



# Outstanding lithium-storage performance of carbon-free $\text{Li}_4\text{Ti}_5\text{O}_{12}$ anode material for rechargeable lithium-ion batteries

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## ABSTRACT

In this work, carbon-free  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  material has been successfully prepared by using a conventional sol-gel method. X-ray diffraction (XRD) result reveals that the as-prepared pristine  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  sample has the spinel-type structure with a cubic lattice. The scanning electron microscopy (SEM) and transmission electron microscopy (TEM) images indicate that the morphology of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  material is well-crystallized with a uniform size distribution in the range of 300–700 nm. Moreover, high-resolution TEM image demonstrates that the surface of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  particle is very smooth without any carbon coating. When used as anode material for lithium-ion batteries, the pristine  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  exhibits outstanding lithium-storage performances such as high reversible capacity, excellent rate capability and stable cycling property. Therefore, we can speculate that the as-prepared  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  sample without any carbon materials is a promising candidate for the application of high-performance anode for lithium-ion batteries.

## 1. Introduction

Nowadays, rechargeable lithium-ion batteries with high energy/power density and good safety have been regarded as one of the most promising energy devices for electric vehicles and hybrid electric vehicles [1,2]. Nevertheless, the search for advanced anode materials to meet the requirements of good rate capability and long-cycle life is still critical for high-performance lithium-ion batteries. Compared with the commercial graphite-based anodes, spinel  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  material has attracted great attention for lithium-ion batteries anode due to its intrinsic characteristics such as high thermal stability, good safety, excellent rate capability and long-cycle life [3–5]. Moreover, it shows a flat  $\text{Li}^+$  insertion/extraction potential plateau at around 1.55 V (vs.  $\text{Li}/\text{Li}^+$ ) and a high theoretical capacity of  $175 \text{ mAh g}^{-1}$  [6,7]. Despite of these advantages, the  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  material suffers from the inherent poor electronic conductivity (ca.  $10^{-13} \text{ S cm}^{-1}$ ) and moderate  $\text{Li}^+$ -ion diffusion coefficient ( $10^{-8} \text{ cm}^2 \text{ s}^{-1}$ ) [8,9], which greatly inhibit its practical application for energy storage.

Up to the present, many strategies have been devoted to solve these issues. Carbon coating [5,7,9–12] has been regarded as the most common approach to improve the lithium-storage performance of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$ . Owing to this carbon layer, which could be able to prevent the growth of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  particles during the sintering process and enhance its apparent electronic conductivity, the high-rate capability and long-cycle stability of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  will be improved. For instance, Lin's group [9] reported that the carbon-coated  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  was prepared by using a facile one-pot sol-gel route and the obtained composite

delivered high discharge capacities of 153.9 and  $147.9 \text{ mAh g}^{-1}$  at 10 and 20 C respectively. Doping with other metal ions [13–15] is another strategy to improve the intrinsic electronic conductivity of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$ . What's more, synthesis of nanosized  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  particles [16,17] is a promising approach to reduce the  $\text{Li}^+$ -ion diffusion path and thus improve the  $\text{Li}^+$ -ion diffusion efficiency. However, it remains a great challenge to prepare nanostructured  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  materials with desirable architectures for fulfilling the requirements of excellent rate capability and stable long-cycle life.

In this current work, we designed and prepared the carbon-free  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  material used as anode for rechargeable lithium-ion batteries by using a simple sol-gel approach. The results showed that the well-crystallized  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  sample had the spinel-type structure and the surface of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  particle was smooth without any carbon coating. The electrochemical measurements revealed that the pristine  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  anode exhibited outstanding lithium-storage performances including rate capability and long-cycle life. Thus, the as-prepared carbon-free  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  material can be considered as a promising anode for high-performance lithium-ion batteries.

