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### **Short Communication**

## Giant enhancement of photocatalytic H<sub>2</sub> production over KNbO<sub>3</sub> photocatalyst obtained via carbon doping and MoS<sub>2</sub> decoration



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#### ABSTRACT

This paper was designed for the first time to improve the photocatalytic activity of KNbO<sub>3</sub> via carbon doping and MoS<sub>2</sub> decoration simultaneously. The efficient photocatalytic hydrogen production was realized on the MoS<sub>2</sub>/C-KNbO<sub>3</sub> composite under simulated sunlight irradiation in the present of methanol and chloroplatinic acid. The optimal composite presents a H<sub>2</sub> production rate of 1300  $\mu$ mol·g<sup>-1</sup>·h<sup>-1</sup>, which reaches 260 times that of pure KNbO<sub>3</sub>. Characterization results of the synthesized composite indicates that the introduction of a small amount of carbon into the KNbO<sub>3</sub> lattice greatly hinders the recombination of electron-hole pairs. The decoration of MoS<sub>2</sub> further induces the separation of charge carriers via trapping the electron in the conduction band of C-KNbO<sub>3</sub>, which is proven by the EIS and transient photocurrent response analyses. The remarkably enhanced separation efficiency of electron-hole pairs is believed to be the origin of the excellent photocatalytic performance, though other changes in surface area and optical property may also contribute the photocatalytic process. This study provides a feasible way for the design and preparation of novel photocatalysts with high efficiency.

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#### Introduction

Hydrogen is considered as one of the ideal alternatives to fossil fuels due to its properties of cleanness, abundance, sustainability, high combustion energy, and null  $CO_2$ 

emission. Compared with the reformation of petroleum and natural gas which process is energy intensive, photocatalytic water splitting is a better and sustainable route to obtain hydrogen, and hence attracts much attention. Since the pioneer work of Fujishima and Honda in 1972 [1], significant efforts have been devoted to designing and synthesizing

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highly efficient photocatalysts for hydrogen generation from water splitting, and a great variety of photocatalysts have been reported, such as NaTaO<sub>3</sub> [2], CdS [3], Ga<sub>2</sub>O<sub>3</sub> [4] and g- $C_3N_4$  [5]. The separation efficiency of electron-hole pairs is widely accepted as one of the key factors in determining the activity of a semiconductor photocatalyst. Some approaches, including noble metal loading and semiconductor coupling, have been developed to improve the property. The potential difference between the photocatalysts and the loaded noble metals (semiconductors) can induce the photogenerated electron to accumulate on the decorated species, and thus hindering the recombination of charge carriers [6-10]. However, this process can only change the distribution of electron on the surface of photocatalyst (the surface charge separation), the bulk recombination of electron-hole pairs which accounts for a large part of the disappearance of the photogenerated electron is not prevented (the bulk charges separation). Actually, inspired by the motivation of achieving highly efficient solar-energy conversion, scientists have found out some approaches to suppress the bulk recombination process. It is recognized that the bulk recombination is closely correlated with the catalyst's structure and the particle size. The photocatalyst with layer structure usually has nonuniform charge distribution, which induces an internal electric field (IEF) and subsequently boosts the bulk charge separation [11]. A small particle size is also beneficial to the process. Additionally, the incorporation of a tiny amount of external elements into the lattice can modify the bulk charge distribution, tune the IEF, and thus increasing the bulk charges separation efficiency [12]. For example, the metal ion doping ((Cr,Ta)-codoped SrTiO<sub>3</sub> [13], Sr-doped NaTaO<sub>3</sub> [14]), nonmetal ion doping (N-TiO<sub>2</sub> [15]; C-TiO<sub>2</sub> [16,17]), and constructing solid solution structures (Ga<sub>1-</sub> <sub>x</sub>In<sub>x</sub>O<sub>3</sub>) [18], Zn<sub>x</sub>Cd<sub>1-x</sub>S [19]) may effectively increase the bulk charge separation efficiency. Clearly, although the reported photocatalytic hydrogen production performance is still not satisfying, the awareness of the recombination process is gradually deepening under the effort of researchers. Meanwhile, based on the obtained achievements, different strategies should be combined to increase both the bulk and surface charge separation efficiency.

