



Emission tuning studies in BaMgSiO₄:RE (RE = Eu²⁺, Sr²⁺) for white LEDs

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HIGHLIGHTS

- The emission of Eu²⁺ in BaMgSiO₄:Eu²⁺ shifts to the green region by increasing Eu concentration.
- The emission of Eu²⁺ in BaMgSiO₄:Eu²⁺, ySr²⁺ shifts to the blue region by substitution Ba²⁺ by Sr²⁺.
- The energy migrates from the Eu_{(I,II)}O₉ to the Eu_{III}O₆ polyhedron efficiently.}

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ABSTRACT

Color tuning is a challenge for improving white light LEDs. In this paper, the color tuning in Ba_{1-x}MgSiO₄:xEu²⁺ (0 ≤ x ≤ 0.07) and Ba_{0.993-y}MgSiO₄:0.007Eu²⁺, ySr²⁺ (0 ≤ y ≤ 0.07) are investigated. The emission can be tuned by increasing Eu²⁺ concentration and substitution of the host lattice cation Ba²⁺ by Sr²⁺. The emission shows a red shift in Ba_{1-x}MgSiO₄:xEu²⁺ with the increase of Eu²⁺ concentration. The red shift results from the preferential occupation of Eu²⁺. On the contrary, the emission of Ba_{0.993-y}MgSiO₄:0.007Eu²⁺, ySr²⁺ shows a blue shift when partial substitution of Ba²⁺ by Sr²⁺. The blue shift is explained by the preferential occupation of Sr²⁺, crystal field splitting of Eu²⁺ and the electron binding energy of Eu-O.

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1. Introduction

White light-emitting diodes (LEDs) is a new generation of illumination as substitutes for conventional fluorescent lamps owing to the long lifetime, low energy consumption, and environmentally friendly characteristics [1–3]. Nowadays, white LEDs are mainly composed by the InGaN blue chip coated on the yellow phosphor Y₃Al₅O₁₂:Ce³⁺ (YAG:Ce³⁺) [4]. This technology has several drawbacks including the low color-rendering index (CRI) and the high color temperature (CCT) [5,6]. To overcome these problems, ultraviolet (UV) or near ultraviolet (NUV) LED red-blue-green (RBG) white LEDs have been suggested for the general illumination. Using the RBG phosphor can provide a higher CRI with good quality white light than the blue chip and YAG:Ce³⁺ method. Moreover, the ability to tune the emission color of a luminescent material is of great importance for practical applications.

BaMgSiO₄:Eu²⁺ is a potential white LED phosphor for it can be efficiently excited by the photon of UV and NUV and emits the photon of blue and green [7–9]. Eu²⁺ and Mn²⁺ co-doped BaMgSiO₄ shows the white light [10]. BaMgSiO₄:Eu²⁺ attracts our attention mainly because of the special crystal structure of BaMgSiO₄ host. The Ba ions in BaMgSiO₄ occupy both the 9-coordinated (Ba_I and Ba_{II}) and 6-coordinated (Ba_{III}) polyhedron [11]. The multi-Ba sites properties in BaMgSiO₄ lead to the different crystalline field intensity of BaO_n polyhedron. The energy level splitting of 5d orbit would be different when Eu²⁺ (outer electron configuration: 4f⁷) replace the different Ba sites, thus the Eu²⁺ in BaMgSiO₄:Eu²⁺ shows the multi-colored emission. But there is a problem that how to control the doped ions to substitute the appointed Ba sites and obtain the expected emission color.

In this work, we mainly discussed the color tuning and the energy transfer of the Eu²⁺ and Sr²⁺ doping ions in BaMgSiO₄ in combining with the photoluminescence spectra of BaMgSiO₄:RE (RE = Eu²⁺, Sr²⁺) under the NUV excitation.

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2. Experimental section

BaCO₃ (99.50%), (MgCO₃)₄·Mg(OH)₂·5H₂O (99.50%), SiO₂(99.50%), Eu₂O₃ (99.99%) and SrCO₃ (99.90%) are used as the starting materials. First, the starting materials are mixed together in stoichiometric amounts. After intimately mixing the starting materials with ethanol, the obtained mixture are fired at 1300 °C for 8 h under a reducing atmosphere (H₂:N₂ = 5%:95%), and then cooled to the room temperature.

All the samples are checked by X-ray diffraction (XRD) using Rigaku diffractometer with Ni-filtered CuK α radiation. The UV and NUV excitation and emission spectra are measured by a Xe lamp (FLS-920T). Time-resolved fluorescence measurements were performed by time correlated single-photon counting on FLS-920T fluorescence spectrometer (Edinburgh Instruments) equipped with a pulsed hydrogen lamp. All the measurements are performed at room temperature.