## 2. Experimental

### 2.1. Material preparation

The pristine  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  sample (P- $\text{Li}_4\text{Ti}_5\text{O}_{12}$ ) was prepared by using a conventional sol-gel method with lithium acetate dehydrate ( $\text{CH}_3\text{COOLi} \cdot 2\text{H}_2\text{O}$ ), tetrabutyl titanate ( $\text{Ti}(\text{OC}_4\text{H}_9)_4$ ) and citric acid

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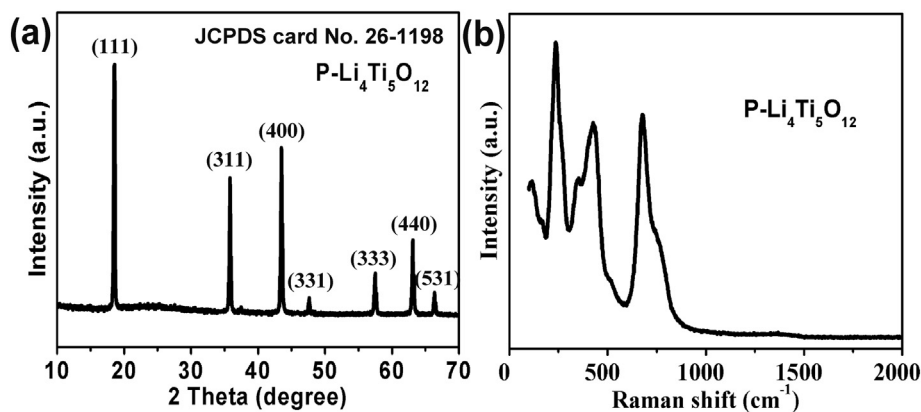


Fig. 1. (a) XRD patterns and (b) Raman spectrum of the as-prepared P-Li<sub>4</sub>Ti<sub>5</sub>O<sub>12</sub> sample.

(C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>·H<sub>2</sub>O) as chelating agent. In a typical synthesis, the CH<sub>3</sub>COOLi·2H<sub>2</sub>O and Ti(OC<sub>4</sub>H<sub>9</sub>)<sub>4</sub> were separately dissolved in ethanol solution with Li:Ti molar ratio of 4.5:5. Afterwards, the CH<sub>3</sub>COOLi solution was slowly added to the Ti(OC<sub>4</sub>H<sub>9</sub>)<sub>4</sub> solution under continuous stirring at 25 °C for 4 h. Then, the mixed solution was further added to the ethanol solution containing C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>·H<sub>2</sub>O. The mixture was stirred at 25 °C for 2 h and gently heated at 70 °C until a gel precursor was formed. Finally, the powder was calcinated at 350 °C for 6 h and then calcinated at 850 °C for 12 h in a flowing air atmosphere to get the P-Li<sub>4</sub>Ti<sub>5</sub>O<sub>12</sub> sample.

## 2.2. Material characterization

The phase and crystal structure of the as-prepared material were investigated by X-ray powder diffraction (XRD) analysis system (Xpert MPD DY1219) equipped with a Cu K $\alpha$  radiation source ( $\lambda = 0.15406$  nm). Raman spectroscopy was carried out on a laser Raman spectrometer (Dongwoo DM500i) with the 532 nm line of an Ar ion laser as the excitation source. The morphology and microstructure of P-Li<sub>4</sub>Ti<sub>5</sub>O<sub>12</sub> sample were recorded by scanning electron microscope (SEM, JEOL-7401) and transmission electron microscopy (TEM, JEOL, JEM-2100F). To further characterize the surface morphology of P-Li<sub>4</sub>Ti<sub>5</sub>O<sub>12</sub> particles, the high-resolution transmission electron microscopy (HRTEM) was also performed. The electronic conductivity of the as-prepared P-Li<sub>4</sub>Ti<sub>5</sub>O<sub>12</sub> material was measured with an RTS-8 linear four-point probe measurement system.

