



## Featured Letter

Resistive switching and ferroelectric properties in BiFeO<sub>3</sub> superlattice filmsMeiyou Guo<sup>a</sup>, Guoqiang Tan<sup>a,\*</sup>, Wei Yang<sup>a</sup>, Long Lv<sup>b</sup>, Mintao Xue<sup>a</sup>, Yun Liu<sup>a</sup>, Huijun Ren<sup>c</sup>, Ao Xia<sup>a</sup><sup>a</sup> School of Materials Science and Engineering, Shaanxi University of Science & Technology, Xi'an 710021, China<sup>b</sup> College of Cryptography Engineering, Engineering University of PAP, Xi'an 710086, China<sup>c</sup> School of Arts and Sciences, Shaanxi University of Science & Technology, Xi'an 710021, China

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## ABSTRACT

Bi<sub>0.89</sub>Ho<sub>0.08</sub>Sr<sub>0.03</sub>Fe<sub>0.97–a</sub>Mn<sub>0.03</sub>Ni<sub>a</sub>O<sub>3</sub>/Bi<sub>0.89</sub>Ho<sub>0.08</sub>Sr<sub>0.03</sub>Fe<sub>0.97–b</sub>Mn<sub>0.03</sub>Ni<sub>b</sub>O<sub>3</sub> (Ni<sub>a</sub>/Ni<sub>b</sub>) superlattice films were prepared via a CSD method. The influences of interface effects on ferroelectric properties and resistance switching behaviors of the superlattice films were investigated. There was a weak interface effect in the superlattice films, which was caused by the misfit dislocation generated at the interfaces between Ni<sub>a</sub> and Ni<sub>b</sub>. The results indicate that the superlattice films show superior ferroelectric properties and good resistance switching behaviors, which is caused by the interface effect. Those superlattice films may have potential applications in multifunctional devices due to their excellent electric properties.

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## 1. Introduction

In recent years, multiferroic BiFeO<sub>3</sub> (BFO) has gained considerable attention, due to its underlying fascinating science and promising industrial applications, such as ferroelectric random access memories and resistive random access memories [1,2]. Compared with the single-phase BFO, ion doping is an effective method to improve the electrical properties of thin films, which makes it useful for the preparation of multifunctional devices [3,4]. However, doping can lead to the increase of the defects which may cause the pinning effect, and the doped thin films show obvious resistance switching (RS) behaviors with the increase of defect density and leakage current [5,6]. To further improve the electrical properties of thin films, the interface engineering is also a common method [7,8]. According to previous studies [9], Bi<sub>0.89</sub>Ho<sub>0.08</sub>Sr<sub>0.03</sub>Fe<sub>0.97–x</sub>Mn<sub>0.03</sub>Ni<sub>x</sub>O<sub>3</sub> (Ni<sub>x</sub>,  $x = 0.01 \sim 0.04$ ,  $a, b \in x$ ,  $a \neq b$ ) thin films have the similar crystal structure with different space groups, which leads to the lattice strain and the interface effect in Ni<sub>a</sub>/Ni<sub>b</sub> superlattice films. Therefore, the enhanced electrical properties are expected from the multi-element co-doping BFO superlattice films.

In this paper, the multi-element co-doping BFO superlattice films were prepared using the chemical solution deposition (CSD)

method. The influences of the interface effect on ferroelectric properties and resistance switching behaviors of the Ni<sub>a</sub>/Ni<sub>b</sub> superlattice films were investigated.

## 2. Experimental procedure

Ni<sub>a</sub>/Ni<sub>b</sub> superlattice films were prepared on the FTO (fluorine doped tin oxide)/glass substrates by the CSD method. The appropriate proportions of Bi(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O, Ho(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O, Sr(NO<sub>3</sub>)<sub>2</sub>, Fe(NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O, C<sub>4</sub>H<sub>6</sub>MnO<sub>4</sub>·4H<sub>2</sub>O and C<sub>4</sub>H<sub>6</sub>NiO<sub>4</sub>·4H<sub>2</sub>O were dissolved in the mixed solution of acetic anhydride and 2-methoxyethanol. Then the solutions were stirred continuously for 2 h at room temperature, stable precursor solutions of Ni<sub>x</sub> were obtained. The Ni<sub>a</sub> precursor solution was initially spin-coated on FTO glass and annealed at 550 °C. Then, the Ni<sub>b</sub> precursor solution was spin-coated on the Ni<sub>a</sub> layer and subsequently annealed at 550 °C. These procedures were repeated for 14 times to obtain the desired thickness of Ni<sub>a</sub>/Ni<sub>b</sub> superlattice films. The Ni<sub>x</sub> films were prepared by the same processes.

