



# Synthesis and photocatalytic properties of bismuth titanate with different structures via oxidant peroxo method (OPM)



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## ABSTRACT

Bismuth titanate ( $\text{Bi}_4\text{Ti}_3\text{O}_{12}$  and  $\text{Bi}_{12}\text{TiO}_{20}$ ) powders were synthesized by the Oxidant Peroxide Method (OPM), and the effect of temperatures on physical and chemical properties of particles was investigated. The results showed that the morphology and average particle size of materials can be successfully controlled by adjusting the temperature. The samples after calcination were characterized by X-ray diffraction, transmission electron microscopy, diffuse reflectance spectroscopy, Raman spectroscopy, and BET isotherms. The photocatalytic activity of materials was also evaluated by studying the degradation of 10 ppm aqueous rhodamine B dye under ultraviolet radiation.

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## 1. Introduction

Currently, advanced oxidation processes (AOP) based on photocatalytic processes have been widely used as an alternative treatment of effluents, especially in the case of water contaminated by organic compounds. Although there are some peculiarities, the AOP mechanism for the degradation of contaminants is well known, as shown by Zhang et al. [1] for degradation of sulfadiazine using heterogeneous photocatalysis. However, new synthesis routes used to obtain improved materials for AOP, as well as studies about the inherent effects of synthesis strategies in the photocatalytic processes, have only been minimally explored.

Bismuth titanate, a large family that includes several phases in the Bi–Ti–O system, is a promising candidate for various technological applications, including effluent treatment [2]. The  $\text{Bi}_4\text{Ti}_3\text{O}_{12}$  compound, which is generally used for applications such as memory storage, optical displays, piezoelectric converters and photocatalysis [3], has a layered perovskite structure built up by the regular intergrowth of  $(\text{Bi}_2\text{O}_2)^{2+}$  layers and a perovskite type layer  $(\text{Bi}_2\text{Ti}_3\text{O}_{10})^{2-}$ , where Bi ions occupy the 12-coordinated sites [4]. Crystals of  $\text{Bi}_{12}\text{TiO}_{20}$ , on the other hand, belong to the sillenite family with crystal structures being formed by Bi–O polyhedra, where Bi ions are coordinated with five oxygens that form an octahedral

arrangement with the Bi–O polyhedron network connected to regular  $\text{TiO}_4$  tetrahedra [5].

In earlier studies, Yao et al. [6,7] showed that sillenite ( $\text{Bi}_{12}\text{TiO}_{20}$ ) and perovskite ( $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ ) show high photocatalytic activity for organic dye degradation. Their photocatalytic activity can be attributed to the special electronic structure and the well dispersed electrons of  $\text{Bi}_{6s}$ , which can form the wide valence band and increase the mobility of electric charges. Bismuth titanates  $\text{Bi}_{12}\text{TiO}_{20}$  and  $\text{Bi}_4\text{Ti}_3\text{O}_{12}$  have different crystal and electronic structures, and may also possess band structures which can be increased in the quantum efficiency of the heterogeneous photocatalytic property by acting as electron (or hole) traps and by altering the  $e^-/h^+$  pair recombination rate.

Several methods, including the solid state reaction [8], microemulsion method [3], co-precipitation [9], hydrothermal processing [10], mechanical synthesis [11] and sol–gel routes [12,13], have been developed for the synthesis of different phases of bismuth titanate powders. Yet, all of these methods have some drawbacks, including the appearance of secondary phases and the presence of non-stoichiometric compositions and undesired impurities [10].

A recent study, describing the oxidant peroxo method of synthesis (OPM), showed several promising advantages when compared to the other synthesis techniques, in terms of purity of the product, simple reactive conditions, and easier control of particle size; moreover this method is free from added contaminants such as chloride ions and organic compounds that could interfere in the properties and photocatalytic process [14–18].

