

## Structural characterization and temperature-dependent luminescence of $\text{CaF}_2:\text{Tb}^{3+}/\text{Eu}^{3+}$ glass ceramics

HU Fangfang (胡芳芳), ZHAO Zhangmei (赵张美), CHI Fengfeng (迟逢逢), WEI Xiantao (韦先涛)\*, YIN Min (尹民)

(Key Laboratory of Strongly-Coupled Quantum Matter Physics, Chinese Academy of Sciences, School of Physical Sciences, University of Science and Technology of China, Hefei 230026, China)

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**Abstract:**  $\text{Tb}^{3+}/\text{Eu}^{3+}$  co-doped transparent glass ceramics containing  $\text{CaF}_2$  nanocrystals were successfully synthesized by high temperature melt-quenching method and subsequent heating. The structure and morphology of the samples were investigated by X-ray diffraction (XRD), transmittance electron microscopy (TEM), high resolution TEM (HRTEM) and selected area electron diffraction (SAED). The photoluminescence properties and energy transfer process from  $\text{Tb}^{3+}$  to  $\text{Eu}^{3+}$  of  $\text{CaF}_2:\text{Tb}^{3+},\text{Eu}^{3+}$  phosphors were also investigated through excitation spectra and decay curves. In addition, the emission spectra of the glass ceramics in a wide temperature range from 21 to 320 K were recorded under the excitation of 485 nm. It was found that the fluorescence intensity ratios of  $\text{Tb}^{3+}$  at 545 nm ( $^5\text{D}_4 \rightarrow ^7\text{F}_5$ ) to  $\text{Eu}^{3+}$  at 615 nm ( $^5\text{D}_0 \rightarrow ^7\text{F}_2$ ) was highly temperature-dependent with an approximate linear relationship, and the temperature sensitivity was about 0.4%/K. It is expected that the investigated  $\text{Tb}^{3+}/\text{Eu}^{3+}$  co-doped  $\text{CaF}_2$  glass ceramics may have prospective application in optical thermometry.

**Keywords:**  $\text{CaF}_2:\text{Tb}^{3+}/\text{Eu}^{3+}$ ; glass ceramics; structural characterization; optical thermometry; rare earths

Temperature is one of the most important fundamental physical parameters, and the accuracy measurement of temperature is crucial in many fields of science, engineering and industry. Recently, optical thermometry based on luminescent materials has been extensively investigated for its non-invasive operating mode, high-spatial resolution and quick response<sup>[1–12]</sup>. To achieve high detection sensitivity, many optical temperature sensing techniques have been developed, such as the spectral shift of emission bands, absolute fluorescent intensity, fluorescence lifetime, and the fluorescence intensity ratio<sup>[13–16]</sup>. Furthermore, sensing temperature with intensity ratio of dual emission bands rather than individual emission peak is regarded as a very promising approach, because it can reduce dependence on measurement conditions and improve sensing accuracy.

The  $\text{Tb}^{3+}$  and  $\text{Eu}^{3+}$  co-doped materials have recently received considerable interests for possible applications in temperature sensing, owing to the temperature-dependent energy transfer from  $\text{Tb}^{3+}$  to  $\text{Eu}^{3+}$ . Self-reference temperature determination based on the intensity ratio of emission bands from different ions rather than a single ion is therefore much preferred, which has been demonstrated with e.g.  $\text{Eu}^{3+}$  and  $\text{Tb}^{3+}$  co-doped metal organic frameworks and nanomaterials<sup>[17–21]</sup>. As the  $^5\text{D}_4$  state (20500  $\text{cm}^{-1}$ ) of  $\text{Tb}^{3+}$  is located near the cen-

ter of  $^5\text{D}_2$  (21500  $\text{cm}^{-1}$ ) and  $^5\text{D}_1$  (19000  $\text{cm}^{-1}$ ) states of  $\text{Eu}^{3+}$ , the energy mismatch between these excited states of  $\text{Tb}^{3+}$  and  $\text{Eu}^{3+}$  is much closer to the maximal phonon energy of the host, resulting in a strong interaction between  $\text{Tb}^{3+}$  and  $\text{Eu}^{3+}$ <sup>[22]</sup>. With the participation of several phonons, the multi-phonon-assisted energy transfer from  $\text{Tb}^{3+}$  to  $\text{Eu}^{3+}$  ions is efficient in co-doped samples and its possibility increases rapidly with temperature. Consequently, the intensity ratio between  $^5\text{D}_4 \rightarrow ^7\text{F}_2$  ( $\text{Tb}^{3+}$  at 545 nm) and  $^5\text{D}_0 \rightarrow ^7\text{F}_2$  ( $\text{Eu}^{3+}$  at 615 nm) was temperature dependent, giving high sensitivity to a temperature change.

