#### Computational Materials Science 77 (2013) 53-60

Contents lists available at SciVerse ScienceDirect

# **Computational Materials Science**

journal homepage: www.elsevier.com/locate/commatsci

# Effect of particle size distribution on debonding energy and crack resistance of polymer composites

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#### ARTICLE INFO

Article history: Received 11 January 2013 Received in revised form 1 March 2013 Accepted 3 April 2013 Available online 14 May 2013

Keywords: Particle-reinforced composites Crack resistance Particle debonding Particle size distribution

#### ABSTRACT

Crack resistance of particle reinforced polymers is affected by the size distribution of particles. Particle debonding is a major dissipation mechanism that contributes itself and triggers other mechanisms such as matrix shear bands or plastic void growth. Assuming the specific debonding energy at the particle/ matrix interface as independent of particle size together with the debonding criterion that depends on the particle size leads to analytical expressions that depend on the parameters of the particle size distribution function as well on the debonding probability function. But numerical results show nearly constant crack resistance by changing mean particle size. Using instead a debonding criterion with the supposition that debonding stress does not depend on particle size reveals that smaller particles increase facture toughness. The increase is significant for composites with particle size distribution functions that show small standard deviations. However, should the debonding energy at the interface be proportional to the particle diameter then the crack resistance remains constant by changing particle size for both debonding criteria.

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### 1. Introduction

Throughout the last decades the subject of improving the mechanical properties of particle filled polymers received large attention and a bulk of publications exists. A survey about this development was given by Fu et al. [1]. The particle size plays a decisive role among the structural and mechanical properties of the components and at the interface that influence the crack resistance, see, for example, Singh et al. [2]. Depending on the used particle or matrix materials crack resistance may increase or decrease with changing particle size distribution parameters. The variation of composite fracture toughness with particle volume fraction for different particle sizes is shown in Fig. 1 for an aluminium-polyester composite. Modelling of crack resistance for composites with particles of mean size was proposed for example by Evans et al. [3], Huang and Kinloch [4], Hsieh et al. [5], Williams [6] and Refs. [7,8]. Evans and Faber [9] developed a basic model for fracture toughness calculation considering particle size distribution, but basic equations are contradictory in the used approximations concerning the particle size dependence.

In the present paper the crack resistance of particle composites caused by particle/matrix debonding was modelled. As in real composites the particle size distribution is considered. The sequence of analysis consists of the presentation of basic equations for crack resistance calculation, followed by the description of the stress field in front of a crack. The size of the dissipation zone was calculated on the basis of linear elastic fracture mechanics. The volume specific debonding energy is calculated as a function of the position in front of a crack for different debonding stress criteria. Subsequent integration over the particle size distribution and the dissipation zone size provides the contribution of particle debonding to the crack resistance of the composite. The analysis of the particle debonding mechanism will be aimed to understand the direct contribution to crack resistance and as a trigger for subsequent mechanisms as, for example, plastic void growth.

#### 2. Crack resistance, fracture toughness

To initiate the propagation of an existing crack, energy must be available. The composite energy release rate *G* (available from the change of the elastic energy and the applied load for an increment of crack growth) must at least be equal to the energy necessary  $R_c$  (crack resistance) to initiate crack propagation. This is expressed usually as:  $G \ge G_c = R_c$  with  $G_c$  as fracture toughness of the composite (all these quantities as energy per unit area of crack growth). The fracture processes act in different zones: there are processes immediately near the crack surfaces, which is termed as process zone. The second group are the more extended structural processes, as for example particle/matrix debonding, which takes place in the debonding zone. These zones are illustrated in Fig. 2. This kind of subdivision of the region in front of the crack can be





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<sup>0927-0256/\$ -</sup> see front matter © 2013 Elsevier B.V. All rights reserved. http://dx.doi.org/10.1016/j.commatsci.2013.04.017

