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## Helium retention behavior in simultaneously He $^+$ -H $_2^+$  irradiated tungsten



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**NUCLEAR MATERIALS** 

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## highlights are the control of

 $\bullet$  High temperature TDS (~1773 K) was used to study He retention in simultaneously He<sup>+</sup>-H $_2^+$  irradiated W.

• Growth of He bubbles was significantly suppressed 1.0 keV  $H_2^+$ .

 $\bullet$  He retention was enhanced when the H $_2^+$  energy was higher due to the additional damage.

### article info

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#### **ABSTRACT** abstract

The purpose of this study is to elucidate helium (He) retention behavior in tungsten (W) under simultaneous He and hydrogen (H) irradiation. Polycrystalline-W was irradiated by He $^+$  and H $^+_2$  simultaneously with the energy of 1.0 keV and 3.0 keV. He<sup>+</sup> fluences were (0.5, 1.0, 10)  $\times$  10<sup>21</sup> He<sup>+</sup> m<sup>-2</sup> and H<sub>2</sub><sup>+</sup> fluence was 1.0  $\times$  10<sup>22</sup> H<sup>+</sup> m<sup>-2</sup>, respectively. After irradiation, He desorption behavior was investigated by high temperature thermal desorption spectroscopy (HT-TDS) in the temperature range of R.T.-1773 K. Micro-structure changes of W after irradiation were observed by TEM. It was found that simultaneous irradiation with different H<sub>2</sub><sup>+</sup> energy significantly changed He retention behavior. 1.0 keV H<sub>2</sub><sup>+</sup> suppressed the He bubble growth and no bubbles can be observed at room temperature. On the other hand, 3.0 keV H $_2^+$  facilitated the formation of He bubbles and increased the He retention due to the additional damage introduction by energetic  $H_2^{\dagger}$ .

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#### 1. Introduction

Due to its low hydrogen isotope retention and high melting point, tungsten (W) is considered as the candidate plasma-facing materials (PFMs) for the ITER divertor and other fusion devices  $[1-3]$  $[1-3]$  $[1-3]$ . In ITER, divertor materials will be exposed to high density ions irradiation with high fluence and low energy ranging from 10 eV to several keV [[4](#page--1-0)]. In the recent long-pulse He discharges in Large Helical Device (LHD), the ion flux of typical divertor plasma were measured to be ~10 $^{23}$  ions m $^{-2}$ s $^{-1}$  and on the other hand, the flux and energy of the incident particles on the first-wall surface

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would be  $\sim$ 2 keV and  $\sim$ 10<sup>19</sup> ions m<sup>-2</sup>s<sup>-1</sup> [\[5\]](#page--1-0). As tritium (T) and deuterium (D) will be used as the fuels in the fusion reactor, irradiation of these energetic ions on W lead to the retention of T in W  $[6-8]$  $[6-8]$  $[6-8]$  $[6-8]$ . Since T resources are very limited and radioactive, T retention results in the fuel cost and causes safety concerns [\[9\]](#page--1-0). Understanding of hydrogen retention and recycling is necessary for the design of fusion reactor. Recycling is the return rate of the cold hydrogen isotope to the plasma, controlling the fuel rate and lowering the plasma temperature [\[3\]](#page--1-0). On the other hand, He is one of the product of D-T fusion reaction:  $D + T \rightarrow He + n$ . He irradiation results in the growth of He bubbles and surface modification in W, i.e. tungsten fuzz, which significantly changes hydrogen isotope retention behavior  $[10-13]$  $[10-13]$  $[10-13]$  $[10-13]$ . In order to clarify the He effect on hydrogen retention, a variety of irradiation experiments on W using plasma devices or ion beams were carried out. Sato et al. [[14\]](#page--1-0) found \* Corresponding author. that large He bubbles were formed near the surface for 3.0 keV He<sup>+</sup> irradiated W at 1173 K with the fluence of  $1.0 \times 10^{21}$  He<sup>+</sup> m<sup>-2</sup>. It was also considered that the existence of He bubbles results in the decrease of W atom ratio on the surface region, which is responsible for the decease of hydrogen retention  $[15-17]$  $[15-17]$  $[15-17]$  $[15-17]$  $[15-17]$ . Recently, the research result from Baldwin et al. [[18\]](#page--1-0) indicated that apart from this effect, the He bubble layers also change the migration energy in W for hydrogen isotope.

To discuss the interaction of He with hydrogen in W, we need more information on He retention behavior. However, most experiments were related to the surface modification or D retention by He irradiation or He-D plasma exposure, systematical study on the He diffusion and retention is rather limited. As observed by Miyamoto et al. [[19\]](#page--1-0), that the depth distribution of He bubbles is deeper than the injected range of He ions in D-He plasma exposure. It is believed that He retention behavior would be quite different in the simultaneous irradiation due to the existence of hydrogen isotopes.

