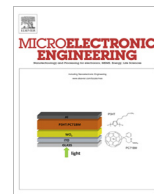




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journal homepage: www.elsevier.com/locate/meeBand alignment at interfaces of few-monolayer MoS₂ with SiO₂ and HfO₂V.V. Afanas'ev^{a,*}, D. Chiappe^b, C. Huyghebaert^b, I. Radu^b, S. De Gendt^b, M. Houssa^a, A. Stesmans^a^aSemiconductor Physics Laboratory, KULeuven, Celestijnenlaan 200D, Leuven 3001, Belgium^bimec, Kapeldreef 75, Leuven 3001, Belgium

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ABSTRACT

Internal photoemission of electrons from 4- and 2-monolayer thick MoS₂ films prepared by sulphurization of metallic Mo on top of SiO₂ or HfO₂/SiO₂ insulating stacks is detected. This enables determination of the energy position of the MoS₂ valence band which is found to be at 4.1–4.2 eV below the SiO₂ conduction band. At the interface with HfO₂, a barrier height of 3.7 eV is found, corresponding to an increase of the electron affinity of MoS₂ by ≈ 0.5 eV as compared to the SiO₂ case. This suggests the presence of interfacial charges (or dipoles) in the interfacial HfO₂ layer.

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1. Introduction

Layered transition metal dichalcogenide semiconductors attract considerable interest because of their tunable properties relevant to a broad range of applications. In particular, few-layer MoS₂ has already been shown to deliver significant advantages in terms of transistor channel downscaling and optoelectronic properties [1]. The performance of the MoS₂-based devices critically depends on the band alignment with other materials since the alignment determines the electrostatics of the stack, height of the tunneling barriers, contact resistance, etc. However, characterization of the barriers at interfaces of few-layer 2D materials generally represents quite an experimental challenge: The electronic transport across the interfaces is strongly affected by hydrocarbon residues of the flake transfer process [2], non-homogeneities (e.g., domain boundaries), and defects in the material itself [3]. For example, it is found that the same metal on MoS₂ may provide both *n*- and *p*-type Schottky barriers, pointing to an inhomogeneous work function distribution [3,4].

Therefore, it is important to find a technique allowing one to reliably determine the MoS₂ bandgap edge energies with respect to other materials. In the present work we will show that this goal can be achieved using the spectroscopy of Internal PhotoEmission (IPE) of electrons from MoS₂ into an insulating layer. By using 2- and 4-monolayer (ML) thick MoS₂ films on top of SiO₂ layer or HfO₂/SiO₂ stack as the prototype interfaces, we observe IPE from

a thin 2D photo-emitter and determine the corresponding energy barriers.

2. Experimental

The studied samples were prepared by thermal evaporation of a thin Mo film on SiO₂(50 nm)/p⁺-Si (B-doped, $n_a \approx 10^{20} \text{ cm}^{-3}$) or HfO₂(2 nm)/SiO₂(50 nm)/p⁺-Si substrates. Next, the Mo/oxide/Si samples were transferred to a furnace and sulfurized in pure H₂S(100 mbar) at 800 °C resulting in the formation of a large-area (in the range of cm²) polycrystalline MoS₂ film with crystallites of ≈ 50 nm size as revealed by atomic force microscopy. Cross-sectional transmission electron microscopy (TEM) images, such as shown in Fig. 1, reveal the characteristic layered structure of a ~ 3 nm thick MoS₂ film (~ 4 molecular planes of 0.65-nm thick each [1]). The Raman spectrum shown in the same figure exhibits two distinct peaks at around 383 cm⁻¹ and 408 cm⁻¹ corresponding to the in-plane (*E*_{2g}) and the out-of-plane (*A*_{1g}) vibrations in MoS₂, respectively [1]. This observation and electron spectroscopy analysis reveals conversion of metallic Mo into 2H-MoS₂ characterized by hexagonal crystalline structure and trigonal prismatic coordination geometry, also supported by plane-view TEM images (not shown). By varying the thickness of the initially deposited Mo, it appears possible to produce samples with 4 or 2 ML thick MoS₂ films without contamination of the MoS₂/oxide interfaces.