#### Nomenclature

$A_d$	debonded particle surface	$R_{dz}$	specific dissipation zone e
$A_{ii}^k, B_i^l$	$K_i, C_{ii}^k$ stress concentration factors, $k = (p, m), i = (x, y, z)$		crack)
B	material parameter as defined in Eq. (5.7)	$R_{pz}$	specific process zone ene
$d_p$	particle diameter and mean particle diameter if no size	•	crack)
	distribution	$S_f$	surface parameter as defin
$d_{p,i}$	particle diameter of the class <i>j</i>	v	particle volume fraction
$d_{p,\min}$	, $d_{p,\text{max}}$ minimum and maximum particle diameter,	$v_i$	particle volume fraction o
17	respectively	Ň	total volume of particles
$E_c$	composite modulus	$V_f$	volume parameter as defi
$E_p$	particle modulus	$\dot{W}_d$	debonding energy of one
$\dot{E_m}$	matrix modulus	β	shape factor of dissipation
$f_n$	normalized particle size distribution function	γd	specific fracture surface en
G	energy release rate (energy per unit area of crack)		(specific debonding energ
$G_c$	critical energy release rate of the composite	κ	stress concentration facto
h <sub>j</sub>	relative frequency of particles of the diameter class j	$\eta_d$	energy density of the de
m	parameter of the debonding probability distribution		per volume)
$N_j$	number of particles with the diameter <i>d<sub>p,j</sub></i>	$v_m, v_p$	Poisson's ratio of matrix a
$N_p$	total number of particles	$\rho, \phi$	radial and angular coord
$n_j$	particle volume density of diameter class <i>j</i>		with the origin at the crac
$n_p$	total particle volume density	$ ho_d$	half width of dissipation a
$P_d$	probability of debonding at the particle/matrix interface	$\sigma_{c,ij}$	composite stress compone
r, θ, ¢	spherical coordinates of the particle	$\sigma_{c,d}$	composite stress ( $\sigma_{c,zz}$ ) for
$R_c, R_n$	n crack resistance of composite and matrix (energy per	$\sigma_{d}$	debonding stress (radial s
	unit area of crack)		matrix interface, stress cr
$R_c^0$	composite crack resistance for mean particle size, stress	$\sigma_d^\gamma$	debonding stress (radial s
	debonding criterion		matrix interface, energy c
$R_{c,en}^0$	composite crack resistance for mean particle size, en-	$\sigma_0$	mean debonding strength
_	ergy debonding criterion		distribution
$R_d^0$	specific debonding energy for mean particle size, stress	$\omega_d$	slope of the function: $\gamma_d$ =
	debonding criterion		
$R_{d,en}^0$	specific debonding energy for mean particle size, energy		
,	debonding criterion		

traced back to the works by Evans and Faber [9]. In the following only the debonding process is taken into account, for other possible mechanisms the same general procedure can be used. The total crack resistance can be calculated by the separate contributions of the mechanisms in the process zone and debonding within the dissipation zone, as:

$$R_{c} = R_{pz} + R_{dz} = R_{pz} + R_{d} = R_{pz} + 2 \int_{0}^{\rho_{d}} \eta_{d}(\rho) d\rho \quad \text{or}$$

$$R_{c} = R_{pz} + 2\eta_{d}\rho_{d}$$
(2.1)



Fig. 1. Normalized fracture toughness of aluminium-polyester composites plotted against particle volume fraction for various particle sizes. Adapted from [2].

$R_{dz}$	specific dissipation zone energy (energy per unit area of	
	crack)	
$R_{pz}$	specific process zone energy (energy per unit area of	
	crack)	
$S_f$	surface parameter as defined in Eq. (5.4)	
ν	particle volume fraction	
$v_i$	particle volume fraction of the diameter class <i>j</i>	
V	total volume of particles	
$V_f$	volume parameter as defined in Eq. $(3.2)$	
Ŵd	debonding energy of one particle	
β	shape factor of dissipation zone	
γd	specific fracture surface energy of fibre/matrix interface	
	(specific debonding energy)	
κ	stress concentration factor	
$\eta_d$	energy density of the debonding mechanism (energy	
	per volume)	
$v_m, v_p$	Poisson's ratio of matrix and particle, respectively	
$ ho$ , $\phi$	radial and angular coordinates of the particle centres	
	with the origin at the crack tip	
$ ho_d$	half width of dissipation zone for debonding	
$\sigma_{c,ij}$	composite stress components, $i = (x, y, z)$	
$\sigma_{c,d}$	composite stress ( $\sigma_{c,zz}$ ) for particle/matrix debonding	
$\sigma_d$	debonding stress (radial stress component) at fibre/	
	matrix interface, stress criterion	
$\sigma_d^{\gamma}$	debonding stress (radial stress component)at fibre/	
u	matrix interface, energy criterion	
$\sigma_0$	mean debonding strength, parameter of Weibull	
	distribution	
$\omega_d$	slope of the function: $\gamma_d = \omega_d d_p$	
	· •	

where  $R_{pz}$  and  $R_{dz}$  are the specific fracture energies of the process and dissipation zone, respectively (as energy per unit area of crack),  $\eta_d$  is the volume density of debonding energy,  $\rho$  is the distance from the crack tip and  $\rho_d$  is the width of the debonding zone. For the special case that  $\eta_d$  is independent of the distance,  $\rho$ , to the crack tip the integral can be carried out directly (second equation of Eq. (2.1)). For the calculation of the width of debonding zone,  $\rho_d$ , the approximation for the plastic zone of homogeneous materials, as used for example by Wetzel et al. [10], is extended for the compos-



Fig. 2. Debonding and process zones in front of a crack; radius of debonding zone  $\rho_d$ , particle size distribution.

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