



Orientation alignment of epitaxial LiCoO₂ thin films on vicinal SrTiO₃ (100) substrates



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HIGHLIGHTS

- LiCoO₂ epitaxial thin films are synthesized on vicinal SrTiO₃ (100) substrates.
- The (104)-oriented films exhibit single-domain structure on the vicinal substrates.
- The orientation alignment is explainable by step-flow growth mechanism.
- The influence of domain boundaries on electrode performance is discussed.

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ABSTRACT

LiCoO₂ is epitaxially grown on SrTiO₃ (100) substrates with (104) orientation. Because the LiCoO₂ film is grown with its *c*-axis parallel to four equivalent (111) axes of the SrTiO₃, the (104)-oriented film exhibits four-domain structure on the SrTiO₃ (100) substrate. Introducing off-cut angle to the substrate surface breaks the equivalency between the four ⟨111⟩ axes of the SrTiO₃ substrate to induce preferential growth of specific orientation with the *c*-axis in a descending direction of off-cut surface. Increasing off-cut angle and lowering deposition rate promote the preferential growth, because they facilitate step-flow growth mode, and finally align the *c*-axes in the domains completely into one ⟨111⟩ direction of the SrTiO₃ substrate. The LiCoO₂ film delivers a discharge capacity of 90 mAh g⁻¹ at a low discharge rate of 0.01 C, and 25% of capacity is kept even at a high rate of discharge with 100 C.

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1. Introduction

Studies on thin-film batteries started in 1980s [1], and they are growing in importance with the births of new micro-devices, e.g. nonvolatile memories, micro electro mechanical systems (MEMS), etc. Another importance of thin-film systems is originated from their simple geometry. Because electrode/electrolyte interfaces and ion-conduction paths can be simplified in the geometry, thin-film systems are beneficial for fundamental research. Especially,

recent materials research has provided new materials with excellent bulk properties. They have often replaced their bulk with their interfaces that govern the battery performance and initiated many studies on interface, where planar interfaces and one-dimensional ionic conduction in the simplified geometry are useful.

Although the electrochemical systems become to have simplified geometry in thin-film systems, more efforts are necessary to gain basic insight into intrinsic interface properties. Grain boundaries affect the ionic conduction, and the interfaces are formed between uncontrolled crystal faces, as far as the films are polycrystalline. Epitaxial films are anticipated to provide ideal interfaces for the fundamental research on the interfaces due to their specified surface. However, even epitaxial films do not always show intrinsic nature of the materials. For example, they often have multi-domain structure, whereas a recent computational study predicts that twin domain boundaries existing in epitaxial films

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affect ion transport in the films [2]. Therefore, ultimate goal is to form these materials in monolithic single-crystal films without domain boundaries, which is equivalent to bulk single crystal, although it is challenging task.

Our previous study on epitaxial growth of LiCoO₂ has revealed that LiCoO₂ is epitaxially grown with (104), (018), and (001) orientations on SrTiO₃ (100), (110), and (111) substrates, respectively [3]. In the study, we demonstrated electrode properties of the (104)-oriented film, because CoO₂ layers in the film are the most upright and thus offer the most accessible interlayers for lithium ions. However, the film has four-domain structures, in which domain boundaries may affect the electrode properties.

Vicinal substrates, surface of which is at off-cut angle to a specific crystal plane, are sometimes utilized in order to improve the crystallinity of epitaxial thin films [4,5], because they are generally believed to enhance step flow growth of thin films and control domain structure. In this study, (104) oriented LiCoO₂ epitaxial thin films are grown on vicinal SrTiO₃ (100) substrates to promote preferential formation of specific domains.

2. Experimental

Epitaxial LiCoO₂ thin films were grown by pulsed laser deposition (PLD). A sintered Li_{1-x}CoO_{2+x} pellet (TOSHIMA Manufacturing Co., Ltd.) was used as the PLD target. A KrF excimer laser (COMPex Pro 50, COHERENT, Inc., $\lambda = 248$ nm) was operated at a repetition rate of 2 Hz or 40 Hz for the target ablation. Laser spot size on the target was 0.031 cm², and the laser fluence was 0.65 J cm⁻². The incident angle of the excimer laser on the target surface was 30°, and the target–substrate distance was set at 58 mm. Oxygen pressure in the chamber and substrate temperature were set at 0.1 Pa and 725 °C, respectively. Detailed information regarding the selection of target composition, laser fluence, and oxygen pressure can be found in Refs. [6,7].

Films were deposited on non-doped SrTiO₃ single crystal substrates, which were exactly-cut SrTiO₃ (100) substrates and vicinal substrates with 3° off-cut toward in-plane [001] direction, and those with 2° and 4° off-cut toward in-plane [011] direction from (100) plane as illustrated in Fig. 1 in order to investigate the effects of the off-cut directions and off-cut angles from (100) plane of SrTiO₃ on the film structure. A vicinal 3 at% La-doped SrTiO₃ (La:SrTiO₃) single crystal substrate with 5° off-cut toward in-plane [011] direction was used in place of the non-doped insulating ones in order to investigate the electrode properties of the LiCoO₂ film grown on the vicinal substrate, because the La-doping provides electronic conduction to the substrates and makes them act as current collectors in the electrochemical measurement. All the substrates were made by SHINKOSHA Co., Ltd., and the substrate size is typically 10 mm in diameter, 10 mm × 10 mm, or 7.5 mm × 7.5 mm with 0.5 mm in thickness.

