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³ Residual layer-free Reverse Nanoimprint Lithography on silicon

and metal-coated substrates

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ABSTRACT

In this work we demonstrate that Reverse Nanoimprint Lithography is a feasible and flexible lithography 25 technique applicable to the transfer of micro and nano polymer structures with no residual layer over 26 areas of cm^2 areas on silicon, metal and non-planar substrates. We used a flexible polydimethylsiloxane 27 stamp with hydrophobic features. We present residual layer-free patterns imprinted using a commercial 28
poly(methylmethacrylate) thermoplastic polymer over silicon, nickel and pre-patterned substrates. Our 29 poly(methylmethacrylate) thermoplastic polymer over silicon, nickel and pre-patterned substrates. Our 29 versatile patterning technology is adaptable to free form nano structuring and has coupling to adhesion 30 technologies. 31

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36 1. Introduction

 Nanoimprint lithography (NIL) dates back to 1995 [\[1\]](#page--1-0) and is currently identified as one of the most promising techniques, due to its potential low-cost, high-resolution and large-area patterning capabilities compared with other lithography techniques. NIL has been used to obtain patterns with sub-10 nm resolution, in ther-42 mal $[2]$ and ultraviolet (UV) nanoimprint $[3]$ modes. Furthermore, this technique is compatible with the roll-to-roll process and it is suitable for manufacturing a range of different devices such as 45 optoelectronic [\[4\]](#page--1-0) or energy devices [\[5\].](#page--1-0)

 Although NIL has been extensively developed, there are still issues that need to be addressed for it to become a mature nano manufacturing technology. One of the challenges is the removal of the residual layer after the imprinting step. Regardless of the type of polymer resists used, a residual layer will always remain between the elevated parts of the stamp and the substrate. Most applications require the elimination of this residual layer via a 53 reactive ion etching (RIE) post processing step [\[6\].](#page--1-0) Although RIE is a well-controlled process this additional step increases the over- all processing complexity, implies an additional cost and decreases the throughput of the nanoimprint process. In addition, it is possi-ble that the edges of the polymer features become rounded during

<http://dx.doi.org/10.1016/j.mee.2014.11.025> 0167-9317/© 2014 Elsevier B.V. All rights reserved. the RIE process because of the low selectivity of the etching pro- 58 cess. Therefore, residual layer-free imprint lithography is indis- 59 pensable for large-area fabrication. The same state of \sim 60

Various methods have been used to create residual layer-free 61 imprints. Attempts include partial UV-curing of polymer followed 62 by an additional chemical development process $[7,8]$, utilizing flex- 63 ible molds to directly contact the substrate in order to eliminate 64 residual layer $[9,10]$ and selective filling into the recessed areas 65 of the mold followed by transferring the polymer patterns to a sub- 66 strate [\[11,12\].](#page--1-0) 67

In thermal NIL, imprinting is typically performed at tempera- 68 tures between 60 and 90 °C above the polymers T_g under pressures 69
as high as 10 MPa [13]. In contrast to thermal NIL reversal imprint 70 as high as 10 MPa $[13]$. In contrast to thermal NIL reversal imprint lithography (RNIL) is a promising residual layer-free lithography 71 technique which requires low pressures and temperatures. 72 Imprinting temperatures around the polymer T_g are sufficient to 73 achieve a reliable pattern transfer. This technique is used to selec- 74 tively imprint thin polymer films over flat and/or pre-patterned 75 surfaces [\[14\]](#page--1-0). 76

This approach has been followed by different authors. Park et al. 77 tried to remove the residual layer from a surfactant-coated silicon 78 mold. By creating and incompatibility between the surface and the 79 resist, they induced a dewetting process. Heating the substrate 80 above the glass transition temperature of the polymer, the top of 81 the mold could be moved to mold trenches [\[12\].](#page--1-0) Kao et al. 82 performed a residue-free RNIL by means of a self-assembly 83

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 monolayer-treated silicon mold [\[11\]](#page--1-0), inducing a selective filling of the mold by spin coating. All these free residual layer imprints were achieved by means of various process steps, involving chem- ical treatments. Furthermore, in all cases hard silicon molds were used, which imply higher imprint pressures and limit the possibil-ity of transferring the process to free-form surfaces.

