

# First-principles calculations of phonon and thermodynamic properties of OsSi<sub>2</sub>



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## ABSTRACT

The structure, lattice dynamics, and some thermodynamic properties of orthorhombic OsSi<sub>2</sub> were investigated using a first-principles density functional theory (DFT). Linear response theory was used to calculate the phonon dispersion relation and phonon density of states for OsSi<sub>2</sub> as well as its infrared and Raman active mode frequencies. In this study, the thermodynamic properties, including vibrational entropy ( $S_{vib}$ ), constant-volume specific heat ( $C_v$ ), and Debye temperature ( $\Theta_D$ ), were predicted theoretically and discussed.

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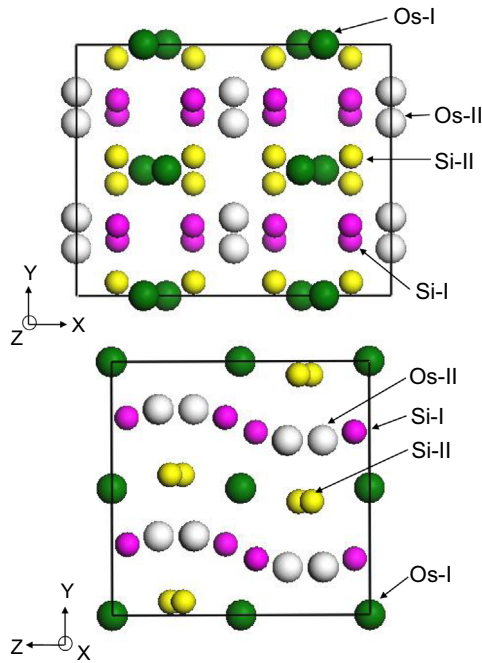
## 1. Introduction

Semiconducting transition silicides have gained considerable attention as potential candidates for optoelectronics, photovoltaics, and thermoelectrics [1]. In particular, to date, iron disilicide ( $\beta$ -FeSi<sub>2</sub>), an environmentally friendly, non-toxic, low-cost material made from abundant elements found in the Earth's crust, has been extensively investigated [2–5]. Despite the fact that osmium disilicide (OsSi<sub>2</sub>) has a crystal structure identical to that of  $\beta$ -FeSi<sub>2</sub>, very few experimental and theoretical studies have been reported for OsSi<sub>2</sub> [6–11]. Fig. 1 shows the crystal structure of OsSi<sub>2</sub>, containing 16 Os and 32 Si atoms per unit cell. The crystal structure of OsSi<sub>2</sub> belongs to a base-centered orthorhombic in the *Cmca* (No. 64) space group, with lattice constants of  $a=10.1496$  Å,  $b=8.1168$  Å, and  $c=8.2230$  Å [12]. There are two crystallographically inequivalent sites (Os<sub>I</sub>, Os<sub>II</sub>, Si<sub>I</sub>, and Si<sub>II</sub>) for both the Os and Si atoms in OsSi<sub>2</sub>. Both the Os<sub>I</sub> and Os<sub>II</sub> atoms are surrounded by 8 nearest-neighbor Si atoms with slightly different distances, and have the crystallographic point symmetries 2 ( $C_2$ ) and  $m$  ( $C_s$ ), respectively. Migas et al. [9] reported on the optical properties of isostructural  $\beta$ -FeSi<sub>2</sub>,

OsSi<sub>2</sub>, and two ternary compounds, Os<sub>0.5</sub>Fe<sub>0.5</sub>Si<sub>2</sub> and Fe<sub>0.5</sub>Os<sub>0.5</sub>Si<sub>2</sub>, in which the Fe<sub>I</sub> or Fe<sub>II</sub> sites were replaced by Os in the  $\beta$ -FeSi<sub>2</sub> structure, respectively, according to the first-principles calculations. They noted the potential for band-gap engineering by composition owing to the Os<sub>0.5</sub>Fe<sub>0.5</sub>Si<sub>2</sub> having a predicted direct band-gap, while  $\beta$ -FeSi<sub>2</sub>, OsSi<sub>2</sub>, and Fe<sub>0.5</sub>Os<sub>0.5</sub>Si<sub>2</sub> have indirect band-gaps. Moreover, the elastic properties such as elastic constants, bulk modulus, shear modulus, and Young's modulus of OsSi<sub>2</sub> were investigated using the first-principles calculations [10,11].

