

## Origin of ferromagnetism in Cu doped rutile TiO<sub>2</sub> - An *ab-initio* approach

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### ABSTRACT

Origin of room temperature ferromagnetism in copper doped rutile TiO<sub>2</sub> has been studied by varying both copper concentrations as well as oxygen vacancies. *Ab-initio* calculations have been carried out within the framework of density functional theory. Copper atom has been doped at two different positions to study the effect of distance between the dopants for ferromagnetism in TiO<sub>2</sub> system. Spin polarized density of states calculation has been performed for each system and the induced magnetic moment has been calculated along with the total free energy and Fermi energy. Spin-spin interaction studies have been performed for two copper doped at Ti site (Ti<sub>22</sub>Cu<sub>2</sub>O<sub>48</sub>) as well as two oxygen vacancies along with Cu doping at Ti site (Ti<sub>22</sub>Cu<sub>2</sub>O<sub>46</sub>). Spin-spin interaction results have been analyzed to understand the possibility of ferromagnetism in the copper doped rutile TiO<sub>2</sub>. It has been found that copper doped system with adjacent oxygen vacancies gives ferromagnetic ordering.

### 1. Introduction

Recently, dilute magnetic semiconductors (DMS) based upon TiO<sub>2</sub> have drawn lots of attention of the researchers due to its potential application in solar cell, opto-electronic, magneto-optic devices [1–4]. High dielectric constant and high refractive index of rutile-TiO<sub>2</sub> make it preferred choice for electronic and optical purposes [5]. TiO<sub>2</sub> nano-materials are widely used as photocatalysts in industrial scale because of its high catalytic efficiency, stability, cost efficiency and environmentally benign nature [6]. Magnetic properties along with intrinsic semiconducting properties can be used for better storage and swift response in electrical devices [7]. This property of ferromagnetic-semiconductors makes them suitable choice for spintronics application [8]. Pristine TiO<sub>2</sub> is a wide-band-gap semiconductor with band gap of ~3 eV [9]. Due to this band gap it can only absorb in UV region but with the incorporation of defects it can be tuned to absorb in the visible part of the spectrum [10]. This large band gap can be reduced by doping TiO<sub>2</sub> with Cu because this doped system has some added states at the top of the valence band [11]. Photocatalytic activity of TiO<sub>2</sub> has been studied by co-doping Mo and Cu in TiO<sub>2</sub> [12]. Numerous studies have been done on defect induced ferromagnetism in oxides (like TiO<sub>2</sub>, ZnO, MgO, SnO<sub>2</sub> etc.) [13–17]. However, in TiO<sub>2</sub> it has been shown both theoretically and experimentally that both cation vacancies [18] as well as anion vacancies [19] can induce room temperature ferromagnetism. Copper being a non magnetic element, Cu doped TiO<sub>2</sub> will give a direction for the study of room temperature ferromagnetism

(RTFM) in wide-band- gap semiconductors. Copper doped TiO<sub>2</sub> can be used as a photocatalytic agent [20] and it also shows bacterial inactivation properties [21]. Duhalde et al., showed that a magnetic moment of about 1.5 μ<sub>B</sub> can be induced per Cu dopant with one oxygen vacancy in (TiO<sub>2</sub>) [22]. However, Torres et al. proposed that Cu doped at Ti site generates a ferromagnetic behavior [23]. Hau et al., observed room temperature ferromagnetism in copper doped TiO<sub>2</sub> thin films. Further, they reported that the ferromagnetic coupling strongly depends upon distance between the dopants and also the magnetism decreases with high dopant concentration [24]. S. A. Ahmed [25] carried out the magnetic measurements for Cu doped TiO<sub>2</sub> prepared by solid state technique and concluded that ferromagnetic property in TiO<sub>2</sub> system is an intrinsic character. Furthermore, magnetic study carried out by Xu et al., for pristine and Cu doped rutile TiO<sub>2</sub> is in agreement to this results and concluded that oxygen vacancies are responsible for inducing ferromagnetism in this system [26]. Despite of all these studies the origin of ferromagnetism in Cu doped TiO<sub>2</sub> system is yet not clear and the results are sometimes self contradictory. Hence, it needs further study to understand the process in detail.

In the present work, we have performed a first principle calculation under the framework of density functional theory. Possibilities of inducing ferromagnetic properties have been studied by considering the different TiO<sub>2</sub> system (i.e. Ti<sub>24-n</sub>Cu<sub>n</sub>O<sub>48-m</sub>, n and m have been assigned values 0, 1, 2, 3 as required in different configurations). Different concentrations of Cu doping and oxygen vacancy have been considered along with the doping distance of two Cu atoms for complete

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understanding of the process.

