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### Letter

# *In-situ* modification of PEDOT:PSS work function using alkyl alcohols as secondary processing solvents and their impact on merocyanine based bulk heterojunction solar cells

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ABSTRACT

The influence of a series of alkyl alcohols on the work function of PEDOT:PSS thin films is systematically investigated by Kelvin probe measurements. We show that the PEDOT:PSS work function can be increased stepwise from 5.2 eV for pristine PEDOT:PSS to 5.61 eV using either alcohols with different alkyl chain length or varying the amount of alcohol in mixtures with chlorobenzene. Moreover, we demonstrate the effect of work function modification on merocyanine based bulk heterojunction solar cells, resulting in improved values for the open-circuit voltage comparable to those obtained with high work function MoO<sub>3</sub>. Thus, the processing method presented herein can potentially serve as a simple, alternative route to adjustable and high work function electrodes while maintaining processability from solution.

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#### 1. Introduction

The extensive research on organic solar cells (OSCs) in the past decade has developed multiple concepts to improve device efficiency comprising synthesis of new photoactive materials, electrode materials, processing conditions and device setups. Power conversion efficiencies (*PCEs*) of about 9% have been achieved for small molecule [1] as well as for polymer [2] based single-junction solar cells.

For bulk heterojunction (BHJ) solar cells the morphology of the 51 active layer is crucial for efficient generation and transport of 52 53 charges. Several solvent additives, such as 1,8-diiodooctane [3], 1,8-octanedithiol [4] or nitrobenzene [5], have been identified to 54 improve the morphology and/or crystallinity of the active layer. 55 56 A common feature of such additives is their higher boiling point compared to the main solvent for active layer deposition combined 57 with different solubilities of the dissolved materials, causing a 58 59 change in morphology during film formation [3–6]. Furthermore, 60 wide band gap materials such as n-type ZnO [1] or TiO<sub>2</sub> [7] have 61 been implemented at the electron collecting side in the device 62 stack to improve OSC performance. These additional layers often

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http://dx.doi.org/10.1016/j.orgel.2015.03.008 1566-1199/© 2015 Published by Elsevier B.V. act as optical spacers, which can increase the fraction of absorbed light within the active layer. Additionally, wide band gap materials can have charge blocking properties preventing the "wrong" charge carriers from reaching the electrode [7]. Recently, conjugated polyelectrolytes (CPEs) emerged as facile method to modify the organic/electrode interface. In most reported cases, thin layers (<5 nm) of CPEs induce a shift in vacuum level that lowers the effective work function of the electrode material and can, therefore, replace low work function metals as top electrodes that are sensitive to air [8]. In terms of OSC characteristics, the reported additives and stack modifications primarily affected the short-circuit current density J<sub>SC</sub> and the fill factor *FF*.

The open-circuit voltage  $V_{OC}$  of a BHJ organic solar cell is limited by the effective photovoltaic gap, i.e. the difference between the highest occupied molecular orbital (HOMO) of the donor and the lowest unoccupied molecular orbital (LUMO) of the acceptor. The synthesis of new donors, therefore, tends toward materials with deep HOMO energy levels to reach high values for  $V_{OC}$ . To obtain high voltages, however, the work functions of the hole and electron collecting contacts need to be adjusted to the BHJ active layer energy levels as well [9]. Typically,  $V_{OC}$  is barely affected by modification of the electron collecting contact most likely due to easily achieved ohmic contact formation of the electrode with the commonly used electron acceptor phenyl- $C_{61}$ -butyric acid methyl ester (PC<sub>61</sub>BM) [10,11].

