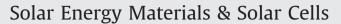
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## Photodegradation of small-molecule organic photovoltaics

Nana Wang<sup>a,d</sup>, Xiaoran Tong<sup>b</sup>, Quinn Burlingame<sup>a</sup>, Junsheng Yu<sup>d</sup>, Stephen R. Forrest<sup>a,b,c,\*</sup>

<sup>a</sup> Department of Electrical Engineering and Computer Science, University of Michigan, Ann Arbor, MI 48109, USA

<sup>b</sup> Department of Materials Science and Engineering, University of Michigan, Ann Arbor, MI 48109, USA

<sup>c</sup> Department of Physics, University of Michigan, Ann Arbor, MI 48109, USA

<sup>d</sup> State Key Laboratory of Electronic Thin Films and Integrated Devices, Department of Optoelectronic Information, University of Electronic Science

and Technology of China (UESTC), Chengdu 610054, PR China

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#### ABSTRACT

We investigate the photostability of organic photovoltaic (OPV) cell active layers comprised of the archetype donor, boron-subphthalocyanine chloride (SubPc), and fullerene acceptors, aged under either AM1.5G illumination or in the dark, and in either air or inert atmosphere. Under long-term exposure to light, we observe significant photobleaching and crystallization of SubPc. Mixing SubPc with  $C_{60}$  as is commonly done in high efficiency small molecule OPVs, the crystallite formation is inhibited and the bleaching is suppressed due to a significantly reduced exciton lifetime in the blends. Furthermore, the spectral dependence of the degradation suggests that photo-dimerization of  $C_{60}$  is an important factor leading to burn-in loss in efficiency previously reported in SubPc/ $C_{60}$  OPVs. The existence of dimerization is supported by Fourier transform infrared spectroscopy data taken both before and after exposure to light. Increasing the fraction of SubPc in a SubPc: $C_{60}$  blend leads to a decrease in the rate of film degradation, providing further evidence for  $C_{60}$  dimerization. Due to its reduced tendency for photo-dimerization,  $C_{70}$  is more stable than  $C_{60}$  when used in small molecule OPVs.

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#### 1. Introduction

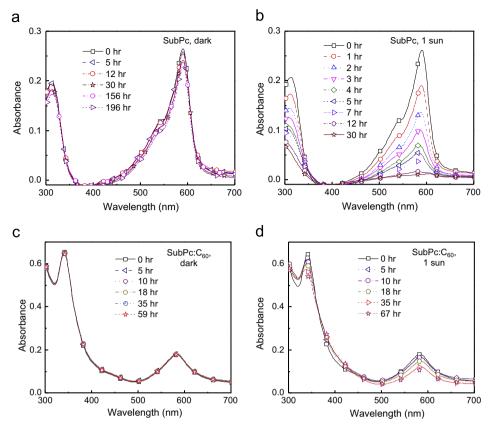
High efficiency, long lifetime, and low cost manufacturing are the three pillars of a successful solar cell technology. Among the various emerging technologies, organic photovoltaic cells (OPVs) are a promising option for renewable energy due to their potential for flexible, low-cost, and large-scale production. While the efficiency of organic photovoltaics is rapidly approaching that required for commercialization [1], their operational lifetimes are below approximately 25 years that is deemed acceptable for practical deployment [2].

Degradation in OPVs can be caused by deterioration of organic molecules [3–9], oxidation of electrodes [10], delamination and reaction of the organic layer with the electrode [11–13], etc. The intrinsic stability of the organic active layer is dependent on both the materials and device structures used. Organic thin film degradation may occur via exciton-induced defects [14–16], morphological changes, photo-oxidation, and other photochemical and thermochemical reactions [4,6–8,17,18]. In polymer solar cells, photo-oxidation in the donors, poly[2-methoxy-5-(2'-ethylhexyloxy)-*p*-phenylenevinylene] (MEH-PPV), poly[2-methoxy-5-(3',7'-dimethyloctyloxy)-1, 4-phenylene vinylene] (MDMO-PPV) and poly(3-hexylthiophene)

(P3HT), has been observed to lead to film bleaching [5,6,19,17]. In addition, the fullerenes are the most commonly used acceptor materials in high efficiency OPVs due to their large absorption coefficients, long diffusion lengths, and high electron mobilities. However,  $C_{60}$  and the substituted fullerene ([6,6]-phenyl- $C_{61}$ -butyric acid methyl ester,  $PC_{60}BM$ ) are known to polymerize upon irradiation [20,21]. In addition, Distler et al. showed that dimerization of  $PC_{60}BM$  leads to the reduction of fill factor and short circuit current in an OPV cell [21]. However, to the best of our knowledge, there have been no reports of the effects of  $C_{60}$  dimerization on device performance.