In our previous paper, we reported the detailed lattice-dynamical properties of  $\beta$ -FeSi<sub>2</sub> based on first-principles calculations [13]. The calculated vibrational entropy, specific heat, and Debye temperature of  $\beta$ -FeSi<sub>2</sub> from these calculations are in agreement with previous experimental data. At a finite temperature, numerous physical properties of solids are greatly influenced by the solid's crystal vibration. Accordingly, it is vital to investigate the vibrational properties of OsSi<sub>2</sub>. However, as far as we know, few experimental and theoretical studies on the phonons of OsSi<sub>2</sub> have been reported. In this study, we report on the detailed lattice-dynamical properties of OsSi<sub>2</sub>, including the phonon dispersion relations, the phonon density of states, the infrared (IR) and Raman active phonon frequencies, the Born effective charge, and the thermodynamic properties, based on first-principles calculations.

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**Fig. 1.** The  $\text{OsSi}_2$  unit cell. The large spheres represent Os and the small spheres represent Si. There are two crystallographically inequivalent sites for both types of atoms ( $\text{Os}_I$ ,  $\text{Os}_{II}$ ,  $\text{Si}_I$ , and  $\text{Si}_{II}$ ).

## 2. Computational details

We used the CASTEP (Accelrys Inc.) software package [14] to perform the density-functional theory (DFT) plane-wave pseudopotential calculations within local density approximations (LDA). The norm-conserving pseudopotential, generated using the optimization scheme of Lin et al. [15], was utilized. The valence electron configurations of the Os and Si atoms were  $5d^66s^2$  and  $3s^23p^2$ , respectively. The Ceperley–Alder form of the exchange correlation potential, as parameterized by Perdew and Zunger [16], was used in the LDA. A base-centered orthorhombic primitive cell having 24 atoms (4  $\text{Os}_I$ , 4  $\text{Os}_{II}$ , 8  $\text{Si}_I$ , and 8  $\text{Si}_{II}$ ) was built. The total energy of the unit cell converged to better than 1 meV/atom using the valence electron wave functions in a plane wave basis, with a kinetic energy cutoff of 560 eV. The density mixing method was used for electronic minimization. In the geometry optimization and lattice-dynamical calculations, the Brillouin-zone integrals were performed using the Monkhorst–Pack grids [17] of a  $3 \times 3 \times 2$  mesh with a  $0.05 \text{ \AA}^{-1}$  k-point spacing. The linear response method [18–20] was applied after the geometrical optimization to calculate phonons, including the longitudinal optical/transverse optical (LO/TO) splitting. A convergence of the calculated phonon frequencies was better than  $1 \text{ cm}^{-1}$ .

## 3. Results and discussion

Table 1 shows the calculated lattice parameters and fractional atomic coordinates of the  $\text{OsSi}_2$  in this study, compared with those obtained from previous experiments [12]. The difference in the three lattice parameters  $a$ ,  $b$ , and

$c$ , between the calculated values herein and those obtained from previous experiments are  $-0.8\%$ ,  $-0.9\%$ , and  $-0.9\%$ , respectively, within the range of typical errors in LDA. The calculated fractional atomic coordinates ( $x$ ,  $y$ ,  $z$ ) of the  $\text{Os}_I$ ,  $\text{Os}_{II}$ ,  $\text{Si}_I$ , and  $\text{Si}_{II}$  atoms are consistent with previous experimental results [12].

