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7YSZ coating prepared by PS-PVD based on heterogeneous nucleation

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Abstract Plasma spray-physical vapor deposition (PS-PVD) as a novel coating process based on low-pressure plasma spray (LPPS) has been significantly used for thermal barrier coatings (TBCs). A coating can be deposited from liquid splats, nano-sized clusters, and the vapor phase forming different structured coatings, which shows obvious advantages in contrast to conventional technologies like atmospheric plasma spray (APS) and electron beam-physical vapor deposition (EB-PVD). In addition, it can be used to produce thin, dense, and porous ceramic coatings for special applications because of its special characteristics, such as high power, very low pressure, etc. These provide new opportunities to obtain different advanced microstructures, thus to meet the growing requirements of modern functional coatings. In this work, focusing on exploiting the potential of gas-phase deposition from PS-PVD, a series of 7YSZ coating experiments with various process conditions was performed in order to better understand the deposition process in PS-PVD, where coatings were deposited on different substrates including graphite and zirconia. Meanwhile, various substrate temperatures were investigated for the same substrate. As a result, a deposition mechanism of heterogeneous nucleation has been presented showing that surface energy is an important influencing factor for coating structures. Besides, undercooling of the interface between substrate and vapor phase plays an important role in coating structures.

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The plasma spray-physical vapor deposition (PS-PVD) process

has been developed with the aim of depositing different struc-

tured functional coatings (such as thin, gas tight, columnar

coatings, etc.) with large area coverage by plasma spray.¹⁻³

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1. Introduction

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The PS-PVD is developed based on low-pressure plasma spray 27 28 (LPPS), where electrical current up to 3000 A, plasma gas flow up to 200 L/min, and an input power level of 180 kW could be 29 achieved. With operation pressure decreasing, the plasma 30 plume expands to a length of more than 2200 mm and a diam-31 eter of 400 mm. Using appropriate parameters, it is possible to 32 evaporate powder feedstock materials, resulting in variant 33 microstructures and non-line-of-sight deposition.^{4,5} Li et al.⁶ 34 reported that using an advanced PS-PVD process, the nano-35 hardness and micro-hardness of prepared dense coatings were 36 37 markedly higher than those of conventional YSZ coatings (i.e., 38 coatings fabricated by atmospheric plasma spray or electron 39 beam-physical vapor deposition), even comparable with those of sintered YSZ polycrystal. Goral et al.⁷ presented that 40 columnar YSZ coatings were deposited from evaporating pow-41 ders during the process of PS-PVD. The microstructures of 42 coatings were affected by the feed rate, chamber pressure, sam-43 44 ple rotation rate, and plasma gas ratio (Ar, He, and N₂). Hos-45 pach et al.⁸ produced columnar-structured YSZ coatings with a diameter between 20 and 750 µm through PS-PVD. Thinner 46 47 and thicker coatings seem to be possible. The geometry and arrangement of a sample and the sample holder have a big 48 influence on the coating quality. 49

Despite much investigation about PS-PVD have been done 50 51 in the past few years, however, the basic process technology, 52 such as heating of powder particles to spray-deposited molten, 53 semi-molten droplets or vapor gas phase onto substrate surface, has remained essentially the same.^{9,10} There are still a 54 55 lot of areas which have not been investigated thoroughly. These areas consist of particle-plasma interactions in the rar-56 efied plasma, particle vaporization and its affect on plasma 57 58 properties, and deposition mechanisms associated with differ-59 ent microstructures. Now, any further understanding in each of these areas will enable the spray community to more easily 60 61 apply the PS-PVD technology to meet emerging coating challenges.11,12 62

To exploit the potential such as gas-phase deposition by PS-63 PVD, the deposition mechanisms and their dependency on 64 process conditions must be better understood. The PS-PVD 65 process can be summarized as three steps⁹: (A) feedstock pro-66 67 cessing in plasma torch; (B) plasma jet formation and materials transport; and (C) deposition and coating growth. The 68 third step mainly includes heterogeneous and homogeneous 69 nucleation depending on the spray distance.¹³ When the spray 70 distance is set at the middle area in the axial direction of the 71 72 plasma flame, coating deposition primarily relies on heteroge-73 neous nucleation on the substrate surface. In this work, different structured coatings based on heterogeneous nucleation 74 have been obtained, and these principles are summarized in 75 this investigation. 76

