



Synthesis and characterization of ϵ -VOPO₄ nanosheets for secondary lithium-ion battery cathode



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Abstract: Vanadium (III) phosphate monoclinic VPO₄·H₂O was synthesized hydrothermally. The ϵ -VOPO₄ nanosheets, formed by the oxidative de-intercalation of protons from monoclinic VPO₄·H₂O, can reversibly react with more than 1 mol lithium atoms in two steps. Crystal XRD analysis revealed that the structure of the ϵ -VOPO₄ nanosheets is monoclinic with lattice parameters of $a=7.2588(4)$ Å, $b=6.8633(2)$ Å and $c=7.2667(4)$ Å. The results show that the ϵ -VOPO₄ nanosheets have a thickness of 200 nm and uniform crystallinity. Electrochemical characterization of the ϵ -VOPO₄ monoclinic nanosheets reveals that they have good electrochemical properties at high current density, and deliver high initial capacity of 230.3 mA·h/g at a current density of 0.09 mA/cm². Following the first charge cycle, reversible electrochemical lithium extraction/insertion at current density of 0.6 mA/cm² affords a capacity retention rate of 73.6% (2.0–4.3 V window) that is stable for at least 1000 cycles.

Key words: lithium-ion battery; nanosheet; vanadium phosphate; cathode material; synthesis; characterization

1 Introduction

During the past decade, the energy shortage and the environment pollution have become serious due to the development of economy and the increase of the population. As a result, the intermittence of the renewable energies such as wind, wave and solar outputs, better energy storage and assistance systems have been the main focuses. However, these renewable energies are dependent on time and season [1]. Undoubtedly, energy storage systems should be developed and used to the reality, which is of vital importance to the social development in the future. Among various storage technologies, the development of battery systems with high energy density as well as long-term cycling stability is a crucial step on the path toward broad implementation of hybrid or electric cars and is a key element for broad adoption of renewable energy technologies [2–4].

Nowadays, lithium-ion batteries have been widely used in many fields as power suppliers for mobile equipment. Most of the present lithium batteries used in electronic devices or hybrid electric vehicles employ transition metal oxides such as LiCoO₂, LiMn₂O₄ or mixed metal analogs such as Li(Ni,Mn,Co)O₂, olivine LiFePO₄ as the active cathode materials [5–8]. The present commercial cathode material, LiCoO₂, possesses some problems in large-scale utilization, due to its high cost and low safety [9–11]. The oxide-based materials (LiCoO₂, LiNiO₂), the spinel LiMn₂O₄, and all their substituted variations have been extensively studied as positive electrode materials [12–17]. Vanadium phosphates have been extensively studied because of their importance in selective catalysis of hydrocarbon oxidation reactions. Vanadium phosphates are also attractive due to their higher free energy of reaction, and the greater possible change of oxidation state.

Seven phases of VPO₄ were synthesized and their lithium intercalation properties were investigated as

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cathode materials [15,17–19]. Several vanadyl phosphates VOPO_4 were investigated as cathode materials [20,21]. Till now, the polymorphs have not satisfactory electrochemical performance, and more efforts are needed to make use of the high operating voltage and reasonable theoretical capacity of VOPO_4 . The $\varepsilon\text{-VOPO}_4$ compound has particularly interesting properties, such as an approximately 4 V flat discharge potential, which is about 0.5 V higher than that of LiFePO_4 , and higher electronic conductivity, leading to the possibility of attaining higher power systems and theoretical energy density compared with LiFePO_4 [8]. This material adopts a stable 3D tunneling structure with the theoretical specific capacity of $\sim 168 \text{ mA}\cdot\text{h/g}$ ($\alpha\text{-LiVOPO}_4$). Compared with LiFePO_4 , $\varepsilon\text{-VOPO}_4$ has a much higher conductivity ($1\times 10^{-6} \text{ S/cm}$ vs $1\times 10^{-10} \text{ S/cm}$ of LiFePO_4) [22,23]. Furthermore, similar to LiFePO_4 , the rate capability of VOPO_4 can be improved by decreasing the particle size [19]. For $\varepsilon\text{-VOPO}_4$, we might intercalate two lithium ions into the structure, which may deliver larger capacity if both lithium ions participate in electrochemistry. Overall, all these advantages make $\varepsilon\text{-VOPO}_4$ a great candidate for next generation of high energy density lithium-ion batteries. Although various preparation methods have been reported for $\varepsilon\text{-VOPO}_4$, there is a possibility that $\varepsilon\text{-VOPO}_4$ with larger capacity and better cycle stability can be obtained by changing the preparation method and crystal structure size. In Ref. [21], we reported the electrochemical behavior of the $\beta\text{-VOPO}_4/\varepsilon\text{-VOPO}_4$ composite and nanostructural $\varepsilon\text{-VOPO}_4$ formed from the monoclinic and tetragonal forms of $\text{VPO}_4\cdot\text{H}_2\text{O}$. In this study, $\varepsilon\text{-VOPO}_4$ nanosheets were synthesized and their electrochemical performances as the cathode materials were investigated.

