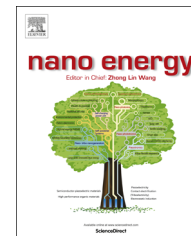


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RAPID COMMUNICATION

A general strategy for the fabrication of high performance microsupercapacitors

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Abstract

We propose a generic strategy for microsupercapacitor fabrication that integrates layers of reduced graphene oxide (rGO) and pseudocapacitive materials to create electrode heterostructures with significantly improved cycling stability and performance. Our approach involves a combination of photolithography and a simple transfer method of free-standing reduced graphene oxide film onto an Au/patterned photoresist bilayer. The resulting stack (rGO/Au/patterned resist/substrate) is then used for the electrochemical deposition of various pseudocapacitive materials before the final step of lift-off. To prove the viability of this method, we have successfully fabricated microsupercapacitors (MSCs) with the following interdigitated electrode heterostructures: MnO_2/rGO , $\text{Co(OH)}_2/\text{rGO}$ and PANI/rGO . These MSCs show better performance and cycling stability compared to the single layer, (i.e., rGO-free) counterparts. The interdigitated electrode heterostructures result in MSCs with energy densities in the range of 3–12 mW h/cm^3 and power densities in the range of 400–1200 mW/cm^3 , which is superior to the Li thin film batteries ($E=10 \text{ mW h/cm}^3$), carbon, and metal oxide based MSCs ($E=1\text{--}6 \text{ mW h/cm}^3$) while device energy densities are in the range of 1.3–5.3 mW h/cm^3 , corresponding power densities are in the range of 178–533 mW/cm^3 . These results can be explained by a facilitated nucleation model, where surface topology of the rGO film creates a favorable environment for the nucleation and growth of pseudocapacitive materials with strong interfacial contacts and enhanced surface area. This approach opens up a new avenue in fabricating MSCs involving a variety of heterostructures combining electrical double layer carbon type with Faradaic pseudocapacitive materials for enhanced electrochemical performance.

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1 Introduction

3 The next generation of energy storage devices is expected
5 to include devices fabricated in the planar format due to
7 their compatible integration with microelectronic devices
9 [1-3]. This reality has fostered the rapid development of on-
11 chip energy storage devices such as thin film batteries and
13 microsupercapacitors (MSCs) [4-7]. MSCs have become
attractive owing to their high power densities and longer
cycle life compared to thin film batteries, which suffer from
poor rate capability and limited cycle lifetimes [4,6].
Additionally, due to their planar configuration, MSCs can
exhibit higher charge-discharge rates compared to their
conventional counterparts [6,7].

15 High surface area porous carbon based MSCs such as
17 carbide derived [8], onion like [9], activated [10] and carbon
nanotubes (CNTs) [11] were fabricated by employing con-
19 ventional photolithography and various deposition methods
including sputtering, electrophoretic, ink-jet printing and
21 spray coating techniques. Further, conducting graphitic
patterns involving 2D graphene and 1D CNTs were grown
23 by chemical vapor deposition (CVD) to fabricate three-
dimensional MSCs [12,13]. Of late, reduced graphene oxide
(rGO), due to its high surface area, conductivity and
25 functionality, has become an attractive material for energy
storage applications [14,15]. For example, Gao et al., have
27 used laser reduction to write reduced graphene oxide
patterns over the GO thin films in order to fabricate rGO
29 MSCs [16]. El-Kady et al., have demonstrated scalable
fabrication of rGO MSCs by ordinary digital video disk
31 (DVD) laser scribing technique [17]. Wu et al., have
employed photolithography and oxygen plasma to create
33 rGO based micropatterns and further demonstrated elec-
trochemical performance of these MSCs [18]. As the
35 mechanism of charge storage in these carbonaceous materi-
als is of non-Faradaic type, these MSCs exhibited limited
37 values of capacitance. Hence, in order to improve the
capacitance values, pseudocapacitive materials such as
39 metal oxides/hydroxides (RuO_2 , MnO_2 and Ni(OH)_2) [19-21]
and even conducting polymers (PPY, PANI) [22-24] which can
41 undergo reversible redox reactions at their surfaces have
been employed to fabricate micro-pseudocapacitors. How-
43 ever, pseudocapacitive materials suffer from poor cycling
performance, which has been circumvented through the
45 fabrication of hybrid electrodes composed of composites of
rGO with pseudocapacitive materials [25-27]. The compli-
47 mentary roles between these two components have been
exploited where reduced graphene oxide serves not only as
49 a conducting network for facile charge transport, but also as
a mechanical support in improving cycling stability of these
51 hybrid electrodes. Recently, there have been efforts in
fabricating pseudocapacitive/reduced graphene oxide elec-
53 trode heterostructures employing micromolding and filtra-
tion methods [28,29].