## 2.3. Electrochemical measurements

CR2032 coin-type cells were used to evaluate the lithium-storage performance of the as-prepared P-Li<sub>4</sub>Ti<sub>5</sub>O<sub>12</sub> anode material. The working electrodes were prepared by spreading the slurry of the active material (85 wt%), super P (10 wt%) and poly(vinylidene fluoride) binder (5 wt%) dissolved in *N*-methylpyrrolidone solvent, which was uniformly spread on a Cu foil and dried at 110 °C overnight to remove the residual solvent. The coin cells were assembled in an Ar-filled glove box using pure lithium foil as the counter electrode, the microporous polypropylene membrane (Celgard 2400) as the separator. The electrolyte was 1 M LiPF<sub>6</sub> solution in ethylene carbonate (EC): diethyl carbonate (DEC): ethyl methyl carbonate (EMC) with a volume ratio of 1:1:1. The mass loading of the working electrode was about 2.8 mg cm<sup>-2</sup>. The charge/discharge tests were carried out over potential ranging from 1.0 V to 2.5 V (vs. Li<sup>+</sup>/Li) at different current rates based on the theoretical capacity of 175 mAh g<sup>-1</sup>. The electrochemical impedance spectroscopy (EIS) measurement was carried out on an electrochemical workstation (PARSTAT) in the frequency range of 0.01 Hz to 100 kHz by applying an AC signal of 5 mV. All the electrochemical measurements were performed at room temperature (25 °C).

## 3. Results and discussion

The XRD patterns of the as-prepared P-Li<sub>4</sub>Ti<sub>5</sub>O<sub>12</sub> sample are shown in Fig. 1a. All of the diffraction peaks of P-Li<sub>4</sub>Ti<sub>5</sub>O<sub>12</sub> can be indexed to a cubic spinel structure with Fd-3m space group, which are in good

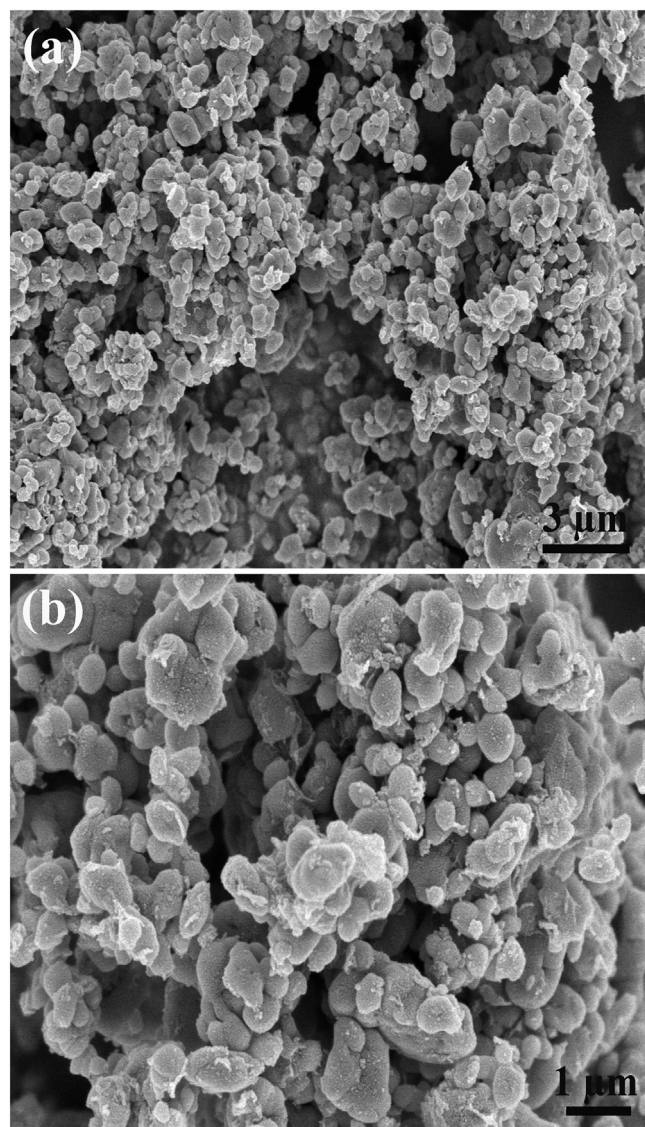


Fig. 2. SEM images of the obtained P-Li<sub>4</sub>Ti<sub>5</sub>O<sub>12</sub> particles.

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