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Research paper

# Temperature dependent selective detection of hydrogen and acetone using Pd doped WO<sub>3</sub>/reduced graphene oxide nanocomposite



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#### ABSTRACT

Reduced graphene oxide (RGO) and Pd doped WO<sub>3</sub> nanocomposites were fabricated by employing electrostatic interactions between poly (diallyldimethylammonium chloride) (PDDA) modified Pd doped WO<sub>3</sub> nanostructures and graphite oxide (GO) and studied for their gas sensing application. XRD, Raman, FTIR, FESEM-EDX, TEM, TGA, XPS and Photoluminescence techniques were used for characterization of as-synthesized samples. Gas sensing studies revealed that the sensor with optimized doping of 1.5 mol% Pd and 1 wt% GO shows temperature dependent selectivity towards hydrogen and acetone. The role of WO<sub>3</sub>, Pd and RGO has been discussed in detail for enhanced sensing performance.

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#### 1. Introduction

In recent years, metal oxide semiconductors have gained significant attention in gas sensing area for detecting various gases including hydrogen (H<sub>2</sub>) and acetone. Among these metal oxides, tungsten trioxide (WO<sub>3</sub>), an n-type semiconductor, has been considered as one of the promising material for gas sensing application due to its noteworthy role in detection of reducing/oxidizing gases, low cost, and for being environment-friendly [1–5]. However, the practical applicability of WO<sub>3</sub> sensor suffers major drawbacks such as poor selectivity and high operating temperatures which leads to excessive power consumption and deterioration of the sample.

In order to overcome these limitations and to improve the gas sensing performance, recent research has been motivated by realizing the importance of noble metals or carbon materials and hybridization of these materials with metal-oxide semiconductor sensors. The presence of noble metal, whether as a catalyst or as a dopant, alter the specific surface area, activation energy and the bulk resistance of metal oxides which eventually offers improved sensing properties [6]. Due to its excellent chemical, electronic and mechanical properties, the incorporation of reduced graphene oxide (RGO) also enhances sensing properties of metaloxides synergically. RGO provide large specific surface area, high conductivity, and electron acceptor ability to hike the sensing properties [7].

In this paper, we are reporting the possibility of hybridizing the advantages of RGO sheets along with merits of Pd doping in WO<sub>3</sub> nanostructures for temperature dependent selective detection of hydrogen ( $H_2$ ) and acetone. It is a well-known fact that  $H_2$  is one of the promising clean and renewable energy sources and is extensively employed in industries, fuel for power generation and transportation, etc [8]. Due to its colorless, odorless and inflammable nature, the demand for early detection of  $H_2$  is extremely important. Palladium (Pd) is one of the best choices for  $H_2$  sensing due to its extraordinary sticking and diffusion coefficient for  $H_2$  [9]. The catalytic dissociation capability of Pd also contributes to excellent sensitivity [6].

Recently, the hybridization of doped metal oxide semiconductors and RGO has emerged as an innovative approach to improve the overall sensing properties of gas sensors [10]. For example, Esfandiar et al. recently reported Pd doped WO<sub>3</sub>/RGO nanocomposites for the detection of hydrogen [11]. Su et al. reported Pd/SnO<sub>2</sub>/ RGO ternary composite for room temperature NH<sub>3</sub> sensing [12]. Toloman et al. also investigated humidity sensing properties of RGO decorated with Fe doped SnO<sub>2</sub> nanoparticles [13]. Recently, our group reported gadolinium doped WO3 and RGO nanocomposite for acetone detection [14]. To the best of our knowledge, there are no reports available in literature on temperature dependent selective behaviour based on Pd doped WO<sub>3</sub> nanostructures and RGO nanocomposites towards H<sub>2</sub> and acetone. To accomplish this, first of all, pure and Pd doped (0.5, 1.0 and 1.5 mol%) WO<sub>3</sub> nanostructures were synthesized using simple acid precipitation method and subsequently studied for gas sensing application. The sensor based on optimum doping amount of Pd doping i.e. 1.5 mol% has

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exhibited temperature dependent selective behaviour. Further improvement in sensing response and reduction in working temperature has been achieved by incorporating RGO. The role of Pd doping and RGO incorporation on enhanced gas sensing performance of WO<sub>3</sub> nanostructures has been discussed in detail.

#### 2. Experimental

The materials used in this work are given in supplementary material (see S2.1). Methodology used to prepare pure WO<sub>3</sub>, Pd doped WO<sub>3</sub> and RGO based nanocomposites is same as described in our previous work reported elsewhere [14]. However, detail synthesis process can be found in supplementary material (see S2.2). Room temperature acid precipitation was used to prepare pure and Pd doped WO<sub>3</sub> nanostructures where Na<sub>2</sub>WO<sub>4</sub>·2H<sub>2</sub>O and PdCl<sub>2</sub> were the precursors of WO<sub>3</sub> and Pd, respectively. Different concentrations of Pd (0.5, 1.0 and 1.5 mol%) were used and labeled as PdO.5, Pd1.0, and Pd1.5, respectively. Graphite oxide (GO) was synthesized by means of modified Hummer's method [15]. The attachment of Pd1.5 on to RGO sheets was attained by electrostatic interactions by using PDDA. Samples with different concentration of GO (0.5, 1.0 and 2.0 wt%) were labelled as PdG0.5, PdG1.0 and PdG2.0, respectively.

The details of all characterization techniques and gas sensing measurements are given in supplementary material (see S2.3).