## 2. Computational methodology

*Ab-initio* calculations under the framework of Density Functional Theory (DFT) have been employed using VASP (Vienna *ab-initio* Simulation Package) Code, along with MedeA Simulation Package [27–30]. The generalized gradient approximation (GGA) with Perdew-Burke-Ernzerhof (PBE) exchange and correlation [31] method has been followed to study the system. Periodic boundary conditions have been introduced along all the basis vectors. A  $2 \times 2 \times 3$  super cell with 72 atoms is build by multiplying the tetragonal unit cell of rutile  $\text{TiO}_2$  with optimized lattice constants of  $a = b = 4.59 \text{ \AA}$  and  $c = 2.95 \text{ \AA}$  and  $\alpha = \beta = \gamma = 90^\circ$ . The structure in each case has been geometrically relaxed until the maximum value of the unbalanced inter-atomic force component (Hellman-Feynman force) is less than  $0.02 \text{ eV/\AA}$ .  $\text{TiO}_2$  system having different atomic concentration ( $\text{Ti}_{24-n}\text{Cu}_n\text{O}_{48-m}$ ) and different doping distance have been taken for detailed understanding of the process. For  $\text{Ti}_{22}\text{Cu}_2\text{O}_{46}$  system ( $\sim 3 \text{ at. \%}$  Cu doping has been considered along with oxygen vacancies ( $V_{\text{O}}$ )), two Ti atoms have been substituted with two Cu atoms and two oxygen atoms are removed from adjacent site of Cu in the pristine  $\text{Ti}_{24}\text{O}_{48}$  structure. Similarly, corresponding changes have been made in  $\text{TiO}_2$  system for generating other systems taken under consideration. In all the cases,  $V_{\text{O}}$  have been created adjacent to the doped atom as it is more stable [22]. Two different positions have been fixed for the doping of Cu atom, one at near distance ( $\sim 3.5 \text{ \AA}$ ) and other one at long position ( $\sim 6.5 \text{ \AA}$ ). To study the doping of three atoms, two different cases have been considered, varying the dopant at middle position. In all calculations,  $400 \text{ eV}$  mesh cut-off energy has been taken into account to expand the plane wave basis set and  $10^{-5} \text{ eV}$  tolerance has been fixed as stopping criteria of the self-consistent loop to reach the electronic ground state. The Brillouin zone (BZ) of the super cells has been divided by  $4 \times 4 \times 4$  Monkhorst-Pack (MP) k-points [32]. Each system has been structure optimized and spin polarized density of states have been calculated along with the induced magnetic moment, Fermi energy and free energy. The ground state energies of ferromagnetic and anti-ferromagnetic ordering have been compared after considering the spin - spin interaction study in the spin polarized condition. The free energies of the different spin polarized systems are compared to get the stability information for corresponding parallel and anti-parallel spin state. Ferromagnetic ordering has been achieved by enforcing all the spins to align in one direction and G-type anti-ferromagnetic state has been achieved for calculating anti-ferromagnetic state.

## 3. Results and discussion

Origin of room temperature ferromagnetism (RTFM) in Cu doped rutile  $\text{TiO}_2$  has been studied for different dopant concentrations. Both the effects of Cu doping and oxygen vacancies ( $V_{\text{O}}$ ) have been considered by varying the number of Cu atoms as well as oxygen vacancies. In reality, there are many defects disorders incorporated in  $\text{TiO}_2$  matrix during its synthesis; e.g., oxygen vacancies, metal vacancies, metal interstitials, formation of  $\text{Ti}^{3+}$ , formation of  $\text{O}^-$  etc [6]. Among them defects due to oxygen vacancies are most common intrinsic defects in the system as the defect formation energy for  $\text{TiO}_2$  system with  $V_{\text{O}}$  is less [18]. For this reason we have studied the Cu doped  $\text{TiO}_2$  systems with oxygen vacancies. Different positions have been chosen for substitution of Cu atoms at Ti site in  $\text{TiO}_2$  super cell. Sites for two Cu atom doping at short distance (1, 2 site, labeled as ‘S’) and for long distance, (1, 3 site, labeled as L) has been chosen as shown in Fig. 1.

The optimized distance between the two Cu atoms doped in  $\text{Ti}_{22}\text{Cu}_2\text{O}_{48}$  is found  $3.57 \text{ \AA}$  for short position and  $6.51 \text{ \AA}$  for long position. Similarly, for three Cu doped system (i.e.,  $\text{Ti}_{21}\text{Cu}_3\text{O}_{48}$ ) two different cases have been considered, one at the near site (1, 2, 4) and other at long site (1, 3, 4) varying the dopant at middle position as

