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ZrO_2 /carbon aerogel composites: A study on the effect of the crystal ZrO_2 structure on cationic dye adsorption



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ABSTRACT

In this study, $ZrO_2/carbon$ aerogel (CA) composites with different amounts of monoclinic and tetragonal ZrO_2 crystallites were successfully prepared by varying the quantity of HMTA in the hydrothermal synthesis. Monoclinic ZrO_2 (m- ZrO_2) structures are successfully transformed to tetragonal ZrO_2 (t- ZrO_2) structures by increasing the amount of HMTA from 0.187 to 1.683 g. As a result, ZrO_2/CA (H 1.683) samples with the addition of 1.683 g of HMTA possess only the t- ZrO_2 structure. Furthermore, ZrO_2/CA (H 1.683) samples possess the highest adsorption capacity for cationic RhB dyes with positive charges, which is attributed to these samples having the largest negative surface charge (-42 mV at pH 7). As a result, ZrO_2/CA (H 1.683) samples have a large potential for use in the industrial removal of the cationic RhB dye. This work provides a basis for the concept of crystalline structure design to improve organic dye adsorption.

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1. Introduction

ZrO₂ is a well-known ceramic material with three different crystalline structures: monoclinic, tetragonal and cubic. These structures are used for various applications, including solid oxide electrolytes [1], drug delivery [2,3], catalysis [4,5], solar cells [6] and adsorption [7,8]. The crystalline structure of ZrO_2 is very important for applications. Take the adsorption application for example; compared with the monoclinic ZrO_2 structure (m-ZrO₂), tetragonal ZrO₂ (t-ZrO₂) possesses a larger negative surface charge, which results in a better adsorption capacity for cationic dyes, such as RhB dyes. However, m-ZrO2 structure is thermally stable below 1200 °C, while the t-ZrO₂ structure is stable between 1200 °C and 2300 °C. Furthermore, cubic ZrO₂ (c-ZrO₂) is thermally stable above 2300 °C [9], which means that t-ZrO₂ and c-ZrO₂ structures are hardly prepared at low temperatures. Some strategies to prepare t-ZrO₂ and c-ZrO₂ structures at low temperatures have been reported in the literature. c-ZrO₂ structures with an excess of oxygen vacancies were successfully prepared by yttrium doping to form cubic yttria-stabilized zirconia (YSZ) [10]. The t-ZrO₂ and c-ZrO₂ nanocrystallites were successfully fabricated in ammonia solutions using a femtosecond laser ablation process [11]. Tahir et al. [12] successfully synthesized c-ZrO₂ nanoparticles in aqueous NaOH solutions at high pH using a hydrothermal process. The results indicate that $\mbox{t-}ZrO_2$ and $\mbox{c-}ZrO_2$ crystallites are

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favored for growth at high pH values and OH⁻ concentrations. In our previous study [13], ZrO₂/carbon aerogel (CA) composites with t-ZrO₂ structures were successfully synthesized with the assistance of phenol formaldehyde (PF) resins that formed during the hydrothermal process. Compared with pure m-ZrO₂ nanoparticles, negatively charged t-ZrO₂/CA composites avoid the particle aggregations, leading to a larger specific surface area and a better adsorption capacity for cationic dyes.

In this study, ZrO_2/CA composites with different amounts of m-ZrO₂ and t-ZrO₂ phases could be controllably generated by the formation of PF resins and adding different amounts of hexamethylenetetramine (HMTA). The effect of the ZrO₂ crystal structure on the adsorption capacity for cationic dyes, such as RhB dyes, was investigated in this work. The results show that t-ZrO₂ possesses a greater negative surface charge, which leads to a better adsorption capacity for cationic dyes, such as RhB dyes. This work illustrates the concept of crystalline structure design for organic dye adsorption.

2. Experimental

2.1. Chemicals

Zirconlyl chloride octahydrate (ZrOCl₂ ²8H₂O, 99%, Merck Company), hexamethylenetetramine (HMTA, 95%, ECHO-Chemical Company) and phenol (99%, ECHO-Chemical Company) were used for the preparation of the ZrO₂/CA composites. Rhodamine B dyes (RhB, 95%, Sigma-Aldrich Company), and de-ionized water were

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Table	1
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 ZrO_2 percentage, percentage of monoclinic and tetragonal crystallites, pore volume, specific surface area and surface charges in the ZrO_2/CA samples as a factor of different amounts of HMTA (0.187 g, 0.561 g and 1.683 g).

_	ZrO ₂ percentage (%)	M (%)	T (%)	Pore volume (cm ³ /g)	Specific surface area (m²/g)	Surface charge (mV) (pH 7)
ZrO ₂ /CA (H 0.187)	27.6	88.4	11.6	0.15	146.5	19.3
ZrO ₂ /CA (H 0.561)	46.9	43.6	56.4	0.35	337.6	27.0
ZrO ₂ /CA (H 1.683)	46.1	0	100	0.41	275.7	42.0

used for the dye absorption experiments. Eye, face or body contacts with RhB dyes should be avoided during the dye absorption experiments.

2.2. Synthesis of ZrO₂/CA composites

The preparation of ZrO_2/CA composites was followed according to our previous studies [13]. Briefly speaking, ZrO_2/CA composites were obtained by the carbonization of ZrO_2 /phenol formaldehyde (PF) resins at 800 °C for 6 h under a nitrogen atmosphere. For the ZrO_2/PF nanostructures, $ZrOCl_2$ (1.1 g), phenol (0.502 g) and three different amounts of HMTA (0.187, 0.561 and 1.683 g) were allowed to react together at 180 °C for 24 h in a hydrothermal process. The resulting ZrO_2/PF nanostructures with different amounts of HMTA (0.187, 0.561 and 1.683 g) were further heated at 800 °C for 6 h to form the ZrO_2/CA (H 0.187), ZrO_2/CA (H 0.561) and ZrO_2/CA (H 1.6831) samples, respectively.