3. Results and discussion

3.1. XRD analysis

The XRD patterns of Ba_{1-x}MgSiO₄:xEu²⁺ ($0 \leq x \leq 0.07$) and

Ba_{0.993-y}MgSiO₄:0.007Eu²⁺, ySr²⁺ ($0 \leq y \leq 0.07$) are respectively shown in the Fig. 1(a) and (b), together with the standard card No.81-2317, the insert is the enlarged image in the region of 27°–29.5°. From Fig. 1(a) and (b), it is observed that the obtained reflection peaks are matched well with the reported hexagonal phase BaMgSiO₄, except for a weak impurity peak at 32.1° which belongs to the Ba₂MgSi₂O₇ phase. The diffraction peaks of Ba_{1-x}MgSiO₄:xEu²⁺ ($0 \leq x \leq 0.07$) slightly shift to the large angle region with increasing the Eu concentration, which is shown in the enlarged image in the Fig. 1 (a). Based on ref. [11], the Ba²⁺ ions in BaMgSiO₄ occupy three sites with the coordination number (CN) 6, 9 and 9. The effective ionic radius of Ba²⁺ is respectively 1.35 Å and 1.47 Å in the hexa- and nine-coordinated polyhedron according to Ref. [12], which is larger than that of Eu²⁺ (1.17 Å and 1.30 Å). Therefore, the doped Eu²⁺ ions to replace Ba²⁺ ions will result in the reduction of the unit cell, and then the diffraction peaks of Ba_{1-x}MgSiO₄:xEu²⁺ ($0 \leq x \leq 0.07$) shift to the large angle region. The similar phenomenon is also observed in the XRD patterns of Ba_{0.993-y}MgSiO₄:0.007Eu²⁺, ySr²⁺ ($0 \leq y \leq 0.07$) in Fig. 1 (b). The reason of the diffraction peak shift in Fig. 1 (b) is same as that of Ba_{1-x}MgSiO₄:xEu²⁺ ($0 \leq x \leq 0.07$) and is the effective ionic radii of Ba²⁺ (1.35 Å and 1.47 Å) are bigger than that of Sr²⁺ (1.18 Å and 1.31 Å) [12]. This phenomenon is consistent with Vegard's law and

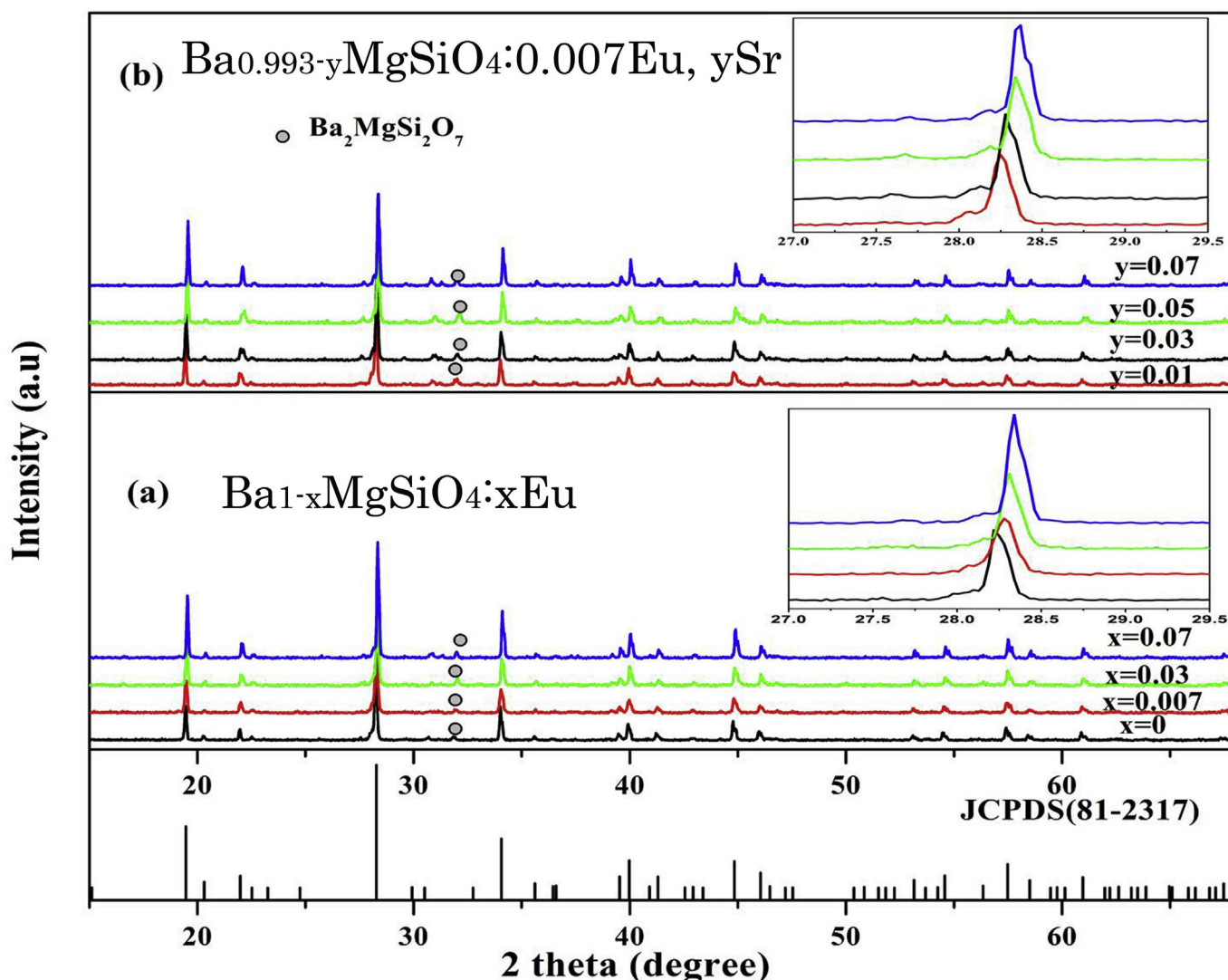


Fig. 1. XRD patterns of Ba_{1-x}MgSiO₄:xEu²⁺ ($0 \leq x \leq 0.07$) (a) and Ba_{0.993-y}MgSiO₄:0.007Eu²⁺, ySr²⁺ ($0 \leq y \leq 0.07$) (b), together with the standard card No.81-2317.

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