#### 3. Results and discussion

#### 3.1. Characterization of as-synthesized samples

The XRD spectra of WO<sub>3</sub> and Pd doped WO<sub>3</sub> nanostructures are displayed in Fig. 1a. The presence of three strong peaks versus  $2\theta$  ranging from 22 to  $25^{\circ}$  correspond to reflection planes ( $2\,0\,0$ ), ( $0\,2\,0$ ) and ( $0\,0\,2$ ) indicating the monoclinic phase of WO<sub>3</sub> nanostructures which is consistent with standard data (JCPDS-83-0951) [16]. In Pd doped WO<sub>3</sub> samples, no peak related to Pd appears in XRD patterns which may be attributed to successful incorporation of Pd into the WO<sub>3</sub> lattice. The intensity of diffraction peaks also decreases with Pd doping which arises from lattice distortion induced by Pd as the ionic radius of host ion is smaller (ionic radii of W<sup>6+</sup> and Pd<sup>2+</sup> are 67 and 78 pm respectively) than dopant ion

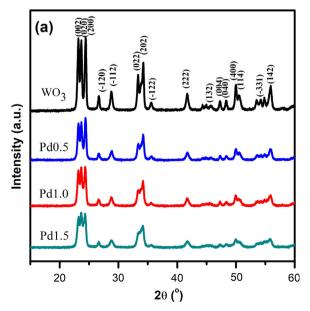


Fig. 1a. XRD patterns of pure and Pd doped WO<sub>3</sub> nanostructures.

[17]. The average crystallite size (D) has been estimated using Scherrer's equation [18] by considering triplet peaks (2 0 0), (0 2 0) and (0 0 2) and the obtained values are 28.70, 18.89, 19.79, and 18.04 nm for pure WO<sub>3</sub>, Pd0.5, Pd1.0, and Pd1.5 respectively. The average crystallite size decreases with Pd doping implying that incorporation of Pd restrains the growth of WO<sub>3</sub> nanostructures which is beneficial in improving sensing performance. Similar reduction in crystallite size on doping has been reported in literature [14,19].

We also performed Rietveld refinements on XRD data of pure and Pd doped WO $_3$  samples with Full Prof program using P 21/n space group. The refined XRD patterns are shown in Fig. 1b and the lattice parameters ('a', 'b', 'c' and ' $\beta$ ' and Chi-square values) extracted from Rietveld analysis are displayed in Table S1 (see supplementary material). Upadhyay et al. have observed similar variations in lattice parameters and cell volume in case of indium doped WO $_3$  [16] and have attributed this deviation to presence of dopant in WO $_3$  lattice.

The XRD patterns of GO and RGO are displayed in Fig. 1c, showing a prominent peak at  $2\theta$  = 10.90° for GO corresponding to (0 0 1) reflection of graphite oxide. This peak vanishes after reduction of GO into RGO [20]. The RGO based nanocomposites exhibit XRD patterns (Fig. 1c) similar to monoclinic WO<sub>3</sub>. No peaks corresponding to Pd, GO or RGO were detected in nanocomposite which may be attributed to their low content and proper exfoliation of RGO sheets [21,22].

Fig. 2a depicts the Raman spectra of pure and Pd doped WO<sub>3</sub> nanostructures. Raman spectrum of pure WO<sub>3</sub> shows the presence of W-O-W stretching (808 and 715 cm<sup>-1</sup>) and O-W-O bending vibration modes (327 and 275 cm<sup>-1</sup>) [16]. The effect of Pd dopant can be seen by slight broadening and shifting of above bands, thus confirming the incorporation of Pd into host lattice. The Pd doped WO<sub>3</sub> nanostructures also exhibited a reduction in intensity in comparison to pure WO<sub>3</sub> which is ascribed to the existence of structural disorder that originates upon doping [19].

Fig. 2b displays the Raman spectra of GO, RGO and RGO based nanocomposites. It is observed that GO displays two notable D and G bands at around 1358 and 1604 cm $^{-1}$ . The D band is ascribed to sp $^3$ - hybridized carbon and presence of defects, whereas, G band is accredited to first-order scattering of  $E_{2g}$  phonons of sp $^2$  carbon atoms [23]. These bands are also present in RGO and nanocomposites but with slight shifting towards lower wavenumbers. The typical bands of WO $_3$  are also present in nanocomposites. Besides this, we also measured the intensity ratio of D and G bands ( $I_D/I_G$ ) which shows an increase in its value for RGO and nanocomposites in comparison to GO. In general, the higher ratio of  $I_D/I_G$  signifies the degree of defects and disorder in graphitic materials and presence of these defects boosts the adsorption of oxygen and gas species which could be beneficial in improving gas sensing response [24].

The FTIR spectra of WO<sub>3</sub>, Pd1.5, GO, and PdG1.0 are shown in Fig. 3a. The spectrum of GO shows a broad band at 3444 cm<sup>-1</sup> and is attributed to O-H stretching vibrations. The characteristic absorption band of C=C stretching vibration is found at 1643 cm<sup>-1</sup> whereas the peak present at 1741 cm<sup>-1</sup> is related to COOH stretching vibrations. The intensity of O-H absorption band decreases in RGO based nanocomposite, thus confirming the successful reduction of RGO. The other absorption bands at 1741, 1371 and 1085 cm<sup>-1</sup> are attributed to a typical stretching vibration of —C=O (carboxyl), deformation vibration of OH and stretching vibration of -C-O (alkoxy), respectively [25]. Except GO, in all other three samples, the bands below 1000 cm<sup>-1</sup> are associated with stretching vibrations of WO<sub>3</sub> [26]. However, with Pd doping, a frequency shift towards lower side related to these W-O-W stretching vibrations (Fig. 3b) has been observed, suggesting the successful substitution of Pd into the WO<sub>3</sub> lattice [27]. Moreover, it is to be noted that in case of RGO based nanocomposite

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