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Controlling the electrochemical properties of an anode prepared from pitch-based soft carbon for Li-ion batteries

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Introduction

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Carbon-based materials are used as negative active materials, most of which are based on graphite and disordered carbon. Disordered carbon (referred to as soft carbon) has been evaluated for its long-term charging and discharging rate characteristics to meet the requirements of electric vehicles (EVs) [1]. However, the reversible capacity of soft carbon is 250 mAh g⁻¹, which is lower than that of graphite (372 mAh g⁻¹). Several criteria exist for the anode material [2,3]. Specifically, low-temperature processing and low-cost precursors are mandatory. When the carbon precursor is carbonized at high temperatures (above 1000 °C), the initial efficiency and rate capability are higher than those of disordered carbon (below 1000 °C). However, carbonaceous materials manufactured at high temperatures consume more energy and have

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ABSTRACT

A petroleum residue-based carbon electrode for use in a lithium-ion battery was prepared through the thermal reaction of pyrolyzed fuel oil. The samples were characterized by their softening point, crystallinity and battery performance. The softening point and pitch crystallinity increased with higher reaction temperatures and longer reaction times. The prepared pitch-based carbon had a 300–350 mAh/g capacity and 71–78% initial efficiency. Notably, the high-softening-point pitch resulted in the developed crystallinity through polymerization and condensation. In conclusion, the high-softening-point pitch had a better initial efficiency and rate capability based on its high crystallinity.

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higher associated costs. Thus, investigating the low-temperature27preparation of carbonaceous materials with outstanding electro-28chemical properties is needed [4–8].29

The petroleum residue derived from the naphtha reforming 30 process is referred to as pyrolyzed fuel oil (PFO) and has an 31 abundant polyaromatic content that adds value to the product 32 [9]. Its low price and abundant sources (1 million tons per year in 33 the Republic of Korea) are the main advantages of PFO. Therefore, 34 PFO has the potential to become a high value product if it is 35 converted into other valuable products. In fact, the price of anode 36 material manufactured using petroleum residue is evaluated at \$8/ 37 kg, and \$12/kg using commercial soft carbon. Pitch is a valuable 38 PFO product that can be prepared by the thermal condensation, 39 polymerization, aromatization and polymerization of aromatic 40 compounds [10,11]. Pitch can be further carbonized to soft carbon 41 (PFO \rightarrow pitch \rightarrow soft carbon). 42

Chung et al. [12] used two types of pitch with different carbon crystallite lattice parameters (L_c and L_a). These researchers compared the initial irreversible capacity of the electrode active material after carbonization. The prepared coke, which was carbonized from the pitch with different softening points, exhibited different crystallite lattice parameters. Kinoshita et al. [13] reported that the irreversible capacity of carbon sources could 43

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be explained using the crystallite lattice parameter equation. Therefore, the lattice parameters (L_a and L_c) can be used to control the initial irreversible capacity of the soft carbon. These carbonaceous materials of the anode can be controlled based on the pitch softening.

The irreversible capacity is believed to be important for full cell performance because of the influence of the reduction of usable lithium on the overall battery capacity. The primary reasons for the decreased electrochemical characteristics involve electrolyte decomposition and the hysteresis phenomenon caused by the nanopores in the prepared carbon [14–20]. Because of the thermodynamically unstable state, a solid electrolyte interphase (SEI) layer forms on the surface of the carbon material via electrochemical reaction accompanied by electrolyte decomposition in the low potential region (approximately 0 V).

Graphite is primarily used as a carbonaceous anode material. However, graphite has disadvantageously poor rate properties. To overcome this disadvantage, graphite is used in anode materials as pitch-coated graphite [21].

69 In this study, we prepared a pitch-based soft carbon using PFO. 70 The pitch was manufactured using different process temperatures 71 and holding times to study the relationship between various pitch 72 characteristics and the electrochemical properties of the pitch-73 based carbon. For the first time, the electrochemical characteristics 74 of pitch-based carbon with a softening point of 100-150 °C were 75 studied. The softening point of the petroleum-based pitch may be a 76 variable that can be used to control the electrochemical properties 77 of pitch-based soft carbon.

78 Experimental details

79 Materials

80 PFO-based pitch preparation

81 PFO derived from the naphtha cracking process was obtained 82 from YNCC (Yeochun NCC CO. LTD), Republic of Korea, and used as 83 the precursor to the anode material. This PFO had 78.6% aromatic 84 and 21.6% resin contents, as determined by SARA (saturates, 85 aromatics, resins, and asphaltenes) analysis. The pitch was prepared 86 via the thermal reaction of PFO in a 5 L autoclave reactor. The PFO 87 was heated at 350, 360 and 370 °C for 1, 2 and 3 h, respectively. 88 Heating was performed in an N₂ atmosphere, and the heating rate 89 was set to 2.5 °C/min. The details of the conditions used are tabulated in Table 2. The pitch prepared at 350 °C for 1 h was named 90 91 '3501', and the other samples were named using the same protocol.

