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# A model for fracture mechanics based prediction of the fatigue strength: Further validation and limitations

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#### ABSTRACT

Recently two of the authors of the present paper proposed a model for a fracture mechanics based prediction of the *S*–*N* characteristics of metallic components with large microstructural defects and supported this by a validation exercise on tensile plates made of an aluminium alloy AL5380 H321. Here the authors extend the study using a number of further data sets from the literature for which data were available at different *R* ratios. These data include two aluminium alloys, Al 2024-T3 and Al 7075-T6, and a ductile cast iron, EN-GJS-400-18-LT. Despite of necessary assumptions for the compensation of partially missing input information the results were fairly reasonable with the exception of one data set. The authors identify high applied stress levels in combination with potential multiple crack initiation as the probable root of the problem and propose a scheme how the model can be extended for taking into account crack initiation.

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#### 1. Introduction

The lifetime of a cyclically loaded structure can be subdivided into three stages: (i) crack initiation; (ii) small and long fatigue crack propagation; and (iii) final fracture. The small crack growth stage can be further subdivided by considering micro-structurally and mechanically small cracks [1]. Micro-structurally small means a crack size in the order of micro-structural features such as the grain size whereas mechanically small refers to the order of mechanical discontinuities such as the plastic zone size or a notch stress field.

In the case of engineering materials with large second phase particles the initiation stage is rather small and the overall lifetime is usually controlled by the extension of small cracks which are – if the initial defects are large enough – mechanically small cracks. In [2] two of the present authors proposed a model for fracture mechanics based determination of the fatigue strength and life based on the assumption of a negligible short crack initiation stage which allowed them to base the analysis on a pre-existing defect which they treated as initial crack. The model is briefly introduced in Section 2 of this paper. In Section 3 the authors provide further validation using literature data which also revealed some limitation of the model, which is finally discussed in Section 4.

#### 2. The model

The scheme of the proposed model is shown in Fig. 1. It is characterized by the following steps:

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Nomenclature	
~	analy longth for surface analysis analy denth
u	initial grady doubth (in the model doubth of a pro-ovicting defect)
u <sub>i</sub>	ft parameter in Eq. (12)
и <sub>0</sub> Р	in parameter in Eq. (15)
D	specificit finckness
C	initial half erack length at surface
C <sub>i</sub> daldN	fatigue graek propagation rate
uu/uiv E	aligue clack propagation falle
E E/	F for plane strain conditions (Eq. (1))
E c	E IOI plane strain conditions (Eq. (1))
J E	tancila force
Г Г	limit load (Eq. (2))
ГY L	$\begin{array}{l} \text{IIIIII IOdu (Eq. (5))} \\ \text{exponent of the } AK = 2 \text{ function } (Eq. (12)) \\ \end{array}$
K V	exponent of the $\Delta x_{th}$ -a function (Eq. (15))
K V	stiess intensity factor in a loading guele
K <sub>max</sub>	K factor above which the crack is open
к <sub>ор</sub> И	stross concontration factor
K <sub>t</sub> K	Sites concentration factor $\kappa$
к <sub>о</sub> I	ligament vielding parameter $(-E/E_{u} - \sigma_{u}/\sigma_{u})$
L <sub>r</sub> N	number of loading cycles
IN N	number of loading cycles to crack initiation
D D	Protio $(-\sigma +  \sigma )$ or $K +  K $
K II	crack closure function $(=\Delta K - \Delta K)$
	crack closure function for the (mechanically) small and the long crack
W	specimen width
v	shape function of the stress intensity factor
αα	local and global constraint parameter (Eq. $(12)$ )
$\Delta F$	tensile load range (= $F_{min}$ - $F_{min}$ )
	cyclic Lintegral (Fq. (2))
$\Delta K^{J}$	$cyclic K factor (=K_{min}-K_{min})$
$\Delta K_{\rm off}$	crack closure effect corrected $\Delta K (K_{max} - K_{op})$
$\Delta K^{J}$	plasticity corrected AK
$\Delta K_{\rm th}$	fatigue propagation threshold
$\Delta K^{J}_{cc}$	plasticity corrected $\Delta K_{\text{eff}}$ (Eq. (1))
$\Delta K_{\rm th off}$	intrinsic fatigue propagation threshold
$\Delta K_{\rm th,lc}$	fatigue propagation threshold, long crack regime
$\Delta K_{\rm th on}$	crack closure effect induced component of $\Delta K_{\rm th}$
$\Delta L_r$	cyclic ligament plasticity parameter (Eq. (3))
$\Delta \sigma_{ m ref}$	cyclic net section reference stress (Eq. (3))
v	Poissons ratio
$\sigma$	stress, general
$\sigma_{ m max}$	maximum stress in a loading cycle
$\sigma_{\rm open}$	stress above which the crack is open
$\sigma_{\rm vv}$	crack opening stress
$\sigma_Y$	yield strength of the material
$\sigma_0$	reference stress in Eq. (11), in the present model the stabilized cyclic $\sigma_{Y}$
√area	crack area
$\sqrt{\text{area}}^*$	crack area at the transition from small to large crack (Fig. 2)
$\sqrt{\text{area}}_i$	crack area of the initial crack

(a) Introduction of an initial crack (in terms of crack depth  $a_i$  or Murakami's parameter  $\sqrt{\text{area}_i}$ ), the size of which is defined by material defects (or statistical defect size distributions). The nature of these defects is strongly dependent on production processes and applications. Examples are non-metallic inclusions, pores and cavities, corrosion pits, welding defects or scratches onto the surface (cf. [3]). In any case it is expected that a very limited number of loading cycles is sufficient to transform the defect into a crack.

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