To understand the He retention behavior, in the earlier stage, Tokitani et al. [[20](#page--1-0)] performed the 2.0 keV He<sup> $+$ </sup> irradiation on 304SS steel with the flux of  $1.0 \times 10^{17}$  He<sup>+</sup> m<sup>-2</sup> s<sup>-1</sup> with various fluences and found that dense He bubbles were formed. Further irradiation in LHD and subsequent thermal desorption spectroscopy (TDS) measurement of SUS316L suggested that He desorption at different temperatures can be roughly divided into three stages according to different trapping sites and most of He desorbed at higher temperature above 1000 K was attributed to the He released from He bubbles [\[5\]](#page--1-0). However, with respect to W material, the experimental data of He retention behavior is scarce. Herein, simultaneous He<sup> $+$ </sup>-H $_2^{\scriptscriptstyle\pm}$  irradiation on W with the ion beam energy of 1.0 keV or 3.0 keV was performed. To study the H effect on He retention behavior at different dose, He fluence was set at  $1.0 \times 10^{21}$  He<sup>+</sup> m<sup>-2</sup> and  $1.0 \times 10^{22}$  He m<sup>-2</sup>, and H fluence was set at  $1.0 \times 10^{22}$  H<sup>+</sup> m<sup>-2</sup>.

The objective of this study is to elucidate the He retention behavior in W, especially He retention behavior in simultaneous He $^+$ -H $^+_2$  irradiation. The other aim is to reveal He and H interaction using different combination of ion energy, which we hope it could be helpful for the understanding of fuel recycling and plasma interaction with materials.

#### 2. Experimental procedure

#### 2.1. Simultaneous He $^+$ -H $^+_2$  irradiation

Polycrystalline W (10 mm $^{\circ}$ , 0.5mm $^{\rm t}$ , 99.99% purity from A.L.M.T. Corp. Ltd) samples were prepared from a rod of tungsten under stress-relieved conditions supplied by Allied Tungsten, whose grains were normal to the surface. The samples were polished mechanically to mirror-finish surfaces and cleaned by an ultrasonic bath with acetone, ethanol and de-ionized water for 5 min each. It was preheated at 1173 K for 30 min under ultrahigh vacuum  $(<10^{-6}$  Pa) to release the stress introduced during polishing. Thereafter, simultaneous He $^+$ -H $_2^+$  irradiation and single He $^+$  irradiation were performed by the triple ion beam device at Shizuoka University [\[21](#page--1-0)]. This device consists of a He gun, a D gun (H $_2^+$  or D $_2^+$ ), and a carbon gun, combined with a TDS. The energy of He $^+$  and H $_2^{\rm +}$ beam can be set in the range of  $1.0-3.0$  keV, while the maximum flux can reach  $5.0 \times 10^{18} \text{ H}^+ \text{ m}^{-2}$  or  $1.0 \times 10^{18} \text{ He}^+ \text{ m}^{-2}$ . A Faraday Cup was attached beside the sample holder. Before each irradiation, the flux was confirmed by the Faraday Cup and then the sample holder was moved to irradiation position. The temperature was monitored by a thermal couple directly attached to the surface of the sample during irradiation. For the comparison of He retention behavior at different irradiation conditions,  $He<sup>+</sup>$  flux was fixed at  $1.0 \times 10^{17}$  He<sup>+</sup> m<sup>-2</sup> s<sup>-1</sup> with the fluence of  $1.0 \times 10^{21}$  He<sup>+</sup> m<sup>-2</sup>

during irradiation. Besides, in the case of reproducibility experiment, higher He flux of  $1.0 \times 10^{18}$  He<sup>+</sup> m<sup>-2</sup> s<sup>-1</sup> was used to achieve the fluence of  $1.0 \times 10^{22}$  He<sup>+</sup> m<sup>-2</sup>.

To study the He transportation through or towards the hydrogen layer, different combination of He $^+$  and H $_2^{\scriptscriptstyle +}$  with the energy of 1.0 and 3.0 keV were used. Fig. 1 shows the results of SRIM calculation of He<sup>+</sup> and H $_2^+$  depth distribution in W with different ion energy. 3.0 keV He $^+$  and 3.0 keV H $_2^+$  show almost the same depths of about 40 nm. On the other hand, the depths of 1.0 keV He $^+$  and 1.0 keV H $_2^{\rm +}$ are 20 nm. Therefore, the influence of hydrogen depth on He retention behavior can be investigated by using different combination of ion energy.

#### 2.2. High-temperature thermal desorption spectroscopy (HT-TDS) measurement

The HT-TDS measurement was performed from room temperature (R.T.). to 1773 K with the ramping rate of 30 K/min at National Science Institute for Fusion Science (NIFS). Before each measurement, the quadruple mass spectrometer (QMS) and He mass position were calibrated by the standard He leak bottle with the leak rate of  $1.4 \times 10^{-7}$  Pa m<sup>3</sup> s<sup>-1</sup> to ensure the accuracy of the data.

#### 2.3. Transmission electron microscope (TEM) observation

Surface of the samples was observed by TEM at Kyushu University. This TEM device can heat the sample to different temperature during observation. Thus, isochronal heating for the irradiated samples before TDS measurement were performed at 573 K  $-1173$  K to observe the microstructure changes during annealing.

#### 2.4. Glow-discharge optical emission spectroscopy (GD-OES) measurement

He depth profile in the irradiated W was measured by GD-OES with Ne plasma at University of Toyama. To maximize the intensity of He emission, high pressure (1200 Pa) and high power (80 W) were used during the measurement. 7 mm diameter anode was used and the time interval was set at 0.05 s. The sputtering rate was determined by measuring the depth of sputtering crater using a surface profiler (Tokyo Seimitsu SURFCOM 1500DX). The detail of the measurement of described in Ref. [\[23\]](#page--1-0).



Fig. 1. Ion depth profile calculated by SRIM code [[22](#page--1-0)].

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