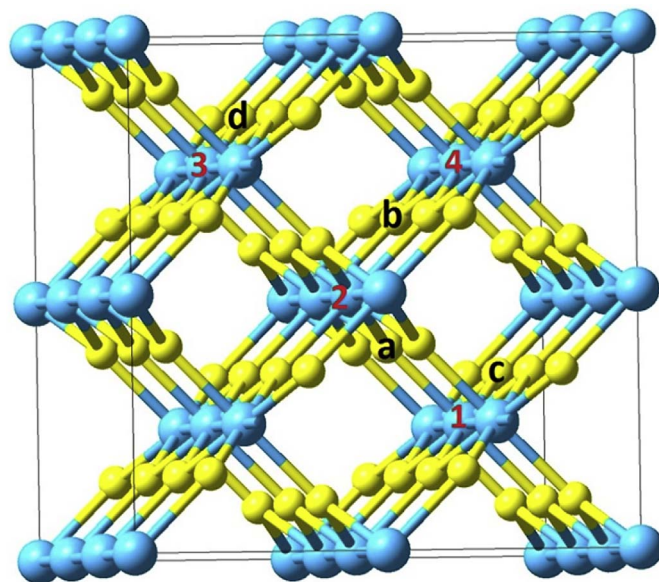


Fig. 1. Structure of pristine  $\text{TiO}_2$  showing positions of doped Cu and  $V_{\text{O}}$ . The blue spheres and yellow spheres represent Ti and O atom respectively in the 72 atom super cell of  $\text{TiO}_2$ .

Table 1

The optimized distance, induced magnetic moment, defect formation energy and Fermi energy shift with respect to pristine  $\text{TiO}_2$ .

$\text{TiO}_2$ System	Position of doped copper	Position of oxygen vacancy	Cu–Cu distance ( $\text{\AA}$ )	Magnetic Moment ( $\mu_{\text{B}}$ )	Defect formation Energy (eV)	Fermi energy Shift (eV)
$\text{Ti}_{23}\text{CuO}_{48}$	2	NA	NA	1.45	12.04	1.23
$\text{Ti}_{23}\text{CuO}_{47}$	2	a	NA	1.01	18.61	1.72
$\text{Ti}_{23}\text{CuO}_{46}$	2	a, b	NA	0.91	28.76	2.81
$\text{Ti}_{22}\text{Cu}_2\text{O}_{48}$	1, 2	NA	3.57	1.26	24.49	1.12
$\text{Ti}_{22}\text{Cu}_2\text{O}_{47}$	1, 2	a	3.58	1.39	29.71	1.17
$\text{Ti}_{22}\text{Cu}_2\text{O}_{46}$	1, 2	a, c	3.48	0.65	34.41	1.69
$\text{Ti}_{22}\text{Cu}_2\text{O}_{48}$	1, 3	NA	6.51	1.68	24.33	1.08
$\text{Ti}_{22}\text{Cu}_2\text{O}_{46}$	1, 3	c, d	6.51	1.96	33.71	1.59
$\text{Ti}_{21}\text{Cu}_3\text{O}_{48}$	(1,2); (2,4); (1,4)	NA	3.57; 3.57; 4.59	0.14	36.93	1.0
$\text{Ti}_{21}\text{Cu}_3\text{O}_{48}$	(1,3); (3,4); (1,4)	NA	6.50; 4.59; 4.59	2.36	36.80	0.96

marked in Fig. 1. The details of the optimized distance, induced magnetic moment, defect formation energy and Fermi energy shift in different Cu doped  $\text{TiO}_2$  systems have been tabulated in Table 1.

Doping of one Cu atom in  $\text{TiO}_2$  produces a significant magnetic moment in  $\text{TiO}_2$ . The Cu atom has  $4s^1 3d^{10}$  as the valence electronic configuration and when  $\text{Ti}^{4+}$  is substituted by Cu it becomes  $\text{Cu}^{2+}$  [33]. Substitution of  $\text{Cu}^{2+}$  in place of  $\text{Ti}^{4+}$  and generation of an oxygen vacancy lead to charge neutrality of the local environment [25]. Though Copper ions can be present in two major oxidation states,  $\text{Cu}^+$  and  $\text{Cu}^{2+}$ , but for substitution of  $\text{Cu}^{2+}$  in place of  $\text{Ti}^{4+}$  in  $\text{TiO}_2$  matrix, generally  $\text{Cu}^{2+}$  is predominant in absence of any complex forming ligand having high affinity for  $\text{Cu}^{2+}$  [34]. Magnetic moment is generated in non magnetic  $\text{TiO}_2$  matrix, due to an unpaired electron in  $\text{Cu}^{2+}$ . The magnetic moment of  $\text{Cu}^{2+}$  ion polarizes the oxygen atoms in its neighboring atmosphere. When these neighboring oxygen atoms are removed, the contribution from these atoms decreases leading to overall decrease in net magnetic moment. After creating oxygen vacancy, magnetic moment decreases to  $1.01 \mu_{\text{B}}$  and  $0.91 \mu_{\text{B}}$  for single  $V_{\text{O}}$  and two  $V_{\text{O}}$  respectively from  $1.45 \mu_{\text{B}}$  in case of Cu doping without vacancy. The substitution of two Cu atoms at short position (i.e., 1, 2

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