



Fine-tuning the structure of cubic indium oxide and their ethanol-sensing properties

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ABSTRACT

Fine-tuning the structure of cubic indium oxide (In_2O_3) was carried out with different calcination temperatures and atmospheres. The structural properties of the products were characterized by X-ray diffraction (XRD), transmission electron microscopy (TEM), Raman spectroscopy, electron paramagnetic resonance spectroscopy (EPR) and X-ray photoelectron spectroscopy (XPS). Ethanol-sensing performance of the products was investigated, and the depletion layer theory was used to explain the sensing mechanism. The experimental results indicated that the ethanol-sensing responses of In_2O_3 increased with the increasing intensity of chemisorbed oxygen, and decreased with the increasing intensity of donor defects. The In_2O_3 annealed at 300°C or in oxygen had higher responses because of more chemisorbed oxygen. However, the In_2O_3 annealed at 600°C or in nitrogen had lower responses due to more donor defects (V_o and In_i). In situ diffuse reflectance infrared spectroscopy (DRIFTS) study revealed that surface reduction and reconstruction of In_2O_3 happened at room temperature upon ethanol adsorption, which may lead our results contradicted previous reports.

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1. Introduction

Indium oxide (In_2O_3) is an important n-type semiconductor with good gas-sensing properties, which makes it a promising candidate for detecting H_2S , CO , NO_2 and ethanol [1–4]. In recent years, lot of work had been focused on In_2O_3 nanomaterials, which exhibited remarkable gas-sensing performance because of their small size and large surface area. Currently, nanostructured design of In_2O_3 mainly focused on their shape, size, and elemental composition [5–8]. However, few studies concerned about enhancing gas-sensing properties by tuning the microstructure of In_2O_3 .

Recently, improving the properties of metal oxides by tuning the microstructures, such as defects and the surface activity has attracted intensive attention [9,10]. Overbury et al. explored the nature of ceria nanocrystals with defined nanoshapes, including rod, cube, and octahedral. The experimental results indicated that although the three ceria nanoshapes were well-structured, they also contained defects on the surface and in the bulk that could have a profound effect during ceria catalysis [11]. Cheung et al. reported an approach to enhance the catalytic activity of ceria nanostructures through engineering high density of oxygen vacancy (V_o)

defects [12]. Moreover, Kumar et al. found the intensity of defect could influence the photoluminescence property of In_2O_3 [13].

Herein, the In_2O_3 nanoparticles synthesized by a facile solvothermal method were treated at different calcination temperatures or in different calcination atmospheres to tune their microstructures. Meanwhile, ethanol was used as a probing molecule to study the sensing performance of the In_2O_3 sensors. The sensing mechanism was proposed and the response differences of the In_2O_3 sensors were explained. Finally, the ethanol adsorption behavior and the structure change of In_2O_3 were investigated by in situ diffuse reflectance infrared spectroscopy (DRIFTS).

2. Experimental

2.1. Synthesis of In_2O_3 nanoparticles

All chemicals ($\text{In}(\text{NO}_3)_3$, ethanol, diethylene glycol (DEG) and $\text{CO}(\text{NH}_2)_2$) used in experimentations were analytical reagents and used without purification. Distilled water was used throughout the experiments.

The synthesis of In_2O_3 was carried out according to the previously reported procedure [14]. In a typical procedure, 2 mmol $\text{In}(\text{NO}_3)_3$ and 2 g $\text{CO}(\text{NH}_2)_2$ were added to the mixture of 35 mL DEG and 2 mL distilled water. The obtained solution was stirred for 3 h. Then the mixture was transferred into a Teflon-lined stainless-steel autoclave and maintained at 200°C for 24 h. In_2O_3 powder was collected from the autoclave, and centrifuged and washed

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several times with ethanol and distilled water. Then, the products were dried by using a vacuum oven at 60 °C. The as-formed In_2O_3 nanoparticles were annealed at 300 °C, 400 °C, 500 °C and 600 °C for 3 h in air, and the obtained samples were denoted as IO300, IO400, IO500 and IO600, respectively. In addition, the In_2O_3 nanoparticles were annealed in different atmospheres (O_2 , N_2 and Air) with 30-standard-cubic-centimeter-per-minute (SCCM) gas flow, and the obtained samples were denoted as IO- O_2 , IO- N_2 and IO-Air, respectively.

2.2. Material characterizations

The In_2O_3 samples were characterized by X-ray diffraction (XRD) with a scanning speed of 10°/min, on a Rigaku D/Max2500VB2+/PC diffractometer using graphite monochromatized Cu K α radiation ($\lambda = 1.54056 \text{ \AA}$). The morphologies of the products were characterized by transmission electron microscopy (TEM, Hitachi-800). Raman spectroscopy was recorded from 200 nm to 800 nm by a 514 nm excitation (Invia Raman Spectrometer manufactured by U.K. Renishaw Corporation). X-ray photoelectron spectroscopy (XPS) measurements were performed on a VG Scientific ESCALAB 250 spectrometer. Electron paramagnetic resonance (EPR) spectra were recorded by using a Bruker EMX spectrometer operating at room temperature with a frequency of 9.853 GHz. The center field of the EPR scans was 3510 G with a sweep width of 1500 G. Specific surface area of the samples