Fig. 2 shows the calculated phonon dispersion relations of  $\text{OsSi}_2$ . The special high-symmetry points of the Brillouin zone (BZ) are denoted by  $\Gamma(0, 0, 0)$ ,  $Z(0, 0, 0.5)$ ,  $T(-0.5, 0.5, 0.5)$ ,  $Y(-0.5, 0.5, 0)$ ,  $S(0, 0.5, 0)$ , and  $R(0, 0.5, 0.5)$ . The phonon frequencies are obtained at these special high-symmetry points, as well as at other intermediate points. All of the calculated phonon frequencies are positive, suggesting that the  $\text{OsSi}_2$  is dynamically stable throughout the BZ. It is clear that the some frequencies at the  $\Gamma$  point depend strongly on direction ( $\mathbf{q} \rightarrow 0$  along  $\Gamma-Z$ ,  $\Gamma-Y$ , and  $\Gamma-S$ ) due to the LO–TO splitting in the anisotropic  $\text{OsSi}_2$  crystal.

Table 2 shows the calculated phonon frequencies from this study of the  $\text{OsSi}_2$  at the  $\Gamma$  point. As the primitive cell contains 24 atoms, there are 72 vibrational modes at the  $\Gamma$  point: 3 acoustic and 69 optical. The  $Cmca$  space group has the crystal point group  $D_{2h}$  in the Schoenflies notation [21]. Therefore, the phonon modes of  $\text{OsSi}_2$  can be classified as 3 acoustic ( $B_{1u} + B_{2u} + B_{3u}$ ), 36 Raman-active ( $9A_g + 9B_{1g} + 9B_{2g} + 9B_{3g}$ ), 25 IR-active ( $9B_{1u} + 9B_{2u} + 7B_{3u}$ ), and 8 silent modes ( $8A_u$ ) [13,22]. The IR-active phonon mode frequencies ( $B_{1u}$ ,  $B_{2u}$ , and  $B_{3u}$ ) are split by the LO–TO splitting, which is computed based on the dynamic Born effective charge [23].

The calculated dynamic Born effective charge tensors ( $Z^*$ ) and static Mulliken charges for the  $\text{Os}_I$ ,  $\text{Os}_{II}$ ,  $\text{Si}_I$ , and  $\text{Si}_{II}$  of the  $\text{OsSi}_2$  are shown in Table 3. All of the off-diagonal components of the  $Z^*$  for  $\text{Os}_I$ ,  $\text{Os}_{II}$ ,  $\text{Si}_I$ , and  $\text{Si}_{II}$  come to zero or negligible values. On the other hand, the diagonal elements of the  $Z^*$  for the  $\text{Os}_I$ ,  $\text{Os}_{II}$ ,  $\text{Si}_I$ , and  $\text{Si}_{II}$  are not symmetric ( $Z^*_{xx} \neq Z^*_{yy} \neq Z^*_{zz}$ ) and have large values,

**Table 1**

Comparison between  $\text{OsSi}_2$ 's structural properties calculated in this study with those obtained in previous experiments [12]. The deviations from the experimental data are shown in parentheses.

| Structural parameters   | Calculated            | Experimental [12]  |         |        |
|-------------------------|-----------------------|--------------------|---------|--------|
| Cell parameters         | $a$ (Å)               | 10.0660<br>(−0.8%) | 10.1496 |        |
|                         | $b$ (Å)               | 8.0445 (−0.9%)     | 8.1168  |        |
|                         | $c$ (Å)               | 8.1487 (−0.9%)     | 8.2230  |        |
|                         | $V$ (Å <sup>3</sup> ) | 659.85 (−2.6%)     | 677.43  |        |
|                         | Atomic coordinates    | $\text{Os}_I$ (8d) | $x$     | 0.2147 |
| $y$                     |                       |                    | 0       | 0      |
| $z$                     |                       |                    | 0       | 0      |
| $\text{Os}_{II}$ (8 f)  |                       | $x$                | 0       | 0      |
|                         |                       | $y$                | 0.1881  | 0.1881 |
|                         |                       | $z$                | 0.1818  | 0.1812 |
| $\text{Si}_I$ (16 g)    |                       | $x$                | 0.3717  | 0.3699 |
|                         |                       | $y$                | 0.2222  | 0.2208 |
|                         |                       | $z$                | 0.0588  | 0.0591 |
| $\text{Si}_{II}$ (16 g) |                       | $x$                | 0.1270  | 0.1280 |
|                         |                       | $y$                | 0.0526  | 0.0534 |
|                         |                       | $z$                | 0.7241  | 0.7252 |

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