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The use of the OPM route to obtain reactive bismuth titanates for AOP can be particularly attractive, since impurities are avoided and crystalline materials can be formed at lower temperatures, excluding a series of interference usually present in different methods of synthesis. In this study, we report an easy method to prepare bismuth titanates ( $\text{Bi}_4\text{Ti}_3\text{O}_{12}$  and  $\text{Bi}_{12}\text{TiO}_{20}$ ) and the influence of the structure on the reaction of photocatalysis under UV radiation. This two-step preparation procedure consists of preparing an amorphous precipitate followed by a firing treatment. The amorphous precipitate was obtained at room temperature through the reaction between some aqueous soluble peroxy-complexes and bismuth ions. The photocatalytic activity of well crystalline bismuth titanates was tested on the degradation of rhodamine B (RhB) dye in aqueous solution, under UV-light irradiation.

## 2. Materials and methods

### 2.1. Synthesis

The amorphous precipitates with molar ratio Bi:Ti of 12:1 and 4:3 were synthesized as described by Camargo and Kakihana [15] through the dissolution of titanium metal powder (98% Aldrich, USA) in an aqueous solution of 60 mL of  $\text{H}_2\text{O}_2$  (30%, analytical grade, Synth, Brazil) and 40 mL of aqueous ammonia solution (30%, analytical grade, Synth, Brazil). After approximately 3 h, a yellow transparent solution of peroxytitanate complexes was obtained. A second solution of  $\text{Bi}_2\text{O}_3$  (99.99%, Aldrich, USA) was added into 10 mL of nitric acid (analytical grade, Synth, Brazil) and slowly added dropwise on the peroxytitanate solution under stirring and cooling in an ice-water bath. This resulted in vigorous gas evolution, forming an amorphous and yellow precipitate that was filtered and washed with diluted ammonia solution to eliminate all nitrate ions. The washed precipitate was dried at 60 °C for 3 h and ground and calcined at several temperatures between 400 and 900 °C for 1 h at a heating rate of 10 °C  $\text{min}^{-1}$  in closed alumina boats.

### 2.2. Characterization

All the powders (amorphous and heat-treated) were characterized at room temperature by X-ray diffraction (XRD) using  $\text{Cu K}\alpha$  radiation (Rigaku D/MAX 200, with a rotary anode operating at 150 kV and 40 mA) in the  $2\theta$  range from 10° to 75° with step scan of 0.02°. Raman spectra were obtained with a Horiba Jobin-Yvon Raman micro-spectrometer LabRAM at room temperature using the 633 nm line of a 5.9 mW He-Ne laser as the excitation source through an Olympus TM BX41 microscope. The Brunauer-Emmett-Teller (BET) isotherms were determined through nitrogen adsorption at 77 K (Micromeritics ASAP 2000). Diffuse reflectance spectra in the ultraviolet-visible region (DRS/UV-Vis) were recorded at room temperature in the 200–800 nm range using a Varian model Cary 5G in the diffuse reflectance mode ( $R$ ) and transformed to a magnitude proportional to the extinction coefficient ( $K$ ) through the Kubelka-Munk function [19]. From these spectra, the band gaps of the samples were determined according to the method proposed by Tauc [20]. The morphology of the powders was characterized by transmission electron microscopy (TEM) FEI/PHILIPS CM120.

### 2.3. Photocatalysis

Eighty milliliters of a 10 mg  $\text{L}^{-1}$  RhB dye solution at pH 6.0 was mixed with 60 mg of the catalyst and irradiated with ultraviolet light placed in a box at 25 °C, which was set at a distance of 30 cm from the beaker containing the catalyst and RhB dye. The

radiation used in this experiment was obtained with a medium-pressure mercury vapor lamp (Osram, HQL 400) using output without a protection bulb to perform the photodegradation reactions. For the adsorption test, we used the same conditions except for the presence of UV radiation. All the tests were performed under vigorous stirring. Reactions were monitored by UV-Vis spectroscopy (JASCO V-660) at 554 nm, using a commercial quartz cuvette. The experiments were performed in quadruplicate.