2 Experimental

The $\varepsilon\text{-VOPO}_4$ was prepared using the method as described by SONG et al [24]. The monoclinic $\text{VPO}_4\cdot\text{H}_2\text{O}$ precursor was synthesized in the aqueous solution, the reactants were 16.7 mL standardized H_3PO_4 solution (2.39 mol/L), 1.60 g VCl_3 , and 20 mL deionized water, with the pH adjusted by triethylmethylammonium hydroxide (20% in water) to 3 under stirring for 7 d. The monoclinic $\text{VPO}_4\cdot\text{H}_2\text{O}$ precursors were obtained in corresponding solutions in 125 mL PTFE-lined Parr autoclaves at 180 °C for 3 d, then ramp-cooled to room temperature at 0.5 °C/min. The $\varepsilon\text{-VOPO}_4$ from the monoclinic phase was obtained by heating the precursor in oxygen at 500 °C for 3 h.

The electrochemical properties of these samples were evaluated in 2325-type coin cells. The cathodes were prepared by grinding 80% (mass fraction) active material with 10% (mass fraction) conductive carbon and

10% (mass fraction) PVDF binder in a mortar and pestle. NMP (1-methyl-2-pyrrolidinone, Aldrich) was added to create a paste, which was laminated onto aluminum foil and vacuum dried at 90 °C before use. The electrolyte was 1 mol/L LiPF_6 (lithium hexafluorophosphate) dissolved in a solution of ethylene carbonate (EC) and dimethyl carbonate (DMC) (LP30 from EM Industries) with a volume ratio of 1:1; a Celgard 2400 separator (Hoechst Celanese) was used. The coin cells were assembled in a glove box filled with pure helium. A Macpile potentiostat was used to cycle the cells at 0.09 mA/cm² between 2 and 4.3 V at room temperature.

3 Results and discussion

The diffraction data were collected for the prepared material over a 12 h period on a Scintag XDS 2000 powder diffractometer using $\text{Cu K}\alpha$ radiation ($\lambda=1.5418 \text{ \AA}$) operating in a $\theta\text{--}\theta$ geometry between 2θ values of 15° and 75° with a step size of 0.02° and a collection time of 8 s per point. The XRD pattern of the prepared material is shown in Fig. 1, which agrees with those previously reported [20,24]. The XRD pattern can be indexed as a monoclinic structure with a space group Cc . Rietveld refinement was carried out using GSAS + EXPGUI [25]. Cell parameters were refined to $a=7.2588(4) \text{ \AA}$; $b=6.8633(2) \text{ \AA}$; $c=7.2667(4) \text{ \AA}$; $\beta=115.391(3)^\circ$; $V=328.26(2) \text{ \AA}^3$ (space group Cc). The XRD data revealed that the sample was a single-phase material. The two main diffraction peaks were observed at 26° and 29°. This agreed with our previous report [20].

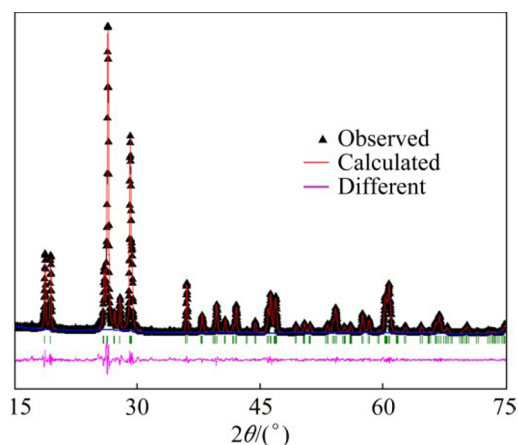


Fig. 1 XRD pattern and Rietveld refinement of $\varepsilon\text{-VOPO}_4$ nanosheets

The $\varepsilon\text{-VOPO}_4$ nanosheets have been derived from monoclinic $\text{VPO}_4\cdot\text{H}_2\text{O}$. The crystal structure of $\varepsilon\text{-VOPO}_4$ consists of corner sharing VO_6 octahedra and PO_4 tetrahedra [26], as shown in Fig. 2. The easy inversion of vanadyl bonds offers a mechanism which allows the VOPO_4 host lattice to fine-adjust for the accommodation

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