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Electrically-driven metal-insulator transition of vanadium dioxide thin films in a metal-oxide-insulator-metal device structure



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ABSTRACT

We report the successful growth of vanadium dioxide (VO₂) films on SiO₂ buffered metal electrode and the fabrication of metal–oxide-insulator–metal (MOIM) junction. The VO₂ film has an abrupt thermal-induced metal–insulator transition (MIT) with a change of resistance of 2 orders of magnitude. The electrically-driven MIT (E-MIT) switching characteristics have been investigated by applying perpendicular voltage to VO₂ based MOIM device at particular temperatures, sharp jumps in electric current were observed in the *I–V* characteristics under a low threshold voltage of 1.6 V. The Ohmic behavior, non-Ohmic super-linear one, and metallic regime are sequentially observed in the MOIM device with the increase of voltage. It is expected to be of significance in exploring ultrafast electronic devices incorporating correlated oxides based MOIM structure.

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1. Introduction

Vanadium dioxide (VO₂) is a phase transition material which undergoes a reversible metal–insulator transition (MIT) near 68 °C accompanied by a lattice structural transition from the monoclinic to tetragonal, an abrupt change of conductivity, and significant changes of the optical properties at almost all wavelengths [1–6]. Furthermore, the MIT of VO₂ can also be triggered by external stimuli such as an applied voltage or optical excitation [7,8]. Particularly, electrically triggered phase transition (E-MIT) in vanadium dioxide is of interest in novel devices

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http://dx.doi.org/10.1016/j.mssp.2014.06.030 1369-8001/© 2014 Elsevier Ltd. All rights reserved. for electric switches, resistance random access memory (ReRAM) networks, and so on [9–12].

Recently, out-of-plane metal–VO₂–semiconductor (metal) structures have attracted much attention [13–16]. It is recently reported that with particular perpendicular structures, for example metal/VO₂/metal structure, the device size can be significant reduced to submicron scale and the OFF (insulating state)/ON (metallic state) switch time can be improved to less than 2 ns [13]. The ultrafast E-MIT is believed to be induced by electronic correlation effects rather than the joule heating, because the heating driven MIT would give rise to a longer switch time [17]. However, the current-driven joule heating effect in a metal/VO₂/metal structure is inevitable once the metallic state of VO₂ is established, which will hinder the On–OFF process of the device. In this work, we proposed a metal–oxide-insulator–metal (MOIM) structure by introducing a thin SiO₂ insulating

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layer between the VO₂ film and the bottom metal layer. With the SiO₂ layer, the current value will be small both before and after the MIT of VO₂, thus the Joule heating effect can be effectively depressed. Furthermore, the SiO₂ buffer layer can eliminate the stress between the VO₂ film and the metal electrode thus can improve the quality of VO₂ film. SiO₂ has excellent thermal, mechanical and optical properties, and is compatible with micro-electromechanical devices, which have great application in the semiconductor process [18,19]. All these features indicate that the SiO₂ based MOIM structure is very desirable in ultrafast electronic devices, and can find applications in optoelectronics and communications technologies.

2. Experimental details

Si/Pt substrate (actually Si/SiO₂/Ti/Pt) were prepared by depositing Ti (20 nm)/Pt (200 nm) layers onto Si/SiO2 using electron-beam evaporation. The SiO₂ buffer was grown directly on Si/Pt substrate using plasma enhanced chemicalvapor-deposition (PECVD). The synthesis conditions of SiO₂ is as following: the substrate temperature (*T*) is fixed at 300 $^{\circ}$ C, the working pressure is 600 mTorr, RF source gun power is 60 W, mixed gas flows: $SiH_4 = 150$ sccm, $N_2O:140$ sccm, N₂:100 sccm. VO₂ thin films were synthesized by using RF magnetron sputtering with a high purity vanadium (V) target (99.99% in purity) on the SiO_2 buffer. A 100 $\mu m \times$ 100 μm Au patch was finally deposited on top of the VO₂ to form the MOIM structure. For VO₂ deposition, the temperature of the substrate and the RF source gun power were set as 550 °C and 200 W, respectively. The operation gas was a mixture of high purity Ar (99.999% pure) and O₂ (99.999% pure). The sputtering pressure was set at 1.0 Pa with various oxygen partial pressures of 3%, 5% and 7% (correspondingly the samples are labeled as 1, 2, and 3, respectively). No post-growth annealing process was carried out.

The crystallography properties of the film were characterized by an x-ray diffractometer (XRD: DX-2700) using Cu Ka radiation. Surface and cross-section morphology of the film were studied by using atomic force microscope (AFM: SPA-300HV) and field-emission scanning electron microscope (FESEM: JSM-7600F). The temperature dependence of electric resistivity for the VO₂ film was measured by a standard four-point measurement method based on a Keithley 2400 sourcemeter, where the sample temperature was controlled by a thin Peltier heater/cooler. Current–voltage (I–V) curve measurements were carried out by using Agilent 4156C to investigate the electrical driven phase transition properties.

3. Results and discussion

Typical XRD patterns were obtained for the bare Si/Pt substrate and the Si/Pt/SiO₂/VO₂ samples deposited in the oxygen partial pressure of p=3%, 5% and 7%, as plotted in Fig. 1. For all the samples, the strong peak at $2\theta=39.7^{\circ}$ is indexed to the diffraction from the Pt (1 1 1) plane. The crystal planes diffraction peaks of the VO₂ films show an intimate dependence on the deposition oxygen pressure. The films grown at p=3% have a weak peak near 27.9°, which corresponds to the (0 1 1) plane of VO₂ phase. As to

Fig. 1. X-ray diffraction (XRD) patterns of the bare Si/Pt and Si/Pt/SiO₂/ VO_2 deposited in the different oxygen partial pressure of 3%, 5% and 7% at the temperature of 550 °C.

O₂/(Ar+O₂): 7%

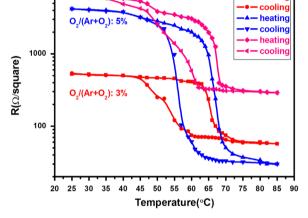
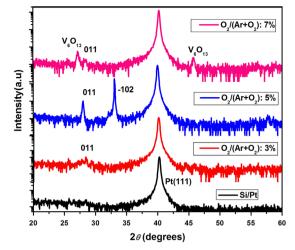


Fig. 2. Resistivity versus temperature (R-T) curves for the Si/Pt/SiO₂/VO₂ samples deposited in the different oxygen partial pressure of 3%, 5% and 7% at the temperature of 550 °C.

the sample deposited at p=5%, two distinct peaks are observed at 27.9° and 33.2°, referring to the (0 1 1) and (-102) planes of VO₂ phase. These results suggest that pure polycrystalline VO₂ phase is formed in sample-2, which is in agreement with the previous reports [20,21]. For the sample-3 grown at p=7%, however, the dominant phase in the film is V₆O₁₃ and only a weak peak of VO₂ (0 1 1) is detected. Anyway, the XRD results confirm that the VO₂ films have been successfully grown on SiO₂ buffered Pt metal film especially at the oxygen partial pressure of 5%.

To confirm the MIT properties of the VO₂ films deposited in the different oxygen partial pressure, the temperature dependence of the electrical resistance were measured and plotted in Fig. 2. An abrupt change of resistance near the transition temperature T_c of 66 °C is clearly observed, which indicates the presence of the VO₂ phase in all the three films. For sample-2, the resistance at low temperature (25 °C) is

- heating



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