## 3. Results and discussion

Generally, bismuth titanates are prepared by the solid-state reaction between  $\text{Bi}_2\text{O}_3$  and  $\text{TiO}_2$  at elevated temperatures; however, owing to some inherent limitations, this process yields large grain size and secondary phases [21,22]. In this study, pure  $\text{Bi}_{12}\text{TiO}_{20}$  and  $\text{Bi}_4\text{Ti}_3\text{O}_{12}$  powders were successfully prepared by the OPM route under mild crystallization conditions.

Hydrogen peroxide is a very attractive oxidant reagent for reactions in aqueous media and has been widely used in several procedures based on the “green chemical” approach [23]. In the OPM method, hydrogen peroxide is replaced by titanium peroxy-complexes synthesized *in situ* that act in a similar fashion toward  $\text{H}_2\text{O}_2$  molecules. This reaction between soluble peroxy-complexes and bismuth ions results in the formation of a non-crystalline precipitate free of contaminants and without the generation of chemical waste. Particularly, a large amount of oxygen is released when this precipitate crystallizes, which increases the local oxygen pressure and strongly affects the final properties of the oxide formed [24]. It should be emphasized that these precipitates can be obtained at any desired Bi:Ti molar ratio, which means that the method can be highly tailored, allowing the synthesis of bismuth titanates with different structures and compositions.

Fig. 1a shows the XRD patterns of samples prepared with Bi:Ti molar ratio of 12:1 fired at different temperatures for 1 h. The absence of any diffraction peak in the XRD curve of the precipitate is of fundamental importance. Sometimes, wet-chemical routes form precipitates partially crystallized with undesired phases, resulting in stable secondary phases virtually impossible to be eliminated. Apart from the presence of a few small peaks of a secondary  $\text{Bi}_4\text{Ti}_3\text{O}_{12}$  phase in the sample annealed at 400 °C (assigned with ●), all peaks in the samples heat-treated at 500 °C or higher were indexed on the basis of body-centered cubic (bcc) sillenite  $\text{Bi}_{12}\text{TiO}_{20}$  (PDF 34-0097). In comparison with other methods of synthesis as disclosed by Ranieri et al. [25] and Thanabodeekij et al. [26], the OPM route requires a lower calcination temperature for obtaining pure phase.

X-ray diffraction patterns of the precipitate and calcined powders with Bi:Ti molar ratio of 4:3 are shown in Fig. 1b. The sample calcined at 900 °C for 1 h presented only the diffraction peaks corresponding to the  $\text{Bi}_4\text{Ti}_3\text{O}_{12}$  phase (PDF 34-0097), without any trace of a secondary phase. However, samples annealed at lower temperatures showed the presence of the  $\text{Bi}_4\text{Ti}_3\text{O}_{12}$  together with the sillenite ( $\text{Bi}_{12}\text{TiO}_{20}$ ) as majority phase. It has been proposed that the synthesis of the  $\text{Bi}_4\text{Ti}_3\text{O}_{12}$  by different routes occurs through the crystallization of sillenite  $\text{Bi}_{12}\text{TiO}_{20}$  phase, before forming the perovskite  $\text{Bi}_4\text{Ti}_3\text{O}_{12}$  [8,27]. It is interesting to note that the same sequence of reactions was observed here during the synthesis of  $\text{Bi}_4\text{Ti}_3\text{O}_{12}$  by the OPM method, corroborating the idea that sillenite phase is always formed first at lower temperatures, and  $\text{Bi}_4\text{Ti}_3\text{O}_{12}$  is only crystallized at higher temperatures, independent of the synthesis method.

Raman spectra of  $\text{Bi}_{12}\text{TiO}_{20}$  samples calcined at 500–700 °C were all similar, confirming the high phase purity of the samples. Obviously, these spectra are quite different from those of the sample calcined at 400 °C (Fig. 2a) due to the presence of the secondary

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