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Sensing performance of zirconia-based gas sensor using titania sensing-electrode added with palladium

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ABSTRACT

The sensing characteristics of a yttria-stabilized zirconia (YSZ)-based sensor using a sensing electrode (SE) consisting of rod-shaped TiO₂ added with 1 wt.% of nano-sized Pd particles (abbreviated as $TiO_2(+Pd)$ -SE) were evaluated, aiming at selective detection of hydrocarbons. The rod-shaped TiO₂-SE without Pd addition (TiO_2-SE) was found to give high sensitivities to both C_3H_8 and C_3H_6 which were used as representatives of alkane and alkene groups, respectively, in hydrocarbons (HCs), while TiO₂(+Pd)-SE gave selective detection of C₃H₈. The evaluation of catalytic activity to the gas-phase oxidation of HCs for the SE materials revealed that the oxidation of C_3H_6 was largely promoted by the Pd addition, leading to the high selectivity to C_3H_8 for the sensor using $TiO_2(+Pd)$ -SE. From the results of polarization-curve measurements, the current value for the cathodic reaction of O_2 was found to be slightly decreased by the Pd addition, while those for the anodic reaction of C_3H_8 and C_3H_6 were largely decreased. This is presumably due to the decrease in the actual concentration of C_3H_8 and C_3H_6 at the SE/YSZ interface owing to the promoted gas-phase oxidation of these HCs through the TiO₂(+Pd)-SE layer. XRD measurements as well as SEM observations confirmed that the added Pd particles were existing as PdO particles on the surface of the rutile-type TiO₂ rods after the SE fabrication process. The stability test revealed that the C_3H_8 sensitivity of the sensor using $TiO_2(+Pd)$ -SE gradually increased and reached the stable value after the 12 day operation. By considering the results of the complex-impedance analysis, we believed that this gradual increase in sensitivity was due to the decrease in the catalytic activity against the gas-phase oxidation of C_3H_8 in the $TiO_2(+Pd)$ -SE layer.

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1. Introduction

Recently, the strict regulations for the automobile exhausts such as hydrocarbons (HCs), carbon monoxide (CO) and nitrogen oxide (NOx) have been settled in various advanced countries. The regulations are becoming more severe year by year. In order to meet such stringent regulations, automobile companies have been trying to develop the advanced engine systems equipped with new catalyst systems, such as selective catalytic reduction (SCR) [1] and NOx storage/reduction (NSR) catalyst [2]. To make such engine systems operate efficiently, high-performance and reliable gas sensors need to be developed for monitoring the concentrations of various exhaust gases upstream and/or downstream of the catalysts. So far, there have been many reports on yttria-stabilized zirconia (YSZ)-based exhaust gas sensors which are classified into three categories based on sensing modes; potentiometric (mixed-potential type) [3–8], amperometric [9–11] and

0167-2738/\$ - see front matter © 2013 Elsevier B.V. All rights reserved. http://dx.doi.org/10.1016/j.ssi.2013.08.028 impedancemetric [12–14]. In general, selection of a SE material is one of the most important points to achieve high-performance of solid-state electrochemical gas sensors.

It has been reported that YSZ-based sensors using composite (oxide + metal)-SE can give sometimes specifically excellent sensing performances toward various exhaust gases even at rather high temperatures [15–17]. Recently, we have also found that the use of composites, such as $\text{ZrCr}_2O_4(+Au)$ and NiO(+Au), as an SE material resulted in high sensitivity and/or high selectivity to CO [18] and C_3H_6 [19], respectively. Thus, in this study, the YSZ-based sensor using such a composite (oxide + metal) SE material, especially rod-shaped TiO₂ added with nano-sized palladium (Pd) particles was fabricated and its sensing characteristics were evaluated, aiming at sensitive and selective detection of HCs.

2. Experimental

2.1. Sensor fabrication

The rod-shaped TiO_2 powders (Ishihara Sangyo, Japan) were drizzled with the commercial colloidal solution consisting of nano-sized Pd

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particles of 3–8 nm (Tanaka Kikinzoku, Japan) and was added to a proper amount of distilled water, followed by agitation for 2 h. Then, the solution was dried at 130 °C overnight and the resulting $TiO_2(+Pd)$ composite powder was sintered at 700 °C for 2 h in air. The final SE material was obtained by thoroughly pulverizing the sintered powder in a mortar.

The obtained composite material was mixed with α -terpineol in a weight ratio of 1:1 and the obtained SE paste was applied on the outer surface of a YSZ tube (8 mol% Y₂O₃ doped ZrO₂, length: 300 mm, inside diameter: 5 mm, outside diameter: 8 mm, Nikkato, Japan). A commercial Pt paste (Tanaka Kikinzoku, Japan) was painted on the inner and outer surfaces of the YSZ tube, respectively. The painted YSZ tube was then sintered at 900 °C for 2 h in air to fabricate TiO₂(+Pd)-SE, Pt reference-electrode (RE) and Pt counter-electrode (CE).