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88 On the hole collecting side of OSCs, metal oxides such as high 89 work function n-type  $MoO_3$  [12] and p-type  $NiO_x$  [13,14] as well 90 as self-assembled monolayers [15] (SAM) of phosphonic acids on 91 ITO have been employed. However, the conductive polymer 92 poly(3,4-ethylenedioxythiophene):poly(styrenesulfonate) PEDOT: 93 PSS still remains the most commonly used material due to its sim-94 ple processability from aqueous suspension. Although easily pro-95 cessed, PEDOT:PSS often lacks the demand of the active layer for 96 high work function. It is therefore desirable to find a way to adjust 97 the properties of PEDOT:PSS to those needed by the active layer. 98 Plenty of parameters that influence the work function (and most often conductivity) of PEDOT:PSS have been reported including 99 UV light exposure [16] and the processing conditions, e.g. by addi-100 tion of secondary solvents [17-20]. Moet et al. showed that the 101 102 work function of PEDOT:PSS, and, thus, the  $V_{OC}$  of the correspond-103 ing solar cells is reduced upon increasing the pH of the casting 104 solution [21]. Graham et al. reported on  $V_{OC}$  enhancement in small molecule based solar cells when adding triethylene glycol to the 105 casting solution of the active layer and suggested an increase of 106 PEDOT:PSS work function due to H-bonding of triethylene glycol 107 108 to the PEDOT:PSS surface [6]. The Friend group recently showed 109 that polymer OSC performance can be improved by exposing the active layer on top PEDOT:PSS to methanol prior to evaporation 110 of the top electrode [22]. This treatment led to an *in-situ* increase 111 112 of the PEDOT:PSS work function and resulted in higher values for 113 *V*<sub>OC</sub> and hole extraction rates [23].

In this publication, we systematically investigate the impact of various alkyl alcohols on the work function of PEDOT:PSS thin films. Furthermore, we show that our findings for bare PEDOT:PSS films can be transferred to merocyanine based OSCs, which are highly sensitive to changes in the work function of the hole collecting contact.

#### 120 2. Experimental section

#### 121 2.1. Materials and device fabrication

122 The syntheses and characterization data of all MC dyes were 123 reported previously [12,24–27].  $PC_{61}BM (\ge 99.5\%)$  was obtained from Nano-C. Solvents (purity ≥98%) were obtained from commer-124 cial sources and used as received. All devices were fabricated on 125 commercial indium-tin oxide (ITO) coated glass (TFD) coated 126 127 with poly(3,4-ethylene dioxythiophene):poly(styrene sulfonate) 128 (PEDOT:PSS) (Clevios P Al 4083, Heraeus; ca. 40 nm). PEDOT:PSS 129 films were heat treated for 2 min at 110 °C to remove residual water 130 and transferred into a N<sub>2</sub> glove box for subsequent treatment with 131 alcohols or device fabrication. The active layers of BHJ solar cells 132 were spin-coated from chlorobenzene (CB) containing the mixture 133 of MC and PC<sub>61</sub>BM (20 mg mL<sup>-1</sup> each in CB). The secondary solvent 134 was added to the blend solution to yield the desired vol%. After 135 evaporation of an aluminum top electrode (120 nm; Alfa Aesar 136 99.9%) through a shadow mask, seven solar cells with an active area 137 of 0.0785 cm<sup>2</sup> were obtained on each substrate.

#### 138 2.2. Measurements and device characterization

139 KP measurements were performed using a system by Besocke Delta phi GmbH in an inert argon atmosphere. The contact potential 140 141 difference between the sample and a vibrating gold grid (refer-142 ence) was measured. The measurement was calibrated with work 143 functions determined by ultraviolet photoelectron spectroscopy 144 (UPS) of various metals as well as pristine PEDOT:PSS films. The 145 IV-characteristics of the solar cells were measured using a 146 Keithley 2425 source measurement unit under simulated AM1.5 147 illumination provided by a filtered Xe lamp. The light intensity of