2.3. RhB dyes adsorption test

 ZrO_2/CA (H 0.187), ZrO_2/CA (H 0.561) and ZrO_2/CA (H 1.6831) samples were utilized for the adsorption of RhB dyes. A total of 10 mg of ZrO_2/CA (H 0.187), ZrO_2/CA (H 0.561) or ZrO_2/CA (H 1.6831) was immersed in 100 ml of RhB dye solutions with differing concentrations at pH 7. The resulting mixtures were further stirred at room temperature for 24 h to reach equilibrium. After a centrifugal separation, the remaining RhB concentration was measured using a UV-visible spectrophotometry, and this concentration was used to obtain the adsorption isotherms. For the regeneration procedures of ZrO_2/CA composites, the used ZrO_2/CA adsorbents were stirred in 100 ml of ethanol solution for 2 h. After centrifugation separation, the washed ZrO_2/CA adsorbents were further used for the second cycle of the dye adsorption.

2.4. Characterizations

The surface morphology and the crystalline phases of the ZrO₂/CA samples were examined via field-emission scanning electron microscopy (FESEM, Hitachi, S-4800N) and powder X-ray diffraction (PXRD, PANalytical X'Pert PRO, PW3040/60), respectively. The t-ZrO₂ crystalline size of ZrO₂/CA (H 0.187), ZrO₂/CA (H 0.561) and ZrO₂/CA (H 1.6831) samples was calculated using the Scheerer equation as follows: $d = \frac{0.9\lambda}{B\cos\theta}$. Where, d is the crystalline size of t-ZrO₂ phase, λ is X-ray wavelength, θ is the Bragg angle and B is the line broadening at half the maximum intensity (FWHM) of the Bragg angle, respectively. The specific surface area and pore volume of the ZrO₂/CA samples were measured based on N₂ adsorption/desorption isotherms (BET, Micromeritics ASAP 2020). The functional groups in the as-prepared samples were characterized using Fourier Transform Infrared spectroscopy (FTIR, Perkin-Elmer Inc., Spectrum One). The concentration of RhB dye in solution was measured via UV-visible spectrophotometry (Hitachi, U-3900/U-3900H).

3. Results and discussion

The crystalline structures and surface morphologies of ZrO_2/CA (H 0.187), ZrO_2/CA (H 0.561) and ZrO_2/CA (H 1.683) samples were



Fig. 1. XRD patterns for the reference (a) tetragonal and (b) monoclinic ZrO₂ crystals, and the ZrO₂/CA structures with different amounts of HMTA: (c) 0.187 g, (d) 0.561 g and (e) 1.683 g. FESEM images of the ZrO₂/CA structures with different amounts of HMTA: (f) 0.187 g, (g) 0.561 g and (h) 1.683 g.

first investigated by FESEM and XRD, as shown in Fig. 1. The reference diffraction patterns for t-ZrO₂ and m-ZrO₂ phases are included in Fig. 1(a) and (b), respectively. Fig. 1(c) and (d) are the XRD diffraction patterns of ZrO₂/CA (H 0.187) and of ZrO₂/CA (H 0.561) samples. The diffraction peaks produced by the ZrO₂/CA (H 0.187) and ZrO_2/CA (H 0.561) samples agree well with those of the reference m-ZrO₂ and t-ZrO₂ crystals, indicating the presence of mixed monoclinic and tetragonal crystalline phases in these samples. Surprisingly, as the quantity of HMTA was increased to 1.683 g, the observed diffraction peaks of the ZrO₂/CA (H 1.683) samples (Fig. 1(e)) corresponded only to those of the tetragonal reference, implying the existence of a pure tetragonal phase in ZrO₂/CA (H 1.683) sample. The monoclinic (M%) and tetragonal (T%) crystallite percentages of the three ZrO₂/CA samples were calculated in the manner following our previous study [13], and the results are listed in Table 1. The monoclinic crystallite percentages are 88.4, 43.6 and 0% for the ZrO₂/CA (H 0.187), ZrO₂/CA (H 0.561) and ZrO₂/CA (H 1.683) samples, respectively. Furthermore, the tetragonal crystallite percentages are 11.6, 56.4 and 100% for the ZrO₂/CA (H 0.187), ZrO₂/CA (H 0.561) and ZrO₂/CA (H 1.683) samples and the t-ZrO₂ crystalline sizes determined by the Scheerer equation are 6.9, 8.5 and 10.8 nm for the ZrO₂/CA (H 0.187), ZrO₂/CA (H 0.561) and ZrO₂/CA (H 1.683) samples, respectively. A possible mechanism for the ZrO₂ monoclinic to tetragonal phase transition was discussed in our previous study [13]. Briefly speaking, higher HMTA concentrations produce more OHions in the reaction solution, and the growth of symmetric tetragonal crystallites are favored at higher OH⁻ concentrations [11,12]. The FESEM images of the three samples (Fig. 1(f-h)) reveal their porous structure. The specific surface areas of the ZrO₂/CA (H 0.187), ZrO₂/CA (H 0.561) and ZrO₂/CA (H 1.683) samples are 146.5, 337.6 and 275.7 m²/g (Table 1), respectively, as obtained from N₂ adsorption/desorption isotherms. The pore volumes of the ZrO₂/CA (H 0.187) (0.15 cm³/g), ZrO₂/CA (H 0.561) (0.35 cm³/g) and ZrO₂/CA (H 1.683) (0.41 cm³/g) samples are also obtained from N_2 adsorption/desorption isotherms, as shown in Table 1.

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