Perovskite-type mixed metal oxides are an important family of materials and exhibit numerous applications in thermal catalysis, electronic device and photonic crystal. Meanwhile, they are good candidates for photocatalytic H<sub>2</sub> production from water splitting due to their availability, low toxicity, and more negative conduction band edge than the  $H^+/H_2$  energy level. As a member of them, KNbO<sub>3</sub> shows great potential in photocatalytic hydrogen generation, and thus attracting scientists' interest [13]. Lots of work about the modification of KNbO3 have been reported. A variety of semiconductors or noble metals, including g-C<sub>3</sub>N<sub>4</sub> [20], CdS [21], Nb<sub>4</sub>O<sub>5</sub> [22], and Au nanoparticles [23], were decorated to improve the surface charges separation of KNbO<sub>3</sub>, while some external elements such as N [24] and Ta [25] were introduced into the KNbO<sub>3</sub> lattice to enhance the bulk charges separation efficiency. However, significantly improved photoactivity has not been observed in these newly-designed KNbO3 based materials. An important reason may be that only a part of the recombination process is retarded.

Therefore, in this work, the strategies of carbon doping and MoS<sub>2</sub> decoration were firstly simultaneously applied to improve the photocatalytic activity of KNbO<sub>3</sub>. Carbon doping has been reported to be highly efficient for the photoactivity improvement [16], Meanwhile, in our knowledge, no work on carbon-doped KNbO<sub>3</sub> has been reported. Due to a similar reason, MoS<sub>2</sub> was chosen to further modify the carbon-doped KNbO<sub>3</sub> because the semiconductor is seen as efficient doper in retarding charge recombination [26–32]. The synthesized MoS<sub>2</sub>/C-KNbO<sub>3</sub> presents significantly higher photocatalytic H<sub>2</sub> production performance than pure KNbO<sub>3</sub>. Thoroughly characterizations and experimental tests are also carried out to reveal the origin of the excellent photocatalytic performance of the sample.

#### **Experimental section**

The detailed information of the preparation, characterization, and photocatalytic test of the  $MoS_2/C-KNbO_3$  catalyst is listed in supplementary materials.

#### **Results and discussion**

SEM analysis (Figure S1) indicates that KNbO<sub>3</sub> and C-KNbO<sub>3</sub> present the morphology of microcube and microtower, respectively. The MoS<sub>2</sub> products are mainly composed of flower-like hierarchical assemblies. The MoS<sub>2</sub>/C-KNbO<sub>3</sub> displays nearly the same morphology as C-KNbO<sub>3</sub>. No MoS<sub>2</sub> is observed in the composite due to its low content. The XRD patterns of KNbO<sub>3</sub>, C-KNbO<sub>3</sub>, MoS<sub>2</sub>, and MoS<sub>2</sub>/C-KNbO<sub>3</sub> composite are shown in Fig. 1a. It can be seen that pure KNbO<sub>3</sub> is in its orthorhombic phase (PDF# 32-0822) and shows strong diffraction peaks at  $2\theta = 22.0^{\circ}$ ,  $31.3^{\circ}$ ,  $45.0^{\circ}$ ,  $50.7^{\circ}$ ,  $55.9^{\circ}$ , indicating its high crystallinity. The addition of glucose during the hydrothermal process shows negligible effects on the XRD patterns. The C-KNbO<sub>3</sub> sample still presents the orthorhombic phase of KNbO<sub>3</sub>. However, the amplified XRD patterns within the range of 30–32° indicate that the peak of C-KNbO<sub>3</sub> shifts to low diffraction angle compared with that of KNbO3. This phenomenon indicates that carbon atom is doped into the lattice of KNbO<sub>3</sub>. XPS analysis (Fig. 2) proves that the doped carbon exists in the form of  $C^{4+}\!\!\!\!$  , indicating that the doped carbon does not replace the position of lattice oxygen. The K<sup>+</sup>,  $Nb^{5+}$  and  $C^{4+}$  radius is about 0.152 nm, 0.064 nm and 0.029 nm, respectively. Given the similarity in radius and charges, it is mostly possible that some Nb<sup>5+</sup> may be substituted by carbon ions. The same peak shift is also observed in the MoS<sub>2</sub>/C-KNbO<sub>3</sub> composite, suggesting the decoration of MoS<sub>2</sub> does not further change the structure of C-KNbO<sub>3</sub>. However, no XRD peaks corresponding to MoS<sub>2</sub> are detected, which may be ascribed to the low MoS<sub>2</sub> content and the weak signals of MoS<sub>2</sub>. Actually, even pure MoS<sub>2</sub> displays very weak XRD diffraction peaks (the inset image). Fig. 1b shows the UV-vis diffuse reflection spectra of the four samples. MoS<sub>2</sub> shows strong absorption in the visible light region, whereas pristine KNbO<sub>3</sub> can only absorb UV light. The band gaps of KNbO<sub>3</sub> and MoS<sub>2</sub> are estimated to be 3.07 and 1.88 eV, respectively (Figure S2). C-KNbO<sub>3</sub> sample presents a similar UV-vis

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