2.2. Evaluation of sensing performance

The electromotive force (*emf*) between SE and RE was measured as a sensing signal by means of a digital electrometer (R8240, Advantest, Japan). SE was exposed to a base gas (21 vol.% $O_2 + 5$ vol.% $H_2O + N_2$ balance) or sample gases (CO, NO, NO₂, C₃H₈, C₃H₆: 100 ppm each, diluted with the base gas), while RE was always exposed to air atmosphere. The total gas flow-rate was fixed at 100 cm³ min⁻¹.

The polarization (current–voltage) curves of the fabricated sensors were measured in the base gas and in the sample gas (100 ppm C_3H_8 or 100 ppm C_3H_6 , diluted with the base gas). The potential scan was carried out in a potentiodynamic mode at a scan rate of 5 mV min⁻¹ by means of an automatic polarization system (HZ-3000, Hokuto Denko, Japan), based on a 3-terminal method.

The complex-impedance of the sensor using $TiO_2(+Pd)$ -SE was measured in the base gas and in 100 ppm C_3H_8 (diluted with the base gas) by means of a complex-impedance analyzer (1255WB, Solartron, UK), based on a 2-terminal method at 0 mV potential difference between SE and RE. The frequency was changed from 0.1 to 10^6 Hz, and the amplitudes of the AC potential and the applied DC potential were fixed at 50 mV and 0 mV, respectively.

2.3. Characterization of SE material

The crystal structures of TiO₂ and TiO₂(+Pd) powders after sintering at 900 °C for 2 h in air were examined by the use of an X-ray diffractometer (XRD, RINT 2100VLR/PC, Rigaku, Japan) with Cu K α radiation ($\lambda = 1.5406$ Å) at angle step of 1° min⁻¹. The morphology of the TiO₂(+Pd)-SE material formed on YSZ was observed by a field-emission scanning electron microscope (FE-SEM, JSM-6340 F, JEOL, Japan).

The catalytic activities to the gas-phase oxidation reaction of 700 ppm C_3H_8 (or 700 ppm C_3H_6 , diluted with synthetic air) for the TiO₂ and TiO₂(+Pd) powders (20 mg each) were evaluated by using a gaschromatography-mass spectroscopic analysis (GC: GC-17A, Shimadzu, Japan; column: RT-m sieve 5A, Restek, USA; MS: MSQP5050A, Shimadzu, Japan) in the temperature range of 25-800°C. Helium was used as a carrier gas at a column-head pressure of 100 kPa. The scanned mass-to-charge (*m*/*z*) ratio was fixed at 39, 41, 42 for C₃H₆ and 27, 29, 43 for C₃H₈ detection.

3. Results and discussion

3.1. Evaluation of SE materials by XRD and SEM

The crystal structures of the TiO_2 and $TiO_2(+Pd)$ powders were investigated by means of XRD and the observed diffraction patterns are given in Fig. 1. The peaks appearing around 36, 39 and 41° observed in the patterns of the both powders revealed that the TiO_2 used in this study has a rutile-type crystal structure (JCPDS no. 73-1765). Only in the patterns of the $TiO_2(+Pd)$ powder, a peak around 34° was seen.

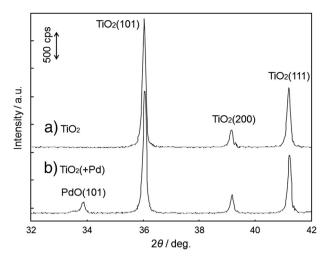


Fig. 1. XRD patterns of (a) TiO₂ powder and (b) TiO₂(+Pd) powder, after calcination at 900 °C for 2 h in air.

This is confirmed to be a (101) peak of palladium oxide (PdO, tetragonal, JCPDS no. 41-1107). Since the strongest (111) peak of metal palladium (Pd) around 40° was not found in the pattern (Fig. 1 (b)), the added Pd is believed to be oxidized to PdO in the sintering process.

To observe the morphology of $TiO_2(+Pd)$ -SE, the SEM measurements were performed and the obtained images are shown in Fig. 2. It

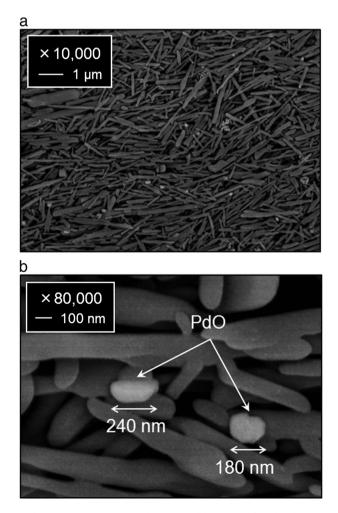


Fig. 2. SEM images (top views, (a):×10,000, (b):×80,000) of TiO₂(+Pd)-SE.

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