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Creep deformation of Co-Re-Ta-C alloys with varying C content – investigated in-situ by simultaneous synchrotron radiation diffraction

Lukas Karge^{a,*}, Ralph Gilles^a, Debashis Mukherji^b, Andreas Stark^d, Premek Beran^c, Norbert Schell^d, Michael Hofmann^a, Pavel Strunz^c, Johannes Häusler^a, Joachim Rösler^b

^aHeinz Maier-Leibnitz Zentrum (MLZ), Technische Universität München, Lichtenbergstr. 1, 85747 Garching bei München, Germany

^bTechnische Universität Braunschweig, Zentrum für Werkstoffe, Langer Kamp 8, 38106 Braunschweig, Germany

^cNuclear Physics Institute of the CAS, Řež 130, 25068 Řež, Czech Republic

^dInstitute of Materials Research, Helmholtz-Zentrum Geesthacht, Max-Planck-Strasse 1, 21502 Geesthacht, Germany

Abstract

The creep deformation of precipitation hardened Co-Re-Ta-C alloys is investigated during in-situ synchrotron diffraction experiment at 1373 K. At room temperature, the alloys have a structure consisting of ϵ -Co (hcp) and metastably retained γ -Co (fcc) and are strengthened by precipitates of the mono-carbide of Ta, which are finely dispersed in the alloy matrix. The alloy exhibits an allotropic $\epsilon \rightarrow \gamma$ -Co phase transformation when heating to > 1173 K. A lower C content in the alloy generally promotes this transformation. It is shown that this transformation is strongly influenced by application of compressive load. The transformation $\epsilon \rightarrow \gamma$ -Co at high temperature under load leads to microstructure refinement and subsequently to dissolution of hardening precipitates. This results in a considerable acceleration of the creep rate. Further, the equilibrium ratio of γ/ϵ -Co phase is significantly altered under compressive load. This behavior is attributed to a volume relaxation as the ϵ - and γ -Co phase have different unit cell volumes.

Keywords: High temperature creep, Co alloys, Synchrotron diffraction, Neutron diffraction, Dilatometer

1. Introduction

Co-Re base alloys show promise for high temperature application in gas turbines, due to their relatively high melting range (1760 K-1830 K) compared to modern Ni-base superalloys [1–4]. One of the primary requisites in such application is a good creep resistance. The Co-Re alloys are precipitation strengthened by a fine (~ 30 nm) dispersion of TaC precipitates. It was shown that TaC particles in this morphology effectively interact with dislocation during high temperature creep exposure [5]. The TaC precipitates are stable in volume fraction up to temperature of 1473 K in the Co-Re matrix [6]. Further, the C/Ta ratio is a crucial parameter for the strengthening, since it influences size and volume fraction of strengthening TaC precipitates [7]. Lower C content results in a lower volume fraction of precipitates but also in a slightly lower coarsening rate of the precipitates.

Pure Co exists in the two allotropic forms ϵ -Co (hexagonal close-packed crystal structure) at temperature ≤ 690 K and γ -Co (face centered cubic structure) at higher temperatures. In alloys containing 17 atomic percent Re the allotropic transformation temperature is above 1480 K, as Re stabilizes the hcp phase [8–10]. The transformation exhibits a hysteresis upon cooling down to 1400 K. During fast cooling, generally some of the γ -Co phase is metastably retained at room temperature (RT). Therefore, the Co-Re matrix is present as a mixture of $\gamma+\epsilon$ -Co phases (with metastable γ -Co) in initial solution heat treated condition. When heating to the temperature interval 1173-1373 K, the metastable γ -Co transforms over time to the thermodynamic equilibrium ϵ -Co phase. Up to a temperature of ~ 1500 K, ϵ -Co phase is dominant in the matrix and only a low amount of γ -Co phase remains in equilibrium. The exact microstructure is sensitive to the holding time at a respective temperature and the C content has an influence on the allotropic matrix phase transformation temperature.

The matrix structure and the evolution of the hardening TaC phase have a strong influence on the creep behavior of the alloy and can only be understood if

*Corresponding author

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