To ensure reliable electrical contacts, optically nontransparent (100-nm thick) Au or Al pads (0.01 mm² area) were evaporated on top of the MoS₂ film. For the sake of comparison, one 4 ML MoS₂ sample was metalized by evaporating semitransparent (13-nm thick) blanket Au electrodes of 0.5 mm² area. The steady-state

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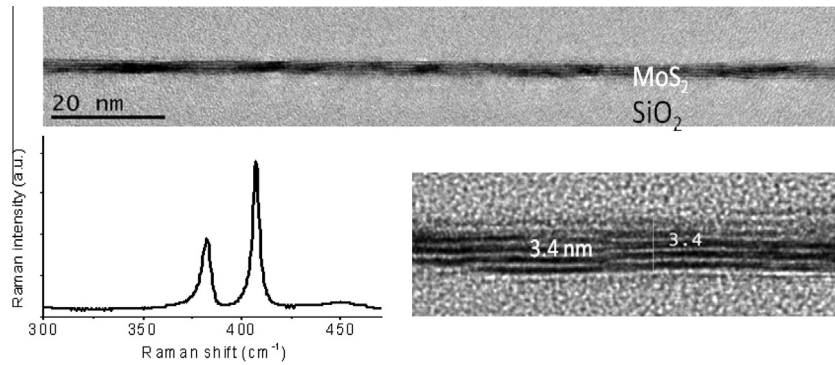


Fig. 1. Cross-sectional TEM images and Raman spectra of a 4 ML MoS₂/SiO₂/Si sample.

value of the IPE current was measured between the Si substrate and the MoS₂ layer by applying a negative bias to MoS₂ to enable electron photoinjection. Generally, the major challenge in the realization of IPE from a 2D material is related to the fact that direct optical transitions result in electrons with predominantly in-plane momentum orientation [1,5] which would not allow them to surmount the interface barrier. Nevertheless, the present study indicates that IPE from the quasi-2D MoS₂ emitter can be detected, suggesting relaxation of the momentum conservation rule.

3. Results and discussion

The IPE quantum yield (Y) measured as a function of the photon energy ($h\nu$) and the $Y^{1/3}$ - $h\nu$ plots used to extract the spectral thresholds are shown in Fig. 2 (open symbols) for a 4 ML MoS₂/SiO₂/Si sample with Au contact pads. The spectral curves exhibit a clearly visible increasing red-shift with increasing negative voltage applied to the MoS₂ film due to the image-force barrier lowering (the Schottky effect). This observation points to IPE of electrons into the conduction band (CB) of SiO₂ since the barrier lowering becomes visible only in the case of ballistic photo-electron transport across the interface barrier. The corresponding spectral thresholds extracted from the $Y^{1/3}$ - $h\nu$ plots [6] shown in Fig. 2(b) obey the image-force model as evidenced by the Schottky plot shown in Fig. 3(a). The barrier height Φ_e between the valence band (VB) of MoS₂ and the oxide CB can be found by extrapolation to zero electric field, giving the value $\Phi_e(\text{MoS}_2) = 4.20 \pm 0.05$ eV, where the image force dielectric constant ϵ_i is found to be

