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High-temperature oxidation behavior and analysis of impedance spectroscopy of 7YSZ thermal barrier coating prepared by plasma spray-physical vapor deposition

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KEYWORDS

14 15 16 Bond coating; 17 Impedance spectroscopy; 18 Isothermal oxidation; 19 **PS-PVD**: TBCs 20

Abstract Quasi-columnar structure 7YSZ (yttria stabilized zirconia) thermal barrier coatings (TBCs) are prepared by plasma spray-physical vapor deposition (PS-PVD) onto pretreated and un-pretreated bond coating, respectively. An isothermal oxidation experiment of 7YSZ TBCs is carried out in the atmosphere of 950 °C in order to simulate the high-temperature oxidation process of engine blades. The isothermal oxidation process of 7YSZ thermal barrier coatings is investigated systematically by impedance spectroscopy. The electrochemical physical model and equivalent circuit of columnar 7YSZ coatings are established. Results show that the isothermal oxidation kinetic curve of columnar 7YSZ thermal barrier coatings appears to follow the parabolic law. A pretreatment of bond coating can reduce the growth rate of the thermally grown oxide (TGO) layer, restraining the initiation and propagation of microcracks between YSZ and TGO layers. The oxidation rate constants of 7YSZ coatings with pretreated and un-pretreated bond coating are 0.101×10^{-12} cm² s^{-1} and $0.115 \times 10^{-13} \text{ cm}^2 \text{ s}^{-1}$, respectively. Impedance analysis shows that the content of oxygen vacancies decreases and the density increases after the TGO layer is oxidized for 150 h. In addition, shrinkage microcracks formed by sintering during the oxidation process is the main reason for an increase of the capacitance and a decrease of the resistance in the grain boundary of YSZ.

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perature and to protect components like blades, which have

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1. Introduction

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Thermal barrier coatings (TBCs) are widely applied to hotcomponents of turbine engines to increase the operation tem-



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a complex structure with a metallic bond coat (MCrAlY, M 26 27 = Ni or Co) on a superalloy substrate for oxidation/corrosion resistance and a ceramic topcoat of yttria stabilized zirconia 28 (YSZ) attached on the bond coating for heat protection.^{1,2} 29 At present, there are mainly two methods to fabricate YSZ 30 topcoats, which are air plasma spraying (APS) and electron 31 32 beam-physical vapor deposition (EB-PVD). However, these two methods also have some disadvantages. Conventional lay-33 ered structured APS TBCs show high deposition rates and 34 good thermal insulation but poor thermal shock resistance. 35 On the contrary, columnar structured EB-PVD TBCs exhibit 36 37 higher strain tolerance and better thermal shock resistance 38 but higher thermal conductivity and lower deposition rates 39 compared with APS TBCs.³

Recently, a new hopeful technique called plasma spray-40 physical vapor deposition (PS-PVD) has been developed based 41 on the low pressure plasma spray process (LPPS) and combi-42 43 nes the advantages of thermal spraying (high deposition rates) 44 and physical vapor deposition (high strain tolerance). The plasma plume of PS-PVD can expand to over 2 m long and 45 nearly 400 mm in diameter under parameters of high power 46 input (about 180 kW) and low work pressure (about 100 47 Pa).⁴ At a high power level, the temperature of the plasma 48 plume can exceed 6000 K.⁵ Therefore, fine grain sized powders 49 are enough to be evaporated and achieve an EB-PVD-like 50 columnar coating. 51

52 High-temperature oxidation is inevitable for TBCs. A layer 53 of thermally grown oxide (TGO) forms at service, which can inhibit the diffusion of oxygen elements into the bond coating 54 and protect the substrate. Moreover, a thermal mismatch due 55 to great differences of the thermal expansion coefficients 56 57 between bond coating and ceramic coatings makes ceramic layer premature failure easy. Much of related literature has 58 studied the high-temperature oxidation behaviors of TBCs, 59 TGO growth evolution, and micro-cracks propagation within 60 61 a ceramic layer. For example, Doleker and Karaoglanli⁶ have compared the oxidation behaviors of YSZ and Gd₂Zr₂O₇ 62 TBCs, which indicate that the Gd₂Zr₂O₇ TBCs have lower 63 64 thermal conductivity, lower oxygen permeability, and higher structural stability at higher temperatures. These advantages 65 66 render the Gd₂Zr₂O₇ a good alternative top coating material for TBCs. Ahmadian and Jordan⁷ have studied the effect of 67 rapid cycling on oxidation, microcracks, and lifetime of APS 68 TBCs. Their results have shown that there must be very signif-69 icant inelasticity presenting in crack formation and crack 70 growth in TBCs. Chen et al.⁸ have investigated the oxidation 71 72 and crack nucleation/growth in an APS TBCs with a NiCrAlY bond coat. Their results have shown that mixed oxides form at 73 the beginning of the thermal exposure in air, along with the 74 formation of the Al₂O₃ layer and cracks initiated mostly in 75 association with the formation of (Cr,Al)₂O₃·Ni(Cr, 76 Al)2O4·NiO. 77