The structures of the films were studied by X-ray diffraction (XRD, Rigaku, D/MAX-2200). The leakage current densities and the hysteresis loops of the BFO films were tested by an Agilent B2901A instrument and a Radiant Multiferroic system. Capacitance–voltage (C–V) measurements were carried out with an Agilent E4980A Concise LCR meter.

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### 3. Results and discussion

Fig. 1(a) shows the XRD patterns of  $\text{Ni}_a/\text{Ni}_b$  superlattice films. The XRD diffraction peaks of all the films are in conformity with the Standard Card of PDF (JCPDS No. 74-2016). All the films have (1 1 0) preferred orientation. Fig. 1(b) shows the Rietveld-refined space group structural and lattice parameters of  $\text{Ni}_x$  films [9]. The  $\text{Ni}_a/\text{Ni}_b$  superlattice film is synthesized by two components,  $\text{Ni}_a$  and  $\text{Ni}_b$ , which are similar in structures as shown in Fig. 1(b). The lattice mismatches ( $f$ ) of the  $\text{Ni}_{0.02}/\text{Ni}_{0.03}$ ,  $\text{Ni}_{0.03}/\text{Ni}_{0.01}$ ,  $\text{Ni}_{0.01}/\text{Ni}_{0.04}$  and  $\text{Ni}_{0.03}/\text{Ni}_{0.04}$  superlattice films are  $-0.46\%$ ,  $0.20\%$ ,  $0.38\%$  and  $0.58\%$ , respectively according to Eq. (1) [10]. This indicates that there is a tensile stress at the vertical interfaces in  $\text{Ni}_{0.02}/\text{Ni}_{0.03}$  superlattice film, and in other  $\text{Ni}_a/\text{Ni}_b$  superlattice films there is a compressive stress. Therefore, the misfit dislocation is generated at the interfaces, resulting in strain relaxation in the superlattice films. While there is a strain in the interior of  $\text{Ni}_x$  films, but without a misfit dislocation ( $f = 0$ ), which cannot relax the strain.

$$f = \frac{2(a_{\text{Ni}_a} - a_{\text{Ni}_b})}{a_{\text{Ni}_a} + a_{\text{Ni}_b}} \quad (1)$$

where  $f$  is a lattice mismatch,  $a_{\text{Ni}_a}$  and  $a_{\text{Ni}_b}$  are the lattice parameters of R3c:H space group in  $\text{Ni}_a$  and  $\text{Ni}_b$  films.

Fig. 2 shows the ferroelectric properties of  $\text{Ni}_a/\text{Ni}_b$  superlattice films. In each diagram, the ferroelectric hysteresis loops ( $P$ - $E$ ) and instantaneous current curves ( $I$ - $E$ ) of  $\text{Ni}_{0.01}$ ,  $\text{Ni}_{0.02}$ ,  $\text{Ni}_{0.03}$ , and  $\text{Ni}_{0.04}$  films are represented by light color lines as control groups. In an applied electric field of 700 kV/cm, the remnant polarizations  $P_r$  ( $-P_r$ ) of  $\text{Ni}_a/\text{Ni}_b$  superlattice films are 122 ( $-118$ ), 100 ( $-84$ ), 70 ( $-57$ ) and 82 ( $-72$ )  $\mu\text{C}/\text{cm}^2$ , respectively. The coercive fields  $E_c$  ( $-E_c$ ) of  $\text{Ni}_a/\text{Ni}_b$  superlattice films are 294 ( $-248$ ), 288 ( $-245$ ), 321 ( $-254$ ) and 314 ( $-249$ ) kV/cm, respectively. The switching currents  $I_s$  ( $-I_s$ ) of  $\text{Ni}_a/\text{Ni}_b$  superlattice films are 1.0 ( $-1.0$ ), 0.8 ( $-0.8$ ), 0.4 ( $-0.5$ ) and 0.6 ( $-0.7$ ) mA, respectively. The ferroelectric leakage currents ( $I_L$ ) of  $\text{Ni}_a/\text{Ni}_b$  superlattice films are less than 0.25 mA. Compared with  $\text{Ni}_x$  films with obvious pinning effects caused

