



## Two-parameter characterization of constraint effect induced by specimen size on creep crack growth



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### ABSTRACT

Experimental and numerical investigations on the effect of constraint induced by different specimen sizes on creep crack growth had been conducted using compact tension specimen with P92 steel. The experimental results revealed that at the same  $C^*$  values, creep crack growth rates increased as specimen size increased and the difference enlarged with increasing  $C^*$  values. In addition, the variations of constraint level  $Q$  during creep were obtained and a modified  $C^*$ - $Q$  approach incorporating constraint effect was proposed to predict creep crack growth. In comparison with conventional single  $C^*$  parameter approach, two parameter approach could provide more accurate crack growth rates.

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### 1. Introduction

Environmental and commercial demands on energy production have led to the development of power plants operable at super critical conditions. Increasing the operating temperature and pressure, the efficiency of conventional steam and gas turbine power plants could be significantly improved and then the fuel consumption and harmful emission of power plants could be remarkably reduced [1,2]. Creep crack propagation is among the most prevalent life inhibitors in high temperature structural components with flaws during manufacturing and servicing. Thus, creep crack growth assessment for structural components serving in elevated-temperature environment is one of the most important tasks to ensure the structure integrity.

To characterize and quantify creep crack propagation behavior, continuum fracture mechanics concepts have to be utilized. Specifically, an appropriate fracture mechanics parameter  $C^*$ , the time dependent analogue of the  $J$  contour integral used in elastic–plastic fracture mechanics, has been employed to correlate the creep crack growth rates, and defined as:

$$C^* = \int_{\Gamma} \left( W^* dy - T \frac{\partial \dot{u}}{\partial x} ds \right) \quad (1)$$

where  $W^*$  is the instantaneous stress–power or energy rate per unit volume;  $\Gamma$  is the path of the integral that encloses the crack tip contour;  $ds$  is the increment in the contour path;  $T$  is outward traction vector on  $ds$ ;  $\dot{u}$  is the displacement rate vector at  $ds$ ;  $x, y, z$  is the rectangular coordinate system and  $T \frac{\partial \dot{u}}{\partial x} ds$  is the rate of stress–power input into the area enclosed by across the elemental length  $ds$ .

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## Nomenclature

$a$	current crack length
$b$	uncracked ligament length
$a_0$	initial crack length
$r$	radial distance from the crack tip
$I_n$	integration constant in the HRR stress field distribution
$K$	stress intensity factor
$F$	applied load in tests
$B$	specimen thickness
$B_n$	net specimen thickness between the bottoms of side grooves
$W$	specimen width
$E$	Young's modulus
$\alpha$	the strain hardening coefficient
$m$	exponent in a Ramberg–Osgood fit to the tensile data
$Q$	dimensionless constraint parameter
$Y_0$	half the distance between the output terminals
$W$	the width of the specimen
$V_0$	the initial voltage
$V$	the actual value of the potential.
$A$	coefficient in the power-law creep strain rate expression
$n$	power-law creep stress exponent
$C^*$	steady state creep fracture mechanics parameter
$D, \phi$	material constants in a correlation with $C^*$
$K$	stress intensity factor
$r_c$	creep process zone size
$J_p$	plastic component of the $J$ -integral
$t_r$	time to rupture in a uniaxial creep test
$t_i$	creep initiation time of crack growth at 0.2 mm
$t_T$	transition creep time
$MSF$	appropriate multiaxial stress factor
$\theta$	crack tip angle
$\nu$	Poisson ratio
$\sigma_0$	yield stress
$\sigma_b$	bending stress effect
$\varepsilon_0$	creep strain at stress $\sigma_0$
$\varepsilon^t, \varepsilon^e, \varepsilon^p, \varepsilon^c$	total, elastic, plastic and creep strain components
$\omega, \dot{\omega}$	creep damage parameter and creep damage rate
$\varepsilon_f$	uniaxial creep failure strain
$\varepsilon_f^*$	multiaxial creep failure strain
$\sigma_m, \sigma_e$	hydrostatic stress and equivalent stress
$\delta_{ij}$	Kronecker delta
$\tilde{\sigma}_{ij}(\theta, n), \tilde{\varepsilon}_{ij}(\theta, n)$	dimensionless stress and strain functions of $n$ and $\theta$
$V_c$	creep load line displacement rate
$\dot{\varepsilon}^c$	equivalent creep strain rate
$\dot{\varepsilon}_A^c$	average creep strain
$\dot{\varepsilon}_{ij}$	strain rate tensor
$\dot{a}$	creep crack growth rate
$\sigma_{\theta\theta}^{HRR}(r, 0)$	analytical crack-opening stress from the HRR field
$\sigma_{\theta\theta}(r, 0)$	numerical crack-opening stress from FEM

In general, the relation between creep fracture parameter and creep crack growth rates could be considered as a material property, which means that it should be independent of size and geometry [3]. Most of the current integrity analyses are based on this property, such as R5, BS7910 standards. However, some experimental and theoretical evidences have shown that crack tip constraints induced by specimen size, crack depth, geometry, and loading condition have great influence on the creep crack behavior. Therefore, many efforts have been made on modeling the constraint effect on the creep crack growth of metals at elevated temperatures [4].

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