78 Impedance spectroscopy (IS) is a cheap, sensitive, and non-79 destructive testing method, which has been used extensively to measure the electrical properties of ceramic materials.^{9–11} In 80 the past decade, impedance spectroscopy has been used to 81 reflect the growth of TGO during oxidation and the effect of 82 cracks propagation of the YSZ layer on YSZ electrical proper-83 ties. For example, Zhang et al.¹² have used impedance spec-84 to measure the relationship between the 85 troscopy microstructure of a top coat and its electrical properties, as 86 well as thickness and compositional changes of the TGO in 87

environments with different oxygen partial pressures at 1050 °C for TBCs produced by EB-PVD. Wang et al.¹³ have studied the TGO growth in APS TBCs after oxidation in air at 1100 °C. Ali et al.¹⁴ have also adopted impedance spectroscopy to investigate the relevance between the microstructure and electrical properties of TGO in APS TBCs oxidation in air at 1150 °C.

At present, all of the studies are aimed at APS TBCs and 95 EB-PVD TBCs and the great mass of oxidation temperatures 96 of TBCs above 950 °C. The present work is the first to evaluate 97 the oxidation behaviors of PS-PVD TBCs in air at 950 °C 98 using impedance spectroscopy. The electrochemical physical 99 model and equivalent circuit of PS-PVD TBCs are established 100 in the work. IS analysis of PS-PVD TBCs has shown four 101 relaxation processes, where three of them correspond to YSZ 102 grains, YSZ grain boundaries, the TGO, and the metal elec-103 trode effect. Furthermore, the oxidation kinetics and behaviors 104 of pretreated and un-pretreated PS-PVD TBCs are compared. 105

2. Experimental

2.1. Sample preparation

Nickel-based super-alloy K417 was cut into columnar speci-108 mens with a dimension of \emptyset 25.3 mm × 6 mm as substrates, 109 which were grit-blasted before PS-PVD. The composition of 110 K417 is given in Table 1. A NiCoCrAlYTa bond coating 111 (about 100 µm) was deposited by low-pressure plasma spray 112 (LPPS) (Guangdong Institute of New Materials, Guangzhou, 113 China) on the surface of superalloy. The preparation of all 114 7YSZ coatings (about 300 µm) was carried out on a PS-PVD 115 system (Sulzer-Metco Multicoat[™], Switzerland) onto pre-116 treated and un-pretreated bond coatings, respectively. The 117 process of surface pretreatment for bond coating is based on 118 the following procedures. Firstly, the surface of bond coating 119 was polished using a polishing machine (Tegramin-25, 120 Struers). Subsequently, the surface of bond coating was heated 121 to 1000 °C through the plasma plume of PS-PVD for 10 min-122 utes in a vacuum chamber with addition of a certain amount of 123 oxygen. The surface temperature of the substrate was main-124 tained at 850 °C to 1000 °C in the process of fabricating 125 7YSZ coatings. The powder information and spraying param-126 eters of PS-PVD are given in Tables 2 and 3, respectively. 127

2.2. High-temperature oxidation testing

All specimens were heated in air to $950 \,^{\circ}$ C with a rate of 10 129 K·min¹ and isothermally oxidized for 10, 50, 100, 150, and 130 250 h, respectively. Three specimens were used for impedance 131 measurements and two for microstructure analysis for each 132 oxidation condition. 133

2.3. Characterization 134

Impedance measurements were implemented at 400 °C using135an Ametek Parstat 4000 electrochemical impedance spectroscopy (EIS) analyzer (Parstat4000, AMETEK Inc, USA),137which was controlled by a computer. Spectra analysis was performed with the help of Zview impedance analysis software to139obtain the electrical properties of TBCs. In this measurement,140

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