55 Patterning of pseudocapacitive/rGO heterostructures is of
significant interest for achieving better electrochemical
57 performance of MSCs. Here, we propose a simple and generic
strategy for the fabrication of pseudocapacitive/rGO planar
59 microsupercapacitor devices using a single step photolitho-
graphy process. Free standing rGO films, obtained through
61 vacuum filtration process, are transferred onto substrates

having metal-coated patterned photoresist layer. The result-
ing stack (rGO/metal/patterned resist/substrate) is selec-
tively coated by metal oxides (MnO_2 and Co(OH)_2) and
conducting polymer (PANI) over the transferred rGO layer,
followed by lift-off process. The result is a MSC with
interdigitated patterns of different pseudocapacitive materi-
als on rGO. As microsupercapacitor is comprised of a few
microns thick film, electrochemical deposition can be a facile
method, resulting in the mesoporous morphology of pseudo-
capacitive materials, required for maximum accessible sur-
face area for the electrolyte ions. Avoiding binders or
additives, while maintaining good interfacial contact through
electrochemical deposition, ensures strong adherence with
current collectors, helps in facile electron transfer across the
pseudocapacitive/current collector interface. This in-plane
design of hybrid electrodes with no separator ensures facile
transport of ions, resulting in high scan rate abilities of
heterostructured microsupercapacitor with improved cycling
stability.

Experimental

Photolithography

Glass substrates (Fisher) were cut into 1×1 in. size, cleaned
with a soap solution to remove the dirt followed by ultra-
sonication in acetone, isopropanol and deionized water
sequentially for 5 min each and then dried by blowing
nitrogen. Photoresist AZ9260 was spun coated at 3000 rpm
for 60 s over the glass substrates to get $10 \mu\text{m}$ thick photoresist
layer. Photoresist coated substrates were soft baked at 110°C
for 3 min. The exposure was done using EVG contact aligner at
a constant dose of $1800 \text{ mJ}/\text{cm}^2$ through the Cr/Glass mask
having the interdigitated patterns. After the exposure, sam-
ples were developed in AZ726 developer solution for 6 min,
which has resulted in the formation of patterns in the
photoresist layer. Metal layers of 200 nm Au/20 nm Ti were
deposited by sputtering (Equipment Support Co., Cambridge,
England) technique over the patterned photoresist layer.
Before the lift-off process, rGO layer was transferred onto
metal-coated patterned photoresist followed by electrodepo-
sition of pseudocapacitive materials. In this study, we have
employed interdigitated finger electrodes (width of each
finger $100 \mu\text{m}$, and spacing between the fingers is $50 \mu\text{m}$);
the total area of all the fingers is 0.25 cm^2 .

RGO preparation and filtration

Graphite oxide was prepared from natural graphite source
using a modified Hummers method [30]. Thus obtained
graphite oxide was exfoliated in de-ionized (DI) water by
sonicating using a bath sonicator (UP400S, Ultrasonic pro-
cessor; Hielscher ultrasound Technology) for 1 h. The result-
ing graphene oxide was then reduced to graphene by
following the method that was reported by Li et al. [31]
Briefly, the homogeneous graphene oxide dispersion
(5.0 mL) was mixed with 5.0 mL of water, $5.0 \mu\text{L}$ of hydra-
zine solution (35 wt% in water) and $100.0 \mu\text{L}$ of ammonia
solution (28 wt% in water). The optimal hydrazine to
graphene oxide weight ratio was 7:10 [31,32]. The mixture

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