Carbonization

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93 Carbonization was performed at 900 °C for 1 h under an N_2 94 atmosphere in a tubular furnace, and the heating rate was 3 °C/ 95 min. Each sample was ground using stainless balls and then passed 96 through a sieve with 25 μ m openings to produce a powder for 97 analysis of its electrochemical properties as an anode material for 98 Li-ion batteries.

99 Characterization

100 Elemental analysis

101 Elemental analysis was performed to confirm the H/C ratio of 102 the carbonized samples using a Thermo Scientific FLASH EA-2000 103 organic elemental analyzer. The results are tabulated in Table 1.

104 X-ray diffraction (XRD) analysis

105The crystalline structure of the samples was determined by XRD106analysis using a Rigaku Ultima IV diffractometer with Cu K α 107radiation. The d-spacing was estimated using the Bragg equation.108To obtain more information of the crystalline structure, L_a and L_c

Table 1		
Elemental	analysis	results.

Sample	С	Н	0	Ν	H/C
3501	96.4	0.7	2.9	-	0.09
3502	95.7	0.9	3.4	-	0.11
3503	95.8	0.7	3.5	-	0.09
3601	96.9	0.7	2.4	-	0.09
3602	97.3	0.6	2.1	-	0.07
3603	97	0.7	2.3	-	0.09
3701	97	0.7	2.3	-	0.09
3702	97.1	0.6	2.3	-	0.07
3703	97.4	0.6	2	-	0.07

(the lattice parameters of the lateral size and stacking height, respectively) were calculated using the following equations [16,17]: 110

$$L_a = \frac{1.84\lambda}{B_a \cos\theta} \tag{1}$$

$$L_c = \frac{0.89\lambda}{B_c \cos \theta} \tag{2}$$

where λ is the wavelength, with a value of 1.54056 Å, and B_a and B_c 114 are the half-widths of the (1 0 0) and (0 0 2) signals, respectively. 115

Softening point measurements

The softening point of the prepared pitches was measured using117a DP-70 (Mettler Toledo) instrument according to ASTM D3416.118The prepared pitch samples were placed on an aluminum plate and119heated to 300 °C at a heating rate of 3 °C/min under an N2 gas flow120of 10 L/min.121

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Electrochemical measurements

The electrochemical properties of each carbon sample were 123 estimated using 2032 coin cells. The mixing ratio of carbon and a 124 polyvinylidene fluoride (PVdF) binder solution (NMP) was 96:4. 125 The conductive materials were excluded to evaluate the perfor-126 mance of the as-prepared sample by itself. After mixing with a 127 Thinky Mixer (MDbros Co.), the resulting slurry was coated onto Cu 128 foils that were used as anode current collectors. The coated 129 electrode was dried at 120 °C in a vacuum oven overnight to 130 remove volatiles. Then, the dried electrode was pressed to a 131 thickness of 70 µm and porosity of 34-5%. Next, a disk with a 132 diameter of 14 mm was punched out from the electrode material. 133 The half cells were assembled in an argon-filled glove box. Li foil 134 was used as the counter electrode. Additionally, 1 M LiPF₆ was 135 dissolved in a mixture of ethylene carbonate (EC), dimethyl 136 carbonate (DMC) (1:1 in volume), and 3 wt% vinylene carbonate 137 (VC) and used as the electrolyte. For the electrochemical test, the 138 aging time of the assembled cells was set to 12 h. The cells were 139 charged at 0.1 C mA from 1.5 V to 0.01 V vs. Li/Li⁺, followed by a 140 constant-voltage charge at 0.01 V. The cut-off current was 141 0.001 C mA. The discharge rate was 0.1C, 0.2C, 0.5C, 1C, and 2C 142 for 5 cycles per current rate, for which we assumed that the specific 143 capacity of each carbon sample was 250 mAh g^{-1} for the 1C rate. 144

1500 °C, but the specific capacity decreased from 300 mAh g^{-1} to

Results and discussion

Structural and elemental analysis

The lattice parameters L_a and L_c of the carbonaceous material,147which were calculated from the XRD measurements, were also148reported by other researchers. Matumura et al. [18] reported that149 L_c of a disordered carbon material is proportional to the heat150treatment temperature and inversely proportional to the half-cell151test capacity. In their report, L_c rapidly increased from 1000 °C to152

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