Oxyfluoride glass ceramics have been widely investigated as host materials for rare earth ions recently because they have the advantages of not only fluorides with comparatively low phonon energies, but also oxide materials with high chemical and mechanical stabilities<sup>[23]</sup>. On the other hand,  $\text{CaF}_2$  is a promising candidate for desired host materials due to its high solubility of both sensitizer and activator rare-earth ions, wide transparent spectral region (about from 0.125 to 10  $\mu\text{m}$ ), good chemical stability, low phonon energy, friendliness to environment, and gives a good match of refractive index with the aluminosilicate glass<sup>[24–28]</sup>.

In this paper, transparent glass ceramics containing  $\text{CaF}_2:5\%\text{Tb}^{3+}/1\%\text{Eu}^{3+}$  nanocrystals were successfully prepared by melt-quenching and subsequent heating. The

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\* **Corresponding author:** WEI Xiantao (E-mail: [wxt@ustc.edu.cn](mailto:wxt@ustc.edu.cn); Tel.: +86-551-63606912)

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fluorescence intensity ratio between the  $\text{Tb}^{3+}:\text{D}_4 \rightarrow \text{F}_5$  transition and the  $\text{Eu}^{3+}:\text{D}_0 \rightarrow \text{F}_2$  one in the glass ceramics was found to be highly temperature-dependent.

## 1 Experimental

The samples were prepared with nominal composition (in mol.%)  $45\text{SiO}_2\text{-}20\text{Al}_2\text{O}_3\text{-}10\text{CaO-}25[(1-x-y)\text{CaF}_2\text{-}x\text{TbF}_3\text{-}y\text{EuF}_3]$  ( $x=0.05, y=0; x=0, y=0.01, 0; x=0.05, y=0.01$ ) by melt-quenching method. The raw materials of  $\text{SiO}_2$  (AR),  $\text{Al}_2\text{O}_3$  (AR),  $\text{CaCO}_3$  (AR) and  $\text{CaF}_2$  (AR) and high purity  $\text{TbF}_3$  (99.99%),  $\text{EuF}_3$  (99.99%) were completely mixed and crashed in an agate mortar. The well ground stoichiometric chemicals were put into an alumina crucible and melted at  $1400\text{ }^\circ\text{C}$  for an hour. The melt was poured into a preheated hot copper mold and then was pressed by another plate to form transparent precursor glass (labeled as PG). The PG was followed by annealing at  $450\text{ }^\circ\text{C}$  for 10 h to release internal stress. Subsequently, PG glass was heat-treated at  $700\text{ }^\circ\text{C}$  for 2 h to form glass ceramics sample, which was labeled as GC700.

Differential scanning calorimetry (DSC) measurement was carried out in a simultaneous thermal analyzer (NETZSCH STA 449F3) at a heating rate of  $10\text{ K/min}$  under  $\text{N}_2$  atmosphere to determine the glass transition ( $T_g$ ) and crystallization ( $T_x$ ) temperatures. The phase structures and the mean crystallite size of samples were identified via the XRD measurement (MAC Science Co. Ltd. Mxp18. AHF, Tokyo, Japan) with nickel-filter  $\text{Cu K}\alpha$  radiation in the range of  $2\theta=10^\circ\text{-}80^\circ$ . The accelerating voltage was  $40.0\text{ kV}$  and the tube current was  $100.0\text{ mA}$ . The microstructure of glass ceramics was analyzed by a HRTEM (FEI Talos F200X, FEI Ltd., Hillsboro, Oregon, United States). The excitation spectra of the samples were recorded with a JY Fluorolog-3-Tou fluorescence spectrophotometer (JobinYvon Ltd., France) using a  $450\text{ W}$  Xenon lamp as the excitation light source. The emission spectra were obtained by a Jobin-Yvon HRD-1 double monochromator equipped with a Hamamatsu R928 photomultiplier. The signals were analyzed by an EG&G 7265 DSP lock-in amplifier and stored into computer memories. For the measurements at low temperature, the powder sample was pressed into a round tablet with thickness of  $0.8\text{ mm}$  and diameter of  $8.0\text{ mm}$ . And the tablet was glued to a copper pedestal with cryogenic glue, then fixed in a closed-cycle cryostat fed by a WC50 helium compressor. Temperature of the sample was controlled over the range of  $21\text{-}300\text{ K}$  by a Lake Shore Model 321 temperature controller.