 The typical silicon molds used in most of the NIL experiments have several disadvantages, such as cost and brittleness. Further- more, these kinds of molds make impossible a proper pattern transfer on free-form surfaces, limiting the application scope. Flex- ible molds, such as those made in polydimethylsiloxane (PDMS) have the advantage of avoiding breakage during direct contact between the mold and substrates and only require low pressure 97 for conformal contact, even in large-area fabrications [\[10\]](#page--1-0).

 Here we show a one-step residual-layer-free patterning method (Fig. 1). By means of the RNIL flexible lithography technique, poly-100 mer structures with no residual layer over cm² areas on silicon and metallic substrates were achieved as well as on pre-patterned sub- strates, leading to the possibility of developing 3D structures. We use a flexible PDMS stamp with honey comb like hydrophobic structures as a proof of concept. These honeycomb structures have a spacing of 500 nm and a height of 800 nm. Due to our experimen- tal design configuration (flexible mold), low pressure (1 bar) and 107 temperatures ($T_{\text{imp}} \approx T_{\text{g}}$) were sufficient enough to achieve the 108 large area $\rm (cm^2)$ pattern transfer.

 This technique, based on the flexibility of the stamp used is transferable to non-planar surfaces and pre-patterned surfaces, given that the stamp can be pressed against a structured surface without break.

 The fabrication approach presented in this work is also transfer- able to any kind of geometry, including those intended for hydro-115 phobic purposes. In these structures the wettability of the resist is a key parameter to take into account, in terms of pattern transfer quality. Furthermore, it is well accepted that wetting phenomena occurring on flat surfaces is quite different from that occurring of 119 patterned ones [\[15–17\]](#page--1-0).

 To study the wettability of a surface, we could take into account 121 two different models, the Cassie–Baxter model [\[18\]](#page--1-0) and the Wen-122 zel model [\[19\]](#page--1-0). The so called Cassie state describes the situation where a droplet partially sits on air pockets whereas the Wenzel state implies a complete wetting of a rough surface, i.e. liquid pen- etrates into grooves constituting a relief. Therefore, for our exper-imental approach, a wetting transition towards a Wenzel state is

needed in order to assure a complete and uniform distribution of 127 the resist along the flexible stamp. Wetting transitions on rough 128 surfaces have been studied in detail by Bormashenko [\[20,21\]](#page--1-0) and 129 constitute an interesting topic regarding wetting behavior and tai- 130 loring of surface properties through external stimuli [\[22\]](#page--1-0). In partic-
131 ular, we have chosen mechanical vibrations as the external source 132 of the required energy to overcome the energetic barrier that sep- 133 arates equilibrium wetting states, from the Cassie towards the 134 Wenzel state. 135

As an example of this approach, we used the negative polarity of 136 the honey comb like structures, due to the fact that the cavities to 137 be filled have a width of 500 nm.

2. Experimental procedure 139 and 139

2.1. PDMS replica fabrication 140

A honeycomb structure was used as mold. Honeycomb designs 141 with 800 nm feature height and 500 nm trench separation were 142 used. Firstly the patterns were defined on a silicon wafer by elec- 143 tron-beam lithography followed by a reactive ion etching step to 144 create the full Si master mold. The Si mold was treated with a fluo- 145 rinated anti-adhesion layer in order to alter the surface hydropho- 146 bicity for easy mold releasing. The coating step was performed by 147 immersing wafers in a low concentration of Optool DSX solution 148 $(1\%$ wt into perfluorohexane) during 1 min. Then, the wafer was 149 placed in a water environment at 65 \degree C for 1 h. Finally, the wafer 150 was rinsed in perfluorohexane during 10 min. 151

The PDMS mold was fabricated by casting the PDMS prepoly- 152 mer against the relief structure of the silicon master mold. We used 153 the PDMS precursor with a mixing ratio of 5:1 (precursor:curing 154 agent) and it cured for 12 h at 60 °C. We optimized the thickness 155 of the PDMS molds to be 3 mm in order to avoid pattern deforma- 156 tion due to the stretching of the mold $[23]$. The cured PDMS mold 157 was peeled off manually from the silicon stamp. 158

2.2. Reverse nanoimprint process 159

The polymer used in this work was poly(methylmethacrylate) 160 $(PMMA)$ with an average molecular weight (MW) of 75,000. In case 161 of a mold with nm features, the thickness of the deposited film is a 162 strong function of the concentration of the solution used for spin 163

Fig. 1. Schematic illustration of the residual-layer-free reversal NIL process.

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