100 mW cm<sup>-2</sup> was determined by using a calibrated inorganic 148 solar cell from the Fraunhofer Institute for solar research (ISE) and 149 a reference P3HT:PC<sub>61</sub>BM cell measured by the same institute. 150 No spectral mismatch factor was included in the calculation of 151 the efficiency. Transmission electron microscopy (TEM) of blend 152 layers was performed with a Leo 912 Omega by Zeiss. UV/visible 153 absorption spectra were taken with a Varian Cary 50 spectrometer 154 in transmission mode. Layer thicknesses were measured by surface 155 profilometry (Veeco). 156

#### 3. Results and discussion

#### 3.1. Effect of alcohols on the work function of PEDOT:PSS thin films 158

The effect of alcohols on the work function  $\Phi$  of PEDOT:PSS thin 159 films was systematically studied by Kelvin probe (KP) measure-160 ments. In a first approach, thin films of PEDOT:PSS on ITO were 161 treated with pure alcohols of increasing alky chain length from 162 methanol (MeOH), to n-pentanol (PeOH). The resulting work func-163 tions are displayed in Fig. 1a. The work function of PEDOT:PSS 164 increases steadily with increasing chain length of the alcohol from 165 5.2 eV for the untreated reference sample (REF) up to 5.61 eV using 166 PeOH, respectively. The only exception from the steady increase is 167 *n*-propanol, yielding a  $\Phi$ -value, which is less than for the two 168 neighboring chain lengths (5.35 eV compared to 5.45 and 169 5.50 eV, respectively); the reason for this finding is currently 170 unknown. Unfortunately, the effect of work function increase by 171 the alkyl alcohols was found to be reversible since washing the 172 alcohol treated substrate with chlorobenzene (CB) - a common sol-173 vent for the subsequent deposition of active layers – restores the 174 work function to 5.15 eV. This finding is in agreement with the 175 176  $\Phi$ -value of ~5.15 eV for pristine PEDOT:PSS treated with pure CB, which will serve as reference value in the following (see Fig. S1 177 in the Supporting Information). 178

To make this finding useful for OSC fabrication, but since most organic materials used in organic photovoltaics are insoluble in alcohols, we tested whether the addition of alcohols to the main casting solvent (here CB) would yield similar results. Indeed, in doing so we were able to increase the work function of PEDOT:PSS even when it is used in mixtures with CB. Expemplarily, Fig. 1b shows the dependence of the PEDOT:PSS work function on the volume fraction of BuOH in mixtures with CB.  $\Phi$  is stepwise increased using small amounts of BuOH until a saturated value of 5.57 eV is reached at 4 vol% BuOH. Thus, BuOH can potentially be added to the casting solution of the subsequent layer for *in-situ* enhancement of the substrate work function during layer deposition. Similar results were obtained with the other alcohols (Fig. 2).

To further investigate the properties of alcohol-treated 193 PEDOT:PSS, substrates treated with 12.5 vol% BuOH/CB, i.e. in the 194 saturation regime, were exposed (i) to elevated temperatures 195 (150 °C, 10 min, N<sub>2</sub> atmosphere), (ii) to air (90 min, room tem-196 perature) and (iii) to vacuum  $(10^{-6} \text{ mbar for } 1 \text{ h})$ , respectively 197 (see Fig. S1 in SI). During heating of the alcohol-treated substrate 198 the work function was restored to 5.15 eV, i.e. the same value as 199 for CB-treated PEDOT:PSS, while a smaller decrease of  $\Phi_{\text{PEDOT}}$  (to 200 5.35 eV) was found when only exposed to air; it can be assumed, 201 however, that extended times would also restore the work func-202 tion to that of CB-treated PEDOT:PSS. Finally, after exposing the 203 BuOH/CB treated film to vacuum the initial work function of 204 PEDOT: PSS of about 5.2 eV was measured. Therefore, due to the 205 reversibility under ultra-high vacuum (UHV) conditions the alco-206 hol modified PEDOT:PSS cannot be studied by ultraviolet photo-207 electron spectroscopy (UPS). This finding opposes the report by 208 Friend et al. for MeOH treated PEDOT:PSS films, where a change 209

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