



## Thermal ablation of metal films by femtosecond laser bursts

Yunpeng Ren<sup>a</sup>, C.W. Cheng<sup>b</sup>, J.K. Chen<sup>a,\*</sup>, Yuwen Zhang<sup>a</sup>, D.Y. Tzou<sup>a</sup>

<sup>a</sup> Department of Mechanical and Aerospace Engineering, University of Missouri, Columbia, MO 65211, USA

<sup>b</sup> ITRI South, Industrial Technology Research Institute, Liujia District, Tainan City 734, Taiwan

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### ABSTRACT

Thermal ablations of metal films by a single femtosecond laser pulse and by a laser burst were investigated using a comprehensive computational model, including a two-temperature model with dynamic optical properties, two phase change models for melting and evaporation under superheating, and a phase explosion criterion for ejection of the mixture of metastable liquid droplets and vapor. Numerical simulations were performed for copper films. The simulation of material ablation by the single pulse showed a good agreement with existing experimental data for a broad range of laser fluence 0.6–30 J/cm<sup>2</sup>. In the study of material ablation by femtosecond laser bursts, it was found that under the same total fluence a laser burst with a pulse separation time of 50 ps or longer can significantly boost the ablation rate, compared to the single pulse. It was also demonstrated that for a given total fluence the ablation rate can be optimized by selecting a proper combination of pulse number and separation time in a laser burst.

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### 1. Introduction

Due to the unique characteristics of material processing, interactions of ultrashort-pulsed laser with metals have received a great deal of attention in the past two decades [1–3]. It has been demonstrated that femtosecond (fs) laser material processing is a very effective means for precise micro/nano machining and surface structure modification of solid materials due to minimal heat affected zone and collateral damage, less debris contamination, and good reproducibility [4–6]. To date, it has become possible to generate any pulse shapes with recently developed optical devices. For example, a pulse train can consist of many bursts with a repetition rate from kHz to MHz, in which each burst consists of several pulses at a separation time of several tens of femtosecond to hundreds of picosecond (ps). The recent advancement of laser burst technology may provide another powerful means for ultrashort-pulsed laser applications.

In theoretical studies, two-temperature models (TTMs) [7–9] have been employed to simulate thermal problems for metals irradiated by ultrashort-pulsed lasers. For the fs-laser pulse train heating, Huang et al. [10] studied energy transport and melting process in a gold film. Temperature-dependent thermo-physical properties were employed; but, constant optical properties were

assumed. No material ablation was studied. Jiang and Tsai [11] and later Sim et al. [12] employed the Drude model to calculate the dynamic changes of optical properties for predicting laser induced thermal response in metal films. Nonetheless, the calculated optical properties may not be accurate because the Drude model generally describes a higher reflectivity for metals. Chowdhury et al. [13] and Du et al. [14] respectively used constant optical properties and thermophysical properties which are valid for relatively low temperature to simulate thermal transport in gold films induced by multiple fs-laser pulses. None of the above works [11–14] investigated ultrafast melting/re-solidification and material ablation.

Besides the models themselves, thermophysical and optical properties of materials are a key factor that governs the solution accuracy in modeling ultrafast laser material interactions [15–17]. The former controls thermal transport and temperature distribution in a heated medium, while the latter dictates laser energy deposition which can affect the resulting thermal response. Of particular importance, the dynamic change in surface reflectivity ( $R$ ) and absorption coefficient ( $\alpha$ ) inside a treated material due to elevated temperature could significantly alter the magnitude and distribution of laser energy deposition, leading to a different thermal response than that described with constant optical properties at room temperature. This may be the critical question which often does not allow directly comparing experimental and modeling data. To match experimental results, particular values for a constant surface reflectivity [18,19] or an absorbed laser fluence [20] together with a constant absorption coefficient at room

\* Corresponding author.

E-mail address: [ChenJnK@missouri.edu](mailto:ChenJnK@missouri.edu) (J.K. Chen).

**Nomenclature**

$A_e$	material constants for electron relaxation time ( $K^{-2} s^{-1}$ )
$B_p$	weighing factor of Lorentz oscillators
$C$	heat capacity ( $J/m^3 K$ )
$c$	speed of light in vacuum (m/s)
$f_1$	normal refractive index
$f_2$	extinction coefficient
$f_{rep}$	repetition rate of pulse bursts (Hz)
$G$	electron–lattice coupling factor ( $W/m^3 K$ )
$\hbar$	reduced Plank constant (J s)
$h_{lv}$	latent heat of evaporation (J/kg)
$h_m$	latent heat of fusion (J/kg)
$J$	laser fluence ( $J/m^2$ )
$k$	thermal conductivity (W/m K)
$k_B$	Boltzmann constant ( $m^2 kg/s^2 K$ )
$L$	film thickness (m)
$m$	electron mass (kg)
$m_{opt}$	effective electron mass (kg)
$q$	phonon wave vector ( $m^{-1}$ )
$R$	reflectivity
$R_g$	specific gas constant (J/kg K)
$s$	longitudinal sound speed (m/s)
$S$	laser heat density ( $W/m^3$ )
$t$	time (s)
$t_p$	FWHM pulse width (s)
$t_{sep}$	separation time of pulses (s)
$T$	temperature (K)
$T_m$	normal melting point (K)

