{\rm oc}$  is almost independent of  $l_{\rm PB}$ . Clearly, the division of the H<sub>2</sub>O<sub>2</sub>-PFC cell in two compartment dramatically enhances the cell performances, and PB contributes to the increase in  $J_{sc}$ . Fig. 2D shows the effect of light intensity at  $\lambda = 365$  nm ( $I_{365}$ ) on the  $J_{sc}$ 

and  $V_{\rm oc}$  for the two-compartment H<sub>2</sub>O<sub>2</sub>-PFC with  $l_{\rm PB} = 0.90 \,\mu\text{m}$ . The  $V_{\rm oc}$  gradually increases with an increase in  $I_{365}$  to approach a saturated value of 940  $\pm$  12 mV. On the other hand, the  $J_{\rm sc}$  increases almost in proportion to  $I_{365}$  at  $I_{365} < 31.2 \,\text{mW cm}^{-2}$ .

The electrons in the valence band (VB) of TiO<sub>2</sub> are excited to the CB by UV-light irradiation. At the mp-TiO<sub>2</sub>/FTO photoanode, the VB-holes with strong oxidation ability oxidize H<sub>2</sub>O<sub>2</sub> in the two-compartment PFC, while they preferentially oxidize H<sub>2</sub>O<sub>2</sub> in the one-compartment PFC [7]. On the other hand, the excited electrons are accumulated in the CB of TiO<sub>2</sub> under open-curcuit conditions, which causes the negative shift in the electrode potential ( $E_{\rm ph}$ ) with respect to the equilibrium value in the dark ( $E_{\rm dark}$ ). Thus, the potential change becomes negative ( $V_{\rm oc} = E_{\rm ph} - E_{\rm dark}$ ), and on stopping the irradiation, the  $V_{\rm oc}$  gardually approaches to zero. The mean lifetime of the excited electrons in the CB of TiO<sub>2</sub> ( $\tau_{\rm n}$ ) can be estimated by analyzing the decay curve by Eq. 2 [15].

$$\tau_{\rm n} = -(k_{\rm B}T/q) \, ({\rm d}V_{\rm oc}/{\rm d}t)^{-1} \tag{2}$$

where  $k_{\rm B}$  is the Boltzmann constant, and q is the elementary charge of electron.

Fig. 3A compares the log  $\tau_n$ - $V_{oc}$  curves for the one-compartment and two-compartment PFCs. At the same  $V_{oc}$ , the  $\tau_n$  for the two-compartment cell is much longer than that for the one-compartment cell. Evidently, the division of the H<sub>2</sub>O<sub>2</sub>-PFC in two compartments leads to a significant extension of the lifetime of the electrons in the CB of TiO2 or the enhancement of the electron transport to the FTO electrode without trapping by the surface complexes. Fig. 3B shows cyclic voltammetry (CV) curves for PB/FTO in 0.1 M KClO<sub>4</sub> without and with 0.1 M H<sub>2</sub>O<sub>2</sub>. In the absence of H<sub>2</sub>O<sub>2</sub>, a pair of current peaks corresponding to the redox reaction between PB and Evaritt's salt (ES, Fe<sup>II</sup><sub>4</sub>[Fe<sup>II</sup>(CN)<sub>6</sub>]<sub>3</sub>) is observed at the formal potential E' = +0.38 V (Eq. 3). The cathodic current sharply weakens at E < +0.2 V as a result of the decrease in the number of  $Fe^{III}$  ions in the PB film. In the presence of  $H_2O_2$ , a large current continues to flow at E < +0.2 V, and the anodic current peak disappears. Thus, the current at E < -0.1 V mainly results from the PB-mediated reduction of H2O2. PB/FTO exhibits electrochromism in aqueous electrolyte solution of KClO<sub>4</sub> [16]. As a result of the cyclic potential scan between +0.8 V and -0.2 V, the color reversibly changes between blue and colorless in the absence of H<sub>2</sub>O<sub>2</sub>, but it remains blue in the presence of H<sub>2</sub>O<sub>2</sub>. These results as well as the data in Fig. 2C strongly suggest that the electrons accumulated in PB are smoothly consumed by the H2O2 reduction, i.e., PB acts as an electrocatalyst for the  $H_2O_2$  reduction (Eq. 4) [11]. In this case, the reduction of H<sub>2</sub>O<sub>2</sub> is suggested to occur at not only the external surface but also the internal surface in the PB nanopores.

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