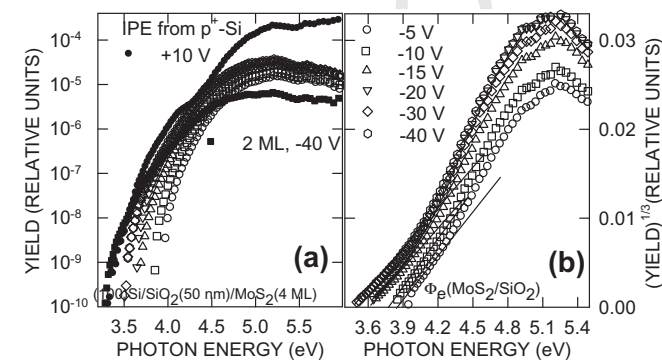


Fig. 2. (a) Quantum yield as a function of photon energy measured on the (100)Si/SiO₂(50 nm)/MoS₂(4 ML) structures under different negative bias voltages applied to the MoS₂ electrode (open symbols). The IPE yield spectra corresponding to electron emission from the p⁺-Si substrate (●), observed under positive bias, and from a 2-ML MoS₂ film (■) are also shown for comparison; (b) Determination of the IPE spectral threshold using linear extrapolation of the $Y^{1/3}$ - $h\nu$ plots.

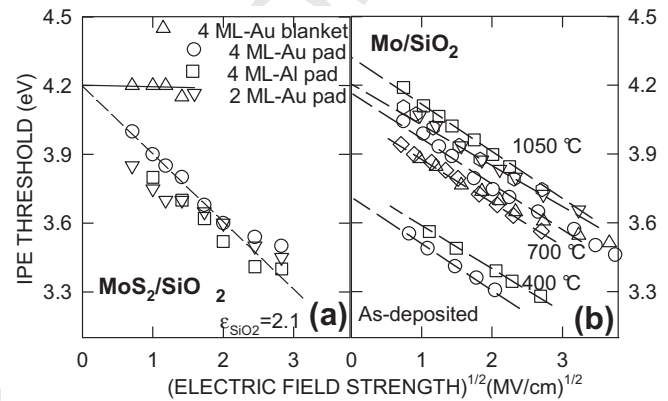


Fig. 3. (a) The Schottky plot of spectral thresholds measured on (100)Si/SiO₂(50 nm)/MoS₂ structures using different contact schemes and metallization materials; (b) Schottky plot of spectral thresholds measured on the (100)Si/SiO₂(10 nm)/Mo(10 nm) structures after 30 s annealing in N₂ at different temperatures (after Ref. [7]).

consistent with the value expected for SiO₂: $\epsilon_i \approx n^2 = 2.1$, where $n = 1.46$ is the refractive index of SiO₂.

For the sake of comparison, in Fig. 2(a) are also shown IPE yield spectra corresponding to electron emission from the p⁺-Si substrate (●, positive bias) and from the 2-ML MoS₂ film (■). Electron IPE from Si can be identified on the basis of the yield drop observed at $h\nu \approx 4.3$ eV corresponding to the excitation of direct optical transitions in the Si substrate crystal (the E_2 singularity [7]). The quantum yield value of electron IPE from Si is approximately 10 times higher than from a 4 ML MoS₂ photoemitter (cf. the range $h\nu > 5$ eV), reflecting the small volume available for the optical excitation in the last case. This explanation is also consistent with even lower yield observed in the case of IPE from a 2 ML MoS₂ electrode [cf. Fig. 2(a)] independently indicating electron IPE from few-ML MoS₂.

Comparison of the field-dependent spectral thresholds measured in the structures with 4 ML MoS₂ electrodes and Au or Al contact pads [Fig. 3(a)] reveals that the metal work function has no significant influence on the results, further supporting the identification of the IPE as constituting electron emission from MoS₂. In the case 2 ML MoS₂, the spectral thresholds are slightly (≈ 150 meV) lower than for the 4 ML MoS₂ layer but still obey the same image-force law. By contrast, as evident from Fig. 3(a), the sample with the Au blanket contact on top of 4 ML MoS₂ shows no measurable barrier lowering while the spectral threshold value exactly corresponds to the zero-field barrier value found in the case of IPE from an unmetallized MoS₂ layer. This effect can also be seen as indication that the IPE occurs by excitation of electrons from the MoS₂ electrode: Since the density of charge carriers in MoS₂ is insufficient to screen a hole left behind by a photoelectron

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