



# Effects of Gd addition on the microstructure, mechanical properties and shape memory effect of polycrystalline Cu-Al-Ni shape memory alloy

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## ABSTRACT

The effects of Gd addition on the microstructure, mechanical properties and shape memory effect of (Cu-13.0Al-4.0Ni)-xGd ( $x=0, 0.3, 0.9, 1.5$  and 6) (wt%) alloys were investigated. The results showed that the grain of (Cu-13.0Al-4.0Ni)-xGd alloys was refined by the Gd addition. The 18R martensite and a small amount of 2 h martensite were present for  $x < 0.9$  wt%, and dual phases containing 18R martensite and hexagonal structured AlCu<sub>4</sub>Gd phase were observed for  $x \geq 0.9$  wt%. In addition, Gd addition significantly enhanced the mechanical and shape memory properties. The compressive fracture strain of 15.3% and reversible strain of 6.6% were obtained in (Cu-13.0Al-4.0Ni)-0.3Gd alloy.

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## 1. Introduction

To date, high-temperature shape memory alloys (HTSMAs) have attracted more and more attentions [1,2]. Cu-Al-Ni HTSMAs have been widely investigated due to its lower cost [3–6] in comparison with some well studied HTSMAs, such as Ti-Ni-Pd, Ti-Ta and Ni-Mn-Ga [7–15], etc. However, polycrystalline Cu-Al-Ni alloys suffer from high brittleness, which is associated with large elastic anisotropy, intergranular cracking and large grain size [16,17]. Some methods were adopted to improve the mechanical properties of polycrystalline Cu-Al-Ni, including grain refinement [18] and fourth element addition [3, 19–21]. Published results showed that rare earth Gd addition can tailor the microstructure and mechanical properties of shape memory alloys [22–24], and accordingly, Gd is added into Cu-13.0Al-4.0Ni HTSMA to improve its plastic in the present work. The effect of Gd content on the microstructure, mechanical properties and shape memory effect (SME) of (Cu-13.0Al-4.0Ni)-xGd alloy is investigated in detail.

## 2. Material and methods

The nominal composition of studied alloys were (Cu-13.0Al-4.0Ni)-xGd ( $x=0, 0.3, 0.9, 1.5$  and 6) (wt%), and they were marked

as Cu0, Cu1, Cu2, Cu3 and Cu4, respectively. Copper, aluminum, nickel and gadolinium with high purity ( $\geq 99.97$  wt%) were melted in a non-consumed vacuum arc furnace to prepare polycrystalline button ingots. The ingots were remelted eight times to ensure homogeneity. The ingots were annealed in a vacuum quartz tubes at 850 °C for 24 h, and subsequently quenched into ice water.

The microstructures were observed by optical microscopy and Quanta 200FEG scanning electron microscopy (SEM). X-ray diffraction (XRD) measurements were performed by Rigaku D/max-rB with Cu  $K_{\alpha}$  radiation. The phase transformation temperatures were determined by Perkin-Elmer Diamond differential scanning calorimetry (DSC) with heating and cooling rate of 20 °C/min. The martensitic transformation temperatures are listed in Table 1. The compression and SME tests were performed at room temperature on Instron 5569 testing system at a crosshead displacement speed of 0.2 mm/min, and the size of the sample was  $\Phi 3 \times 5$  mm. The

