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# The influence of pre-compression on the creep deformation and failure behaviour of Type 316H stainless steel \*



A. Mehmanparast<sup>a,\*</sup>, C.M. Davies<sup>a</sup>, D.W. Dean<sup>b</sup>, K.M. Nikbin<sup>a</sup>

<sup>a</sup> Department of Mechanical Engineering, Imperial College London, South Kensington Campus, London SW7 2AZ, UK <sup>b</sup> EDF Energy, Barnett Way, Barnwood, Cloucester GL4 3RS, UK

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#### ABSTRACT

The influence of pre-compression to 8% plastic strain at room temperature has been examined on the tensile properties, uniaxial creep deformation, creep crack initiation and growth behaviour of Type 316H stainless steel at 550 °C. Uniaxial creep and crack growth tests have been performed on the pre-compressed (PC) material and the results compared to existing long term (>15,000 h) and short term test data on as-received (AR) (i.e. uncompressed) material. Pre-compression has been found to increase the materials subsequent vield stress in tension. Therefore the extent of non-linearity observed on the load-displacement curves of uniaxial creep rupture and crack growth tests on PC material is limited compared to AR material. In addition pre-compression causes a significant reduction in creep ductility and rupture time, although similar average and minimum creep strain rates are found in PC and AR materials. The creep crack growth (CCG) data on PC and AR materials have been characterised using the steady state creep  $C^*$  parameter employing appropriate validity criteria and geometry dependent fracture mechanics parameter solutions. The CCG results are compared to the creep crack growth prediction models. Based on the creep properties, creep ductility and metallurgical observations of the fracture behaviour of the AR and PC materials, it has been shown that short term creep crack growth tests on PC material may be used to predict long term creep crack initiation (CCI) and CCG behaviour of AR material at 550 °C.

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#### 1. Introduction

The creep deformation and crack growth behaviour of Type 316H stainless steel (SS) is of great concern, particularly due to its widespread use in the UK's advanced gas cooled reactor (AGR) power stations. Recently there has been significant interest in characterising the influence of plastic pre-strain on the creep deformation and crack growth properties in 316H SS [1–3] and other materials [4] since pre-compression is often employed to introduce tensile residual stress fields in a fracture sample [5] and plastic strain is generated during component fabrication processes such as bending and welding. The influence of prior plastic pre-straining, introduced in tension, on the subsequent creep deformation behaviour of a range of materials has been examined in [6–8] and smaller values of creep ductility are found in the pre-conditioned material compared to the asreceived (AR) material. It has also been shown in [6] that at a given temperature and stress, the creep ductility reduces as the

Corresponding author.
E-mail address: ali.mehmanparast@imperial.ac.uk (A. Mehmanparast).

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Nomenclature	
а	crack length
a a	initial crack length
a (or da/	dt) creep crack growth rate
a <sub>NSW-MOD</sub>	creep crack growth rate predicted by NSW-MOD model
$\Delta a$	increment of crack growth
A	creep stress coefficient in minimum creep strain law
A <sub>4</sub>	creep stress coefficient in average creep strain law
B	specimen thickness
B <sub>n</sub>	net specimen thickness between side grooves
Br	coefficient of rupture law
C*	steady state creep fracture mechanics parameter
D	constant coefficient in creep crack growth correlation with C*
Ε	elastic (Young's) modulus
E'	effective Young's modulus
In	non-dimensional function of <i>n</i>
K	stress intensity factor
п	creep stress exponent in minimum creep strain law
n <sub>A</sub>	creep stress exponent in average creep strain law
P	load
r <sub>c</sub>	creep process zone
t <sub>0.2</sub>	time for 0.2 mm crack extension
t <sub>0.5</sub>	time for 0.5 mm crack extension
t <sub>f</sub>	creep crack growth test duration
t <sub>i</sub>	initiation time
$t_{red}^{eng}$	engineering definition of the redistribution time
t <sub>r</sub>	time to rupture in a uniaxial creep rupture test
$t_T$	transition time from small scale creep to widespread creep conditions
W	specimen width
η	geometry function in C* relation
$\epsilon_{f}$	uniaxial creep strain at failure (creep ductility)
$\widetilde{\mathcal{E}}_{f}^{*}$	multiaxial creep ductility
E <sub>e</sub>	non-dimensional function of $\theta$ and $n$
$\mathcal{E}_{s}$	steady state creep strain rate
ε <sub>A</sub>	average creep strain rate
v <sub>r</sub>	even a correlation of creen crack growth rate with C*
$\varphi$	load line displacement
4-	creen load line displacement
$\Delta_c$	elastic load line displacement
Å	load line displacement rate
$\dot{\Lambda}^{c}$	component of displacement rate directly associated with the accumulation of creep strains
$\overline{\dot{\Lambda}}^i$	component of displacement rate directly associated with instantaneous (elastic and plastic) strains
$\overline{\dot{\Lambda}}_{a}^{i}$	component of displacement rate directly associated with instantaneous elastic strains
$\dot{\Delta}^{\tilde{T}}$	total load line displacement rate
$\sigma$	stress
$\sigma_{0.2}$	0.2% proof stress
$\sigma_{ref}$	reference stress
θ	crack tip angle
AR	as-received material
CCG	creep crack growth
CCI	creep crack initiation
LLD	load line displacement
NSW	Nikbin, Smith and Webster creep crack growth model
NSW-MO	DD modified version of the NSW creep crack growth prediction model
PC	pre-compressed material
PE	plane strain
PS	plane stress
ROA	reduction of area
SS	stainiess steel
015	uitimate tensile strength

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