$u$	interfacial velocity (m/s)
$u_0$	maximum solid–liquid interfacial velocity (m/s)
$v_F$	Fermi velocity (m/s)
$x$	coordinate (m)

**Greek symbols**

$\alpha$	absorption coefficient ( $m^{-1}$ )
$\gamma$	total electron collision rate (1/s)
$\Gamma_p$	Lorentz oscillator spectral width (rad/s)
$\bar{\epsilon}$	emissivity factor
$\epsilon$	dielectric function
$\epsilon_1$	real part of dielectric function
$\epsilon_2$	imaginary part of dielectric function
$\epsilon_F$	Fermi energy (eV)
$v_{e,p}$	collision rate between electron and phonon ( $s^{-1}$ )
$\rho$	mass density ( $kg/m^3$ )
$\sigma$	Stefan–Boltzmann constant ( $W/m^2 K^4$ )
$\tau_e$	electron relaxation time (s)
$\phi_p$	Lorentz oscillator phase
$\omega$	laser frequency (Hz)
$\omega_D$	plasma frequency (Hz)
$\Omega_p$	Lorentz oscillator strength (rad/s)

**Subscripts**

e	electron
l	lattice
l	liquid
lv	liquid–vapor interface
s	solid
sl	solid–liquid interface

temperature are frequently assumed in modeling ultrafast laser material ablation. Strictly speaking, they can only be considered as a trial-and-error approach. The dynamic changes in optical properties indeed could play an important role in the laser energy absorption, particularly for metal films heated by fs-laser bursts due to the incubation effect from the repeated laser pulse heating.

In this paper, thermal ablations of metal films by a single fs-laser pulse and by a laser burst are investigated using a comprehensive computational model, including a TTM with dynamic optical properties, two phase change models for melting and evaporation under superheating, and a phase explosion criterion for ejection of the mixture of metastable liquid droplets and vapor. A critical point model with three Lorentzian terms demonstrated by Ren et al. [17] is incorporated into the TTM to characterize temperature-dependent optical properties. All the thermophysical properties employed are valid for a wide range of temperature. Numerical analysis is performed for copper films. The material ablation is first studied for a single fs-laser pulse, and the simulated ablation depths are compared with existing experimental data. Then, the burst modes are investigated for their performance of material ablation. The numerical results of thermal response and ablation depth generated by the fs-laser bursts having different pulse number and separation time are presented and discussed.

**2. Mathematical models**

Consider that a free standing metal film of thickness  $L$  and initial temperature  $T_i$  is irradiated at its front surface ( $x = 0$ ) by a single laser pulse or by a pulse train. Fig. 1 depicts the structure of a laser pulse train consisting of two bursts with a repetition rate  $f_{rep}$ , in which each burst consists of three pulses with a separation time  $t_{sep}$ . All the laser pulses are Gaussian in time with a full width at half

maximum (FWHM)  $t_p$ . For simplicity, the problem is approximated to be one dimensional since laser spot sizes are much larger than the thermally affected depth.

**2.1. Two-temperature model**

In the TTM for pure metals, heat conduction in lattice is often neglected due to the fact that the thermal conductivity of lattice is much smaller than that of electrons. However, heat conduction in the lattice could also be of importance for precisely describing lattice thermal response, particularly when phase change(s) are considered. In that case, a TTM that includes heat conduction in the lattice [9] should be employed:

$$C_e \frac{\partial T_e}{\partial t} = \frac{\partial}{\partial x} \left( k_e \frac{\partial T_e}{\partial x} \right) - G(T_e - T_l) + S \quad (1)$$

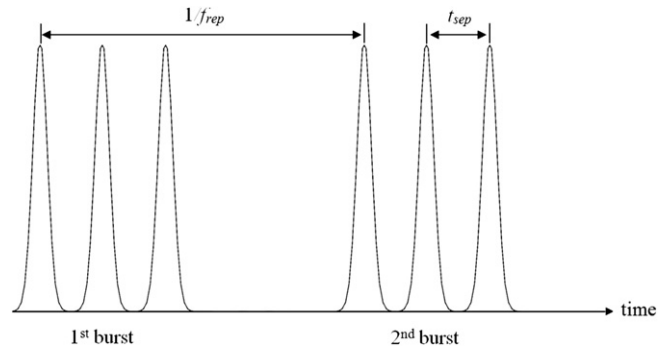


Fig. 1. Illustration of a laser pulse train.

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