An X-ray diffractometer (XRD, SmartLab, Rigaku Corp.)

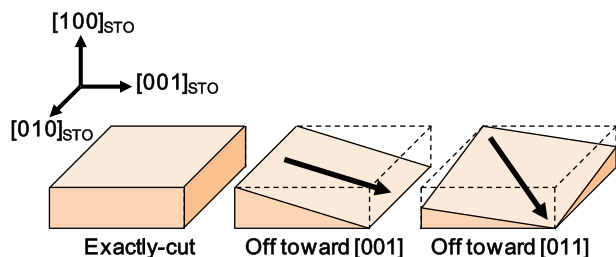


Fig. 1. Schematics of SrTiO₃ single crystal substrates with “Exactly-cut”, off-cut toward [001] direction, and off-cut toward [011] direction.

equipped with a rotating Cu target was used for crystallographic and thickness analyses of obtained thin films, which include in-plane pole figure measurement [8]. The pole figures were taken for reflections of LiCoO₂ 003, as well as LiCoO₂ 104, 110 and SrTiO₃ 100, 110, and 111 with a step of 1.0°, where non-monochromated X-ray was used to obtain high diffraction intensity. Prior to the pole figure measurement, asymmetric diffraction measurements were performed with 2-bounce Ge 220 monochromated CuK α_1 radiation in order to determine the 2θ value precisely for each reflection.

Transmission electron microscopy (TEM) was employed to investigate the structural features of the films. Cross-sectional samples were made by wedge polishing. Bright field (BF) and dark-field (DF) transmission electron micrographs were taken on using a JEOL ARM-200F microscope at 200 kV acceleration voltage.

Electrode performance of the obtained films was investigated in an all-solid-state cell. Powder Li_{3.25}Ge_{0.25}P_{0.75}S₄ (*thio*-LISICON [9]) was used as the solid electrolyte. An In–Li alloy was selected as the counter electrode and formed by attaching a piece of lithium (2 mg) to an indium foil (60 mg). The surface of LiCoO₂ film was coated with a 20-nm-thick Li₃PO₄ layer to reduce the interfacial resistance to be low enough for the electrochemical measurement [10,11]. The Li₃PO₄-coated LiCoO₂ film on the La:SrTiO₃ substrate and the In–Li alloy were attached to respective sides of a *thio*-LISICON layer (150 mg) and they were pressed together at around 500 MPa to form a three-layered pellet with a 10 mm diameter.

The cell was charged and discharged at room temperature using a potentiogalvanostats (VSP, Bio-Logic SAS, and PS-08, Toho Technical Research Co., Ltd.). Because the electrode potential of the In–Li alloy counter electrode is 0.62 V vs. Li⁺/Li, the upper cutoff voltage was set at 3.58 V in order to charge the epitaxial LiCoO₂ thin film up to 4.2 V vs. Li⁺/Li. The charging rate was fixed at 0.01 C, while the discharging rate was varied from 0.01 C to 100 C, where the rate of 137 mA g⁻¹ was defined as 1 C. The specific capacity was estimated based on the film thickness determined by X-ray reflectivity measurement and the theoretical density of 5.0 g cm⁻³.

3. Results and discussion

Our previous study, in which exactly-cut SrTiO₃ substrates were used, has revealed that LiCoO₂ epitaxial thin films grow on SrTiO₃ (100) surface with (104) orientation. The films exhibit four-domain structure, in which the four kinds of domains whose *c*-axes of LiCoO₂ are parallel to the four equivalent $\langle 111 \rangle$ directions of SrTiO₃ substrates appear with equal probability [3]. The LiCoO₂ films in this study are also (104) oriented regardless of the exactly-cut or vicinal SrTiO₃ (100) substrates; however, the four kinds of domains do not appear with equal probability on the vicinal substrates.

Fig. 2 shows X-ray pole figures for 003 reflections of the LiCoO₂ films, and Fig. 3 indicates azimuthal phi scan curves ($\alpha = 36^\circ$ and $\phi = \beta$, α : angle from substrate surface) extracted from the pole figures with off-cut angle adjustment. The LiCoO₂ film grown on the exactly-cut SrTiO₃ (100) substrate gives four poles in $\langle 111 \rangle$ directions with an equal intensity, as indicated in Fig. 2a and the bottom curve in Fig. 3, which support that the four kinds of domains appear with the equal probability.

Such domain structure is dramatically changed on the vicinal SrTiO₃ (100) substrates, while the out-of-plane (104) orientation is preserved. The off-cut toward [001] direction prefers two orientations out of the four, as can be recognized in the pole figure in Fig. 2c and the second curve from the bottom in Fig. 3. The XRD data obtained from the LiCoO₂ thin film grown on a 3° off-cut substrate toward [001] direction at a laser repetition rate of 40 Hz clearly indicate that the off-cut promotes two orientations with the *c*-axis parallel to $[111]$ and $[\bar{1}\bar{1}1]$ directions of the SrTiO₃ substrate, i.e. in a descending direction of in-plane off-cut direction (arrows in Fig. 1)

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