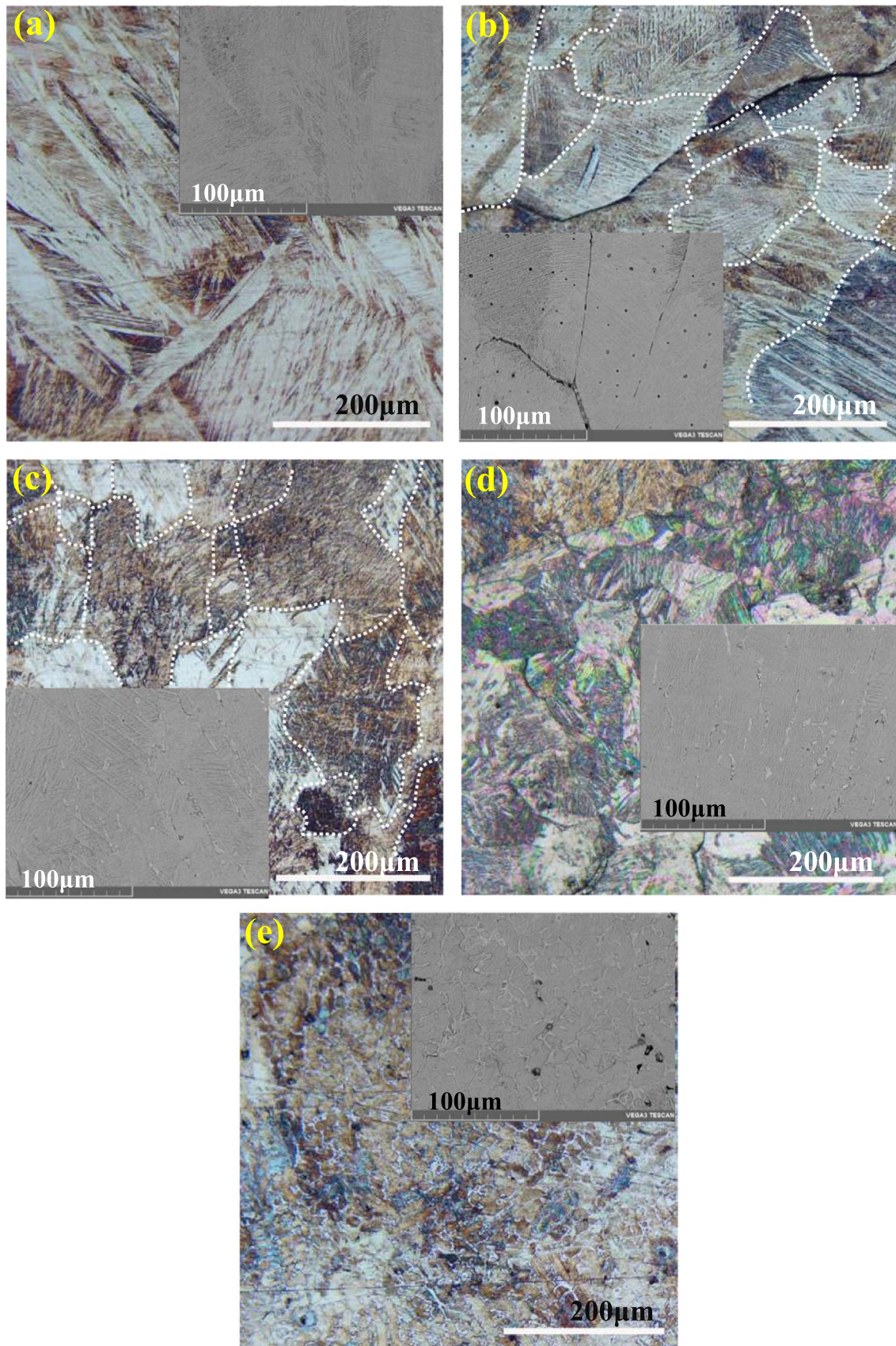
**Table 1**

The transformation temperatures of (Cu-13.0Al-4.0Ni)-xGd alloys (°C).

Sample	$A_s$	$A_f$	$M_s$	$M_f$
Cu0	325.27	376.57	228.59	210.10
Cu1	307.70	362.22	202.72	149.60
Cu2	319.19	353.25	228.19	211.73
Cu3	319.08	360.75	189.16	177.92
Cu4	315.43	364.37	234.78	205.31

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**Fig. 1.** Optical and SEM micrographs of solution treated (Cu-13.0Al-4.0Ni)-xGd alloys, (a)  $x=0$ ; (b)  $x=0.3$ ; (c)  $x=0.9$ ; (d)  $x=1.5$ ; (e)  $x=6$ .

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