77 2. Experimental procedure

The experimental set-up is based on an Oerlikon-Metco Mul-78 ticoat [™] PS-PVD system together with an O3CP plasma torch 79 80 mounted on a robot manipulator of ABB inside a 81 \emptyset 2.5 m × 4.5 m vacuum chamber. The PS-PVD system is obtained through a comprehensive reconstruction of an exist-82 ing conventional LPPS system. In particular, the system is 83 equipped with an additional vacuum pumping unit, a large 84 vacuum blower to provide sufficient pumping capacity at low 85

pressures and enhanced cooling capacity, additional power sources, a new torch transfer arc system, and new operational control units. In terms of the powder feeding system, two powder injectors are located in the cylindrical section of the O3CP nozzle (diameter = 12.5 mm) close to the divergent part.

Feedstock agglomerated 7YSZ powders (Metco 6700, Oerlikon-Metco) were used and their grain sizes ranged from 5 to 22 μ m. 7YSZ coatings were deposited on graphite, sintered zirconia, and nickel-based superalloy K417 substrates (size \emptyset 25.4 mm × 5 mm and surface roughness < 2 μ m) at a spray distance of 950 mm, where the Ar-He hybrid plasma was operated at a 67 kW net power of O3CP under an operation pressure of 150 Pa. Meanwhile, the substrate pre-heating temperatures were controlled at 850 °C and room temperature prior to deposition of 7YSZ coating, respectively. During the pre-heating or deposition process, the substrate remained still, while the plasma gun moved at a speed of 1000 mm/s. The detailed spray parameters are shown in Table 1.

The microstructures of 7YSZ coatings were characterized by field emission-scanning electron microscopy (FE-SEM, Nova-Nono430, FEI) and transmission electron microscopy (TEM, JEM2100F, JEOL). Additionally, before TEM characterization, test samples were prepared by focused ion beam (FIB, 450S, FEI) milling.

3. Results and discussion

3.1. Variation of coating microstructure

3.1.1. Effect of substrate materials

Taking graphite as a substrate without pre-heating by plasma 113 flame (namely, the substrate temperature is controlled at room 114 temperature), 7YSZ coatings prepared by PS-PVD show typi-115 cal columnar microstructure, as seen in Fig. 1(a). The interface 116 between the coating and the substrate has good bonding with-117 out crack. Among columns, there exist different sizes of gaps. 118 Between columns, many small particles were formed, which 119 were resulted from condensation of the vapor phase^{14,15}, as 120 shown in the magnified Fig. 1(b). When the graphite substrate 121 was replaced by sintered zirconia, similar columnar 7YSZ 122 coating was generated with the same parameters at room tem-123 perature, as presented in Fig. 2(a) and (b). According to the 124 result of comparison, it can be known that the horizontal 125 width of a single column is larger than that of a column depos-126 ited on graphite. Moreover, the deposition rate of 7YSZ on 127 graphite is higher than that on sintered zirconia due to differ-128 ent thermal conductivities between graphite $(129 \text{ W}/(\text{m}\cdot\text{K}))$ 129 and zirconia $(2.2 \text{ W}/(\text{m}\cdot\text{K}))$. During the deposition process, 130 the temperature gradient on graphite is larger than that on zir-131 conia, which results in a higher growth driving force on gra-132 phite. Thus, graphite has a higher deposition rate. Due to 133 similar properties, the interface between 7YSZ coating and sin-134 tered zirconia has a better bonding. Between columns, there is 135 no apparent vertical gap, and no small particle as well appears 136 in the gaps. Besides, the columns are made of fine grains and 137 denser than those deposited on graphite. 138

3.1.2. Effect of substrate temperature

With a net power of 60 kW, the inner O3CP gun provided high140plasma energy density, and the electron excitation temperature141is more than 10,000 °C, so that most of the 7YSZ powders can142

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