Previously, we found that exciton-induced traps in  $C_{60}$  are responsible for the deterioration in power conversion efficiency in fullerene-based small molecule OPVs during early stage burn-in [16]. In this work, we examine the photostability of boron subphthalocyanine chloride (SubPc) and  $C_{60}$  thin films employed in that initial study, and find that the burn-in degradation is due to photoinduced dimerization along with photobleaching of the SubPc. Adding either  $C_{60}$  or  $C_{70}$  to the SubPc to form a mixed donor/acceptor active region greatly reduces photo-degradation by facilitating rapid electron transfer to the acceptor before the high energy excitons can cause chemical reactions in the photoactive molecules that result in trap formation. Furthermore, the use of a  $C_{70}$  acceptor results in more stable OPVs due to its reduced tendency to photo-polymerize.

<sup>\*</sup> Corresponding author. *E-mail addresses: jsyu@uestc.edu.cn* (J. Yu), stevefor@umich.edu (S.R. Forrest).



**Fig. 1.** Time evolution of absorbance spectra in air for (a) boron-subphthalocyanine chloride (SubPc), kept in the dark, (b) SubPc, under 1 sun illumination, (c) SubPc: $C_{60}$  (1:4), kept in the dark, (d) SubPc: $C_{60}$  (1:4) under 1 sun illumination.

This paper is organized as follows: in Section 2, we present experimental methods, in Section 3 we provide results, and in Section 4 we discuss the physical origins of photobleaching and OPV degradation. In Section 5 we present conclusions.

#### 2. Experimental

The SubPc and fullerene films were deposited on Si or ITOcoated glass substrates that were previously solvent cleaned followed by CO<sub>2</sub> snow cleaning using procedures described previously [22]. Following ultraviolet (UV)-ozone treatment, substrates were loaded into a high-vacuum (base pressure  $< 4 \times 10^{-7}$  Torr) thermal evaporation chamber where organic materials were deposited at a rate of 1 Å/s. All deposition rates and thicknesses were measured using a quartz crystal thickness monitor. Thicknesses were determined following growth using variable-angle spectroscopic ellipsometry.

To obtain information about the composition of the degradation products, we measured the Fourier transform infrared absorption spectra (Nicolet 6700 FTIR spectrometer) from 100 cm<sup>-1</sup> to 4000 cm<sup>-1</sup> for 200 nm thick films deposited on CaF<sub>2</sub> windows (ESCO Products). The windows were cleaned using a dilute detergent solution, rinsed with deionized water, dried under N<sub>2</sub>, and further cleaned for 4 min with air plasma (Plasma Etch, PE-50). The films were aged for 1 week at ambient temperature. The FTIR sample stage was purged with N<sub>2</sub> prior to and during data collection, and the spectra were corrected for atmospheric water interference and baseline anomalies. Each spectrum was collected using 32 scans with a resolution of 4 cm<sup>-1</sup>.

A subset of deposited films was encapsulated with a glass lid sealed to the substrate using epoxy in an ultrahigh-purity  $N_2$ 

 $(<1 \text{ ppm O}_2 \text{ and } H_2\text{O})$  environment. Aging was done in the dark or at  $35 \pm 5$  °C using a simulated AM1.5G illumination source at 1 sun intensity. Optical intensity was calibrated using an NRELtraceable Si reference cell. Absorbance was recorded by a Lambda 1500 spectrophotometer (Perkin Elmer). X-ray diffraction and atomic force microscopy measurements on as-grown and aged films were obtained in air.

### 3. Results

The absorption spectra of SubPc (11 nm thick) samples aged in air are shown in Fig. 1 For the SubPc kept in the dark, there is a moderate decrease of the peak absorption at a wavelength of  $\lambda$ =590 nm after 30 h. However, when the SubPc sample exposed to 1 sun intensity, AM1.5G illumination, the same absorption peak disappears after 30 h, indicating complete bleaching of the SubPc film. In contrast, a SubPc:C<sub>60</sub> (1:4, 55 nm thick) blend shows no degradation when kept in the dark, and the decrease in absorption for a similar sample under illumination indicates that the rate of photobleaching is significantly slower, with only ~60% loss of the absorption peak at  $\lambda$ =590 nm after 67 h.

We also measured the time evolution of N<sub>2</sub>-packaged SubPc and SubPc:C<sub>60</sub> (1:4) film absorption as shown in Fig. 2. After 2000 h, the reduction in SubPc film absorption is only ~10%, which is considerably less than the sample exposed to air (Fig. 2a). However, after 660 h, we observe a red shift of SubPc absorption at  $\lambda$ =590 nm, while the intensity of the shoulder at  $\lambda$ =540 nm increases slightly along with a similar wavelength shift. Similar trends are observed for the SubPc film aged under 1 sun illumination. After mixing SubPc with C<sub>60</sub> (1:4, SubPc:C<sub>60</sub>), the absorption Download English Version:

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