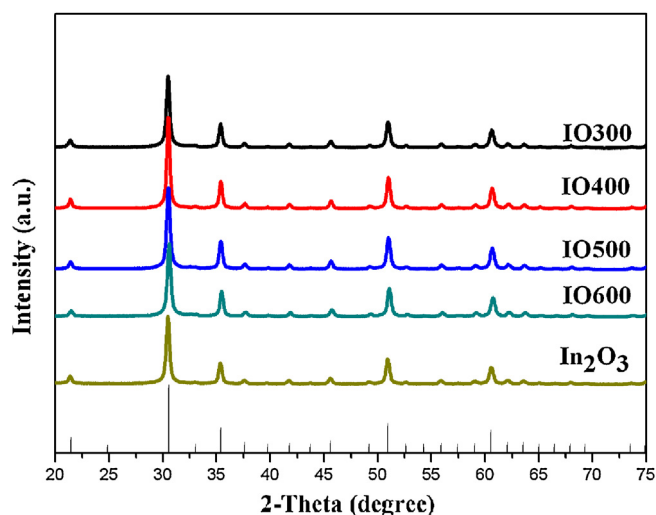


Fig. 1. XRD patterns of the as-formed In_2O_3 and the In_2O_3 annealed at different temperatures.

was measured by nitrogen adsorption (BET analysis, Micromeritics ASAP 2010).

DRIFTS spectra were collected using a BRUKER VERTEX 70 spectrometer. The In_2O_3 sample (~300 mg) was pretreated in He flow (20 mL/min) at 300 °C for 1 h (heating rate 10 °C/min) and then

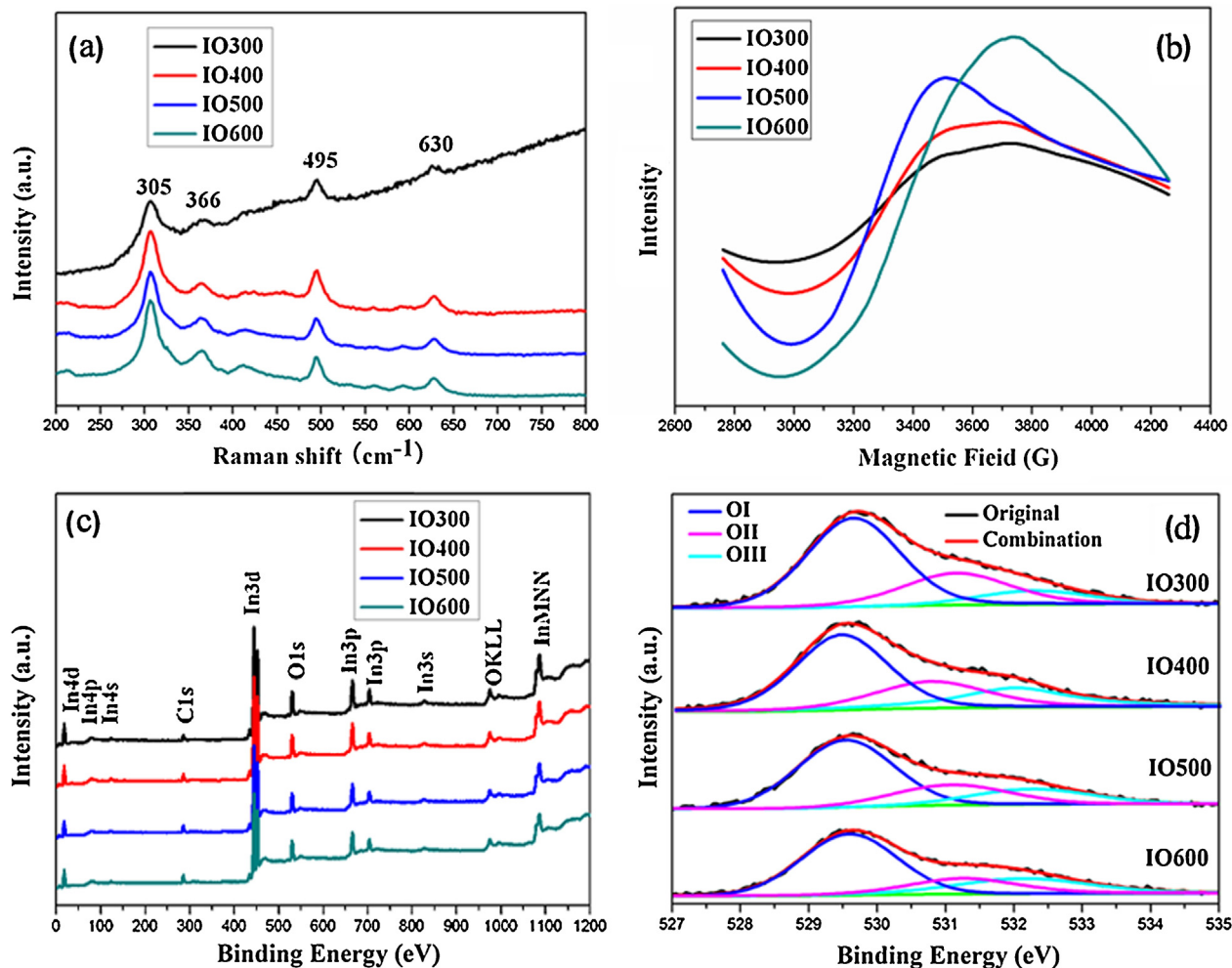


Fig. 2. Raman (a), EPR (b), XPS (c) and high-resolution O1s (d) spectra of In_2O_3 nanoparticles treated at different temperatures.

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