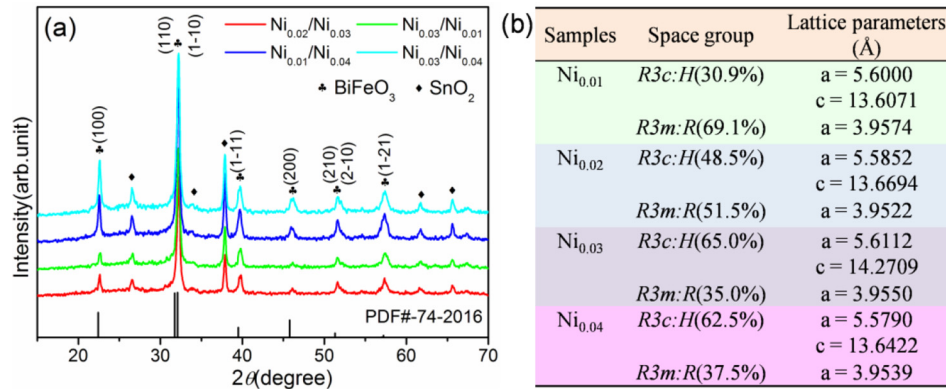


Fig. 1. (a) XRD patterns of  $\text{Ni}_a/\text{Ni}_b$  superlattice films; (b) Details of the Rietveld-refined structural parameters of  $\text{Ni}_x$  films.

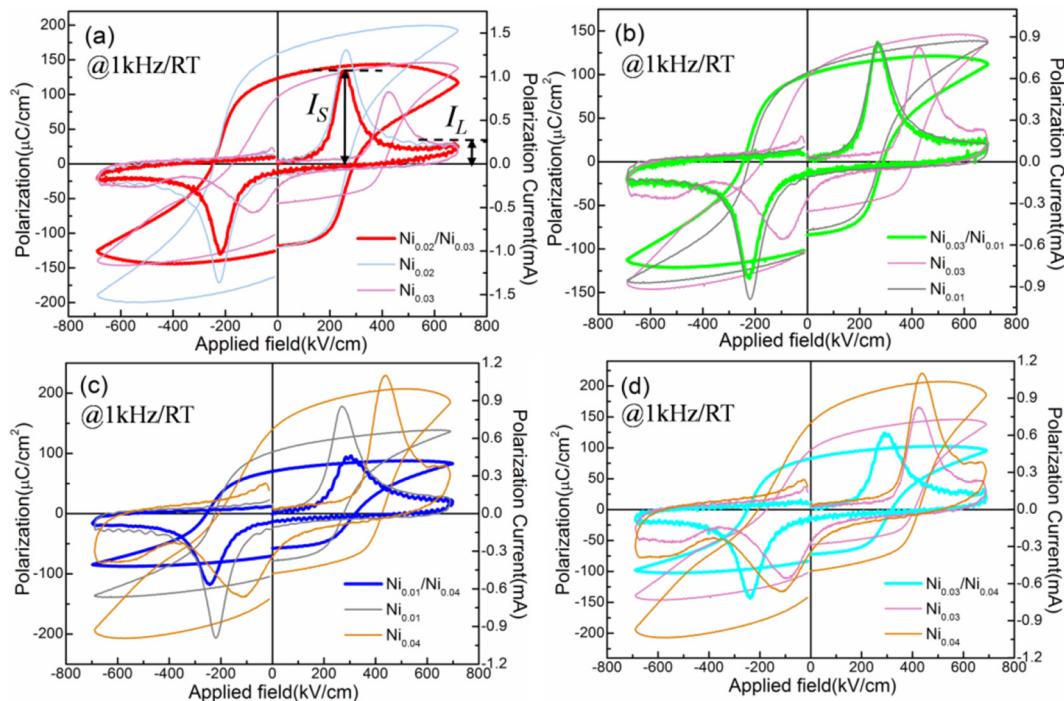


Fig. 2. Ferroelectric hysteresis loops and polarization current curves of (a)  $\text{Ni}_{0.02}/\text{Ni}_{0.03}$ ,  $\text{Ni}_{0.02}$  and  $\text{Ni}_{0.03}$ ; (b)  $\text{Ni}_{0.03}/\text{Ni}_{0.01}$ ,  $\text{Ni}_{0.03}$  and  $\text{Ni}_{0.01}$ ; (c)  $\text{Ni}_{0.01}/\text{Ni}_{0.04}$ ,  $\text{Ni}_{0.01}$  and  $\text{Ni}_{0.04}$ ; (d)  $\text{Ni}_{0.03}/\text{Ni}_{0.04}$ ,  $\text{Ni}_{0.03}$  and  $\text{Ni}_{0.04}$  films.

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