## 2 Results and discussion

Fig. 1 shows the DSC curve of the glass with  $(x, y)=(0.05, 0.01)$  composition. The glass transition temperature  $T_g$ , the  $\text{CaF}_2$  crystallization temperature  $T_x$ , and the

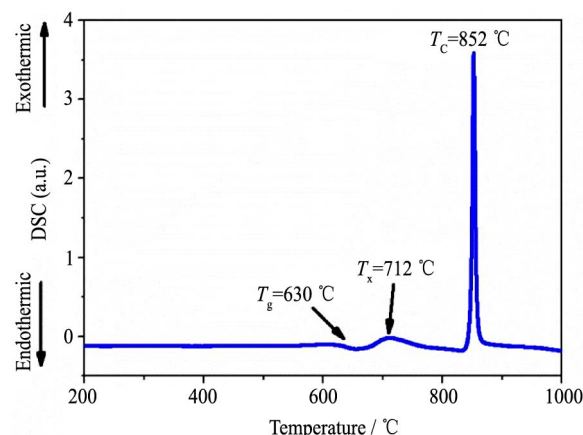


Fig. 1 Differential thermal analysis curve of  $45\text{SiO}_2\text{-}0\text{Al}_2\text{O}_3\text{-}10\text{CaO-}25[(1-x-y)\text{CaF}_2\text{-}x\text{TbF}_3\text{-}y\text{EuF}_3]$  ( $x=0.05, y=0.01$ ) glass

crystallization temperature of oxide matrix  $T_c$  were estimated to be  $630, 712,$  and  $852\text{ }^\circ\text{C}$ , respectively. Fig. 2 shows the XRD patterns of as-synthesized  $5\text{ mol.}\% \text{Tb}^{3+}$ ,  $1\text{ mol.}\% \text{Eu}^{3+}$ , and  $5\text{ mol.}\% \text{Tb}^{3+}/1\text{ mol.}\% \text{Eu}^{3+}$  co-doped  $\text{CaF}_2$  PG and GC700. PG is determined to be structurally amorphous, characterized by two diffuse humps without any sharp peaks. After heat treatment at  $700\text{ }^\circ\text{C}$  for 2 h, four strong diffraction peaks unambiguously indexed to cubic  $\text{CaF}_2$  phase (JCPDS No. 35-816) with space group  $Fm\text{-}3m$  (No. 225) emerge in the XRD pattern for GC700, suggesting that the  $\text{CaF}_2$ -based glass ceramics have been successfully synthesized. Based on the diffraction peak widths, the mean crystalline size of  $5\% \text{Tb}^{3+}/1\% \text{Eu}^{3+}$  GC700 nanocrystals were estimated to be  $15\text{ nm}$  with Scherrer equation:<sup>[29]</sup>

$$D = \frac{k\lambda}{\beta \cos \theta} \quad (1)$$

where  $D$  is the crystal size,  $k=0.89$ ,  $\lambda$  ( $0.154056\text{ nm}$ ) represents the wavelength of  $\text{Cu K}\alpha$  radiation,  $\theta$  is the Bragg angle of XRD peak, and  $\beta$  is the full-width at half-maximum of the diffraction peak.

For investigating the morphology of GC700 sample with  $5\text{ mol.}\% \text{Tb}^{3+}/1\text{ mol.}\% \text{Eu}^{3+}$  co-doped  $\text{CaF}_2$ , TEM,

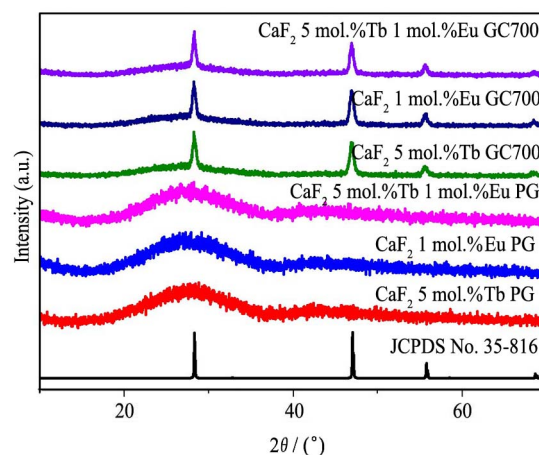


Fig. 2 XRD patterns of PG and GC700 with  $5\text{ mol.}\% \text{Tb}^{3+}$ ,  $1\text{ mol.}\% \text{Eu}^{3+}$ , and  $5\text{ mol.}\% \text{Tb}^{3+}/1\text{ mol.}\% \text{Eu}^{3+}$  co-doped  $\text{CaF}_2$

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